The invention provides a process for producing a modified conjugated diene based polymer which attains favorable interaction between a rubber component and carbon black and/or silica, thereby improving dispersibility of the fillers, and which exhibits excellent properties such as heat-buildup-suppressing performance, fracture characteristics, and wear resistance; a modified conjugated diene based polymer produced through the process; a rubber composition containing the diene polymer; and a tire produced from the rubber composition and exhibiting the above properties. The process for producing a modified conjugated diene based polymer includes a step (a) of reacting a silicon compound with a conjugated diene based polymer having an active end so that the reaction takes place at the active end, the silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydroxycarbonyloxy group and a reactive group including a hydroxycarbonyloxy group are bonded, to thereby modify the active end, and a step (b) of performing condensation reaction which involves the compound having a bi-functional silicon atom, in the presence of a titanium compound serving as a titanium-based condensation-accelerating agent.
PROCESS FOR PRODUCING MODIFIED CONJUGATED DIENE BASED POLYMER, MODIFIED CONJUGATED DIENE BASED POLYMER PRODUCED BY THE PROCESS, RUBBER COMPOSITION, AND TIRE

TECHNICAL FIELD

[0001] The present invention relates to a process for producing a modified conjugated diene based polymer, to a modified conjugated diene based polymer produced through the process, to a rubber composition, and to a tire. More particularly, the invention relates to a process for producing a modified conjugated diene based polymer which attains favorable interaction between a rubber component and carbon black and/or silica, thereby improving dispersibility of the fillers, which exhibits excellent properties such as heat-buildup-suppressing performance, fracture characteristics, and wear resistance, and which is remarkably consistent in quality; to a modified conjugated diene based polymer produced through the process; to a rubber composition containing the diene polymer; and to a tire produced from the rubber composition and exhibiting the above properties.

BACKGROUND ART

[0002] In recent years, social demand with respect to energy conservation and concerns about environmental problems have increased, and emission of carbon dioxide is now controlled more and more rigorously throughout the world. In the midst of such a trend, demand has increased for reducing fuel consumption of automobiles. In order to satisfy such demands, performance of tires must be enhanced; particularly, rolling resistance must be reduced. Previously, optimization of the structure of tires was investigated for reducing rolling resistance. However, at present, rolling resistance is generally reduced through employment of a low-heat-buildup rubber composition.

[0003] In order to produce such a low-heat-buildup rubber composition, a variety of modified rubbers containing silica or carbon black serving as a filler have been developed. Among the techniques for producing such modified rubbers, Japanese Patent Publication (kokoku) Nos. 6-53763 and 6-57767 and other documents disclose particularly effective approaches in which the polymerization active end of a conjugated diene based polymer produced through anionic polymerization in the presence of an organic lithium is modified with an alkoxysilane derivative having a functional group interacting with a filler.

[0004] Although, these approaches are generally effective for polymers having a stable polymerization living end, modification of a rubber composition containing silica or carbon black has not been satisfactorily attained. In addition, when a conventional modification technique is employed, in many cases, branching of the backbone of the polymer is insufficient. Therefore, when such a modified rubber is used in practice, problematic cold flow occurs. In this case, partial coupling is performed so as to prevent cold flow and, as a result, the effect of modification is reduced.

[0005] Thus, an approach for overcoming the above drawbacks and enhancing the effect of modification has been proposed. WO 03/087171 discloses a method in which a condensation-accelerating agent is added to a reaction system during modification of the active end of a conjugated diene based polymer with an alkoxysilane. Although the method successfully prevents loss of a silica filler in the rubber composition, loss of a carbon black filler cannot be satisfactorily prevented.

DISCLOSURE OF THE INVENTION

[0006] Under such circumstances, an object of the present invention is to provide a process for producing a modified conjugated diene based polymer which attains favorable interaction between a rubber component and carbon black and/or silica, thereby improving dispersibility of the fillers, and which exhibits excellent properties such as heat-buildup-suppressing performance, fracture characteristics, and wear resistance. Another object of the invention is to provide a modified conjugated diene based polymer produced through the process. Still another object of the invention is to provide a rubber composition containing the diene polymer. Yet another object is to provide a tire produced from the rubber composition and exhibiting the above properties.

[0007] The present inventors have carried out extensive studies in order to attain the above objects, and have found that the objects can be attained by two-part process, including the step of modification reaction in which the active end of a conjugated diene based polymer is modified with a compound containing a bi-functional silicon atom, which compound has at least an amino group of a specific structure in the molecule thereof and a hydroxyloxycarbonyl group bonded to the silicon atom; and the step of condensation reaction performed in the presence of a specific titanium compound serving as a condensation-accelerating agent. The present invention has been accomplished on the basis of this finding.

[0008] Accordingly, in a first aspect of the present invention, there is provided

[0009] a process for producing a modified conjugated diene based polymer, the process comprising

[0010] a step (a) of reacting a silicon compound with a conjugated diene based polymer having an active end so that the reaction takes place at the active end, the silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydroxyloxycarbonyl group and a reactive group including a hydroxyloxycarbonyl group are bonded, to thereby modify the active end, and

[0011] a step (b) of performing condensation reaction which involves the compound having a bi-functional silicon atom, in the presence of a titanium compound serving as a titanium-based condensation-accelerating agent;

[0012] The process may further include a deprotection step (c) of hydrolyzing a group which is bonded to the active end of the conjugated diene based polymer and which has been derived from the compound having a bi-functional silicon atom, whereby the protected primary amino group contained in the group bonded to the active end is converted to a free amino group.

[0013] The compound having a bi-functional silicon atom employed in the step (a) is a silicon compound represented by formula (I):

\[
R'O\left[\begin{array}{cc}
R^1 & R^2 \\
SiR^3R^4 & SiR^5R^6
\end{array}\right]_A SiR^7R^8
\]

(wherein each of R^1 and R^2 represents a hydrocarbon group having 1 to 20 carbon atoms, each of R^3 to R^8 represents a
hydrocarbon group having 1 to 20 carbon atoms, \( R^8 \) represents a alkylene group having 1 to 12 carbon atoms. \( A \) represents a reactive group, and \( f \) is an integer of 1 to 10)

\[ R^8 \rightarrow \begin{array}{c} S \rightarrow N \rightarrow S \rightarrow R^9 \end{array} R^{11} \]  
(II)

(wherein each of \( R^7 \) to \( R^{11} \) represents a hydrocarbon group having 1 to 20 carbon atoms, and \( R^{12} \) represents an alkylene group having 1 to 12 carbon atoms; or

[0015] a silicon compound represented by formula (III):

\[ R^8 \rightarrow \begin{array}{c} S \rightarrow R^9 \rightarrow N \rightarrow S \rightarrow R^{13} \end{array} \]
(III)

(wherein each of \( R^1 \) and \( R^2 \) represents a hydrocarbon group having 1 to 20 carbon atoms, each of \( R^3 \) to \( R^9 \) represents a hydrocarbon group having 1 to 20 carbon atoms, \( R^8 \) represents an alkylene group having 1 to 12 carbon atoms, \( R^{12} \) represents an alkylene group having 1 to 12 carbon atoms, \( A \) represents a reactive group, and \( f \) is an integer of 1 to 10).

[0016] The group \( A \) in formula (I) is a halogen atom or an alkoxyl group having 1 to 20 carbon atoms.

[0017] The conjugated diene based polymer having an active end may be produced through anionic polymerization, in the presence of an organic alkali metal compound serving as a polymerization initiator, of a conjugated diene compound singly or of a conjugated diene compound and an aromatic vinyl compound in combination.

[0018] The conjugated diene compound is at least one species selected from the group of 1,3-butadiene, isoprene, and 2,3-dimethyl-1,3-butadiene.

[0019] The aromatic vinyl compound is styrene.

[0020] The conjugated diene based polymer has a polymer unit derived from an aromatic vinyl compound in an amount of 0 to 55 mass % based on the total polymer unit present in the conjugated diene based polymer and a vinyl bond content of 7 to 65 mass % based on the total conjugated diene portion.

[0021] The condensation-accelerating agent employed in the step (b) is at least one species selected from among a titanium alkoxide, a titanium carboxylate salt, a titanium acetylacetonate complex salt, and a self mixture thereof.

[0022] In a second aspect of the present invention, there is provided a modified conjugated diene based polymer produced by the above process.

[0023] In a third aspect of the present invention, there is provided a rubber composition comprising the modified conjugated diene based polymer.

[0024] The rubber composition may comprise a rubber component containing 20 mass % or more of the modified conjugated diene based polymer in an amount of 100 parts by mass and silica and/or carbon black in the total amount of 20 to 120 parts by mass.

[0025] The rubber component may comprise the modified conjugated diene based polymer in an amount of 15 to 100 mass % and at least one species selected from among a natural rubber, a synthetic isoprene rubber, a butadiene rubber, a styrene-butadiene rubber, an ethylene-o-olefin copolymer rubber, an ethylene-o-olefin-diene copolymer rubber, a chloroprene rubber, a halogenated butyl rubber, and a styrene-isobutylene copolymer having a halomethyl group, in an amount of 85 to 0 mass %.

[0026] In a fourth aspect of the present invention, there is provided a pneumatic tire employing the rubber composition.

[0027] The pneumatic tire may employ the rubber composition as a tread, a base tread, or a sidewall thereof.

[0028] According to the present invention, there can be provided a process for producing a modified conjugated diene based polymer which attains favorable interaction between a rubber component and carbon black and/or silica, thereby improving dispersibility of the fillers, which exhibits excellent properties such as heat-buildup-suppressing performance, fracture characteristics, and wear resistance, and which is remarkably consistent in quality; a modified conjugated diene based polymer produced through the process; a rubber composition containing the diene polymer; and a tire produced from the rubber composition and exhibiting the above properties.

**BEST MODE FOR CARRYING OUT THE INVENTION**

[0029] The process of the present invention for producing a modified conjugated diene base polymer includes

[0030] a step (a) of reacting a silicon compound with a conjugated diene based polymer having an active end so that the reaction takes places at the active end, the silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydrocarboxy group and a reactive group including a hydrocarboxy group (hereinafter the compound may be referred to as “an alkoxysilane compound”) are bonded, to thereby modify the active end, and

[0031] a step (b) of performing condensation reaction which involves the compound having a bi-functional silicon atom, in the presence of a titanium compound serving as a titanium-based condensation-accelerating agent.

[0032] The condensation-accelerating agent is generally added after modification reaction in which the alkoxysilane compound is reacted with the active end of the conjugated diene based polymer, and before condensation reaction. Alternatively, the condensation-accelerating agent may be added before addition of the alkoxysilane compound (i.e., before modification reaction), followed by adding the alkoxysilane compound for modification and performing condensation reaction.

[0033] No particular limitation is imposed on the method for producing a conjugated diene based polymer having an active end employed in the present invention, and a conjugated diene based polymer may be produced from a diene monomer or a diene monomer and another comonomer. Examples of the mode of polymerization include solution polymerization, gas-phase polymerization, and bulk polymerization. Of these, solution polymerization is particularly
preferred. The polymerization may be performed in a batch manner or a continuous manner.

**[0034]** The active site in the molecule of a conjugated diene based polymer is preferably at least one metal species selected from alkaline metals and alkaline earth metals. Of these, alkali metals are preferred, with lithium being particularly preferred.

**[0035]** In the solution polymerization, the polymer of interest may be produced through, for example, anionically polymerizing a conjugated diene compound alone or a conjugated diene compound with an aromatic vinyl compound in the presence of an organic alkali metal compound, particularly an organic lithium compound as a polymerization initiator. In the specification, the term "conjugated diene based polymer" refers not only to a polymer formed from a conjugated diene but also to a polymer formed from a conjugated diene and an aromatic vinyl compound.

**[0036]** In addition, in an effective manner, a halogen-containing monomer is employed, and a halogen atom contained in the formed polymer is activated by an organic metal compound. For example, a bromine site of a copolymer containing an isobutylene unit, a p-methylstyrene unit, or a p-bromomethylstyrene unit is lithiated to thereby provide an active site.

**[0037]** Examples of the aforementioned conjugated diene compound includes 1,3-butadiene, isoprene, 1,3-pentadiene, 2,3-dimethyl-1,3-butadiene, 2-phenyl-1,3-butadiene, and 1,3-hexadiene. These dienes may be used singly or in combination of two or more species. Among them, 1,3-butadiene, isoprene, and 2,3-dimethyl-1,3-butadiene are particularly preferred.

**[0038]** Examples of the aromatic vinyl compound used in copolymerization with these conjugated diene compounds include styrene, α-methylstyrene, 1-vinylcyclohexene, 3-vinyltoluene, ethylvinylbenzene, divinylbenzene, 4-cyclohexylbenzene, and 2,4,6-trimethylstyrene. These compounds may be used singly or in combination of two or more species. Among them, styrene is particularly preferred.

**[0039]** In the case where the conjugated diene compound and the aromatic vinyl compound are used as comonomers, use of 1,3-butadiene and styrene are particularly preferred, from the viewpoint of practical aspects including availability, and anionic polymerization characteristics including a living property.

**[0040]** When solution polymerization is employed, the monomer concentration of the solution is preferably 5 to 50 mass %, more preferably 10 to 30 mass %. When the conjugated diene compound and the aromatic vinyl compound are used as comonomers, the monomer mixture preferably has an aromatic vinyl compound content falling within a range of 0 to 55 mass %.

**[0041]** No particular limitation is imposed on the lithium compound serving as a polymerization initiator, and hydrocarbyllithium and a lithiumamido compound are preferably used. When hydrocarbyllithium is used, a conjugated diene based polymer which has a hydrocarbyl group at a polymerization-initiating end and a polymerization active site at the other end is produced, whereas when the lithiumamido compound is used, a conjugated diene based polymer which has a nitrogen-containing group at a polymerization-initiating end and a polymerization active site at the other end is produced.

**[0042]** The hydrocarbyllithium is preferably a compound having a C2 to C20 hydrocarbyl group. Specific examples include ethylolithium, n-propylolithium, isopropylolithium, n-butylithium, sec-butylithium, tert-octylithium, n-decylithium, phenyllithium, 2-naphthyllithium, 2-butyl-phenyllithium, 4-phenylbutyllithium, cyclohexyllithium, cyclo-
These randomizers may be used singly or in combination of two or more species. The randomizer is preferably employed in an amount 0.01 to 1000 mole equivalents per mole of the lithium compound.

The polymerization reaction is preferably carried out at 0 to 150°C, more preferably 20 to 130°C. The polymerization reaction may be carried out under generated pressure. In a general procedure, the pressure is preferably selected such that the monomer is maintained virtually as a liquid phase. That is, a higher pressure may be employed in accordance with needs, although depending on the individual substances to be polymerized, polymerization solvent, and polymerization temperature. Such pressure may be obtained through an appropriate method such as applying pressure to a reactor by use of gas inert to the polymerization reaction.

In the polymerization, all the raw materials involved in polymerization such as a polymerization initiator, a solvent, monomers, etc. are preferably employed after removing reaction-inhibiting substances such as water, oxygen, carbon dioxide, and protic compounds.

In order to produce an elastomeric polymer, the formed polymer or copolymer preferably has a glass transition temperature (Tg) of −95 to −15°C, as determined through differential thermal analysis. Through controlling of the glass transition temperature to fall within the above range, increase in viscosity is prevented, whereby a polymer which can be easily handled can be obtained.

In the present invention, the active end of the thus-produced conjugated diene based polymer is modified through reaction with a silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydrocarbyloxy group and a reactive group including a hydrocarbyloxy group are bonded.

Examples of the silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydrocarbyloxy group and a reactive group including a hydrocarbyloxy group are bonded (hereinafter the compound may be referred to as “a modifying agent”) include the following compounds:

a silicon compound represented by formula (I):

\[
\text{R}^1\text{O} - \text{Si} - \text{R}^2 - \text{N} - \text{Si} - \text{R}^3 - \text{Si} - \text{R}^5
\]

(wherein each of \( R^1 \) and \( R^2 \) represents a hydrocarbon group having 1 to 20 carbon atoms, each of \( R^3 \) to \( R^5 \) represents a hydrocarbon group having 1 to 20 carbon atoms, \( R^6 \) represents an alkylene group having 1 to 12 carbon atoms, and \( f \) is an integer of 1 to 10);

a silicon compound represented by formula (II):

\[
\text{R}^1\text{O} - \text{Si} - \text{R}^8 - \text{N} - \text{Si} - \text{R}^9\text{R}^{10}\text{R}^{11}
\]

(wherein each of \( R^7 \) to \( R^{11} \) represents a hydrocarbon group having 1 to 20 carbon atoms, and \( R^{12} \) represents an alkylene group having 1 to 12 carbon atoms); and

a silicon compound represented by formula (III):

\[
\begin{align*}
\text{R}^1\text{O} & - \text{Si} - \text{R}^2 - \text{N} - \text{Si} - \text{R}^3 - \text{Si} - \text{R}^4 - \text{N} - \text{Si} - \text{R}^5 - \text{Si} - \text{R}^6 - \text{Si} - \text{R}^7 - \text{Si} - \text{R}^8 - \text{Si} - \text{R}^9 - \text{Si} - \text{R}^{10} - \text{Si} - \text{R}^{11} \\
& \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A} \quad \text{A}
\end{align*}
\]

(wherein each of \( R^1 \) and \( R^2 \) represents a hydrocarbon group having 1 to 20 carbon atoms, each of \( R^3 \) to \( R^5 \) represents a hydrocarbon group having 1 to 20 carbon atoms, \( R^6 \) represents an alkylene group having 1 to 12 carbon atoms, \( R^7 \) represents an alkylene group having 1 to 12 carbon atoms, \( R^8 \) represents a reactive group, and \( f \) is an integer of 1 to 10).

In the above formulas (I) to (III), examples of preferred alkylene groups having 1 to 12 carbon atoms represented by \( R^8 \) or \( R^{12} \) include a methylene group; an ethylene group, and a propylene group. Examples of hydrocarbons having 1 to 20 carbon atoms include alkyl groups such as methyl, ethyl, and propyl; and aryl groups such as aralkyl groups including phenyl, tolyl, naphthyl, and benzyl.

In formula (I), any two groups of \( R^2 \), \( R^3 \), and \( R^4 \) may be linked to each other, to thereby form, together with the silicon atom bonded thereto, a 4- to 7-membered ring. Similarly, in formula (II), any two groups of \( R^4 \), \( R^{10} \), and \( R^{11} \) may be linked to each other, to thereby form, together with the silicon atom bonded thereto, a 4- to 7-membered ring. \( R^{13} \) represents an alkylene group having 1 to 12 carbon atoms.

Examples of the silicon compound having a protected primary amino group and a bi-functional silicon atom to which at least an alkylene group is bonded include \( N,N\text{-bis}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxyasilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxyasilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxyasilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxyasilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxyasilane}, and 1\text{-trimethylsilyl}-2\text{-ethoxyethyl}1\text{-aza}-2\text{-cyclopentane}.

Examples of such compounds in which \( A \) is a halogen atom include \( N,N\text{-bisa}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxychlorosilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxychlorosilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxychlorosilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxychlorosilane}, and \( N,N\text{-bis}((\text{trimethylsilyl})\text{aminoethylmethyldiethoxychlorosilane}.

Of these, \( N,N\text{-bis}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxychlorosilane}, N,N\text{-bis}((\text{trimethylsilyl})\text{aminopropyl)methyldiethoxychlorosilane}, and 1\text{-trimethylsilyl}-2\text{-ethoxyethyl}1\text{-aza}-2\text{-cyclopentane} are preferred.

These modifying agents may be used singly or in combination of two or more species, and may be a partial condensate.

As used herein, the term “partial condensate” means a condensation product in which a part (not entirety) of SiOR moieties are condensed to form Si—O—Si bonds.

The polymer subject to the modification reaction preferably contain at least 10% of living polymer chains.

Reaction between a living polymerization end, for example, \( P^{-1}t\text{i}^{-1} \) and a modifying agent represented by formula (I) (\( f = 1 \)) is represented by the following scheme:
In the present invention, a specific condensation-accelerating agent is employed in order to accelerate condensation reaction involving the aforesaid alkoxysilane compound serving as a modifying agent.

The condensation-accelerating agent employed in the invention may be added to the reaction system before the aforementioned modification reaction. However, preferably, the agent is added to the reaction system after modification reaction and before condensation reaction. When the agent is added before modification reaction, in some cases, the agent is directly reacted with the active end, thereby failing to introduce a hydrocarboxyloxy group to the active end.

When the agent is added after initiation of condensation reaction, in some cases, the condensation-accelerating agent is not uniformly dispersed in the reaction system, thereby deteriorating the catalyst performance.

The timing of addition of the condensation-accelerating agent is generally 5 minutes to 5 hours after initiation of modification reaction, preferably 15 minutes to 1 hour after initiation of modification reaction.

Specific examples of the condensation-accelerating agent include titanium ethylhexyldioeleate, titanium 2-ethylhexylhexoxide, titanium di-n-butoxide (bis-2,4-pentadionate), titanium isobutoxide, titanium stearyloxide, titanium oxide bis(tetramethylethapentanedionate), titanium oxide bis (pentanedionate), and titanium lactate. Of these, titanium ethylhexyl dioeleate, titanium 2-ethylhexoxide, and titanium di-n-butoxide (bis-2,4-pentanedionate) are preferred.

During modification reaction, the above modifying agent is preferably employed in an amount of 0.5 to 200 mmol/kg (conjugated diene based polymer), more preferably 1 to 100 mmol/kg (conjugated diene based polymer), particularly preferably 2 to 50 mmol/kg (conjugated diene based polymer). In the unit of the amount, the "conjugated diene based polymer" means the mass of polymer not containing additives such as an anti-aging agent added during or after the production of the diene polymer. Through controlling the amount of the modifying agent employed so as to fall within the above ranges, high dispersibility of fillers can be attained, and mechanical characteristics, wear resistance, and heat-buildup-suppressing performance after vulcanization can be enhanced.

No particular limitation is imposed on the method of adding the above modifying agent, and one batch addition, divided addition, continuous addition, etc. may be employed. Among them, one batch addition is preferred.

The modifying agent may act on any of a polymerization-initiating end, a polymerization-terminating end, a polymer backbone, and a polymer side chain. From the viewpoint of improvement of the heat-buildup-suppressing performance by preventing energy loss from a polymer end, the modifying agent is preferably introduced into the polymerization-initiating end or the polymerization-terminating end.
The amino group derived from a modifying agent for producing the modified conjugated diene based polymer of the present invention may be protected, or deprotected to be a primary amine. Both cases are preferred. In the case where a protected group is deprotected, the following procedure is performed.

Specifically, silyl protective groups on the protected amino group are hydrolyzed, to thereby form the corresponding free amino group. Through removal of the solvent from the thus-deprotected polymer, the corresponding dried polymer having a primary amino group is obtained. Needless to say, in any step from a step including the condensation to a step of removing solvent to produce a dried polymer, deprotection of the protected primary amino group derived from the modifying agent may be performed in accordance with needs.

In the present invention, after completion of the condensation, a deprotection step (c) may be performed. In the step (c), a group which is bonded to the active end of the conjugated diene based polymer and which has been derived from a compound having a bi-functional silicon atom is hydrolyzed, whereby the protected primary amino group in the end group is converted to a free amino group. Thus, a modified conjugated diene based polymer of interest can be produced.

The modified conjugated diene based polymer produced in the present invention preferably has a Mooney viscosity (ML(1+4), 100°C) of 10 to 150, more preferably 15 to 100. Though controlling the Money viscosity to fall within the above range, a rubber composition exhibiting excellent kneadability and mechanical strength after vulcanization can be produced.

The rubber composition of the present invention preferably contains, as a rubber component, the aforementioned modified conjugated diene based polymer in an amount of at least 20 mass %. The rubber component more preferably contains the modified conjugated diene based polymer in an amount of 30 mass % or more, particularly preferably 40 mass % or more. Through controlling the modified conjugated diene based polymer content of the rubber component to be 15 mass % or more, the rubber composition is endowed with a physical property of interest.

The modified conjugated diene based polymer species may be used singly or in combination of two or more species. Examples of the additional rubber component employed in combination with the modified conjugated diene based polymer include natural rubber, synthetic isoprene rubber, butadiene rubber, styrene-butadiene rubber, ethylene-octene copolymer rubbers, ethylene-octene-diene copolymer rubbers, acrylonitrile-butadiene copolymer rubber, chloroprene rubber, halogenated butyl rubbers, and mixtures thereof. These rubber species may be treated with a multifunctional modifying agent such as tin tetrachloride or silicon tetrachloride, to thereby have a branch structure.

The rubber composition of the present invention preferably contains, as a filler, silica and/or carbon black.

No particular limitation is imposed on the type of silica, and any of the silica species conventionally employed as rubber reinforcing fillers may be used.

Examples of the silica species include wet silica (hydrated silicic acid), dry silica (anhydrous silicic acid), calcium silicate, and aluminum silicate. Among them, wet silica is preferred, since the silica can remarkably improve both fracture characteristics and wet grip performance.

No particular limitation is imposed on the type of carbon black, and SRF, GPF, FEE, HAF, ISAF, SAF, etc. may be employed. The carbon black employed in the invention preferably has an iodine absorption (1A) of 60 mg/g or more and a dibutyl phthalate oil absorption (DBP) of 80 mL/100 g or more. By use of carbon black, grip performance and fracture characteristics can be greatly improved. From the viewpoint of wear resistance, HAF, ISAF, and SAF are particularly preferred.

The silica and/or carbon black may be used singly or in combination of two or more species.

The rubber composition preferably contains silica and/or carbon black in an amount of 20 to 120 parts by mass with respect to 100 parts by mass of the rubber component. From the viewpoint of reinforcing effects and improvement of physical properties, the amount is more preferably 25 to 100 parts by mass. Through controlling the amount of carbon black and/or silica to fall within the above range, a rubber composition exhibiting excellent operability in factories such as kneadability and fracture characteristics of interest can be produced.

The rubber composition of the present invention, when silica is employed as a reinforcing filler, a silane coupling agent may be incorporated into the composition in order to further enhance reinforcing performance and heat-buildup-suppressing performance.

Examples of the silane coupling agent includes bis(3-triethoxysilylpropyl)tetrasulfide, bis(3-triethoxysilylpropyl)trisulfide, bis(3-triethoxysilylpropyl)disulfide, bis(2-triethoxysilylhexyl)tetrasulfide, bis(3-triethoxysilylpropyl) tetrasulfide, bis(2-triethoxysilylhexyl)disulfide, 3-mercaptopropyltrimethoxysilane, 3-mercaptopropyltriethoxysilane, 2-mercaptopropyltrimethoxysilane, 2-mercaptopropyltriethoxysilane, 3-triethoxysilylpropyl-N,N,N-dimethylthiocarbamoyl tetrasulfide, 3-triethoxysilylpropyl-N,N,N-dimethylthiocarbamoyl tetrasulfide, 2-triethoxysilylhexyl-N,N-dimethylthiocarbamoyl tetrasulfide, 3-triethoxysilylpropylbenzothiazoyl tetrasulfide, 3-triethoxysilylpropylmethacrylate monosulfide, 3-triethoxysilylpropyl methacrylate monosulfide, bis(3-diolcondensedtrimethoxysilane, 3-mercaptopropylmethoxymethylsilane, dimethoxymethylsilylethynyl-N,N-dimethylthiocarbamoyl tetrathifide, and dimethoxymethylsilylethynylpropylbenzothiazoyl tetrasulfide. Among them, bis(3-triethoxysilylpropyl) polysulfide and 3-triethoxysilylpropylbenzothiazoyl tetrasulfide are preferred from the viewpoint of an effect for improving the reinforcing property.

These silane coupling agents may be used singly or in combination of two or more species.

The rubber composition of the present invention employs, as a rubber component, a modified polymer in which a functional group having a high affinity to silica is introduced into an active site of the molecule thereof. Therefore, the amount of the silane coupling agent can be reduced as compared to the general cases. The amount of the silane coupling agent, which varies depending on the type of the agent, is preferably 1 to 20 mass % based on the silica. When the amount is less than 1 mass %, the effect of the coupling agent cannot sufficiently be attained, whereas when the amount is in excess of 20 mass %, the rubber component may be gelated. From the viewpoint of fully attaining the effect of coupling agent and prevention of gelation, the amount of the silane coupling agent is preferably 5 to 15 mass %.
So long as the object of the present invention is not impeded, the rubber composition of the present invention may further contain, in accordance with needs, a variety of chemicals usually used in the rubber industry. Examples of the chemicals include vulcanizing agents, vulcanization-accelerating agents, process oils, anti-aging agents, antioxidants, scorch preventives, zinc oxide, and stearic acid.

The rubber composition of the present invention is produced through kneading by means of an open kneader such as a roller or a closed kneader such as a Banbury mixer. The kneaded rubber composition is molded and, subsequently, vulcanized, to thereby provide a wide range of rubber products. Examples of such rubber products include tire-related uses such as tire treads, under treads, carcass sidewalls, and bead portion; vibration-insulating rubber; headers; belts; hoses; and other industrial products. Particularly, the rubber composition of the invention, exhibiting well-balanced heat buildup-suppressing performance, wear resistance, and fracture strength, is suitably employed as fuel-saving tires, large-scale tires, and tires for high-performance tires.

EXAMPLES

Next, the present invention will be described more specifically with reference to examples in the following. However, the present invention is not limited to the examples.

In the Examples, physical properties of the samples were determined by the following procedures.

1. Vinyl Content of Conjugated Diolefin Portion (% Based on the Entirety of the Diolefin Portion)

2. Vinyl content was determined by 270 MHz 1H-NMR.

3. Bonded Styrene Content (Mass % in Polymer)

Bonded styrene content was determined by 270 MHz 1H-NMR.

4. Weight Average Molecular Weight

Weight average molecular weight was determined, by gel permeation chromatography (GPC) (by means of a chromatograph, HLC-8220GPC, product of Tosoh Corporation). The result was expresses as the value of corresponding polystyrene as the reference.

5. Mooney Viscosity (ML1+4, 100°C)

Mooney viscosity was determined in accordance with JIS K6300 (use of an L rotor, preheating for one minute, rotor operation for four minutes, and temperature of 100°C).


Physical properties of vulcanized rubber samples were determined by the following methods (i) and (ii).

(i) tanδ (50°C): By means of a dynamic spectrometer (product of Rheometrics Co., Ltd.), tanδ (50°C) was determined under a tensile strain of 1%, a frequency of 10 Hz, and 50°C, and represented by an index. The higher the index, the smaller the rolling resistance (the better the quality).

(ii) Wear resistance (Lamborn wear index): By means of a Lamborn type abrasion tester, the wear resistance at a percent slip of 25% was measured and represented by an index. The measurement was performed at room temperature. The higher the index, the better the wear resistance.

<Synthesis of Modifying Agent>

Synthesis Example 1

Synthesis of N,N-bis(trimethylsilyl)aminopropymethyldiethoxysilane

Under nitrogen, 3-aminopropymethyldiethoxysilane (product of Gelest) (36 g) for forming an aminosilane moiety was added to dichloromethane (solvent) (400 mL) placed in a glass flask equipped with an agitator. Subsequently, trimethylsilane chloride (product of Aldrich) (48 mL) and triethylamine (53 mL) for forming a protective moiety were added to the solution, followed by stirring the mixture at room temperature for 17 hours. The reaction mixture was evaporated by means of an evaporator, to thereby remove solvent from the mixture. The thus-obtained reaction mixture was distilled under reduced pressure (5 mmHg), to thereby yield 40 g of N,N-bis(trimethylsilyl)aminopropymethyldiethoxysilane as a 130-135°C fraction.

Synthesis Example 2

Synthesis of 1-trimethylsilyl-2-ethoxymethyl-1-aza-2-sila-cyclopentane

The procedure of Synthesis Example 1 was repeated, except that 2-ethoxymethyl-1-aza-2-cyclopentane (28 g) for forming an aminosilane moiety and trimethylsilane chloride (24 mL) for forming a protective moiety were employed, to thereby yield 1-trimethylsilyl-2-ethoxymethyl-1-aza-2-sila-cyclopentane.

Synthesis Example 3

Synthesis of 3-(2,2,5,5-tetramethyl(1-aza-2,5-disila-cyclopentane)-1-yl)-propymethyldiethoxysilane

The procedure of Synthesis Example 1 was repeated, except that 1,2-bis(chloro-dimethylsilyl)-ethane (product of Gelest) (44 mL) for forming a protective moiety was employed, to thereby yield 3-(2,2,5,5-tetramethyl(1-aza-2,5-disila-cyclopentane)-1-yl)-propymethyldiethoxysilane.

Synthesis Example 4

Synthesis of N,N-bis(trimethylsilyl)aminopropyldimethylethoxysilane

The procedure of Synthesis Example 1 was repeated, except that 3-aminopropyldimethylethoxysilane (product of Gelest) (30 g) for forming an amino moiety was employed, to thereby yield N,N-bis(trimethylsilyl)aminopropyldimethylethoxysilane.

Synthesis Example 5

Synthesis of N-methyl(trimethylsilyl)aminopropymethyldiethoxysilane

The procedure of Synthesis Example 1 was repeated, except that N-methyl-3-aminopropymethyldiethoxysilane (33 g), which had been synthesized through a procedure disclosed in Organic letters (2002), 4(13), 2117 to 2119, for forming an aminosilane moiety and trimethylsilane
chloride (24 mL) for forming a protective moiety were employed, to thereby yield N-methyl(trimethylsilyl)amino-
propylmethyldiethoxyxilane.

Synthesis Example 6
Synthesis of N,N-dimethyl-3-amino propylmethyldiethoxyxilane

[0116] The compound was synthesized in accordance with a procedure disclosed in Japanese Patent Application Laid-
Open (kokai) No. 2003-155381.

Synthesis Example 7
Synthesis of N,N-bis(trimethylsilyl)aminopropytrimethoxyxilane

[0117] Under nitrogen, 3-aminopropyltrimethoxyxilane (product of Gelest) (41 g) for forming an aminosilane moiety
was added to dichloromethane (solvent) (400 mL) placed in a glass flask equipped with an agitator. Subsequently, trimeth-
silane chloride (product of Aldrich) (48 mL) and triethyl-
amine (53 mL) for forming a protective moiety were to
the solution, followed by stirring the mixture at room temperature for 17 hours. The reaction mixture was evaporated by
means of an evaporator, to thereby remove solvent from the mixture. The thus-obtained crude reaction mixture was dist-
tilled under reduced pressure (5 mmHg), to thereby yield 40
g of N,N-bis(trimethylsilyl)aminopropyltrimethoxyxilane as a
125-130°C fraction.

Example 1
(Synthesis of Copolymer A)

[0118] To an autoclave reactor (inner volume: 5 L) whose atmosphere had been purged with nitrogen, cyclohexane
(2,750 g), tetrahydrofuran (41.3 g), styrene (125 g), and 1,3-butadiene (375 g) were placed. The content of the reactor
was adjusted to 10°C, and n-butyllithium (215 mg) was added thereto, to thereby initiate polymerization. The polymeri-
zation was carried out under adiabatic conditions. The temperature of the polymerization reached 85°C (maximum).

[0119] When percent conversion in polymerization reached 99%, butadiene (10 g) was added to the polymerization system,
followed by polymerization for a further 5 minutes. The resultant polymer solution was removed from the reactor,
and a small aliquot of the solution was sampled and added to a methanol (1 g) in cyclohexane (30 g). N,N-bis
(trimethylsilyl)aminopropylmethyldiethoxyxilane (1,129
mg), obtained in Synthesis Example 1, was added thereto, and
modification reaction was performed for 15 minutes. Subse-
sequently, titanium ethylhexyl dineate (8.11 g) was added to
the reaction mixture, followed by stirring for 15 minutes.
Finally, after completion of reaction, 2,6-di-tert-butyl-p-cres-
ol was added thereto. The mixture was subjected to steam stripping, to thereby remove solvent. The thus-formed rubber
was dried by means of a hot roller (maintained at 110°C), to
thereby yield copolymer A. Table 1 shows the polymerization formula for producing copolymer A, and Table 2 shows physical properties of the copolymer.

Example 2
(Synthesis of Copolymer B)

[0120] The procedure of Example 1 was repeated, except that titanium ethylhexyl dineate was replaced by titanium
di-n-butoxide(bis-2,4-pentanedionate), to thereby yield copolymer B. Table 1 shows the polymerization formula for
producing copolymer B, and Table 2 shows physical properties of the copolymer.

Example 3
(Synthesis of Copolymer C)

[0121] The procedure of Example 1 was repeated, except that titanium ethylhexyl dineate was replaced by titanium
2-ethylhexoxide, to thereby yield copolymer C. Table 1 shows the polymerization formula for producing copolymer
C, and Table 2 shows physical properties of the copolymer.

Example 4
(Synthesis of Copolymer D)

[0122] The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxyxilane
was replaced by 1-trimethylsilyl-2-ethoxymethyl-1-aza-
2-silacyclopentane, obtained in Synthesis Example 2, to
thereby yield copolymer D. Table 1 shows the polymerization formula for producing copolymer D, and Table 2 shows physical properties of the copolymer.

Example 5
(Synthesis of Copolymer E)

[0123] The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxyxilane
was replaced by 3-(2,2,5,5-tetramethyl(1-aza-2,5-distila-
cyclopentane)-1-yl)-propylmethyldiethoxyxilane, obtained in
Synthesis Example 3, to thereby yield copolymer E. Table
1 shows the polymerization formula for producing copolymer
E, and Table 2 shows physical properties of the copolymer.

Comparative Example 1
(Synthesis of Copolymer F)

[0124] The procedure of Example 1 was repeated, except that no titanium ethylhexyl dineate was added, to thereby
yield copolymer F. Table 1 shows the polymerization formula for producing copolymer F, and Table 2 shows physical properties of the copolymer.

Comparative Example 2
(Synthesis of Copolymer G)

[0125] The procedure of Example 1 was repeated, except that titanium ethylhexyl dineate was replaced by tin 2-ethyl-
hexancate, to thereby yield copolymer G. Table 1 shows the polymerization formula for producing copolymer G, and
Table 2 shows physical properties of the copolymer.

Comparative Example 3
(Synthesis of Copolymer H)

[0126] The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxyxilane
was replaced by N,N-bis(trimethylsilyl)aminopropyl-
methyldiethoxyxilane, obtained in Synthesis Example 4, to
thereby yield copolymer H. Table 1 shows the polymerization formula for producing copolymer H, and Table 2 shows physical properties of the copolymer. Comparative Example 4 (Synthesis of copolymer I)
The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxysilane was replaced by N-methyl(trimethylsilyl)aminopropylmethyldiethoxysilane, obtained in Synthesis Example 5, to thereby yield copolymer I. Table 1 shows the polymerization formula for producing copolymer I, and Table 2 shows physical properties of the copolymer.

### Comparative Example 5  
(Synthesis of Copolymer J)

The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxysilane was replaced by N,N-dimethyl-3-aminopropylmethyldiethoxysilane, obtained in Synthesis Example 6, to thereby yield copolymer J. Table 1 shows the polymerization formula for producing copolymer J, and Table 2 shows physical properties of the copolymer.

### Comparative Example 6  
(Synthesis of Copolymer K)

The procedure of Example 1 was repeated, except that N,N-bis(trimethylsilyl)aminopropylmethyldiethoxysilane was replaced by N,N-bis(trimethylsilyl)aminopropyltriethoxysilane, obtained in Synthesis Example 7, to thereby yield copolymer K. Table 1 shows the polymerization formula for producing copolymer K, and Table 2 shows physical properties of the copolymer.

#### Table 1

<table>
<thead>
<tr>
<th>Comp. Ex. 1</th>
<th>Comp. Ex. 2</th>
<th>Comp. Ex. 3</th>
<th>Comp. Ex. 4</th>
<th>Comp. Ex. 5</th>
<th>Comp. Ex. 6</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>B</td>
<td>C</td>
<td>D</td>
<td>E</td>
<td>F</td>
</tr>
<tr>
<td>G</td>
<td>H</td>
<td>I</td>
<td>J</td>
<td>K</td>
<td></td>
</tr>
</tbody>
</table>

| Modified conjugated diene based polymer | Ex. 1 | Ex. 2 | Ex. 3 | Ex. 4 | Ex. 5 | Comp. Ex. 1 | Comp. Ex. 2 | Comp. Ex. 3 | Comp. Ex. 4 | Comp. Ex. 5 | Comp. Ex. 6 |
| Solvent: cyclohexane (g) | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 | 2750 |
| Vinyl content regulator | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 | 41.3 |
| Monomers | | | | | | | | | | | |
| styrene (g) | 125 | 125 | 125 | 125 | 125 | 125 | 125 | 125 | 125 | 125 | 125 |
| butadiene (g) | 375 | 375 | 375 | 375 | 375 | 375 | 375 | 375 | 375 | 375 | 375 |
| Polymerization initiator | | | | | | | | | | | |
| Modifying agent | | | | | | | | | | | |
| N-Si-1* (mg) | 1129 | 1129 | 1129 | — | — | 1129 | 1129 | — | — | — | — |
| N-Si-2* (mg) | — | — | — | — | — | — | — | — | — | — | 699 |
| N-Si-3* (mg) | — | — | — | — | — | — | — | — | — | — | 1075 |
| N-Si-4* (mg) | — | — | — | — | — | — | — | — | — | — | 1027 |
| N-Si-5* (mg) | — | — | — | — | — | — | — | — | — | — | 902 |
| N-Si-6* (mg) | — | — | — | — | — | — | — | — | — | — | 675 |
| N-Si-7** (mg) | — | — | — | — | — | — | — | — | — | — | 1231 |
| Condensation-accelerating agent | | | | | | | | | | | |
| Ti(EHDO)_{4}* (g) | 8.11 | — | 8.11 | 8.11 | — | 8.11 | 8.11 | 8.11 | 8.11 | 8.11 | 8.11 |
| Sn(EHA)_{4}* (g) | — | — | — | — | — | — | — | — | — | — | — |

**Note:**  
*N-Si-1: N,N-bis(trimethylsilyl)aminopropylmethyldiethoxysilane  
*N-Si-2: 1-trimethylsilyl-2-methylethoxy-1-aza-2-cyclopentane  
*N-Si-3: (2,2,5,5-tetramethyl(1-aza-2,5-disilacyclopentane)-1-yl)-propylmethyldiethoxysilane  
*N-Si-4: N-methyl(trimethylsilyl)aminopropylmethyldiethoxysilane  
*N-Si-5: N,N-dimethyl-3-aminopropylmethyldiethoxysilane  
*N-Si-6: N,N-bis(trimethylsilyl)aminopropylmethyldiethoxysilane  
*Ti(EHDO)_{4}: Titanium ethyl enol diolate  
*Sn(EHA)_{4}: Titanium di-n-butoxide (bis-2,4-pentanedionate)  
*Ti(OE)_{4}: Titanium 2-ethylhexoxide  
**Sn(EHA)_{4}: Tin 2-ethylhexanoate
Examples 6 to 10 and Comparative Examples 7 to 12

Each of the modified diene polymers A to K shown in Table 1, produced in Examples 1 to 5 and Comparative Examples 1 to 6, was blended with additives shown in Table 3 (formulation I), to thereby prepare a carbon-black-blended rubber composition through the below-described procedure. The rubber composition was vulcanized at 160°C for 15 minutes, and physical properties of the vulcanized rubber were determined.

The results are shown in Table 4. In Table 4, each of the heat-buildup-suppressing performance (tanδ: 50°C) and wear resistance is shown with an index with respect to the corresponding value of the sample of Comparative Example 7, which is taken as 100. The greater the value of an index, the more excellent the corresponding property.

<Formulation I: Carbon-Black-Blend Formulation>

To a blend (100 parts) containing each modified conjugated diene based polymer of the present invention shown in Table 2 (80 parts) and polyisoprene rubber (20 parts), carbon black, aromatic oil, stearic acid, and an anti-aging agent 6C were added at the corresponding proportions shown in Table 3 (formulation I), to thereby prepare a master batch. To the master batch, zinc oxide, vulcanization accelerators DPG, DM, and NS, and sulfur were added, to thereby prepare a carbon-black-blended rubber composition.

### Table 2

<table>
<thead>
<tr>
<th>Comp. Ex. 1</th>
<th>Comp. Ex. 2</th>
<th>Comp. Ex. 3</th>
<th>Comp. Ex. 4</th>
<th>Comp. Ex. 5</th>
<th>Comp. Ex. 6</th>
</tr>
</thead>
<tbody>
<tr>
<td>Ex. 1</td>
<td>Ex. 2</td>
<td>Ex. 3</td>
<td>Ex. 4</td>
<td>Ex. 5</td>
<td>Ex. 6</td>
</tr>
<tr>
<td>Bonded styrene content (%)</td>
<td>20</td>
<td>20</td>
<td>21</td>
<td>20</td>
<td>20</td>
</tr>
<tr>
<td>Vinyl content (%)</td>
<td>56</td>
<td>56</td>
<td>55</td>
<td>55</td>
<td>55</td>
</tr>
<tr>
<td>Mooney viscosity</td>
<td>32</td>
<td>31</td>
<td>41</td>
<td>33</td>
<td>25</td>
</tr>
</tbody>
</table>

### Table 3

<table>
<thead>
<tr>
<th>Formulation (parts by mass)</th>
<th>Formulation</th>
<th>Formulation</th>
</tr>
</thead>
<tbody>
<tr>
<td>1st stage</td>
<td>I</td>
<td>II</td>
</tr>
<tr>
<td>Modified conjugated diene based polymer</td>
<td>80</td>
<td>80</td>
</tr>
<tr>
<td>Polyisoprene rubber</td>
<td>20</td>
<td>20</td>
</tr>
<tr>
<td>Aromatic oil</td>
<td>10</td>
<td>10</td>
</tr>
<tr>
<td>Carbon black</td>
<td>50</td>
<td>—</td>
</tr>
<tr>
<td>Silica</td>
<td>—</td>
<td>55</td>
</tr>
<tr>
<td>Silane coupling agent</td>
<td>—</td>
<td>5.5</td>
</tr>
<tr>
<td>Stearic acid</td>
<td>2.0</td>
<td>2.0</td>
</tr>
<tr>
<td>Anti-aging agent 6C</td>
<td>1.0</td>
<td>1.0</td>
</tr>
</tbody>
</table>

| 2nd stage | Zinc Oxide | 3.0 | 3.0 |
| Vulcanization accelerator DPG | 0.5 | 1 |
| DMP | 0.5 | 1 |
| NS | 0.5 | 1 |
| Sulfur | 1.5 | 1.5 |

Note:
1. Modified conjugated diene based polymer: Shown in Table 1
2. Polyisoprene rubber: JR2200, product of JSR
3. Aromatic oil: Arenax #3, product of Fuji Kenan Co., Ltd.
4. Carbon black: DIABLACK N339, product of Mitsubishi Chemical Corporation
5. Silica: AQ, product of Tosoh Silica
6. Silane coupling agent: S-690, product of Degussa

### Table 4

<table>
<thead>
<tr>
<th>Comp. Ex. 1</th>
<th>Comp. Ex. 2</th>
<th>Comp. Ex. 3</th>
<th>Comp. Ex. 4</th>
<th>Comp. Ex. 5</th>
<th>Comp. Ex. 6</th>
</tr>
</thead>
<tbody>
<tr>
<td>Ex. 6</td>
<td>Ex. 7</td>
<td>Ex. 8</td>
<td>Ex. 9</td>
<td>Ex. 10</td>
<td>Ex. 11</td>
</tr>
<tr>
<td>Modified conjugated diene based polymer</td>
<td>A</td>
<td>B</td>
<td>C</td>
<td>D</td>
<td>E</td>
</tr>
<tr>
<td>Vulcanize characteristics (carbon black blend)</td>
<td>tanδ (50°C) (index)</td>
<td>131</td>
<td>121</td>
<td>108</td>
<td>129</td>
</tr>
<tr>
<td>Wear resistance (index)</td>
<td>128</td>
<td>116</td>
<td>107</td>
<td>127</td>
<td>121</td>
</tr>
</tbody>
</table>
Examples 11 to 15 and Comparative Examples 13 to 18

Each of the modified diene polymers A to K shown in Table 1, produced in Examples 1 to 5 and Comparative Examples 1 to 6, was blended with additives shown in Table 3 (formulation II), to thereby prepare a silica-blended rubber composition through the below-described procedure. The rubber composition was vulcanized at 160 °C for 15 minutes, and physical properties of the vulcanized rubber were determined.

The results are shown in Table 5. In Table 5, each of heat-buildup-suppressing performance (tanδ: 50 °C) and wear resistance is shown by an index with respect to the corresponding value of the sample of Comparative Example 13, which is taken as 100. The more the value of an index, the more excellent the corresponding property.

TABLE 5

<table>
<thead>
<tr>
<th>Comp. Ex. 11</th>
<th>Comp. Ex. 12</th>
<th>Comp. Ex. 13</th>
<th>Comp. Ex. 14</th>
<th>Comp. Ex. 15</th>
<th>Comp. Ex. 16</th>
<th>Comp. Ex. 17</th>
<th>Comp. Ex. 18</th>
</tr>
</thead>
<tbody>
<tr>
<td>tanδ (50 °C) (index)</td>
<td>126</td>
<td>112</td>
<td>119</td>
<td>124</td>
<td>124</td>
<td>100</td>
<td>94</td>
</tr>
<tr>
<td>Wear resistance (index)</td>
<td>124</td>
<td>109</td>
<td>117</td>
<td>121</td>
<td>120</td>
<td>100</td>
<td>92</td>
</tr>
</tbody>
</table>

As is clear from Tables 4 and 5, the rubber composition samples of the present invention (Examples 6 to 10) containing carbon black serving as a filler and the modified conjugated diene based polymer produced through a step (a) of modifying a conjugated diene based polymer with a silicon compound essentially having a protected primary amino group and a bi-functional silicon atom to which an alkyoxyl group is bonded, and a step (b) of performing condensation reaction in the presence of a titanium compound serving as a condensation-accelerating agent exhibit excellent heat-buildup-suppressing performance (low loss property) and wear resistance, as compared with the rubber composition sample (Comparative Example 7) produced through the step (a) (not including the step (b)); the rubber composition sample (Comparative Example 8) produced through the step (a) and the step (b) but employing a different condensation accelerator; and the rubber composition samples (Comparative Examples 9 to 12) produced through the step (b) and the step (a) but employing a different modifying agent.

Particularly, the effect of the invention is remarkable, as compared with the rubber composition samples of Comparative Examples 9 to 12 produced from a modifying agent differing from the modifying agent employed in the invention.

The compositions shown in Table 5 were prepared from silica serving as a filler, instead of carbon black. As is clear from Table 5, use of silica instead of carbon black can also provide excellent heat-buildup-suppressing performance and wear resistance.

INDUSTRIAL APPLICABILITY

The rubber composition of the invention containing a modified copolymer produced through the method of the present invention attains excellent interaction between the rubber component and carbon black and/or silica, whereby dispersibility of carbon black and/or silica in the composition can be improved, and tires exhibiting excellent heat-buildup-suppressing performance, fracture characteristics, wear resistance, etc. can be provided. Particularly, the rubber composition can be effectively employed as tire tread coating for rubber low-fuel-consumption automobiles.

1. A process for producing a modified conjugated diene based polymer, the process comprising:

(1) a step (a) of reacting a silicon compound with a conjugated diene based polymer having an active end so that the reaction takes place at the active end, the silicon compound having a protected primary amino group in the molecule thereof and a bi-functional silicon atom to which a hydrocarboxyloxy group and a reactive group including a hydrocarboxyloxy group are bonded, to thereby modify the active end, and

(2) a step (b) of performing condensation reaction which involves the compound having a bi-functional silicon atom, in the presence of a titanium compound serving as a titanium-based condensation-accelerating agent.

2. A process for producing a modified conjugated diene based polymer as described in claim 1, which further includes a deprotection step (c) of hydrolyzing a group which is bonded to the active end of the conjugated diene based polymer and which has been derived from the compound having a bi-functional silicon atom, whereby the protected primary amino group contained in the group bonded to the active end is converted to a free amino group.

3. A process for producing a modified conjugated diene based polymer as described in claim 1, wherein the compound having a bi-functional silicon atom employed in the step (a) is a silicon compound represented by formula (1):
(wherein each of R' and R" represents a hydrocarbon group having 1 to 20 carbon atoms, each of R³ to R⁵ represents a hydrocarbon group having 1 to 20 carbon atoms, R⁶ represents an alkylene group having 1 to 12 carbon atoms, A represents a reactive group, and f is an integer of 1 to 10); a silicon compound represented by formula (II):

(II)

(II)

(wherein each of R' to R" represents a hydrocarbon group having 1 to 20 carbon atoms, and R¹³ represents an alkylene group having 1 to 12 carbon atoms); or a silicon compound represented by formula (III):

(III)

(III)

(III)

(wherein each of R¹ and R² represents a hydrocarbon group having 1 to 20 carbon atoms, each of R³ to R⁵ represents a hydrocarbon group having 1 to 20 carbon atoms, R⁶ represents an alkylene group having 1 to 12 carbon atoms, R¹³ represents an alkylene group having 1 to 12 carbon atoms, A represents a reactive group, and f is an integer of 1 to 10).

4. A process for producing a modified conjugated diene based polymer as described in claim 3, wherein the group A in formula (I) is a halogen atom or a alkoxyl group having 1 to 20 carbon atoms.

5. A process for producing a modified conjugated diene based polymer as described in claim 1, wherein the conjugated diene based polymer having an active end is produced through anionic polymerization, in the presence of an organic alkali metal compound serving as a polymerization initiator, of a conjugated diene compound singly or of a conjugated diene compound and an aromatic vinyl compound in combination.

6. A process for producing a modified conjugated diene based polymer as described in claim 5, wherein the conjugated diene compound is at least one species selected from the group of 1,3-butadiene, isoprene, and 2,3-dimethyl-1,3-butadiene.

7. A process for producing a modified conjugated diene based polymer as described in claim 5, wherein the aromatic vinyl compound is styrene.

8. A process for producing a modified conjugated diene based polymer as described in claim 5, wherein the conjugated diene based polymer has a polymer unit derived from an aromatic vinyl compound in an amount of 0 to 55 mass % based on the total polymer unit present in the conjugated diene based polymer and a vinyl bond content of 7 to 65 mass % based on the total conjugated diene portion.

9. A process for producing a modified conjugated diene based polymer as described in claim 1, wherein the condensation-accelerating agent employed in the step (b) is at least one species selected from among a titanium alkoxide, a titanium carboxylate salt, a titanium acetylacetonate complex salt, and a salt mixture thereof.

10. A modified conjugated diene based polymer produced by a process as recited in claim 1.


12. A rubber composition as described in claim 11, which comprises a rubber component containing 20 mass % or more of the modified conjugated diene base polymer in an amount of 100 parts by mass and silica and/or carbon black in the total amount of 20 to 120 parts by mass.

13. A rubber composition as described in claim 11, wherein the rubber component comprises the modified conjugated diene based polymer in an amount of 15 to 100 mass % and at least one species selected from among natural rubber, a synthetic isoprene rubber, a butadiene rubber, a styrene-butadiene rubber, an ethylene-octyl copolymer rubber, an ethylene-octene-diene copolymer rubber, a chloroprene rubber, a halogenated butyl rubber, and a styrene-ethylene copolymer having a halomethyl group, in an amount of 85 to 0 mass %.

14. A pneumatic tire employing a rubber composition as recited in claim 11.

15. A pneumatic tire employing a rubber composition as recited in claim 11 as a tread, a base tread, or a sidewall thereof.