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(71) Applicant: BASELL POLIOLEFINE ITALIA S.R.L. [IT/IT]; Via Soperga 14/A, 20127 Milano (IT).

(72) Inventors: MARCHINI, Roberta; C/O Basell Poliolefine Italia S.r.l., P.le G. Donegani, 12, 44122 Ferrara (IT). SPATARO, Stefano; C/O Basell Poliolefine Italia S.r.l., P.le G. Donegani, 12, 44122 Ferrara (IT). PASQUALI, Stefano; C/O Basell Poliolefine Italia S.r.l., P.le G. Donegani, 12, 44122 Ferrara (IT). MUSACCHI, Gianluca; C/O Basell Poliolefine Italia S.r.l., P.le G. Donegani, 12, 44122 Ferrara (IT).

(74) Agent: GAVERINI, Gaetano; c/o Basell Poliolefine Italia S.r.l., P.le Donegani 12, 44122 Ferrara (IT).

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(54) Title: SOFT AND FLEXIBLE POLYOLEFIN COMPOSITION

(57) Abstract: A polyolefin composition having a melting enthalpy ΔH_{fus} , measured by DSC at the second heating scan, of from 7 to 30 J/g, comprising: A) from 63 to 78% by weight of a copolymer of butene-1 with ethylene having a copolymerized ethylene content of up to 18% by mole and no melting peak detectable at the DSC at the second heating scan; B) from 22 to 37% by weight of (i) a propylene homopolymer, or (ii) a propylene copolymer, or (iii) a mixture of two or more of (i) and (ii), having a melting temperature T_m , measured by DSC at the second heating scan, of from 130°C to 165°C; wherein the amounts of A) and B) are referred to the total weight of A) + B).

SOFT AND FLEXIBLE POLYOLEFIN COMPOSITION

FIELD OF THE INVENTION

[0001] The present disclosure relates to a flexible and thermoplastic polyolefin composition having low values of shore hardness in combination with good tensile and elastic properties.

BACKGROUND OF THE INVENTION

[0002] Polyolefin compositions having high flexibility while maintaining a good thermoplastic behavior are disclosed in the art. They have been used in many application fields, due to the valued properties which are typical of polyolefins (such as chemical inertia, mechanical properties and nontoxicity).

[0003] They generally comprise crystalline and amorphous portions, in various relative amounts.

[0004] Such crystalline and amorphous portions can be present in the same polymer chain, and/or in distinct phases.

[0005] Depending upon the chemical composition of such portions, their relative amounts and the way in which they are combined in the polyolefin composition, different sets of properties are obtained.

[0006] However it is always difficult to achieve a good balance of flexibility, softness and thermoplastic behavior.

[0007] An example of flexible and thermoplastic composition is provided in PCT Publication No. WO2009080485, wherein low values of flexural modulus and Shore hardness are obtained by combining a crystalline copolymer of propylene and hexene-1 with a highly soluble, thus highly amorphous ethylene copolymer.

[0008] It has now been found that by combining specific butene-1 polymers with reduced amounts of propylene polymers, it is possible to obtain a flexible polyolefin composition having an excellent and unusual set of properties.

[0009] Such properties make it possible to use the compositions for preparing liners for closures, providing a tight and durable seal.

SUMMARY OF THE INVENTION

[0010] Thus the present invention provides a polyolefin composition having a melting enthalpy ΔH_{fus} , measured by DSC (Differential Scanning Calorimetry) at the second heating scan, of from 7 to 30 J/g, preferably from 8 to 28 J/g, comprising:

- A) from 63 to 78% by weight, preferably from 64 to 76% by weight, of a copolymer of butene-1 with ethylene having a copolymerized ethylene content of up to 18% by mole and no melting peak detectable at the DSC at the second heating scan;
- B) from 22 to 37% by weight, preferably from 24 to 36% by weight, of (i) a propylene homopolymer, or (ii) a propylene copolymer, or (iii) a mixture of two or more of (i) and (ii), having a melting temperature T_m , measured by DSC at the second heating scan, of from 130°C to 165°C, preferably from 131 to 165°C, more preferably from 131 to 160°C; wherein the amounts of A) and B) are referred to the total weight of A) + B) and the DSC second heating scan is carried out with a heating rate of 10°C per minute.

[0011] The composition provided herein has high softness (Shore A lower than 90), good tensile properties (elongation at break of 900 -1000%) and elastic properties (compression set at 22°C lower 50) while still maintaining a thermoplastic behavior due to the presence of a crystalline fraction, as revealed by the melting enthalpy values.

DETAILED DESCRIPTION OF THE INVENTION

[0012] The polyolefin composition as provided herein has low values of flexural modulus, preferably equal to or lower than 150 MPa, more preferably equal to or lower than 100 MPa in particular from 150 or from 100 to 20 MPa.

[0013] Particularly preferred relative amounts of the previously defined polymer components A) and B) are, due to the low compression set values achieved, of 69% by weight or lower of A) and of 31% by weight or higher of B), thus from 63 to 69% by weight of A) and from 31 to 37% of B), or from 64 to 69% by weight of A) and from 31 to 36% by weight of B), referred to the total weight of A) + B), even more preferably from 63 to 68% by weight of A) and from 32 to 37% of B), or from 64 to 68% by weight of A) and from 32 to 36% by weight of B), referred to the total weight of A) + B).

[0014] In combination with such preferred amounts, component B) is most preferably a propylene copolymer (i) or a mixture (iii) of a propylene homopolymer and a propylene copolymer.

[0015] The polyolefin compositions comprising from 70 to 78% by weight, preferably from 70 to 76% by weight of A) and from 22 to 30%, preferably from 24 to 30% of B), referred to the total weight of A) + B) are still particularly valuable in the preparation of extruded articles and in particular of liners for closures.

[0016] Moreover, the polyolefin composition as provided herein has a melting temperature T_m which is equal or close to the melting temperature T_m of the propylene homopolymer or

copolymer component B), namely from 130°C to 165°C, preferably from 132 to 165°C, more preferably from 132 to 160°C.

[0017] Generally a single melting peak is detected in the second DSC scan of the propylene homopolymer or copolymer component B) and in the second DSC scan of the polyolefin composition in the said temperature range.

[0018] Should more than one peak be detected, the temperature of the most intense melting peak in the said temperature range is to be taken as the T_m value for both component B) and the polyolefin composition comprising A) and B).

[0019] Thus the ΔH_{fus} value for the polyolefin composition as provided herein is preferably given by the area of the melting peak or the total area of melting peaks (if more than one) in the DSC temperature range from 130° to 160°C.

[0020] Preferred values of MIE for the said composition are from 0.5 to 8 g/10 min., where MIE is the melt flow index at 190°C with a load of 2.16 kg, determined according to ISO 1133.

[0021] Preferred Shore A values for the said compositions are of lower than 90, particularly equal to or lower than 88, the lower limit being of 70.

[0022] Correspondingly, the Shore D values are equal to or or lower than 20, in particular from 20 to 5, more preferably lower than 20, in particular from lower than 20 to 5.

[0023] The butene-1 copolymer component A) just after it has been melted and cooled does not show a melting peak at the second heating scan, however it is crystallizable, i.e. after about 10 days that it has been melted the polymer shows a measurable melting point and a melting enthalpy measured by DSC. In other words the butene-1 copolymer shows no melting temperature attributable to polybutene-1 crystallinity (T_m II) DSC, measured after cancelling the thermal history of the sample, according to the DSC method described herein below in the experimental part.

[0024] Moreover, the butene-1 copolymer component A) can have at least one of the following additional features:

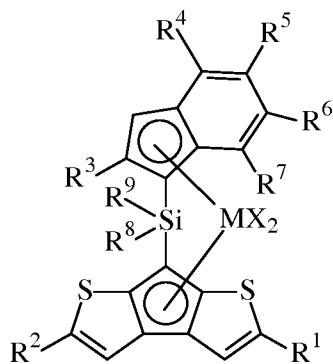
- MIE of from 0.5 to 3 g/10 min.;
- a lower limit of the copolymerized ethylene content of 12% by mole;
- a Shore A value equal to or lower than 80, more preferably equal to or lower than 70, in particular from 80 to 40 or from 70 to 40;
- a Shore D value equal to or lower than 20, in particular from 20 to 5, more preferably lower than 20, in particular from lower than 20 to 5;
- a M_w/M_n value, where M_w is the weight average molar mass and M_n is the number average molar mass, both measured by GPC, equal to or lower than 3, in particular from 3 to 1.5.

- a tension set of less than 30% at 100% of deformation at 23°C (ISO 2285), more preferably equal to or less than 20%, wherein the lower limit is of 5;
- a percentage of butene-1 units in form of isotactic pentads (mmmm%) greater than 80%, preferably equal to or greater than 85%, even more preferably equal to or greater than 90%, wherein the upper limit is 99%;
- tensile stress at break, measured according to ISO 527, of from 3 MPa to 20 MPa, more preferably from 4 MPa to 13 MPa;
- tensile elongation at break, measured according to ISO 527, of from 550% to 1000%; more preferably from 700% to 1000%;
- intrinsic viscosity (I.V.) equal to or higher than 1dl/g; more preferably equal to or higher than 1.5 dl/g, wherein the upper limit is of 3 dl/g;
- crystallinity of less than 30% measured via X-ray, more preferably of less than 20%;
- density of 0.895 g/cm³ or less, more preferably of 0.875 g/cm³ or less; wherein the lower limit is of 0.86 g/cm³;
- content of xylene insoluble fraction at 0°C of less than 15% by weight, wherein the lower limit is of 0%.

[0025] The butene-1 copolymer component A) can be obtained by polymerizing the monomer(s) in the presence of a metallocene catalyst system obtainable by contacting:

- a stereorigid metallocene compound;
- an alumoxane or a compound capable of forming an alkyl metallocene cation; and, optionally,
- an organo aluminum compound.

[0026] Preferably the stereorigid metallocene compound belongs to the following formula (I):



wherein:

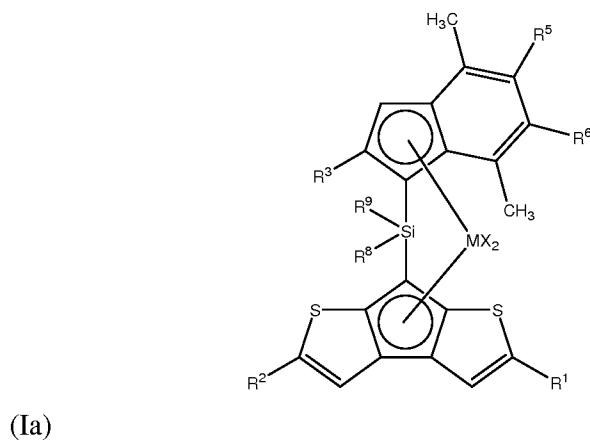
M is an atom of a transition metal selected from those belonging to group 4; preferably M is zirconium;

X, equal to or different from each other, is a hydrogen atom, a halogen atom, a R, OR, OR'O, OSO₂CF₃, OCOR, SR, NR₂ or PR₂ group wherein R is a linear or branched, saturated or unsaturated C₁-C₂₀-alkyl, C₃-C₂₀-cycloalkyl, C₆-C₂₀-aryl, C₇-C₂₀-alkylaryl or C₇-C₂₀-arylalkyl radical, optionally containing heteroatoms belonging to groups 13-17 of the Periodic Table of the Elements; and R' is a C₁-C₂₀-alkylidene, C₆-C₂₀-arylidene, C₇-C₂₀-alkylarylidene, or C₇-C₂₀-arylalkylidene radical; preferably X is a hydrogen atom, a halogen atom, a OR'O or R group; more preferably X is chlorine or a methyl radical;

R¹, R², R⁵, R⁶, R⁷, R⁸ and R⁹, equal to or different from each other, are hydrogen atoms, or linear or branched, saturated or unsaturated C₁-C₂₀-alkyl, C₃-C₂₀-cycloalkyl, C₆-C₂₀-aryl, C₇-C₂₀-alkylaryl or C₇-C₂₀-arylalkyl radicals, optionally containing heteroatoms belonging to groups 13-17 of the Periodic Table of the Elements; or R⁵ and R⁶, and/or R⁸ and R⁹ can optionally form a saturated or unsaturated, 5 or 6 membered rings, said ring can bear C₁-C₂₀ alkyl radicals as substituents; with the proviso that at least one of R⁶ or R⁷ is a linear or branched, saturated or unsaturated C₁-C₂₀-alkyl radical, optionally containing heteroatoms belonging to groups 13-17 of the Periodic Table of the Elements; preferably a C₁-C₁₀-alkyl radical;

R³ and R⁴, equal to or different from each other, are linear or branched, saturated or unsaturated C₁-C₂₀-alkyl radicals, optionally containing heteroatoms belonging to groups 13-17 of the Periodic Table of the Elements; preferably R³ and R⁴ equal to or different from each other are C₁-C₁₀-alkyl radicals; more preferably R³ is a methyl, or ethyl radical; and R⁴ is a methyl, ethyl or isopropyl radical.

[0027] Preferably the compounds of formula (I) have formula (Ia):



Wherein:

M, X, R¹, R², R⁵, R⁶, R⁸ and R⁹ have been described above;

R³ is a linear or branched, saturated or unsaturated C₁-C₂₀-alkyl radical, optionally containing heteroatoms belonging to groups 13-17 of the Periodic Table of the Elements; preferably R³ is a C₁-C₁₀-alkyl radical; more preferably R³ is a methyl, or ethyl radical.

[0028] Specific examples of metallocene compounds are dimethylsilanediyl{(1-(2,4,7-trimethylindenyl)-7-(2,5-dimethyl-cyclopenta[1,2-b:4,3-b']-dithiophene)}Zirconium dichloride and dimethylsilanediyl{(1-(2,4,7-trimethylindenyl)-7-(2,5-dimethyl-cyclopenta[1,2-b:4,3-b']-dithiophene)}Zirconium dimethyl.

[0029] Examples of alumoxanes are methylalumoxane (MAO), tetra-(isobutyl)alumoxane (TIBAO), tetra-(2,4,4-trimethyl-pentyl)alumoxane (TIOAO), tetra-(2,3-dimethylbutyl)alumoxane (TDMBAO) and tetra-(2,3,3-trimethylbutyl)alumoxane (TTMBAO).

[0030] Examples of compounds able to form an alkylmetallocene cation are compounds of formula D^+E^- , wherein D^+ is a Brønsted acid, able to donate a proton and to react irreversibly with a substituent X of the metallocene of formula (I) and E^- is a compatible anion, which is able to stabilize the active catalytic species originating from the reaction of the two compounds, and which is sufficiently labile to be able to be removed by an olefinic monomer. Preferably, the anion E^- comprises of one or more boron atoms.

[0031] Examples organo aluminum compound are trimethylaluminum (TMA), triisobutylaluminium (TIBAL), tris(2,4,4-trimethyl-pentyl)aluminum (TIOA), tris(2,3-dimethylbutyl)aluminium (TDMBA) and tris(2,3,3-trimethylbutyl)aluminum (TTMBA).

[0032] Examples of the said catalyst system and of polymerization processes employing such catalyst system can be found in WO2004099269 and WO2009000637.

[0033] In general, the polymerization process for the preparation of the butene-1 copolymer component A) can be carried out according to known techniques, for example slurry polymerization using as diluent a liquid inert hydrocarbon, or solution polymerization using for example the liquid butene-1 as a reaction medium. Moreover, it may also be possible to carry out the polymerization process in the gas-phase, operating in one or more fluidized bed or mechanically agitated reactors. The polymerization carried out in the liquid butene-1 as a reaction medium is preferred.

[0034] As a general rule, the polymerization temperature is generally of from -100°C to 200°C, preferably from 20°C to 120°C, more preferably from 40°C to 90°C, most preferably from 50°C to 80°C.

[0035] The polymerization pressure is generally comprised between 0.5 and 100 bar. The polymerization can be carried out in one or more reactors that can work under same or different reaction conditions such as concentration of molecular weight regulator, comonomer concentration, temperature, pressure etc.

[0036] The propylene homopolymer or copolymer component B) is a semicrystalline polymer, as demonstrated by the said melting point values, and has a stereoregularity of isotactic type.

[0037] It has preferably a solubility in xylene at room temperature (about 25°C) equal to or lower than 25% by weight, the lower limit being of 0.5% by weight.

[0038] Moreover it has preferably MIL values of from 0.5 to 9 g/10 min, more preferably from 1 to 8 g/10 min., where MIL is the melt flow index at 230°C with a load of 2.16 kg, determined according to ISO 1133.

[0039] Preferred examples of copolymers B) are the copolymers of propylene with one or more comonomers selected from ethylene, C₄-C₁₀ alpha-olefins and their combinations.

[0040] From the above definition it is evident that the term "copolymer" includes polymers containing more than one kind of comonomers.

[0041] Preferred amounts of comonomers in B) are of from 1 to 15% by weight, in particular from 2 to 10% by weight, referred to the weight of the concerned copolymer.

[0042] The said C₄-C₁₀ alpha-olefins are selected from olefins having formula CH₂=CHR wherein R is an alkyl radical, linear or branched, or an aryl radical, having from 2 to 8 carbon atoms.

[0043] Specific examples of C₄-C₁₀ alpha-olefins are butene-1, pentene-1, 4-methylpentene-1, hexene-1 and octene-1.

[0044] The preferred comonomers in the propylene copolymer B) are ethylene, butene-1 and hexene-1.

[0045] The propylene homopolymer or copolymer component B) can be prepared by using a Ziegler-Natta catalyst or a metallocene-based catalyst system in the polymerization process.

[0046] Typically a Ziegler-Natta catalyst comprises the product of the reaction of an organometallic compound of group 1, 2 or 13 of the Periodic Table of elements with a transition metal compound of groups 4 to 10 of the Periodic Table of Elements (new notation). In particular, the transition metal compound can be selected among compounds of Ti, V, Zr, Cr and Hf and is preferably supported on MgCl₂.

[0047] Particularly preferred catalysts comprise the product of the reaction of said organometallic compound of group 1, 2 or 13 of the Periodic Table of elements, with a solid catalyst component comprising a Ti compound and an electron donor compound supported on MgCl₂.

[0048] Preferred organometallic compounds are the aluminum alkyl compounds.

[0049] Thus preferred Ziegler-Natta catalysts are those comprising the product of reaction of:

- 1) a solid catalyst component comprising a Ti compound and an electron donor (internal electron-donor) supported on MgCl₂;
- 2) an aluminum alkyl compound (cocatalyst); and, optionally,

3) an electron-donor compound (external electron-donor).

[0050] The solid catalyst component (1) contains as electron-donor a compound generally selected among the ethers, ketones, lactones, compounds containing N, P and/or S atoms, and mono- and dicarboxylic acid esters.

[0051] Catalysts having the above mentioned characteristics are well known in the patent literature; particularly advantageous are the catalysts described in US patent 4,399,054 and European patent 45977.

[0052] Particularly suited among the said electron-donor compounds are phthalic acid esters, preferably diisobutyl phthalate, and succinic acid esters.

[0053] Other electron-donors particularly suited are the 1,3-diethers, as illustrated in published European patent applications EP-A-361 493 and 728769.

[0054] As cocatalysts (2), one preferably uses the trialkyl aluminum compounds, such as Al-triethyl, Al-triisobutyl and Al-tri-n-butyl.

[0055] The electron-donor compounds (3) that can be used as external electron-donors (added to the Al-alkyl compound) comprise the aromatic acid esters (such as alkylic benzoates), heterocyclic compounds (such as the 2,2,6,6-tetramethylpiperidine and the 2,6-diisopropylpiperidine), and in particular silicon compounds containing at least one Si-OR bond (where R is a hydrocarbon radical).

[0056] Examples of the said silicon compounds are those of formula $R^1_aR^2_bSi(OR^3)_c$, where a and b are integer numbers from 0 to 2, c is an integer from 1 to 3 and the sum (a+b+c) is 4; R^1 , R^2 and R^3 are alkyl, cycloalkyl or aryl radicals with 1-18 carbon atoms optionally containing heteroatoms.

[0057] Useful examples of silicon compounds are $(tert\text{-butyl})_2Si(OCH_3)_2$, $(cyclohexyl)(methyl)Si(OCH_3)_2$, $(phenyl)_2Si(OCH_3)_2$ and $(cyclopentyl)_2Si(OCH_3)_2$.

[0058] The previously said 1,3-diethers are also suitable to be used as external donors. In the case that the internal donor is one of the said 1,3-diethers, the external donor can be omitted.

[0059] The catalysts may be precontacted with small quantities of olefin (prepolymerization), maintaining the catalyst in suspension in a hydrocarbon solvent, and polymerizing at temperatures from room to 60 °C, thus producing a quantity of polymer from 0.5 to 3 times the weight of the catalyst.

[0060] The operation can also take place in liquid monomer, producing, in this case, a quantity of polymer up to 1000 times the weight of the catalyst.

[0061] The polymerization process, which can be continuous or batch, is carried out in the presence of said catalysts following known techniques and operating in liquid phase, in the presence or not of inert diluent, or in gas phase, or by mixed liquid-gas techniques.

[0062] Polymerization reaction time, pressure and temperature are not critical, however it is best if the temperature is from 20 to 100°C. The pressure can be atmospheric or higher.

[0063] The regulation of the molecular weight, resulting into the said MFR values, is carried out by using known regulators, hydrogen in particular.

[0064] Preferred examples of metallocene-based catalyst systems are disclosed in US20060020096 and WO98040419.

[0065] The polymerization conditions for preparing the homopolymer or copolymer component B) with metallocene-based catalyst systems in general do not need to be different from those used with Ziegler-Natta catalysts.

[0066] The polyolefin composition of the present invention can also contain additives commonly employed in the art, such as antioxidants, light stabilizers, heat stabilizers, colorants and fillers.

[0067] It can also contain additional polyolefins, in particular crystalline ethylene homopolymers and copolymers of ethylene with propylene and/or a C₄-C₁₀ α-olefin, such as HDPE, LLDPE or LDPE.

[0068] Other additional polyolefins than can be present are elastomeric copolymers, like copolymers of ethylene with propylene and/or higher alpha-olefins, like in particular butene-1, hexene-1 or octene-1. Such copolymer materials are commonly known as EPR or EPDM copolymers.

[0069] Preferred amounts of said additional polyolefins are from 1 to 10% by weight, more preferably from 3 to 7% by weight with respect to the total weight of the polyolefin composition.

[0070] The polyolefin composition may be manufactured by mixing the components together, extruding the mixture, and pelletizing the resulting composition using known techniques and apparatus.

[0071] The present invention also provides final articles made of or comprising the said polyolefin composition.

[0072] Such articles are soft and flexible.

[0073] In general, the polyolefin compositions of the present invention are suited for preparing final articles by extrusion, with equipment and processes well known in the art.

[0074] Specific examples of such final articles are liners, in particular liners for closures, preferably plastic and/or metal closures.

[0075] The liners for closures are widely used in twist closures for containers like jars and bottles, generally made of glass or plastic materials.

[0076] The twist closures are commonly in form of caps of circular shape, generally made of metal or plastics, hosting the liner on the inner surface facing the opening in the threaded circular neck of the container.

[0077] The liner is used to achieve a tight seal on the rim of the opening.

[0078] By twisting (rotating) the closure it is thus possible to close and open the container.

[0079] However, the Press-on/Twist-off® caps are first pressed on the container to close it by elastic deformation of the liner against the threading elements of the neck of the container, and then twisted to open.

[0080] So, depending upon the type of closure, the liner can have various kinds of shapes and functions.

[0081] In particular, while it generally has a sealing function, in the Press-on/Twist-off® caps it has both a sealing and a holding function.

[0082] In other kinds of closures, the holding function is performed by threads or lugs in the body of the closure itself.

[0083] Moreover, depending upon the use of the container, the closures are subject to different and often demanding requirements, in the fulfillment of which a critical role is played by the gasket.

[0084] The said closures are particularly for use in food and pharmaceutical packaging.

[0085] By using the polyolefin composition of the invention, air and liquid tight liners are obtained.

[0086] Such liners, due to their content of crystalline polymer are also able to withstand high temperature treatments (sterilization).

[0087] In general, liners are prepared from the polyolefin composition of the invention by a process comprising the following steps:

- a) laying down the polyolefin composition in the molten state on the inner surface of the closure;
- b) forming the laid polyolefin composition.

[0088] The above defined step a) is generally carried out by using extruders and metering devices well known in the art, so to achieve a controlled laying down of the polyolefin composition. For “controlled” it is meant that amount and shape of the laid material are controlled.

[0089] Extrusion temperatures generally applied in step a) are from 160 to 220°C.

[0090] Before carrying out the step a), the inner surface of the closure can be coated with a protective film of a varnish or a lacquer.

[0091] The above defined step b) is carried out by compression molding the molten polyolefin composition against the inner surface of the closure.

[0092] The above described processes and equipment are well known in the art, as disclosed, for example, in US5451360.

[0093] The resulting liners can have different shapes, like in particular “o-ring” or flat film of variable thickness depending upon the final use of the closure.

[0094] From the previously described properties of the polyolefin composition of the invention, from which the gasket are obtained, it is immediately evident that such liners are soft and elastic, hence they provide a tight and durable seal even after long use.

[0095] Another important advantage is that the said properties are obtained without adding softening agents. Such softening agents are in general made of low molecular weight materials, like mineral oils, and are easily extractable by contact with free fat/oil components of foods, for instance.

[0096] Such liners, due to their content of crystalline polymer are also able to withstand high temperature treatments (sterilization) which generally occurs at temperatures of 110-125°C.

EXAMPLES

[0097] The practice and advantages of the various embodiments, compositions and methods as provided herein are disclosed below in the following examples. These Examples are illustrative only, and are not intended to limit the scope of the invention in any manner whatsoever.

[0098] The following analytical methods are used to characterize the polymer compositions.

[0099] Thermal properties (melting temperatures and enthalpies)

[0100] Determined by Differential Scanning Calorimetry (DSC) on a Perkin Elmer DSC-7 instrument.

The melting temperatures of the butene-1 copolymer A) were determined according to the following method:

- TmII (measured in second heating scan): a weighted sample (5-10 mg) obtained from the polymerization was sealed into aluminum pans and heated at 200°C with a scanning speed corresponding to 10°C/minute. The sample was kept at 200°C for 5 minutes to allow a complete melting of all the crystallites thus cancelling the thermal history of the sample. Successively, after cooling to -20°C with a scanning speed corresponding to 10°C/minute, the peak temperature was taken as crystallization temperature (T_c). After standing 5 minutes at -20°C, the sample was heated for the second time at 200°C with a scanning speed corresponding to 10°C/min. In this second heating run, the peak temperature, when present is taken as the melting

temperature of the polybutene-1 (PB) crystalline form II (TmII) and the area as global melting enthalpy (ΔH_{fII}).

The butene-1 copolymer component A) of the polyolefin composition of the invention does not have a TmII peak.

- In order to determine the TmI, the sample was melted, kept at 200°C for 5 minutes and then cooled down to 20°C with a cooling rate of 10°C/min.

The sample was then stored for 10 days at room temperature. After 10 days the sample was subjected to DSC, it was cooled to -20°C, and then it was heated at 200°C with a scanning speed corresponding to 10°C/min. In this heating run, the first peak temperature coming from the lower temperature side in the thermogram was taken as the melting temperature (TmI).

The melting temperatures of the propylene homopolymer or copolymer component B) and of the overall composition comprising the polymer components A) and B) were measured at the second heating scan under the same conditions as above reported for the determination of TmII of the butene-1 copolymer component A).

Both component B) and the overall composition of the examples show a single melting peak between 130 and 165°C, corresponding to the melting temperature T_m .

The area of such melting peak of the overall composition is taken as the melting enthalpy ΔH_{fus} of the polyolefin composition.

[0101] Flexural elastic modulus

[0102] According to norm ISO 178, measured 10 days after molding.

[0103] Shore A and D

[0104] According to norm ISO 868, measured 10 days after molding.

[0105] Tensile stress and elongation at break

[0106] According to norm ISO 527 on compression molded plaques, measured 10 days after molding.

[0107] Tension set

[0108] According to norm ISO 2285, measured 10 days after molding.

[0109] Compression set

[0110] According to norm ISO 815, measured 10 days after molding;

[0111] MIE

[0112] Determined according to norm ISO 1133 with a load of 2.16 kg at 190 °C.

[0113] MIL

[0114] Determined according to norm ISO 1133 with a load of 2.16 kg at 230 °C.

[0115] Intrinsic viscosity

[0116] Determined according to norm ASTM D 2857 in tetrahydronaphthalene at 135 °C.

[0117] Density

[0118] Determined according to norm ISO 1183 at 23°C.

[0119] Comonomer contents

[0120] Determined by IR spectroscopy or by NMR.

[0121] Particularly for the butene-1 copolymers the amount of comonomer is calculated from ^{13}C -NMR spectra of the copolymers. Measurements were performed on a polymer solution (8-12 wt%) in dideuterated 1,1,2,2-tetrachloro-ethane at 120 °C. The ^{13}C NMR spectra were acquired on a Bruker AV-600 spectrometer operating at 150.91 MHz in the Fourier transform mode at 120 °C using a 90° pulse, 15 seconds of delay between pulses and CPD (WALTZ16) to remove ^1H - ^{13}C coupling. About 1500 transients were stored in 32K data points using a spectral window of 60 ppm (0-60 ppm).

[0122] Copolymer Composition

[0123] Diad distribution is calculated from ^{13}C NMR spectra using the following relations:

- $\text{PP} = 100 \frac{I_1}{\Sigma}$
- $\text{PB} = 100 \frac{I_2}{\Sigma}$
- $\text{BB} = 100 \frac{(I_3 - I_{19})}{\Sigma}$
- $\text{PE} = 100 \frac{(I_5 + I_6)}{\Sigma}$
- $\text{BE} = 100 \frac{(I_9 + I_{10})}{\Sigma}$
- $\text{EE} = 100 \frac{(0.5(I_{15} + I_6 + I_{10}) + 0.25(I_{14}))}{\Sigma}$
- Where $\Sigma = I_1 + I_2 + I_3 - I_{19} + I_5 + I_6 + I_9 + I_{10} + 0.5(I_{15} + I_6 + I_{10}) + 0.25(I_{14})$
- The molar content is obtained from diads using the following relations:
- $\text{P (m\%)} = \text{PP} + 0.5(\text{PE} + \text{PB})$
- $\text{B (m\%)} = \text{BB} + 0.5(\text{BE} + \text{PB})$
- $\text{E (m\%)} = \text{EE} + 0.5(\text{PE} + \text{BE})$

$I_1, I_2, I_3, I_5, I_6, I_9, I_6, I_{10}, I_{14}, I_{15}, I_{19}$ are integrals of the peaks in the ^{13}C NMR spectrum (peak of EEE sequence at 29.9 ppm as reference). The assignments of these peaks are made according to J.C. Randal, *Macromol. Chem Phys.*, **C29**, 201 (1989), M. Kakugo, Y. Naito, K. Mizunuma and T. Miyatake, *Macromolecules*, **15**, 1150, (1982), and H.N. Cheng, *Journal of Polymer Science, Polymer Physics Edition*, **21**, 57 (1983). They are collected in Table A (nomenclature according to C.J. Carman, R.A. Harrington and C.E. Wilkes, *Macromolecules*, **10**, 536 (1977)).

Table A.

I	Chemical Shift (ppm)	Carbon	Sequence
1	47.34 - 45.60	S _{αα}	PP
2	44.07 - 42.15	S _{αα}	PB
3	40.10 - 39.12	S _{αα}	BB
4	39.59	T _{δδ}	EBE
5	38.66 - 37.66	S _{αγ}	PEP
6	37.66 - 37.32	S _{αδ}	PEE
7	37.24	T _{βδ}	BBE
8	35.22 - 34.85	T _{ββ}	<u>XBX</u>
9	34.85 - 34.49	S _{αγ}	<u>BBE</u>
10	34.49 - 34.00	S _{αδ}	<u>BEE</u>
11	33.17	T _{δδ}	<u>EPE</u>
12	30.91 - 30.82	T _{βδ}	<u>XPE</u>
13	30.78 - 30.62	S _{γγ}	XEX
14	30.52 - 30.14	S _{γδ}	EEE
15	29.87	S _{δδ}	EEE
16	28.76	T _{ββ}	XPX
17	28.28 - 27.54	2B ₂	XBX
18	27.54 - 26.81	S _{βδ} + 2B ₂	BE, PE, BBE
19	26.67	2B ₂	EBE
20	24.64 - 24.14	S _{ββ}	XEX
21	21.80 - 19.50	CH ₃	P
22	11.01 - 10.79	CH ₃	B

[0124] For the propylene copolymers the comonomer content is determined by infrared spectroscopy by collecting the IR spectrum of the sample vs. an air background with a Fourier Transform Infrared spectrometer (FTIR). The instrument data acquisition parameters are:

- purge time: 30 seconds minimum;
- collect time: 3 minutes minimum;
- apodization: Happ-Genzel;
- resolution: 2 cm⁻¹.

[0125] Sample Preparation

[0126] Using a hydraulic press, a thick sheet is obtained by pressing about g 1 of sample between two aluminum foils. If homogeneity is in question, a minimum of two pressing operations are recommended. A small portion is cut from this sheet to mold a film. Recommended film thickness ranges between 0.02-:0.05 cm (8 – 20 mils).

[0127] Pressing temperature is 180±10°C (356°F) and about 10 kg/cm² (142.2 PSI) pressure for about one minute. Then the pressure is released and the sample is removed from the press and cooled the to room temperature.

[0128] The spectrum of a pressed film of the polymer is recorded in absorbance vs. wavenumbers (cm^{-1}). The following measurements are used to calculate ethylene and butene-1 content:

- Area (At) of the combination absorption bands between 4482 and 3950 cm^{-1} which is used for spectrometric normalization of film thickness.
- If ethylene is present, Area (AC2) of the absorption band between 750-700 cm^{-1} after two proper consecutive spectroscopic subtractions of an isotactic non additivated polypropylene spectrum and then, if butene-1 is present, of a reference spectrum of a butene-1-propylene random copolymer in the range 800-690 cm^{-1} .
- If butene-1 is present, Height (DC4) of the absorption band at 769 cm^{-1} (maximum value), after two proper consecutive spectroscopic subtractions of an isotactic non additivated polypropylene spectrum and then, if ethylene is present, of a reference spectrum of an ethylene-propylene random copolymer in the range 800-690 cm^{-1} .

In order to calculate the ethylene and butene-1 content, calibration straight lines for ethylene and butene-1 obtained by using samples of known amount of ethylene and butene-1 are needed.

[0129] Mw/Mn determination by GPC

[0130] The determination of the means Mn and Mw, and Mw/Mn derived therefrom was carried out using a Waters GPCV 2000 apparatus, which was equipped with a column set of four PLgel Olexis mixed-gel (Polymer Laboratories) and an IR4 infrared detector (PolymerChar). The dimensions of the columns were 300×7.5 mm and their particle size $13 \mu\text{m}$. The mobile phase used was 1,2,4-trichlorobenzene (TCB) and its flow rate was kept at 1.0 ml/min. All the measurements were carried out at 150°C . Solution concentrations were 0.1 g/dl in TCB and 0.1 g/l of 2,6-diterbutyl-*p*-chresole were added to prevent degradation. For GPC calculation, a universal calibration curve was obtained using 10 polystyrene (PS) standard samples supplied by Polymer Laboratories (peak molecular weights ranging from 580 to 8500000). A third order polynomial fit was used for interpolate the experimental data and obtain the relevant calibration curve. Data acquisition and processing was done using Empower (Waters). The Mark-Houwink relationship was used to determine the molecular weight distribution and the relevant average molecular weights: the *K* values were $K_{PS} = 1.21 \times 10^{-4}$ dL/g and $K_{PB} = 1.78 \times 10^{-4}$ dL/g for PS and PB respectively, while the Mark-Houwink exponents $\alpha = 0.706$ for PS and $\alpha = 0.725$ for PB were used.

[0131] For butene-1/ethylene copolymers, as far as the data evaluation is concerned, it is assumed that the composition is constant in the whole range of molecular weight and the *K* value of the Mark-Houwink relationship is calculated using a linear combination as reported below:

$$K_{EB} = x_E K_{PE} + x_B K_{PB}$$

[0132] where K_{EB} is the constant of the copolymer, K_{PE} (4.06×10^{-4} , dL/g) and K_{PB} (1.78×10^{-4} dL/g) are the constants of polyethylene and polybutene, x_E and x_B are the ethylene and the butene-1 weight% content. The Mark-Houwink exponents $\alpha = 0.725$ is used for all the butene-1/ethylene copolymers independently of their composition.

[0133] Fractions soluble and insoluble in xylene at 0°C (XS-0°C)

[0134] 2.5 g of the polymer sample are dissolved in 250 ml of xylene at 135°C under agitation. After 30 minutes the solution is allowed to cool to 100°C, still under agitation, and then placed in a water and ice bath to cool down to 0°C. Then, the solution is allowed to settle for 1 hour in the water and ice bath. The precipitate is filtered with filter paper. During the filtering, the flask is left in the water and ice bath so as to keep the flask inner temperature as near to 0°C as possible. Once the filtering is finished, the filtrate temperature is balanced at 25°C, dipping the volumetric flask in a water-flowing bath for about 30 minutes and then, divided in two 50 ml aliquots. The solution aliquots are evaporated in nitrogen flow, and the residue dried under vacuum at 80°C until constant weight is reached. The weight difference in between the two residues must be lower than 3%; otherwise the test has to be repeated. Thus, one calculates the percent by weight of polymer soluble (Xylene Solubles at 0°C = XS 0°C) from the average weight of the residues. The insoluble fraction in o-xylene at 0°C (xylene Insolubles at 0°C = XI%0°C) is:

$$XI\%0^\circ C = 100 - XS\%0^\circ C.$$

[0135] Fractions soluble and insoluble in xylene at 25 °C (XS-25°C)

[0136] 2.5 g of polymer are dissolved in 250 ml of xylene at 135°C under agitation. After 20 minutes the solution is allowed to cool to 25°C, still under agitation, and then allowed to settle for 30 minutes. The precipitate is filtered with filter paper, the solution evaporated in nitrogen flow, and the residue dried under vacuum at 80°C until constant weight is reached. Thus, one calculates the percent by weight of polymer soluble (Xylene Solubles - XS) and insoluble at room temperature (25°C).

[0137] The percent by weight of polymer insoluble in xylene at room temperature (25°C) is considered the isotactic index of the polymer. This value corresponds substantially to the isotactic index determined by extraction with boiling n-heptane, which by definition constitutes the isotactic index of polypropylene polymers.

[0138] Determination of isotactic pentads content

[0139] 50 mg of each sample were dissolved in 0.5 ml of $C_2D_2Cl_4$.

[0140] The ^{13}C NMR spectra were acquired on a Bruker DPX-400 (100.61 MHz, 90° pulse, 12s delay between pulses). About 3000 transients were stored for each spectrum; the mmmm pentad peak (27.73 ppm) was used as reference.

[0141] The microstructure analysis was carried out as described in literature (Macromolecules 1991, 24, 2334-2340, by Asakura T. et Al. . and Polymer, 1994, 35, 339, by Chujo R. et Al.).

[0142] The percentage value of pentad tacticity (mmmm%) for butene-1 copolymers is the percentage of stereoregular pentads (isotactic pentad) as calculated from the relevant pentad signals (peak areas) in the NMR region of branched methylene carbons (around 27.73 ppm assigned to the BBBBB isotactic sequence), with due consideration of the superposition between stereoirregular pentads and of those signals, falling in the same region, due to the comonomer.

[0143] Determination of X-ray crystallinity

[0144] The X-ray crystallinity was measured with an X-ray Diffraction Powder Diffractometer using the Cu-K α 1 radiation with fixed slits and collecting spectra between diffraction angle $2\Theta = 5^\circ$ and $2\Theta = 35^\circ$ with step of 0.1° every 6 seconds.

[0145] Measurements were performed on compression molded specimens in the form of disks of about 1.5-2.5 mm of thickness and 2.5-4.0 cm of diameter. These specimens are obtained in a compression molding press at a temperature of $200^\circ\text{C} \pm 5^\circ\text{C}$ without any appreciable applied pressure for 10 minutes, then applying a pressure of about 10Kg/cm^2 for about few second and repeating this last operation for 3 times.

[0146] The diffraction pattern was used to derive all the components necessary for the degree of crystallinity by defining a suitable linear baseline for the whole spectrum and calculating the total area (Ta), expressed in counts/sec $\cdot 2\Theta$, between the spectrum profile and the baseline. Then a suitable amorphous profile was defined, along the whole spectrum, that separate, according to the two phase model, the amorphous regions from the crystalline ones. Thus it is possible to calculate the amorphous area (Aa), expressed in counts/sec $\cdot 2\Theta$, as the area between the amorphous profile and the baseline; and the crystalline area (Ca), expressed in counts/sec $\cdot 2\Theta$, as $\text{Ca} = \text{Ta} - \text{Aa}$

The degree of crystallinity of the sample was then calculated according to the formula:

$$\% \text{Cr} = 100 \times \text{Ca} / \text{Ta}$$

[0147] Examples 1-4 and Comparative Example 1

[0148] Materials used in the examples

PB-1: butene-1/ethylene copolymer containing 16% by moles of copolymerized ethylene, produced according to the process described in WO2009000637 and in-line blended with a propylene copolymer composition (I) added in amount of 7%

by weight with respect to the total weight of the butene-1/ethylene copolymer and the propylene copolymer composition (I).

Such propylene copolymer composition (I) has MFRL of 5.5 g/10 min., total copolymerized ethylene content of 3% by weight, total copolymerized butene-1 content of 6% by weight; XS-25°C of 19% by weight and T_m of 133°C, and is made of the following two components:

I') 35% by weight of a copolymer of propylene with ethylene (3.2% by weight in the copolymer), and

I'') 65% by weight of a copolymer of propylene with ethylene (3.2% by weight in the copolymer) and butene-1 (6% by weight in the copolymer);

wherein the amounts of I') and I'') are referred to the total weight of I') + I'');

PP-1: propylene homopolymer having T_m of 158°C, MFRL of about 7 g/10 min., XS-25°C of 3% by weight;

PP-2: copolymer of propylene with ethylene, containing 6% by weight of ethylene, having T_m of 133°C, MFRL of about 7 g/10 min., XS-25°C of 20% by weight;

Stabilizers: blend of 0.05% by weight of pentaerythritol tetrakis(3-(3,5-di-tert-butyl-4-hydroxyphenyl)propionate (Irganox® 1010, sold by BASF) and 0.05% by weight of tris (2,4-di-tert-butylphenyl) phosphite (Irgafos® 168, sold by BASF), said percent amounts being referred to the total weight of the polyolefin composition;

Lubricants: blend of 1% by weight of erucamide (Crodamide® ER, sold by Croda), 1% by weight of Oleamide (Crodamide® OR, sold by Croda) and 1% by weight of Glyceryl Stearate (Atmer® 129, sold by Croda), said percent amounts being referred to the total weight of the polyolefin composition;

Pigment: Titanium dioxide Ti-Pure® R-104, sold by DuPont.

[0149] Due to the reduced content of propylene polymers, no melting peak is detected in the DSC analysis (second scan) of the above described PB-1.

[0150] The said materials are melt-blended in a co-rotating twin screw extruder Coperion ZSK40SC, with screw diameter of 40 mm and screw length/diameter ratio of 43:1, under the following conditions:

- extrusion temperature of 180-200°C;
- screw rotation speed of 220 rpm;
- production rate of 60 kg/hour.

[0151] The properties of the so obtained final compositions are reported in Table 1.

[0152] In Table 1 are also reported the properties of the above described PB-1 for comparison purpose (Comparison Example 1).

Table I

Example		1	2	3	4	Comp. 1
PB-1	Weight %	72.1	72.1	69.7	67.61	100
PP-1	Weight %	24.0	-	-	-	-
PP-2	Weight %	-	24.0	26.9	29.0	-
Stabilizers	Weight %	0.1	0.1	0.1	0.1	-
Lubricants	Weight %	3	3	3	3	-
Pigment	Weight %	0.8	0.8	0.3	0.3	-
Amount of A)*	Weight %	69.8	69.8	67.1	65.1	93
Amount of B)*	Weight %	30.2	30.2	32.9	34.9	7
Composition Properties						
ΔH_{fus}	J/g	25	10.8	12.4	11.5	0
T _m	°C	159.1	132.3	132.8	133.6	-
Shore A		87.2	78.8	81	83.5	60
Shore D		22.8	20	19	21.8	< 20
Flexural Modulus	MPa		40	40	40	< 10
MIE	gr/10'	2.14	2.07	1.75	3.65	1.4
Stress at Break	MPa	16.4	14.9	16.6	15.2	11
Elongation at Break	%	1000	1010	990	1100	790
Compression Set 22 hours 23°C after 10 min. in Autoclave	%	55	47	49	44	32
Compression Set 22 hours 70°C after 10 min. in Autoclave	%	79	84	72	71	100
Compression Set 22 hours 100°C after 10 min. in Autoclave	%	88	88			100

Note: * weight % with respect to the total weight of A) + B).

CLAIMS

What is claimed is:

1. A polyolefin composition having a melting enthalpy ΔH_{fus} , measured by DSC at the second heating scan, of from 7 to 30 J/g, preferably from 8 to 28 J/g, comprising:
 - A) from 63 to 78% by weight, preferably from 64 to 76% by weight, of a copolymer of butene-1 with ethylene having a copolymerized ethylene content of up to 18% by mole and no melting peak detectable at the DSC at the second heating scan;
 - B) from 22 to 37% by weight, preferably from 24 to 36% by weight, of (i) a propylene homopolymer, or (ii) a propylene copolymer, or (iii) a mixture of two or more of (i) and (ii), having a melting temperature T_m , measured by DSC at the second heating scan, of from 130°C to 165°C, preferably from 131 to 165°C, more preferably from 131 to 160°C;wherein the amounts of A) and B) are referred to the total weight of A) + B) and the DSC second heating scan is carried out with a heating rate of 10°C per minute.
2. The polyolefin composition of claim 1, having MIE of from 0.5 to 3 g/10 min., where MIE is the melt flow index at 190°C with a load of 2.16 kg, determined according to ISO 1133.
3. The polyolefin composition of claims 1 and 2, having flexural modulus equal to or lower than 150 MPa.
4. The polyolefin composition of claims 1 and 2, wherein the butene-1 copolymer component A) has Shore A equal to or lower than 80.
5. The polyolefin composition of claims 1 and 2, comprising from 63 to 69% by weight of A) and from 31 to 37% by weight of B).
6. The polyolefin composition of claim 1, wherein the butene-1 copolymer component A) has at least one of the following additional features:
 - MIE of from 0.5 to 3 g/10 min.;
 - a lower limit of the copolymerized ethylene content of 12% by mole;

- a Shore A value equal to or lower than 80, in particular from 80 to 40;
- a Shore D value equal to or lower than 20, in particular from 20 to 5;
- a Mw/Mn value, where Mw is the weight average molar mass and Mn is the number average molar mass, both measured by GPC, equal to or lower than 3, in particular from 3 to 1.5.
- a tension set of less than 30% at 100% of deformation at 23°C (ISO 2285), more preferably equal to or less than 20%;
- a percentage of butene-1 units in form of isotactic pentads (mmmm%) greater than 80%, preferably equal to or greater than 85%, even more preferably equal to or greater than 90%;
- tensile stress at break, measured according to ISO 527, of from 3 MPa to 20 MPa, more preferably from 4 MPa to 13 MPa;
- tensile elongation at break, measured according to ISO 527, of from 550% to 1000%; more preferably from 700% to 1000%;
- intrinsic viscosity (I.V.) equal to or higher than 1dl/g; more preferably equal to or higher than 1.5 dl/g, wherein the upper limit is of 3 dl/g;
- crystallinity of less than 30% measured via X-ray, more preferably of less than 20%;
- density of 0.895 g/cm³ or less, more preferably of 0.875 g/cm³ or less;
- content of xylene insoluble fraction at 0°C of less than 15% by weight.

7. The polyolefin composition of claim 1, wherein propylene homopolymer or copolymer component B) has MIL values of from 0.5 to 9 g/10 min, preferably from 1 to 8 g/10 min., where MIL is the melt flow rate at 230°C with a load of 2.6 kg, determined according to ISO 1133.
8. A formed articles comprising the polyolefin composition of claims 1 to 7.
9. Formed article according to claim 8, in form of liner.
10. A twist closure comprising the liner of claim 9.

11. The twist closure of claim 10, for use in food containers.
12. The twist closure of claims 10 or 11, in form of Press-on/Twist-off cap.
13. A process for preparing the liner of claim 9, comprising the following steps:
 - a) laying down the polyolefin composition in the molten state on the inner surface of the closure;
 - b) forming the laid polyolefin composition.

INTERNATIONAL SEARCH REPORT

International application No
PCT/EP2015/076770

A. CLASSIFICATION OF SUBJECT MATTER
INV. C08L23/20 B65D41/00
ADD.

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)
C08L C08J B29C B32B B65D

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

EPO-Internal, WPI Data

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	EP 2 631 271 A1 (BASELL POLIOLEFINE SRL [IT]) 28 August 2013 (2013-08-28) claim 1 paragraphs [0013] - [0015], [0018] - [0025], [0032], [0056], [0057] -----	1-8
X	WO 2012/152803 A1 (BASELL POLIOLEFINE SRL [IT]; GOBERTI PAOLO [IT]; PELLEGATTI GIAMPAOLO) 15 November 2012 (2012-11-15) page 11, line 22 - page 13, line 4 page 13, line 27 - line 31 page 14, line 5 - line 8 ----- -/-	1-9

Further documents are listed in the continuation of Box C.

See patent family annex.

* Special categories of cited documents :

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"P" document published prior to the international filing date but later than the priority date claimed

"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention

"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone

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Date of the actual completion of the international search	Date of mailing of the international search report
18 January 2016	25/01/2016
Name and mailing address of the ISA/ European Patent Office, P.B. 5818 Patentlaan 2 NL - 2280 HV Rijswijk Tel. (+31-70) 340-2040, Fax: (+31-70) 340-3016	Authorized officer Lippert, Sigrid

INTERNATIONAL SEARCH REPORT

International application No
PCT/EP2015/076770

C(Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	WO 2010/069775 A1 (BASELL POLIOLEFINE SRL [IT]; DI PIETRO FABIO [IT]; DE PALO ROBERTO [IT] 24 June 2010 (2010-06-24) claims 1,2 page 2, line 10 - line 24 page 3, line 24 - page 4, line 13 -----	1-8
A	EP 2 551 109 A1 (BASELL POLIOLEFINE SRL [IT]) 30 January 2013 (2013-01-30) the whole document -----	1-13
A	US 2004/211744 A1 (MARZOKAH RADWAN [US]) 28 October 2004 (2004-10-28) paragraphs [0002], [0011] - [0013], [0021] claims 1,10 -----	10-13

INTERNATIONAL SEARCH REPORT

Information on patent family members

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