METAL ORGANIC FRAMEWORKS FOR ELECTRONIC GAS STORAGE

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Related U.S. Application Data

Continuation of application No. 14/828,097, filed on Aug. 17, 2015, which is a continuation of application No. 14/450,994, filed on Aug. 4, 2014, now Pat. No. 9,138,720.

Provisional application No. 61/862,238, filed on Aug. 5, 2013.

ABSTRACT

A metal organic framework (MOF) includes a coordination product of a metal ion and an at least bidentate organic ligand, where the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/l for an electronic gas. A porous organic polymer (POP) includes polymerization product from at least a plurality of organic monomers, where the organic monomers are selected to provide a deliverable adsorption capacity of at least 70 g/l for an electronic gas.
FIG. 1e  
$\text{Zn}^{2+} + \text{CO}_2\text{H} + \text{N} \longrightarrow \text{ZnBDCDABCO}$
(pillared-paddlewheel framework; ZnBDCDABCO)
(BDC equals benzene dicarboxylate)
(DABCO equals 1,4-diazabicyclo[2.2.2]octane)

FIG. 1f  
$\text{Cu}^{2+} + \text{CO}_2\text{H} \longrightarrow \text{Cu-BTC}$
(paddlewheel framework; Cu-BTC)
(BTC equals benzene tricarboxylate)

FIG. 1g  
$\text{Cu}^{2+} + \text{N} \longrightarrow \text{NU-125}$(rhi framework; NU-125)
FIG. 1h  \[ \text{Zn}^{2+} + \text{CO}_2\text{H} \quad \rightarrow \quad \text{(Zn}_2\text{O framework; MOF-5)} \]

FIG. 1i  \[ \text{Zr}^{4+} + \text{CO}_2\text{H} \quad \rightarrow \quad \text{(Zr}_6\text{O}_3\text{O}_2 \text{ or Zr}_7\text{O}_3\text{ framework; UIO-66)} \]

FIG. 1j  \[ \text{Zn}^{2+} + \text{H}_2\text{N}_2\text{O}_2 \quad \rightarrow \quad \text{(heterocycle framework; ZIF-8)} \]
M = any tetrahedral atom including B, N, Al, C, Si, Sn, Ge, etc
R = any generic group attached at any position

NuMatPOP1
Each M-M is attached by a biphenyl group, as illustrated in the one example. (tetrahedral POP framework)
FIG. 8a

FIG. 8b

FIG. 8c
METAL ORGANIC FRAMEWORKS FOR ELECTRONIC GAS STORAGE

RELATED APPLICATIONS

[0001] This application is a continuation of U.S. application Ser. No. 14/882,097, filed Aug. 17, 2015, which is a continuation of U.S. application Ser. No. 14/450,994, filed Aug. 4, 2014, now U.S. Pat. No. 9,138,720, which claims the benefit of U.S. Provisional Application No. 61/862,238, filed Aug. 5, 2013, all of which are hereby incorporated by reference in their entirety.

FIELD

[0002] The present invention is directed to porous materials including metal-organic frameworks and porous organic polymers, specifically metal organic frameworks and porous organic polymers for electronic gas storage.

BACKGROUND

[0003] Extensive research over the past few years has been focused on the synthesis and characterization of microporous materials with high internal surface areas. Metal-Organic Frameworks (MOFs), a crystalline subset of these materials, have shown promise in a wide range of applications from gas storage and separation applications. MOFs are composed of at least ditopic organic linkers and one metal ion. Metal ions of MOFs include, but are not limited to, Li⁺, Na⁺, Rb⁺, Mg²⁺, Cu²⁺, Zn²⁺, Ba²⁺, Sc³⁺, Ti⁴⁺, Zr⁴⁺, Ta⁴⁺, Cr³⁺, Mo³⁺, W⁶⁺, Mn³⁺, Fe³⁺, Fe⁴⁺, Ru³⁺, Ru⁴⁺, Os³⁺, Os⁴⁺, Co³⁺, Co⁴⁺, Ni²⁺, Ni³⁺, Pd²⁺, Pd³⁺, Pt²⁺, Pt³⁺, Cu²⁺, Cu³⁺, Au¹⁺, Zn²⁺, Al³⁺, Ga³⁺, In³⁺, Si⁴⁺, Si⁵⁺, Ge⁴⁺, Ge⁵⁺, Sn⁴⁺, Sn⁵⁺, Bi³⁺, Bi⁴⁺, Cd²⁺, Mn²⁺, Tb³⁺, Gd³⁺, Ce⁴⁺, La⁵⁺ and Cr⁶⁺ and combinations thereof. MOFs (Metal Organic Framework) are porous materials with compelling capabilities for gas storage (FIG. 1). Their development has accelerated in the past decade [1-3] because of favorable performance characteristics as a result of their high surface area, porosity and stability [3-8]. Additionally, porous organic polymers (POPs), are porous materials made from only organic building units, have favorable performance characteristics as a result of their high surface area, porosity, extreme stability, and short range crystallinity.

SUMMARY

[0004] Embodiments of the invention are drawn to (a) the development of novel MOF structures for the storage of electronic gas at pressures below 760 torr, and (b) the integration of these structures into a gas delivery system, enabling a significant increase in delivered storage capacity over existing solutions. An embodiment relates to a metal organic framework (MOF) including the coordination product of a metal ion and an at least bidentate organic ligand, wherein the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas.

[0005] Another embodiment relates to a method of making a metal organic framework (MOF) including reacting a metal ion and an at least bidentate organic ligand, wherein the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas.

[0006] Another embodiment relates to a method of using a metal organic framework (MOF) comprising at least a plurality of organic monomers, including filling a cylinder with a MOF, charging the MOF-filled cylinder with an electronic gas at pressures below 760 torr, storing the electronic gas in the MOF-filled cylinder, and dispensing the electronic gas under vacuum. The MOF includes a deliverable adsorption capacity of at least 70 g/L for the electronic gas.

[0007] Further embodiments of the invention are drawn to (a) the development of novel porous organic polymers (POP) for the storage of electronic gases at pressures below 760 torr, and (b) the integration of these structures into a gas delivery system, enabling a significant increase in delivered storage capacity over existing solutions. An embodiment relates to a POP comprising the polymerization from at least a plurality of organic monomers and comprising at least a plurality of linked organic repeating units, wherein the linked organic repeating units are selected to provide a porous material and selected to provide a porous material with a deliverable adsorption capacity of at least 70 g/L for an electronic gas.

[0008] Another embodiment relates to a method of making a porous organic polymer (POP) including reacting a plurality of organic monomers, wherein the organic monomers are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas.

[0009] Another embodiment relates to a method of using a porous organic polymer (POP) comprising at least a plurality of organic monomers, including filling a cylinder with a POP, charging the POP-filled cylinder with an electronic gas at pressures below 760 torr, storing the electronic gas in the POP-filled cylinder, and dispensing the electronic gas under vacuum. The POP includes a deliverable adsorption capacity of at least 70 g/L for the electronic gas.

BRIEF DESCRIPTION OF THE DRAWINGS

[0010] The patent of application file contains at least one drawing executed in color. Copies of this patent or patent application publication with color drawing(s) will be provided by the officie upon request and payment of the necessary fee.

[0011] FIGS. 1e-1f are ball and stick illustrations of metal organic frameworks including (a) CoMOF74, (b) NU-110, (c) NU-111 and (d) NU-125. FIGS. 1e-1k are illustrations of sorbent materials including the formation of metal organic frameworks (e) ZnBDCDAABCO (pillared-paddlewheel framework), (f) Cu-BTC (paddlewheel framework), (g) NU-125 (rht framework), (h) Zn2O-BDC (Zn2O framework), (i) Zr-BDC (ZrO2 or ZrO2 framework), (j) ZIF-8 (heterocycle framework) and the formation of porous organic polymers (k) PAF-1 (tetrahedral POP framework).

[0012] FIG. 2a is a schematic illustration of a method of self-assembly of a MOF according to an embodiment.

[0013] FIG. 2b is a micrograph of a MOF according to an embodiment.

[0014] FIG. 2c is a plot comparing simulated and experimental hydrogen adsorption as a function of pressure of an NU-100 MOF.

[0015] FIGS. 3a, 3c, 3e, 3g and 3i are plots of simulated deliverable capacity in g/L as a function of pore size including (a) AsH3, (c) PdH3, (e) SnH4, (g) BiH3 and (i) BF3 for a large library of MOFs.

[0016] FIGS. 3b, 3d, 3f, 3h and 3j are plots of simulated deliverable capacity in g/L as a function of gravimetric
surface area including (b) AsH₃, (d) PH₃, (f) SbH₃, (h) B₂H₆, and (j) BF₃ for a large library of MOFs.

FIG. 4a illustrates the chemical formulas of protonated precursors of linkers used to construct a MOF series.

FIG. 4b is a simulated crystal structure of MOF UiO68.

FIG. 5a illustrates the chemical formulas of hexa-protonated precursors of the linker used to construct the isostructural copper-based rht MOF series.

FIG. 5b is a simulated crystal structure of a copper-based rht MOF with a 1.1 precursor.

FIGS. 6a-6c illustrate the simulated delivery capacity of arsenic as a function of (a) pore size, (b) pore volume and (c) gravimetric surface area for a series of Zr6O52 framework MOFs.

FIGS. 7a-7c illustrate the simulated delivery capacity of arsenic as a function of (a) pore size, (b) pore volume and (c) gravimetric surface area for a series of rht framework MOFs.

FIGS. 8a-8c illustrate MOF or POP based gas storage and delivery system including (a) a pellet filled tank, (b) a disk filled tank and (c) a monolithic filled tank.

FIGS. 9a-9c are plots illustrating the storage capacity of electronic gases in various adsorbents as a function of pressure (a) BF₃ in Cu-BTC, (b) AsH₃ in ZnBDCDABCO, and (c) GeF₄ in Cu-BTC.

DETAILED DESCRIPTION

Conventionally, lower-performing porous materials, such as zeolites and activated carbon, have been used for the storage of electronic gases. While significant research has explored the use of MOFs for the storage of commodity gases such as methane or hydrogen, little attention has been paid to the use of MOFs as a commercially viable adsorption solution for electronic gases used in the semiconductor industry. As used herein, “electronic gases” are source gases used for fabrication of solid state devices, such as transistors, diodes, light emitting diodes, lasers, solar cells, capacitors, etc. The electronic gases may be used for doping (e.g., as an ion implantation source gases or chemical vapor deposition (CVD) or atomic layer deposition (ALD) gas phase dopant sources) or layer deposition (e.g., such as CVD or ALD source gases) of semiconductor (e.g., Group IV, III-V, II-VI or other compound semiconductors), insulators (e.g., silicon oxide, silicon nitride, etc.) and conductors (e.g., tungsten, etc.) in solid state devices.

MOF-based adsorbents may be used to store electronic gases in high concentrations between 0-250 bar, which enables the optimization of storage capacity and safety trade-offs for different gases. While activated carbon has been used for sub-atmospheric storage of these gases, their concentrations in the cylinders is low due to the weak binding of the electronic gases, such as arsenic and phosphine, to the pores of this adsorbent. At similar pressure regimes, MOF and POPs adsorbents exhibit improved capacity and gas densities when compared to activated carbon. Thus, the use of MOFs and POPs provides improved electronic gas storage methods and delivery systems. This performance improvement significantly reduces storage device costs and promotes safety as a result of reduced cylinder change-outs and worker contact.

The electronic gas industry uses heavy metallic cylinders to store gases, which results in significant compression, storage, handling, and delivery costs. Chemical, semiconductor manufacturers and nanomaterials fabrication facilities purchase a wide range of industrial and electronic gases for use in production and manufacturing equipment.

Historically, ion implantation has been considered a hazardous process in a semiconductor fabrication facility. For example, electronic gases used in implantation or vapor deposition, such as arsine, are fatal at concentrations over 25 parts per million. Phosphine is fatal at concentrations over 50 parts per million. As a result, ion implanters have been isolated to minimize potential exposure to the toxic gases used during implant. High-pressure cylinders located in a confined area inside the implanter have presented a risk for semiconductor manufacturers. In response, the semiconductor industry has learned to reduce the risk of leaks through the use of source isolation, ventilation, gas-detector technology, improved gas delivery components and electronic treatment systems to prevent discharges-to-atmosphere above allowable limits and through extensive personnel training. While the risk of using highly toxic gases for ion implantation is reduced through the use of embedded mechanical controls to mitigate cylinder pressure, and by removing pressure from the gas delivery system, both require a vacuum to be in place before gas is delivered from the cylinder to a semiconductor manufacturing tool.

While high compression cylinders are still in use today, sub-atmospheric pressure gas sources (SAGs) were developed to enable semiconductor manufacturers to overcome the risk to workers posed by high-pressure toxic gases. These SAGs satisfied health and safety requirements and regulations. This adsorbent technology stores toxic electronic gases below atmospheric pressure, removing the concern of catastrophic releases of high-pressure toxic electronic gases. More specifically, the SAGs include a gas source package that stores and delivers gas at sub-atmospheric pressures. This package includes a gas cylinder and outlet valve, operated by reversibly adsorbing electronic gases (such as arsine, phosphine, boron trifluoride and germanium tetrafluoride) with a relatively high surface area matrix. The adsorbed gas is in a lower energy state and exhibits a significant vapor-pressure reduction. The adsorbent loading (the saturation of the gas into the adsorbent) is 30-50% by weight and the pressure is 650 torr at 25 degrees Celsius. A vacuum provides the motive force to displace the gas/solid equilibrium and then convey the gas to the point of use.

However, a major limitation of SAGs is that the adsorbents, comprised primarily of activated carbon, have limited surface areas. Activated carbon adsorbents have amorphous (random shapes and sizes) pores, in which some pore spaces are too tight, some are too large and others cannot be accessed. Thus, activated carbon has inherent limitations and there is a need for porous materials with more uniform well-defined pore size, pore accessibility, and greater storage capacity such sorbents include MOFs and/or POPs. Additionally, to maximize storage of toxic electronic gases in current SAG cylinders, activated carbon sorbents are pressed into large disk-shaped monoliths. As a result, cylinders need to be built around the sorbents using a two-step welding process, adding significant complexity and system cost.

MOFs have the highest surface area per gram (believed to be up to 14,500 m²/g) in comparison to any known adsorbent materials. In fact, MOFs have an internal
Surface area that significantly exceeds activated carbon,[10, 11] and MOFs have greater design flexibility in comparison to zeolites. MOF adsorbents have significantly higher surface area and a higher effective storage capacity than the currently used activated carbon materials used in SAGs. MOFs enable safer gas packaging at improved economics by reducing the number of cylinder change-outs required. This results in a meaningful safety benefit as workers limit their interaction with cylinders, and also provides a cost benefit in the form of reduced machine down time as higher capacity cylinders require less change-outs.

MOF adsorbents can be used in portable gas cylinders to allow for gas storage at significantly lower pressures. Thus, the use of MOFs will result in (1) reduced filling pressures, (2) lowered bill-of-materials through use of off-the-shelf storage containers as compared to welded cylinders for disk monoliths (3) reduced handling and delivery costs (4) improved worker safety, and (5) reduced machine downtime as a result of reduced cylinder change-outs. Further, MOFs can be used to significantly increase storage capacity of high value electronic gases that currently suffer from poor storage economics using high-pressure, or lower performing sorbents such as activated carbon or zeolites.

An embodiment of the present invention includes computationally designed and synthesized MOF’s for the storage of electronic gases. An advantage of MOFs is that the molecular building blocks of MOF’s, organic linkers and metal ions, self-assemble in a predictable way into uniform crystals (see FIG. 2a). This advantage can be leveraged to rapidly search and screen for optimal materials in silico through use of a computational MOF generator using few organic building blocks which ultimately builds a large library of MOF’s.

FIG. 2a illustrates how MOF’s self-assemble from molecular “building blocks” into predictable structures. After designing a MOF on a computer, as in the case of NU-100, the MOF may be synthesized, as shown in FIG. 2b, which is a micrograph of a NU-100 crystal taken with an optical microscope. Subsequent measurements of hydrogen storage capacity in the NU-100 MOF were in excellent agreement with simulations as illustrated in the plot of adsorbed hydrogen as a function of pressure illustrated in FIG. 2c.

Simulation results indicate that MOF’s with gravimetric surface areas ranging between 1,000-14,500 m²/g and pore sizes ranging from 2-25 Å between an operating pressure range of 0-760 torr may have favorable performance characteristics for the storage of the electronic gases, including, but not limited to the list shown in Table 1.

<table>
<thead>
<tr>
<th>Electronic gases [12]</th>
</tr>
</thead>
<tbody>
<tr>
<td>Ammonia</td>
</tr>
<tr>
<td>Boron trifluoride</td>
</tr>
<tr>
<td>Chlorine</td>
</tr>
<tr>
<td>Deuteromethane</td>
</tr>
<tr>
<td>Fluorine</td>
</tr>
<tr>
<td>Tetrafluoromethane</td>
</tr>
<tr>
<td>Methyl fluoride</td>
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<tr>
<td>Hydrogen</td>
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<tr>
<td>Hydrogen Chloride</td>
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<tr>
<td>Krypton</td>
</tr>
<tr>
<td>Neon</td>
</tr>
<tr>
<td>Nitrogen</td>
</tr>
<tr>
<td>Silane</td>
</tr>
<tr>
<td>Stibine</td>
</tr>
<tr>
<td>Silicon (SiH₄) (e.g., Silicon)</td>
</tr>
<tr>
<td>Tungsten</td>
</tr>
<tr>
<td>Hexafluoride</td>
</tr>
<tr>
<td>Acetylene</td>
</tr>
<tr>
<td>Trichlorosilane</td>
</tr>
</tbody>
</table>

The deliverable adsorption capacity, measured between 5 Torr (final discharging pressure) and 650 Torr (final charging pressure), of a library of possible MOFs obtained from a MOF computer simulation, was predicted using Grand Canonical Monte Carlo (GCMC) simulations. FIGS. 3a-3d illustrate how many MOFs, given their ranging pore sizes and gravimetric surface areas, have a 2-3 fold deliverable capacity increase for both AsH₃ and PH₃ relative to that of activated carbon. In FIGS. 3a-3f, each point signifies the deliverable capacity of one MOF derived from the MOF generator. Furthermore, the deliverable capacities for stibine, diborane, and boron trifluoride were also predicted using similar methods (FIGS. 3e-3j). An embodiment includes a metal organic framework (MOF) comprising the coordination product of a metal ion and an at least bidentate organic ligand, wherein the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/l, such as 190 g/l for an electronic gas. In an embodiment, the deliverable adsorption capacity is 70 to 840 g/l for the electronic gas.

Another embodiment includes the method of using a MOF comprised of a metal ion and an at least bidentate organic ligand where the MOF is used as an adsorbent in a cylinder to store and deliver electronic gases. The MOF has a storage capacity of at least 70 g/L and at most 840 g/L measured at 650 torr and 25° C.

In an attempt to gain more detailed insights, the direct effect of the linker length on the surface area, pore size, and pore volume were computationally studied while relating them with arsenic deliverable adsorption capacities. Two series of MOFs were investigated: MOFs with zirconium metal nodes (FIGS. 4a-4b, UIO type MOFs) and copper-based MOFs with the rht topology (FIGS. 5a-5b, L1-L6 type MOFs). In the case of the zirconium-based MOFs, both known and hypothetically simulated MOFs were used to identify optimal pore sizes, pore volumes, and gravimetric surface areas for maximum deliverable adsorption capacity of arsenic. For the rht topology series of MOFs, the MOF’s studied herein are previously known from literature references. Combined, the optimal parameters of both families of MOFs investigated herein illustrate common
parameters which provide high arsine adsorption and these findings can be expanded to many other MOF families. In both series of MOFs, the ability to tune pore volumes and pore sizes, in addition to surface area, illustrate the ability to control and maximize arsine adsorption. These findings were clearly evident in the UiO series of MOFs (FIGS. 6a-c) as deliverable capacity of arsine increases with increasing pore sizes, pore volumes, and gravimetric surface areas starting with UiO66 (70 g/L, 8 and 10 Å, 0.13 cm²/g, and 1680 m²/g respectively) until the arsine deliverable capacity reached a maximum in UiO68 (775 g/L, 15 and 8 Å, 1.1 cm²/g, and 4300 m²/g respectively). However beyond UiO68, the deliverable capacity of arsine drastically decreases with further increasing pore sizes and pore volumes which is illustrated by UiO48h (120 g/L, 18 and 24 Å, and 1.9 cm²/g respectively). An embodiment includes a MOF which has a first type of pore having an average pore size between 15 Å and 25 Å. In another embodiment, the MOF further includes a second type of pore, the second type of pore having a smaller pore size between 8 Å and 15 Å. In an embodiment, the MOF has a gravimetric surface area between 1,000 and 14,500 m²/g and a pore volume between 1 and 3 cm³/g. In an embodiment, the metal ion comprises Zr⁶⁺ and the at least bidentate organic ligand is selected from UiOace2ph and UiOaoz.

Gravimetric surface does not appear to be as dominant a factor as either pore sizes or pore volumes in the deliverable capacity of arsine. This is illustrated in the rth series of analyzed MOFs (FIGS. 7a-c). This family of MOFs have similar gravimetric surface areas ranging from 4000 to 5500 m²/g in MOFs based on L1 to L6 ligand precursors. Although they have similar surface areas, the deliverable capacity of arsine increases with increasing pore sizes and pore volumes reaching a maximum in L4 (840 g/L, 23 Å, and 2.8 cm³/g respectively). However, beyond the optimal parameters of L4, the deliverable capacity of arsine decreases significantly as the pore size and pore volumes increase as shown by L6 (70 g/L, 27 Å, and 3.3 cm³/g respectively). An embodiment includes a MOF in which the metal ion comprises Cu²⁺ and the at least bidentate organic ligand is selected from rth ligands of precursors L1, L2, L3, L4 and L5.

A selection of MOFs with different types of frameworks (FIGS. 1e-f) were synthesized, i.e. pillared-paddlewheel (ZnBDCDABCO), paddlewheel (Cu-BTC), rth (NU-125), ZnO₆ (MOF-5), ZrO₂ (UiO-66), and heterocycle frameworks (ZIF-8). Additionally, a porous organic polymer (POP) was also synthesized (FIG. 16). These materials were filled with electronic gases at pressures below 760 torr and the capacities were measured and compared to the modeled simulations. For example, the experimental capacity of BF₃ in Cu-BTC measured at 650 torr and 25 °C was found to be 440 milligrams of BF₃ per gram of adsorbent (i.e. 440 mg/g) and this result showed very good agreement with the simulated BF₃ isotherm (FIG. 9a). This agreement was further observed for the experimentally measured sub-atmospheric storage of AsH₃ in ZnBDCDABCO (600 mg/g measured at 650 torr and 25 °C) as shown in FIG. 9b.

An embodiment includes using MOFs as sorbents wherein the MOF causes an increase in density for the adsorbed electronic gas measured at 650 torr and 25 °C. More specifically, the MOF has a fill density for arsine of 0.33 to 3.8 grams of arsine per gram of MOF (i.e. g/g) and 172 to 850 grams of arsine per liter of volume (i.e. g/L). Moreover, the embodiment includes using MOF as an adsorbent in cylinder with a fill density for boron trifluoride of 0.35 to 3.5 g/g and a fill density of 150 to 600 g/L. Furthermore, the embodiment includes a MOF with a fill density for phosphine of 0.17 to 1.7 g/g and 70 to 400 g/L. The embodiment includes MOFs able to store and deliver germanium tetrafluoride having a fill density of 0.8 to 8.0 g/g and 400 to 2,000 g/L. The embodiment includes MOFs with storage capacities for a variety of different gases used to manufacture semiconductors and other electronic components selected from a group consisting of hydride gases, halide gases, and organometallic Group V gaseous compounds including, but not limited to, ammonia, arsine, boron trichloride, boron trifluoride, carbonyl sulfide, chlorine, deuterium, diborane, dichlorosilane, dichlorosilane, difluoromethane, disilane, fluorsilane, germane, germanium tetrafluoride, hexafluoroethane, hydrogen bromide, hydrogen chloride, hydrogen fluoride, hydrogen selenide, hydrogen telluride, hydrogen sulfide, methyl fluoride, methyl silane, neon, nitric organic, nitrogen trifluoride, perfluoropropane, phosphine, silane, silicon tetrafluoride, tetrafluoromethane, tetr methylsilane, silicon tetrafluoride, stibine, sulfur hexafluoride, trichlorosilane, trifluoromethane, trimethylsilane, tungsten hexafluoride, acetylene, and organometallic gaseous reagents.

The embodiment also includes using MOFs as sorbents for electronic gas wherein the metal ion(s) used to form the MOF are selected from Li⁺, Na⁺, K⁺, Rb⁺, Cs⁺, Mg²⁺, Ca²⁺, Sr²⁺, Ba²⁺, Sc³⁺, Y³⁺, Ti⁴⁺, Zr⁴⁺, Hf⁴⁺, V⁵⁺, V⁷⁺, Nb⁵⁺, Ta⁵⁺, Cr³⁺, Ce³⁺, Mo⁶⁺, W⁶⁺, Mn³⁺, Fe³⁺, Fe⁵⁺, Ru³⁺, Ru⁵⁺, Os³⁺, Os⁵⁺, Co³⁺, Co⁵⁺, Ni²⁺, Ni⁴⁺, Pd²⁺, Pd⁴⁺, Pt²⁺, Pt⁴⁺, Cu²⁺, cu⁺, Ag⁺, Au⁺, Zn²⁺, Al³⁺, Ga³⁺, In³⁺, Si³⁺, Ge⁴⁺, Ge²⁺, Sn⁴⁺, Sn²⁺, Bi⁴⁺, Bi²⁺, Cd²⁺, Mn²⁺, Mn⁴⁺, Th³⁺, Gd³⁺, Ce³⁺, La³⁺ and Ce⁴⁺, and combinations thereof. The embodiment includes observations from the modeling simulations where the maximum storage capacity for electronic gases at below atmospheric pressure is found in a series of MOFs having pore size distributions of 3 to 32 Å (see FIGS. 3a, 3c, 3e, 3g and 3j). Additionally, the embodiment includes MOFs comprising gravimetric surface areas between 1,000 and 14,500 m²/g (see FIGS. 3b, 3d, 3f, 3h, and 3j for simulations comparing deliverable capacity and surface area). The embodiment includes MOFs having different physical forms. Typically MOFs are synthesized in powder form however this embodiment also includes forms such as powder, microcrystals, crystals, granules, pellets, spheres, and combinations of these forms wherein the physical form has an average diameter between 0.1 to 10.0 mm. The physical forms of the MOFs have various bulk densities after filled inside the cylinder. These bulk densities range from 0.2 to 2.5 g/L. Another embodiment also includes forming the powder into pellets, disks, or a monolithic body (as illustrated in FIGS. 8a to 8c).

The embodiment also includes using MOFs that are composed of different framework types including MOFs with open coordination sites allowing substrate to metal interactions or binding. Additionally the embodiment includes MOFs that have metal nodes where substrate-metal interactions are suppressed because the metals in the nodes are coordinatively saturated because of maximized interactions with the ligands in the framework. More specifically these types of MOF frameworks, included in the embodiment, are listed as pillared-paddlewheel (such as ZnBDCDABCO in FIG. 1e), paddlewheel (such as Cu-BTC in FIG.
Once the MOFs were filled and charged with the electronic gases, the purity of the storage gas was measured. In the case of ZnBDCDABCO MOF, PH₃ was filled to a pressure of 650 torr at 25 °C and the adsorbed PH₃ was desorbed and analyzed using a Residual Gas Analyzer (RGA) which measured trace gas impurities. The discharged PH₃ gas had 95 ppm of total impurities. Similarly, ZnBDCDABCO MOF was charged with AsH₃ to pressures below 760 torr. Under vacuum, the adsorbed AsH₃ was removed and the discharged gas had 176 ppm of impurities. Having a very high purity of discharged gas is often difficult with adsorbent materials given the highly reactive nature of AsH₃. AsH₃ will react rapidly with oxidizing species to produce hydrogen and other impurities that drastically increase the pressure of filled cylinder. The resulting high pressure disqualifies the sub-atmospheric safety benefits. Moreover, impurities above 2000 ppm are undesirable for ion implantation. In the present case, the ZnBDCDABCO MOF does not cause major decomposition given the nature of the metal node, wherein each zinc metal ion is fully saturated with organic ligands that tie the framework of the MOF together. The fully coordinatively saturated metal node prevents metal-arsine binding that ultimately could result in reduction/oxidation mechanisms that produce unwanted impurities.

Holding the zincBDCDABCO MOF with AsH₃ pressures below atmospheric pressure, the gas was nearly completely discharged under vacuum. Ultimately, greater than 95% of the adsorbed gas was fully removed from the MOF-based AsH₃ filled cylinder. For example, 96-99% may be fully removed. This is highly advantageous compared to SAGs, which often have a “heel” (gas that is unable to be extracted at discharge pressures of 5 torr) exceeding 10% of the total adsorbed gas in a given system.

Another embodiment includes a MOF wherein the electronic gas dispensed from the adsorbent contains less than 2,000 ppm of trace impurities including water, carbon dioxide, hydrogen, nitrogen, oxygen, and argon. As mentioned above, the low levels of impurities are a result of a lack of hydrolyzing species or oxidizing species within the MOF framework, also included in this embodiment. Additionally, an embodiment includes a MOF where the dispensed gas is greater than or equal to 50% of the total adsorption capacity.

Additionally, a porous organic polymer (POP) was synthesized from a tetrahedral monomer (FIG. 1k where R = H and M = C) and this POP was measured for AsH₃ capacity. This material, named PAF-1, stored 0.60 g/g of AsH₃ measured at 650 torr at 25 °C. After charging the cylinder with AsH₃, the gas was removed under vacuum and the impurity content was found to be 168 ppm. The relatively high arsine capacity for these POP frameworks composed of tetrahedral monomers is due to a combination of high porosity of the parent materials (surface areas ranging from 1,880 to 7,100 m²/g, pore volumes ranging from 0.7 to 3.0 cc/g) and the short-range crystallinity of the diamondoid-like networks. The well-defined short-crystallinity of these diamondoid is in contrast to the ill-defined nature of typical activated carbon sorbents.
prevent the loss of deliverable capacity due to heat generation or heat loss. The MOF or POP-filled tank may be incorporated into gas delivery system that can include, but is not limited to, a pressure regulator, pressure transducer, filter, mass flow controller, valve, pipe or other structures. An embodiment includes a method of using a metal organic framework (MOF) comprising a metal ion and an at least bidentate organic ligand.

[0053] Any suitable tank may be used, such as a high pressure tank, an atmospheric tank or sub-atmospheric gas storage tank, such as the SDS® or SAGE® brand sub-atmospheric gas storage cylinders from ATMI, Inc. may be used.

[0054] A further embodiment includes the manufacturing or method of making of MOFs or POPs for the purpose of storing and delivering electronic gases with deliverable storage capacities of at least 70 g/L and ranging between 70 to 840 g/L. A further embodiment includes the manufacturing of MOFs or POPs where the list of electronic gases stored and delivered are selected from a group consisting of hydride gases, halide gases, and organometallic Group V gaseous compounds including, but not limited to ammonia, arsine, boron trichloride, boron trifluoride, carbonyl sulfide, chlorine, deuterium, diborane, dihydrogen disilane, dichlorosilane, difluoromethane, disilane, fluorine, germane, germanium tetrafluoride, hexafluoroethane, hydrocarbon monoxide, hydrogen bromide, hydrogen chloride, hydrogen fluoride, hydrogen selenide, hydrogen telluride, hydrogen sulfide, methyl fluoride, and silane, neon, nitric organic, nitrogen trifluoride, perfluoropropane, phosphine, silane, silicon tetrachloride, tetrafluoroethane, tetramethylsilane, silicon tetrafluoride, stibine, sulfur hexafluoride, trichlorosilane, trifluoromethane, trimethylsilane, tungsten hexafluoride, acetylene, and organometallic gaseous reagents.

[0055] Furthermore an embodiment includes the method of making MOFs to store and deliver electronic gases wherein the MOF is comprised of metal ions selected from selected from Li⁺, Na⁺, K⁺, Rb⁺, Cs⁺, Mg²⁺, Ca²⁺, Sr²⁺, Ba²⁺, Sc³⁺, Ti⁴⁺, Zr⁴⁺, Hf⁴⁺, V⁵⁺, Nb⁵⁺, Ta⁵⁺, Cr³⁺, Cr⁴⁺, Mo⁶⁺, W⁶⁺, Mn⁷⁺, Fe⁷⁺, Ru⁷⁺, O₆⁺, Os⁶⁺, Co⁷⁺, Np⁷⁺, Pu⁷⁺, Pa⁷⁺, Pr⁷⁺, P₃⁺, Pu⁷⁺, Cm⁷⁺, An⁷⁺, An³⁺, Bi³⁺, Bi⁵⁺, Cd²⁺, Cu²⁺, Tb³⁺, Ce³⁺, La³⁺ and Cr⁴⁺, and combinations thereof. A further embodiment includes the method of making MOFs with the purpose of storing and delivering electronic gases where the MOF is characterized by a surface area between 1,000 and 14,500 m²/g, a pore volume between 1 and 3 cc/g, and pore sizes ranging between 3 and 32 A.

[0056] More specifically, another embodiment includes making MOFs to store electronic gases at pressures below ambient wherein the MOFs are synthesized from metal ions and precursor organic building blocks: Zr⁴⁺ ions and terephthalic acid (as found in FIG. 1f), Zr⁴⁺ and diphenyl dicarboxylic acid (as shown in UIO67 in FIG. 4a), and Cu²⁺ ions and trimesic acid. Further embodiments include making MOFs for storing and delivering arsine or boron trifluoride where the MOF is synthesized from organic ligands comprising terephthalic acid and/or DABCO (also named 1,4-diazabicyclo[2.2.2]octane). Another embodiment includes MOFs to store and deliver an electronic gas other than arsine, phosphine, boron trifluoride, or germanium tetrafluoride.

[0057] A further embodiment includes taking a gas storage device such as a cylinder, tanks, or gas dispensing apparatus and filling the device with a sorbent material, either a MOF or a POP, and the sorbent material meets criteria via supra. The embodiment includes the storage and delivery apparatus wherein the sorbent material, either a MOF or a POP, is formed into small pellets or granules, larger disks with a shape similar to a hockey puck, or even larger monolith shapes that encompasses near the entire size of the vessel.

[0058] The following references teach aspects of the fabrication of MOFs and are hereby incorporated by reference:


[0072] Although the foregoing refers to particular preferred embodiments, it will be understood that the invention is not so limited. It will occur to those of ordinary skill in the art that various modifications may be made to the disclosed embodiments and that such modifications are intended to be within the scope of the invention. All of the publications,
What is claimed is:

1. A metal organic framework (MOF) comprising the coordination product of a metal ion and an at least bidentate organic ligand, wherein the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas, and wherein the electronic gas is selected from a group consisting of hydride gases, halide gases and organometallic Group V gaseous compounds.

2. The MOF of claim 1, wherein the metal ion and the organic ligand are selected to provide the deliverable adsorption capacity of at least 190 g/L for an electronic gas measured at 650 torr and 25°C.

3. The MOF claim 2, wherein the deliverable adsorption capacity is 190 to 840 g/L for the electronic gas.

4. The MOF of claim 1, wherein:
   - the electronic gas is selected from a group consisting of ammonia, arsine, boron trichloride, boron trifluoride, diphosphorus, dichlorosilane, difluoromethane, disilane, fluorine, germanium, germanium tetrafluoride, hexafluoroethane, hydrogen bromide, hydrogen chloride, hydrogen fluoride, hydrogen sulfide, methyl fluoride, methyl silane, nitrogen trifluoride, perfluoropropane, phosphine, silicon, silicon tetrachloride, tetrafluoromethane, tetramethylsilane, silicon tetrachloride, stibine, sulfur hexafluoride, trichlorosilane, trifluoromethane, trimethylsilane, tungsten hexafluoride and acetylene; and
   - the metal ion is selected from Li+, Na+, K+, Rb+, Cs+, Mg2+, Ca2+, Sr2+, Ba2+, Sc3+, Y3+, Ti4+, Zr4+, Hf4+, V5+, V4+, V3+, Nb5+, Ta5+, Cr5+, Cr4+, Mo3+, W3+, Mn3+, Fe3+, Fe2+, Ru3+, Ru2+, Os3+, Os2+, Co3+, Co2+, Ni2+, Pd2+, Pd+, Pt2+, Pt+, Cu4+, Cu+, Ag+, Au+, Zn2+, Al3+, Ga3+, In3+, Si4+, Si2+, Ge4+, Ge2+, Sn4+, Sn2+, Bi5+, Bi3+, Cd2+, Mn2+, Tb3+, Gd3+, Ce3+, La3+ and Cr4+, and combinations thereof.

5. The MOF of claim 4, wherein the electronic gas is selected from a group consisting of arsenic, phosphine, diborane and boron trifluoride.

6. The MOF of claim 5, wherein the MOF has a fill density for the arsenic measured at 25°C and 650 torr that is greater than 172 g/L and less than 850 g/L.

7. The MOF of claim 5, wherein the MOF has a fill density for the boron trifluoride measured at 25°C and 650 torr that is greater than 150 g/L and less than 600 g/L.

8. The MOF of claim 5, wherein the MOF has a fill density for the phosphine measured at 25°C and 650 torr that is greater than 70 g/L and less than 400 g/L.

9. The MOF of claim 1, wherein the MOF is formed into pellets, disks or a monolithic body.

10. The MOF of claim 1, wherein the MOF is in powder form.

11. The MOF of claim 1, wherein the MOF comprises a pore size distribution between 3 Å and 32 Å.

12. The MOF of claim 11, wherein the MOF comprises a gravimetric surface area between 1,000 and 14,500 m²/g and a pore volume between 0.3 and 3 cm³/g.

13. A gas storage and dispensing apparatus comprising a tank filled with the MOF of claim 1.

14. The apparatus of claim 13, further comprising the electronic gas adsorbed on the MOF.

15. A metal organic framework (MOF) comprising the coordination product of a metal ion and an at least bidentate organic ligand, wherein the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas, and wherein the MOF is formed into pellets, disks or a monolithic body.

16. The MOF claim 15, wherein the deliverable adsorption capacity is 190 to 840 g/L for the electronic gas measured at 650 torr and 25°C.

17. The MOF of claim 15, wherein:
   - the electronic gas is selected from a group consisting of ammonia, arsine, boron trichloride, boron trifluoride, diphosphorus, dichlorosilane, difluoromethane, disilane, fluorine, germanium, germanium tetrafluoride, hexafluoroethane, hydrogen bromide, hydrogen chloride, hydrogen fluoride, hydrogen sulfide, methyl fluoride, methyl silane, nitrogen trifluoride, perfluoropropane, phosphine, silicon, silicon tetrachloride, tetrafluoromethane, tetramethylsilane, silicon tetrachloride, stibine, sulfur hexafluoride, trichlorosilane, trifluoromethane, trimethylsilane, tungsten hexafluoride and acetylene; and
   - the metal ion is selected from Li+, Na+, K+, Rb+, Cs+, Mg2+, Ca2+, Sr2+, Ba2+, Sc3+, Y3+, Ti4+, Zr4+, Hf4+, V5+, V4+, V3+, Nb5+, Ta5+, Cr5+, Cr4+, Mo3+, W3+, Mn3+, Fe3+, Fe2+, Ru3+, Ru2+, Os3+, Os2+, Co3+, Co2+, Ni2+, Pd2+, Pd+, Pt2+, Pt+, Cu4+, Cu+, Ag+, Au+, Zn2+, Al3+, Ga3+, In3+, Si4+, Si2+, Ge4+, Ge2+, Sn4+, Sn2+, Bi5+, Bi3+, Cd2+, Mn2+, Tb3+, Gd3+, Ce3+, La3+ and Cr4+, and combinations thereof.

18. The MOF of claim 17, wherein the electronic gas is selected from a group consisting of arsine, phosphine, diborane and boron trifluoride.

19. The MOF of claim 18, wherein the MOF comprises a pore size distribution between 3 Å and 32 Å, a gravimetric surface area between 1,000 and 14,500 m²/g and a pore volume between 0.3 and 3 cm³/g.

20. A gas storage and dispensing apparatus comprising a tank filled with the MOF of claim 15.

21. A metal organic framework (MOF) comprising the coordination product of a metal ion and an at least bidentate organic ligand, wherein:
   - the metal ion and the organic ligand are selected to provide a deliverable adsorption capacity of at least 70 g/L for an electronic gas; and
   - the MOF comprises a pore size distribution between 3 Å and 32 Å, a gravimetric surface area between 1,000 and 14,500 m²/g and a pore volume between 0.3 and 3 cm³/g.

22. The MOF claim 21, wherein the deliverable adsorption capacity is 190 to 840 g/L for the electronic gas measured at 650 torr and 25°C.

23. The MOF of claim 21, wherein:
   - the electronic gas is selected from a group consisting of ammonia, arsine, boron trichloride, boron trifluoride, diphosphorus, dichlorosilane, difluoromethane, disilane, fluorine, germane, germanium tetrafluoride, hexafluoroethane, hydrogen bromide, hydrogen chloride, hydrogen fluoride, hydrogen sulfide, methyl fluoride, methyl silane, nitrogen trifluoride, perfluoropropane, phosphine, silicon, silicon tetrachloride, tetrafluoromethane, tetramethylsilane, silicon tetrachloride, stibine, sulfur...
hexafluoride, trichlorosilane, trifluoromethane, trimethylsilane, tungsten hexafluoride and acetylene; and the metal ion is selected from Li+, Na+, K+, Rb+, Be2+, Mg2+, Ca2+, Sr2+, Ba2+, Sc3+, Y3+, Ti4+, Zr4+, Hf4+, V5+, V4+, V3+, Nb3+, Ta3+, Cr3+, Cr2+, Mo3+, W3+, Mn3+, Fe3+, Fe2+, Ru3+, Ru2+, Os3+, Os2+, Co3+, Co2+, Ni2+, Ni+, Pd2+, Pd+, Pt2+, Pt+, Cu2+, Cu+, Ag+, Au+, Zn2+, Al3+, Ga3+, In3+, Si4+, Si2+, Ge4+, Ge2+, Sn4+, Sn2+, Bi5+, Bi3+, Cd2+, Mn2+, Tb3+, Gd3+, Ce3+, La3+ and Cr4+, and combinations thereof.

24. The MOF of claim 23, wherein the electronic gas is selected from a group consisting of arsine, phosphine, diborane and boron trifluoride.

25. The MOF of claim 21, wherein the MOF is formed into pellets, disks or a monolithic body.

26. The MOF of claim 21, wherein the MOF is in powder form.

27. A gas storage and dispensing apparatus comprising a tank filled with the MOF of claim 21.