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(71) Demandeurs/Applicants:
LANXESS CORPORATION, US;
LANXESS DEUTSCHLAND GMBH, DE

(72) Inventeurs/Inventors:
NASREDDINE, VICTOR, US;
SODDEMANN, MATTHIAS, DE

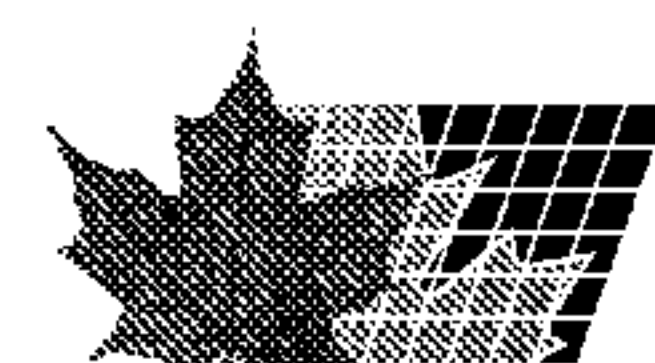
(74) Agent: OGILVY RENAULT LLP/S.E.N.C.R.L.,S.R.L.

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EXCELLENTE APTITUDE A LA MISE EN OEUVRE ET UNE EXCELLENTE RESISTANCE AUX FLUIDES
AGRESSIFS

(54) Title: HNBR COMPOSITIONS WITH VERY HIGH FILLER LEVELS HAVING EXCELLENT PROCESSABILITY AND
RESISTANCE TO AGGRESSIVE FLUIDS

(57) **Abrégé/Abstract:**

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(71) Applicants (for all designated States except US):

LANXESS CORPORATION [US/US]; 111 RIDC Park
West Drive, Pittsburgh, Pennsylvania 15275 (US).
LANXESS DEUTSCHLAND GMBH [DE/DE];
D-51368 Leverkusen (DE).

(72) Inventors; and

(71) Applicants : **NASREDDINE, Victor** [CA/US]; 1005
Stockton Ridge, Cranberry Township, Pennsylvania
16066 (US). **SODDEMANN, Matthias** [DE/DE]; Ber-
gische Landstrasse 89b, D-51503 Rosrath (DE).

(74) Agents: **SENG, Jennifer R.** et al.; LANXESS Corpora-
tion, 111 RIDC Park West Drive, Pittsburgh, Pennsylva-
nia 15275 (US).

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HNBR COMPOSITIONS WITH VERY HIGH FILLER LEVELS HAVING EXCELLENT PROCESSABILITY
AND RESISTANCE TO AGGRESSIVE FLUIDS

FIELD OF THE INVENTION

The present invention relates to an elastomeric composition containing a nitrile rubber, preferably hydrogenated, and a filler. Compositions according to the present invention have high filler levels, excellent processability, and very good resistance to aggressive fluids and fuels compared to known hydrogenated nitrile rubber compositions.

BACKGROUND OF THE INVENTION

There is a continuous demand in the automotive, heavy duty, and chemical industry for high performance elastomers that have improved resistance to aggressive oils and fuels. Hydrogenated nitrile butadiene rubber (HNBR) is a high performance specialty elastomer used in a variety of demanding engineering applications: automotive, chemical and heavy duty industries, oil well exploration, and mining. HNBR, is made by the selective hydrogenation of the double bonds in nitrile rubber. It has a combination of unique properties: high tensile strength and very good mechanical properties even at elevated temperatures, excellent abrasion resistance, low compression set, excellent heat resistance, very good resistance to ozone, weathering, and high energy radiation, low permeability to vapors and gases. In addition, HNBR has very good resistance to oils, fluids, diesel, fuels, sour gasoline, lubricating oils with aggressive alkaline additives, as well as good resistance to crude oil even in the presence of hydrogen sulfide, amines, and corrosion inhibitors.

US Patent Nos. 7,381,781; 6,841,623; and 6,780,939 assigned to LANXESS recently introduced to the rubber industry a novel HNBR technology that

enables the production of low viscosity HNBR polymers, previously commercially impossible to manufacture. The commercial product is sold by LANXESS under the trade name THERBAN® Advanced Technology (AT). These low Mooney viscosity HNBR grades have several benefits over standard HNBR products, including: retention of physical properties known for HNBR polymer plus the added benefit of improved processability; combined with a narrow molecular weight distribution to maintain excellent mechanical properties; faster black incorporation time and lower mix temperature; faster mold filling and shorter cycle times in injection molding; faster output and compounds of smoother surfaces and sharper edges are obtained in extrusion; lower compound costs by using higher fillers levels.

The present invention provides novel HNBR compositions that have very high filler levels resulting in very good mechanical properties, processability, and especially excellent resistance and minimal swelling in fluids known to be aggressive to nitrile polymers such as methyl ethyl ketone (MEK), toluene, and certain fuels. Such compositions and their properties are not known in the art.

SUMMARY OF THE INVENTION

The present invention relates to an elastomeric composition containing a nitrile rubber, preferably hydrogenated, and a filler. Compositions according to the present invention have high filler levels, excellent processability, and very good resistance to aggressive fluids and fuels compared to known HNBR compositions.

The present invention also provides a novel HNBR composition containing high acrylonitrile low Mooney viscosity HNBR polymers and high filler levels. The resulting compositions have very good mechanical properties, processability, and especially excellent resistance and minimal swelling in fluids known to be aggressive to nitrile polymers such as methyl ethyl ketone (MEK) and toluene.

The novel HNBR compositions of the present invention enable the use of HNBR in new applications requiring resistance to these media as well as in a variety of fuels such as gasoline, flex-fuel, and biodiesel. Such compositions and their properties are not known in the art.

DETAILED DESCRIPTION OF THE INVENTION

The present invention therefore provides for an elastomer composition containing a nitrile rubber, preferably hydrogenated, and a filler.

As used throughout this specification, the term "nitrile rubber" or NBR is intended to have a broad meaning and is meant to encompass a copolymer having repeating units derived from at least one conjugated diene, at least one α,β -unsaturated nitrile and optionally further one or more copolymerizable monomers. The conjugated diene may be any known conjugated diene such as a C_4 - C_6 conjugated diene. Preferred conjugated dienes include butadiene, isoprene, piperylene, 2,3-dimethyl butadiene and mixtures thereof. More preferred C_4 - C_6 conjugated dienes are butadiene, isoprene and mixtures thereof. The most preferred C_4 - C_6 conjugated diene is butadiene. The α,β -unsaturated nitrile may be any known α,β -unsaturated nitrile, such as a C_3 - C_5 α,β -unsaturated nitrile. Preferred C_3 - C_5 α,β -unsaturated nitriles include acrylonitrile, methacrylonitrile, ethacrylonitrile and mixtures thereof. The most preferred C_3 - C_5 α,β -unsaturated nitrile is acrylonitrile. The preparation of nitrile rubbers via polymerization of the above referenced monomers is well known to a person skilled in the art and is extensively described in the literature (i.e, Houben-Weyl, Methoden der Organischen Chemie, Vol. 14/1, Georg Thieme Verlag Stuttgart, 1961).

Hydrogenated in this invention is preferably understood by more than 50 % of the residual double bonds (RDB) present in the starting nitrile polymer/NBR being hydrogenated, preferably more than 90 % of the RDB are hydrogenated, more preferably more than 95 % of the RDB are hydrogenated and most preferably more than 99 % of the RDB are hydrogenated. The hydrogenation of nitrile rubber is well known in the art and described in, for

example, U.S. Patents No. 3,700,637, U.S. Patent No. 4,464,515 and U. S. Patent No. 4,503,196.

The present invention also includes the use of carboxylated nitrile rubbers. As used throughout this specification, the term "carboxylated nitrile rubber" or XNBR is intended to have a broad meaning and is meant to encompass a copolymer having repeating units derived from at least one conjugated diene, at least one α,β -unsaturated nitrile, at least one alpha-beta-unsaturated carboxylic acid or alpha-beta-unsaturated carboxylic acid derivative and optionally further one or more copolymerizable monomers α,β -unsaturated mono- or dicarboxylic acids, or their esters or amides. Preferred α,β -unsaturated mono- or dicarboxylic acids here are fumaric acid, maleic acid, acrylic acid and methacrylic acid. Preferred esters used of the α,β -unsaturated carboxylic acids are their alkyl esters and alkoxyalkyl esters. Particularly preferred esters of the α,β -unsaturated carboxylic acids are methyl acrylate, ethyl acrylate, butyl acrylate, butyl methacrylate, 2-ethylhexyl acrylate, 2-ethylhexyl methacrylate and octyl acrylate.

Suitable nitrile rubbers according to the present invention include low Mooney viscosity hydrogenated nitrile rubbers. Suitable rubbers have a Mooney viscosity of below 55 (ML 1+4 100C). Preferably the low Mooney viscosity polymers have a Mooney viscosity between 0 – 55. More preferably, the low Mooney viscosity polymers have a Mooney viscosity between 5 – 50, most preferably between 10 – 40. Even more preferably, the low Mooney viscosity polymers have a Mooney viscosity below 30 and even more preferably below 10. Suitable low Mooney viscosity polymers can be prepared by processes disclosed in U.S. Patent No. 7,381,781, U.S. Patent No. 6,841,623 and U. S. Patent No. 6,780,939. The subject matter thereof being incorporated by reference.

Suitable nitrile rubbers according to the present invention should have a medium to high acrylonitrile content (ACN) for an acceptable degree of fluid and fuel resistance. Preferably, the nitrile rubbers according to the present invention have a acrylonitrile content greater than 15 %, more preferably

greater than 30 %, even more preferably greater than 39 % and most preferably, greater than 43 %. Suitable nitrile rubbers are partially or fully hydrogenated and contain less than 10% of residual double bonds. Preferably, the nitrile rubbers are fully saturated and contain less than 1% of residual double bonds.

The composition of the present invention includes the use of high filler levels, including high white filler levels and/or black filler levels. According to the present invention, black fillers, such as carbon black is present in the composition in an amount of in the range of from 200 to 500 phr (parts per hundred rubber), preferably in the range of from 250 to 350 phr. The carbon blacks useful in the present invention are preferably prepared by the lamp black, furnace black or gas black process and have preferably BET (DIN 66 131) specific surface areas in the range of from 20 to 200 m²/g, e.g. SAF, ISAF, HAF, FEF or GPF carbon blacks.

Compositions of the present invention also include the use of over 100 phr of white fillers. Suitable white mineral fillers for use in the present invention include:

- highly dispersed silicas, prepared e.g. by the precipitation of silicate solutions or the flame hydrolysis of silicon halides, with specific surface areas of in the range of from 5 to 1000 m²/g, and with primary particle sizes of in the range of from 10 to 400 nm; the silicas can optionally also be present as mixed oxides with other metal oxides such as those of Al, Mg, Ca, Ba, Zn, Zr and Ti;
- synthetic silicates, such as aluminum silicate and alkaline earth metal silicate like magnesium silicate or calcium silicate, with BET specific surface areas in the range of from 20 to 400 m²/g and primary particle diameters in the range of from 10 to 400 nm;
- natural silicates, such as kaolin and other naturally occurring silica;
- glass fibers and glass fiber products (matting, extrudates) or glass microspheres;

Preferred white fillers are acircular or nonisometric materials with an aspect ratio of 2:1 to 200:1 (including clays, talcs, micas). The aspect ratio of these platy materials is defined as the ratio of mean diameter of a circle of the same area as the face of the plate to the mean thickness of the plate. More preferable are needle-like structures with an aspect ratio of 2:1 to 20:1. The aspect ratio of for needle and fiber shaped fillers is the ratio of length to diameter.

Composition according to the present invention may also include the use of synergistic stabilizer system. Suitable systems include those disclosed in U.S. Patent Publication Nos. 2003/176587(A1), 2001/0018479, 2003/0170549, 2004/0992634 and 2005/014352; the subject matter thereof being incorporated herein by reference.

The composition according to the present invention can also include the use of acid acceptors such as metal oxides including magnesium oxide, calcium oxide, and carbonates with alkaline pH. The acid acceptor is added to the composition of the present invention in an amount in the range of 0-200phr, more preferably 0-100phr, and most preferably 0-10phr.

The rubber elastomer according to the present invention can contain further auxiliary products for rubbers, such as reaction accelerators, vulcanizing accelerators, vulcanizing acceleration auxiliaries, antioxidants, foaming agents, anti-aging agents, heat stabilizers, light stabilizers, ozone stabilizers, processing aids, plasticizers, tackifiers, blowing agents, dyestuffs, pigments, waxes, extenders, organic acids, inhibitors, metal oxides, and activators such as triethanolamine, polyethylene glycol, hexanetriol, etc., which are known to the rubber industry. The rubber aids are used in conventional amounts, which depend inter alia on the intended use. Conventional amounts are e.g. from 0.1 to 50 phr.

According to the present invention, the composition can contain in the range of 0.1 to 20 phr of an organic fatty acid as an auxiliary product, such as a unsaturated fatty acid having one, two or more carbon double bonds in the

molecule which more preferably includes 10% by weight or more of a conjugated diene acid having at least one conjugated carbon-carbon double bond in its molecule. Those fatty acids can have in the range of from 8-22 carbon atoms, or for example from 12-18. Examples include stearic acid, palmitic acid and oleic acid and their calcium-, zinc-, magnesium-, potassium- and ammonium salts.

According to the present invention, the composition can contain in the range of 5 to 50 phr of an acrylate as an auxiliary product. Suitable acrylates are known from EP-A1-0 319 320, in particular p. 3, l. 16 to 35, from U.S. Patent No. 5,208,294, see Col. 2, l. 25 to 40, and from U.S. Patent No. 4,983,678, in particular Col. 2, l. 45 to 62. Reference is made to zinc acrylate, zinc diacrylate or zinc dimethacrylate or a liquid acrylate, such as trimethylolpropane-trimethacrylate (TRIM), butanedioldimethacrylate (BDMA) and ethylenglycoldimethacrylate (EDMA). It might be advantageous to use a combination of different acrylates and/or metal salts thereof. It may also be advantageous to use metal acrylates in combination with a Scorch-retarder such as sterically hindered phenols (e.g. methyl-substituted aminoalkylphenols, such as 2,6-di-tert.-butyl-4-dimethylaminomethylphenol).

An antioxidant may be used in preparing a compound according to the present invention. Examples of suitable antioxidants include p-dicumyl diphenylamine (NAUGARD[®] 445), VULKANOX[®] DDA (a diphenylamine derivative), VULKANOX[®] ZMB2 (zinc salt of methylmercapto benzimidazole), VULKANOX[®] HS (polymerized 1,2-dihydro-2,2,4-trimethyl quinoline) and IRGANOX[®] 1035 (thiodiethylene bis(3,5-di-tert.-butyl-4-hydroxy) hydrocinnamate or thiodiethylene bis(3-(3,5-di-tert.-butyl-4-hydroxyphenyl)propionate supplied by Ciba-Geigy. VULKANOX is a trademark of Bayer AG.

Similarly, in preparing compounds according to the present invention it is useful to employ a crosslinking agent, including commercially available agents including sulfur/sulfur accelerator systems, diamines and peroxides. Most preferred are the peroxide based vulcanizing agents due to the excellent

thermal stability conveyed by the carbon-carbon linkages between polymer chains. Useful peroxide crosslinking agents, include dicumyl peroxide (Di-Cup 40KE), di-tert.-butyl peroxide, benzoyl peroxide, 2,2'-bis (tert.-butylperoxy) diisopropylbenzene (VULCUP[®] 40KE), benzoyl peroxide, 2,5-dimethyl-2,5-di(tert-butylperoxy)-hexyne-3, 2,5-dimethyl-2,5-di(benzoylperoxy)hexane, (2,5-bis(tert.-butylperoxy)-2,5-dimethyl hexane and the like. Preferred curing agents are readily determined by means of a few preliminary experiments, which is within the scope of one skilled in the art. A preferred peroxide curing agent is commercially available under the tradename Di-Cup 40KE. The peroxide curing agent (60% active) is suitably used in an amount of 0.1 to 15 parts per hundred parts of rubber (phr), preferably 4 to 10 phr. Too much peroxide may lead to undesirably violent reaction.

Vulcanizing co-agents can also be added to the composition of the present invention. Mention is made of triallyl isocyanurate (TAIC), commercially available under the trademark DIAK 7 from DuPont Or N,N'-m-phenylene dimaleimide known as HVA-2 (DuPont Dow), triallyl cyanurate (TAC) or liquid polybutadiene known as Ricon D 153 (supplied by SARTOMER). Amounts can be equivalent to the peroxide curative or less, preferably equal.

The present invention also includes the use of activators such as zinc peroxide (50% on an inert carrier) using Struktol ZP 1014 in combination with the peroxide. Amounts can be from 0.1 to 15, preferably from 4 to 10phr.

The ingredients of the elastomer composition are often mixed together, suitably at an elevated temperature that may range from 25 °C to 200 °C. Normally the mixing time does not exceed one hour and a time in the range from 2 to 30 minutes is usually adequate. Mixing is suitably carried out in an internal mixer such as a Banbury mixer, or a Haake or Brabender miniature internal mixer. A two roll mill mixer also provides a good dispersion of the additives within the elastomer. An extruder also provides good mixing, and permits shorter mixing times. It is possible to carry out the mixing in two or more stages, and the mixing can be done in different apparatus, for example

one stage in an internal mixer and one stage in an extruder. However, it should be taken care that no unwanted pre-crosslinking (= scorch) occurs during the mixing stage. For compounding and vulcanization see also: Encyclopedia of Polymer Science and Engineering, Vol. 4, p. 66 et seq. (Compounding) and Vol. 17, p. 666 et seq. (Vulcanization).

The elastomeric composition according to the present invention is especially suitable for injection molding articles such as the present invention relates to shaped articles, such as seals, hoses, bearing pads, stators, well head seals, valve plates, cable sheathing, wheels, rollers, pipe seals, couplings, as well as for alternative fuels applications such as flex-fuels (gasoline-alcohol blends) and biodiesels (i.e., fatty acid methyl esters such as Soybean Methyl Ester/SME and Rapeseed Methyl Ester/RME).

EXAMPLES

Description of tests:

Cure rheometry:

Vulcanization testing was carried out on a Moving Die Rheometer (MDR 2000(E)) using a frequency of oscillation of 1.7 Hz and a 1° arc at 180°C for 60 minutes total run time. The test procedure follows ASTM D-5289.

Compound Mooney Viscosity and Scorch:

A large rotor was used for these tests in compliance with the ASTM method D-1646. The compound Mooney viscosity was determined at 100°C by preheating the sample 1 minute and then, measuring the torque (Mooney viscosity units) after 4 minutes of shearing action caused by the viscometer disk rotating at 2 r.p.m.. Mooney scorch measurements taken as the time from the lowest torque value to a rise of 5 Mooney units (t05) were carried out at 135°C.

Stress-strain:

Samples were prepared by curing a macro sheet at 180°C for 13 minutes. Afterwards, samples were die cut into standard ASTM die C dumbbells. The test was conducted at 23°C and complies with ASTM D-412 Method A.

Hardness:

All hardness measurements were carried out with an A-2 type durometer following the procedure outlined in ASTM D-2240.

Tear resistance:

A tensile sheet cured 14 minutes at 180°C was used to prepare appropriate samples of Die B and Die C geometries. Both tests are designed to give an indication of the resistance to tear of the rubber. The test procedure complies with ASTM D 624.

Din Abrasion:

Abrasion resistance is determined according to test method DIN 53 516. The volume loss by rubbing the rubber specimen with an emery paper of defined abrasive power is measured and reported.

Compression Set:

This testing complies with ASTM D395 (Method B). Solid button type samples were cured for 20 minutes at 180°C and the sample subjected to a 25% compression deflection during hot air aging.

Preparations of Examples:

A laboratory size Banbury BR-82 (1.6 L capacity) internal mixer cooled at 30°C was used to prepare the Examples. Rotor speed was held constant during mixing at 50 rpm. At 0 seconds, 1A and 1B (See Table 1) ingredients were added. At 30 seconds, the 1C ingredients were added to the mixer. A sweep was performed at 120 seconds then 1D ingredients were added at 200 seconds followed by a sweep at 320 seconds. The mix was stopped at 420 seconds. The dropped mix was allowed to cool for four hours prior to addition

of curatives. The curatives (2A ingredients) were both added on a 10" by 20" two roll mill cooled at 30°C.

Table 1. HNBR formulations with high filler levels.

Formulation (PHR)		A	B	C	D	E	F	G
THERBAN AT A4304	1A	100	100	100	100	100	100	100
NAUGARD 445	1B	1.1	1.1	1.1	1.1	1.1	1.1	1.1
VULKANOX ZMB-2	1B	0.4	0.4	0.4	0.4	0.4	0.4	0.4
MAGLITE D	1B	5	5	5	5	5	5	5
KADOC 920C	1B	3	3	3	3	3	3	3
N-990	1C	50	75	100	150	-	-	-
ZEOLEX 23	1C	-	-	-	-	50	-	-
HISIL 532 EP	1C	-	-	-	-	-	50	-
MINSTRON vapor talc	1C	-	-	-	-	-	-	50
STRUKTOL SCA 972	1C	-	-	-	-	5	5	5
PLASTHALL 226	1C	10	10	10	10	10	10	10
N-990	1D	50	75	100	150	-	-	-
ZEOLEX 23	1D	-	-	-	-	50	50	50
STRUKTOL SCA 972	1D	-	-	-	-	5	5	5
PLASTHALL TOTM	1D	10	10	10	10	10	10	10
DIAK 7	2A	1.5	1.5	1.5	1.5	1.5	1.5	1.5
VULCUP 40KE	2A	11	11	11	11	11	11	11
Mooney Viscosity, ML(1+4)@100C		33	42	61	111	51	61	42
Mooney scorch, MS, Large @ 135C, t05 (min)		17	16	14	16	15	17	12
MDR								
MH (dN.m)		32.2	40.2	48.6	64.3	53.9	52.7	44.5
ML (dN.m)		0.40	0.46	0.79	1.77	1.28	1.84	0.77
DELTA MH-ML (dN.m)		31.8	39.7	47.8	62.5	52.6	50.8	43.8
t' 95 (min)		5.47	5.77	5.92	6.02	4.23	3.55	5.03
Stress strain								
Hardness Shore A2 (pts.)		66	76	85	92	84	87	85
Ultimate Tensile (MPa)		16.2	15.5	13.7	12.2	19.1	20.7	19.2
Ultimate Elongation (%)		305	223	169	93	156	155	165
Stress @ 100 (MPa)		4.6	7.8	12.1	-	13.2	14	15.6
DIE C TEAR								
Tear Strength (kN/m) , 23C		34	35.2	35.1	32.7	34.7	35.1	37.2
Tear Strength (kN/m) , 150C		8.9	8.2	10.1	9.5	13.3	13.9	16.7
Compression @ 150C								
168hrs		32	30	25	25	41	40	38
504hrs		51	40	43	43	52	54	52
Fluid aging 70hrs/23C, MEK								
Chg. Hard. Shore A2 (pts.)		-15	-18	-19	-20	-16	-17	-15
Chg. Ulti. Tens. (%)		-64	-41	-33	-18	-47	-44	-39
Chg. Ulti. Elong. (%)		-61	-45	-43	-23	-44	-44	-54

Vol. Change (%)		58	46	35	25	32	28	38
<i>70hrs/23C, TOLUENE</i>								
Chg. Hard. Shore A2 (pts.)		-15	-18	-19	-20	-15	-18	-13
Chg. Ulti. Tens. (%)		-54	-43	-28	-8	-48	-47	-34
Chg. Ulti. Elong. (%)		-52	-40	-35	-23	-46	-45	-48
Vol. Change (%)		47	38	31	26	37	35	40
<i>70hrs/40C, FUELC</i>								
Chg. Hard. Shore A2 (pts.)		-11	-13	-13	-13	-11	-12	-10
Chg. Ulti. Tens. (%)		-31	-23	-9	-5	-34	-31	-21
Chg. Ulti. Elong. (%)		-29	-16	-14	-9	-30	-28	-32
Vol. Change (%)		20	17	15	12	20	20	22
CARBON BLACK N 660	Carbon Black from Cabot Tire Blacks							
CARBON BLACK N 990	Carbon Black from Cabot Tire Blacks							
DIAK® #7	Triallylisocyanurate available from R.T. Vanderbilt Company							
DISFLAMOL DPK	Diphenyl cresyl phosphate plasticizer from LANXESS Deutschland GmbH							
EDENOR® C, 18 98-100	Stearic acid from Cognis Corp							
HI-SIL® 532 EP	Precipitated hydrated amorphous silica from PPG Industries							
HYCITE® 713	Magnesium/Aluminum Hydro-talcite from Sued-Chemie AG							
KADOX® 920	Zinc oxide from Horsehead Company							
LUVOMAXX CDPA	p-dicumyl diphenylamine from Lehmann & Voss Company							
MAGLITE® D	Magnesium oxide from Hallstar Innovations Corp.							
MISTRON® VAOPR TALC	Magnesium aluminum silicate/talcum from Luzenac America, Inc.							
MISTRON® VAPOR	Magnesium silicate (Talc) from Luzenac America Inc.							
PERKADOX 14-40	40% active di-(tertbutylperoxyisopropyl) benzene from AKZO Chemie Nederland B.V.							
PLASTHALL® 226	DBEEA from Hallstar Innovations Corp.							
PLASTHALL® TOTM	Trioctyl trimellitate available from Hallstar Innovations Corp.							
SPIDER SULFUR	Sulfur from Hallstar Innovations Corp.							
STRUKTOL® SCA 972	VINYL SILANE from Schill & Seilacher "Struktol" AG							
THERBAN® AT A4304	Low Mooney viscosity HNBR with high acrylonitrile from LANXESS Deutschland GmbH; 43% ACN, 0.9%max RDB, 39 ML(1+4)@100C							
THERBAN® AT A5005	Low Mooney viscosity HNBR with Ultra-high acrylonitrile from LANXESS Deutschland GmbH; 49% ACN, 0.9%max RDB, 55 ML(1+4)@100C							
VULKANOX® ZMB 2/C5	Zinc-4-and 5-methyl-2-mercaptobezimidazole from LANXESS Deutschland GmbH							
ZEOLEX® 23	Synthetic Sodium Aluminum Silicate from J.M. Huber Corporation							

Table 1 illustrates that up to 300phr of carbon black filler can be used in the HNBR formulations presented and the compounds have very good processability and scorch safety. The Mooney viscosity increases as a function of filler level and mineral fillers such as HiSil 532 EP, ZEOLEX 23, and Talc result in higher viscosities than carbon black N990.

As the carbon black N990 level increases, hardness increases, tensile decreases, and modulus increases. Even at high fillers levels these black-filled compounds still have good mechanical strength (Ultimate tensile > 10MPa). On the other hand, mineral filled compounds have higher hardness, tensile, and modulus but lower elongation than black-filled compounds. All compounds show excellent tear and compression set properties.

As the N990 level increases, there is a significant improvement in the aging resistance of these HNBR compounds in MEK and toluene demonstrated by better retention in tensile, elongation, and lower swelling (reaching < 30%). At the same level, mineral fillers seem to have lower swelling than N990. ZEOLEX 23 and HiSil 532 EP give the lowest swell in MEK and toluene. Increasing the N990 level also results in better tensile and elongation retention as well as lower swelling. All fillers have comparable swell in Fuel C.

Table 2. Comparison of High and Ultra-high acrylonitrile HNBR polymers.

Formulation (PHR)		H	I
Therban AT A 4304	1A	100	-
Therban AT A 5005VP	1A	-	100
LUVOMAXX CDPA	1B	1.5	1.5
Vulkanox ZMB2	1B	0.5	0.5
MAGLITE DE	1C	2	2
N-990	1C	200	200
Hycite 713	1C	15	15
DISFLAMOLL DPK	1C	30	30
DIAK 7	2A	1.5	1.5
PERKADOX 14-40 B-PD	2A	9.5	9.5
Mooney Viscosity, ML(1+4)@100C		53.9	59.7
MDR			
MH (dN.m)		39.4	36.0
ML (dN.m)		0.8	1.1
Delta MH-ML (dN.m)		38.7	34.9
t' 95 (min)		7.4	8.0
Mooney scorch, MS, Large @ 135C, t05 (min)		19	20
Stress strain			
Hardness Shore A2 (pts.)		84	86
Ultimate Tensile (MPa)		12	12

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Ultimate Elongation (%)		223	223
Stress @ 100 (MPa)		11	11
Fluid aging			
<i>168hrs/100C</i>			
<i>ULSD (Ultra Low Sulfur Diesel)</i>			
Chg. Hard. Shore A2 (pts.)		-1.8	1.3
Chg. Ulti. Tens. (%)		-4.9	-4.1
Chg. Ulti. Elong. (%)		-12.1	-7.2
Vol. Change (%)		4	1.7
<i>168hrs/100C,</i>			
<i>B100 (100% SME)</i>			
Chg. Hard. Shore A2 (pts.)		-2.2	1.3
Chg. Ulti. Tens. (%)		-3.3	-1.7
Chg. Ulti. Elong. (%)		-11.7	-16.1
Vol. Change (%)		1.8	-1.9
<i>168hrs/100C</i>			
<i>B20 (20% SME/80% ULSD)</i>			
Chg. Hard. Shore A2 (pts.)		-2.4	1.3
Chg. Ulti. Tens. (%)		-6.6	-4.1
Chg. Ulti. Elong. (%)		-9.0	-11.7
Vol. Change (%)		4.0	1.2
<i>168hrs/100C</i>			
<i>B100 (100% SME)</i>			
Chg. Hard. Shore A2 (pts.)		-2.2	1.3
Chg. Ulti. Tens. (%)		-3.3	-1.7
Chg. Ulti. Elong. (%)		-11.7	-16.1
Vol. Change (%)		1.8	-1.9
<i>168hrs/40C</i>			
<i>E85 (85% Ethanol/15%Fuel C)</i>			
Chg. Hard. Shore A2 (pts.)		-15	-14
Chg. Ulti. Tens. (%)		-19	-21
Chg. Ulti. Elong. (%)		-20	-17
Vol. Change (%)		9	5

Table 2 shows that the use of these high acrylonitrile content polymers (>39%), and ultra-high acrylonitrile (>43%) low Mooney viscosity polymers with high filler levels is possible and yields very good mechanical and processing properties as well very good performance and minimal swelling in biodiesel and flex-fuel.

In the Claims:

1. An elastomeric composition comprising a hydrogenated nitrile rubber (HNBR) and a filler, wherein, the filler is present in an amount greater than 200 parts per rubber.
2. The composition according to Claim 1, wherein the hydrogenated nitrile rubber has a Mooney viscosity below 55 (ML(1+4)@100° C).
3. The composition according to Claim 2, wherein the hydrogenated nitrile rubber has a Mooney viscosity below 30 (ML(1+4)@100° C).
4. The composition according to Claim 3, wherein the hydrogenated nitrile rubber has a Mooney viscosity below 10 (ML(1+4)@100° C).
5. The composition according to Claim 1, wherein the filler is carbon black.
6. The composition according to Claim 1, wherein the filler is a mineral filler selected from the group consisting of silicas, silicates, talcs, and blends thereof.
7. The composition according to Claim 1, wherein the filler is present in an amount in the range of from 250 to 350phr.
8. The composition according to Claim 1, wherein the hydrogenated nitrile rubber is carboxylated.
9. A composition according to Claim 1, wherein the hydrogenated nitrile rubber has an acrylonitrile content (ACN) greater than 39%.
10. A composition according to Claim 9, wherein the hydrogenated nitrile rubber has an acrylonitrile content (ACN) greater than 43%.
11. A composition according to 9, wherein the hydrogenated nitrile rubber is fully saturated and contains less than 1% of residual double bonds.

12. An article comprising the composition according to Claim 1, wherein the article has excellent processability and heat resistance of fluids and fuels.