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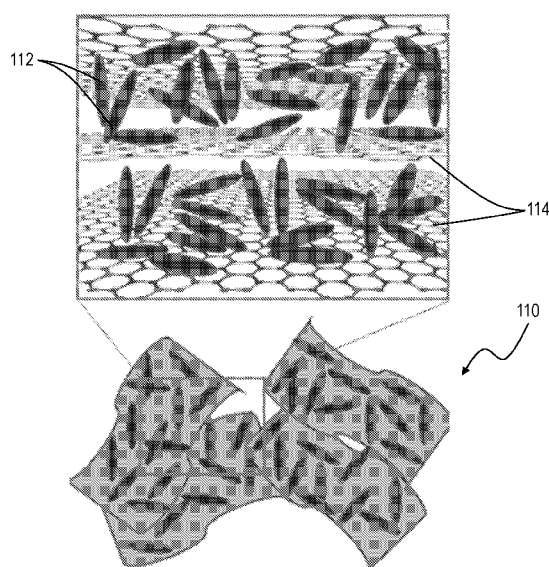


FIG. 2

(57) Abstract: An improved nanocomposite cathode material for lithium-ion batteries and method of making the same. The nanocomposite cathode material includes lithium iron silicate based nanoparticles with a conductive matrix of graphene sheets. The nanoparticles may be doped with at least one anion or cation.

Declarations under Rule 4.17:

- *as to applicant's entitlement to apply for and be granted a patent (Rule 4.17(ii))*
- *as to the applicant's entitlement to claim the priority of the earlier application (Rule 4.17(iii))*

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LITHIUM SILICATE CATHODES FOR LITHIUM-ION BATTERIES

CROSS REFERENCE TO RELATED APPLICATION

[0001] This application claims priority to U.S. Provisional Patent Application Serial No. 62/814,396, filed March 6, 2019, the disclosure of which is hereby expressly incorporated by reference herein in its entirety.

FIELD OF THE DISCLOSURE

[0002] The present disclosure relates generally to lithium-ion (Li-ion) batteries and, more particularly, to lithium iron silicate ($\text{Li}_2\text{FeSiO}_4$) based cathodes for Li-ion batteries.

BACKGROUND OF THE DISCLOSURE

[0003] Since the introduction of Li-ion batteries, the demand for increasingly higher specific capacity and specific energy batteries has steadily increased with the advance of portable electronics, electric vehicles (EVs), hybrid electric vehicles (HEVs) and the like. Likewise, the need for alternative fuel sources has grown over the last decades, due to such factors as the rise of oil prices, the increase in global population, and the pollution generated by internal combustion vehicles. As world population continues to grow, so will the number of vehicles and the demand for more efficient vehicles that require fewer natural resources and generate less pollution.

[0004] Advancement in battery technology has made the dream of replacing internal combustion engines with electric motors a reality, reducing the consumption of liquid hydrocarbon fuels. Implementation of battery-powered EVs still faces stiff opposition as they carry a higher cost, still have limited range, and suffer weight parity issues when compared to traditional internal combustion vehicles. Further, the batteries of choice, Li-ion batteries, suffer from short cycle lives and exhibit significant degradation over time, making battery powered vehicles less attractive.

[0005] The cathode materials of most Li-ion batteries include transition metal compounds, oxides, or complex oxides. Such transition metal compounds have layered crystal structures (e.g., lithium cobalt oxide (LiCoO_2)), spinel crystal structures (e.g., lithium manganese oxide (LiMn_2O_4)), or olivine crystal structures (e.g., lithium iron phosphate (LiFePO_4)). The transition metal cations typically display four- and/or six-fold coordination with oxygen anions, anionic clusters, or ligands, classified into the full octahedron, full tetrahedron as well as octahedron/tetrahedron hybrid structures.

[0006] In operation, Li^+ ions are inserted via an electrochemical intercalation reaction. While Li^+ ions occupy the space between adjacent layers or unoccupied octahedral or tetrahedral sites, an equal number of electrons enter the available d orbitals of the transition metal cations in the host crystal. Essentially, the oxidation state of metal ions keeps change with the deinsertion accompanying the phase change of these compounds while the Li^+ ions remain in ionic state. These materials have some common characteristics: (1) chemical stability, (2) structural stability, and (3) channels allowing the effective diffusion of Li^+ ions within the solid oxides. The chemical stability of the cathode material ensures that the host of the cathode does not decompose during the delithiation process while structural stability allows the repeated deintercalation of Li^+ ions into the lattices of the host materials. Channels within the materials lead to the high-rate delithiation process within the materials, which in turn is essential for the high rate performance of Li-ion batteries.

[0007] The performance characteristics of various cathode materials are summarized in **Table 1** below.

Table 1

Cathode Type		LiMn_2O_4	LiCoO_2	LiFePO_4	$\text{LiNi}_x\text{Mn}_y\text{Co}_z\text{O}_2$ (NMC)	$\text{Li}_2\text{FeSiO}_4$
Potential (V vs Li/Li^+)	Theoretical	4.2	4.0	3.5	3.7	3.5
	Practical	4.0	3.8	3.3	3.5	3.0
No. of Li^+ Intercalated	Theoretical	1	1	1	1	2
	Practical	1	0.5	1	0.5	2
Specific Capacity (mAh/g)	Theoretical	148	274	175	280	331
	Practical	120	145	165	170	305
Specific Energy (Wh/kg)	Theoretical	622	1096	613	1036	1158
	Practical	480	551	545	595	915
Relative Cost (\$/kg)		30	60	30	65	30

[0008] A particularly attractive cathode material for Li-ion batteries is $\text{Li}_2\text{FeSiO}_4$, which has a high specific capacity with two Li^+ ion insertion and a high specific energy (See **Table 1**). Additionally, $\text{Li}_2\text{FeSiO}_4$ is chemically stable, safe, nontoxic, and affordable (See **Table 1**). However, the practical applications of V_2O_5 have been limited due to: (1) low electrical conductivity due to its tetrahedron structure, (2) slow Li^+ ion diffusion, (3) structural instability during the repeated delithiation processes due to weaknesses in its tetrahedron structure, and (4) interfacial instability due to the high charging voltage for driving the Li^+ ions into the stable FeSiO_4 network. Efforts have been made to improve the performance of $\text{Li}_2\text{FeSiO}_4$, including porous nanostructures, cation doping in crystals, carbon coating or fabrication of carbonaceous matrices, incorporation of reduced graphene, conductive inorganic compounds, or organic polymers. Although these efforts have improved the performance of $\text{Li}_2\text{FeSiO}_4$ to a certain degree, these efforts have not significantly improved the specific capacity, structural stability, and cycle life to the level of practical application.

[0009] For these reasons, there is a need for a cathode material for Li-ion batteries that takes advantage of the benefits of $\text{Li}_2\text{FeSiO}_4$ while addressing its inherent drawbacks in a comprehensive manner.

SUMMARY

[0010] The present disclosure provides an improved nanocomposite cathode material for lithium-ion batteries and method of making the same. The nanocomposite cathode material includes $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles with a conductive matrix of graphene sheets. The nanoparticles may be doped with at least one anion or cation.

[0011] According to an embodiment of the present disclosure, an electrode material is provided including a conductive matrix comprising a plurality of graphene sheets, and a plurality of $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles coupled to the conductive matrix.

[0012] According to another embodiment of the present disclosure, a method is provided for making an electrode material, the method including: preparing a solution containing a lithium oxide, a silicon oxide, and iron; adding graphene oxide to the solution; heating the solution to

produce a plurality of $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles over the graphene oxide; and sintering the nanoparticles to reduce the graphene oxide to graphene and produce a nanocomposite material.

BRIEF DESCRIPTION OF THE DRAWINGS

[0013] The above-mentioned and other features and advantages of this disclosure, and the manner of attaining them, will become more apparent and will be better understood by reference to the following description of embodiments of the invention taken in conjunction with the accompanying drawings, wherein:

[0014] FIG. 1 is a schematic view of a Li-ion battery of the present disclosure;

[0015] FIG. 2 is a schematic view of an exemplary $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite material of the present disclosure;

[0016] FIG. 3 is a flow chart illustrating an exemplary method for preparing the $\text{Li}_2\text{FeSiO}_4$ -based/Graphene material;

[0017] FIGS. 4A-4C are transmission electron microscopy images of $\text{Li}_2\text{FeSiO}_4$ -based particles alone, with graphene, and with graphene and fluorine dopants, respectively, in accordance with the examples;

[0018] FIGS. 5A-5D graphically illustrate electrochemical test results for electrodes made from the $\text{Li}_2\text{FeSiO}_4$ -based particles of FIGS. 4A-4C, in accordance with the examples;

[0019] FIG. 6 graphically illustrates capacity data at different fluorine doping ratios in accordance with the examples;

[0020] FIG. 7 graphically illustrates cycle life data at different fluorine doping ratios in accordance with the examples;

[0021] FIGS. 8A-8C graphically illustrate electrochemical test results for electrodes made with graphene and fluorine dopants in accordance with the examples;

[0022] FIGS. 9A-9C graphically illustrate electrochemical test results for electrodes made with graphene and manganese dopants in accordance with the examples;

[0023] FIG. 10 graphically illustrates capacity data at different manganese doping ratios in accordance with the examples;

[0024] FIG. 11 graphically illustrates cycle life data at different niobium doping ratios in accordance with the examples; and

[0025] FIG. 12 graphically illustrates cycle life data at different titanium doping ratios in accordance with the examples.

[0026] Corresponding reference characters indicate corresponding parts throughout the several views. The exemplifications set out herein illustrate exemplary embodiments of the invention and such exemplifications are not to be construed as limiting the scope of the invention in any manner.

DETAILED DESCRIPTION

[0027] FIG. 1 provides a Li-ion battery 100 including an anode 102 coupled to an anode current collector 103, a cathode 104 coupled to a cathode current collector 105, and an electrolyte-filled separator 106. In one particular example, the anode 102 comprises Li metal or graphite and the electrolyte comprises a solution of LiPF_6 in a mixture of ethylene carbonate (EC) and ethyl methyl carbonate (EMC), but sulfone-based, ionic liquid-based, nitrile-based, or other electrolytes can also be used. The illustrative battery 100 is a pouch cell, but the battery 100 may also be a cylindrical cell, a coin cell, or a prismatic cell, for example. The battery 100 may be configured for use in a portable electronic device, an electric vehicle, an energy storage device, or other electronic devices.

[0028] FIG. 2 provides an improved $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 for cathode 104 of battery 100 (FIG. 1). Advantageously, the $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 may have an initial specific energy of 600 Wh/kg, may have a cycle life of at least 1,000 cycles, and may lack cobalt, nickel, and manganese.

[0029] As shown in FIG. 2, the $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 includes $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 (e.g., nanorods) formed upon a conductive matrix of graphene sheets 114. The graphene sheets 114 may improve the electrical

conductivity of the nanocomposite cathode material 110 compared to the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 alone. Additionally, the graphene sheets 114 may provide a structural matrix to anchor and stabilize the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 and reduce grain stress during charge/discharge cycling, leading to better cycle life. As shown in FIG. 2, each graphene sheet 114 includes a single-layer of graphene with sp^2 -bonded carbon atoms arranged in a honeycomb crystal structure and can be viewed as an individual atomic plane of a graphite structure. Each carbon atom in the graphene uses 3 of 4 valence band ($2s$, $2p$) electrons (which occupy the sp^2 orbitals) to form 3 covalent bonds with the neighboring carbon atoms in the same plane. Each carbon atom in the graphene contributes its fourth lone electron (occupying the p_z orbital) to form a delocalized electron system, a long-range π -conjugation system shared by all carbon atoms in the graphene plane. Such a long-range π -conjugation in graphene yields extraordinary electrical, mechanical, and thermal properties in the nanocomposite cathode material 110. The nanocomposite cathode material 110 exhibits improved intraparticle electronic conduction because of good electrical conductivity of graphene, and Li^+ ion diffusion is improved because diffusion length is shortened. Furthermore, the small grain size of the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 in the nanocomposite cathode material 110 reduces internal stress, leading to better structure stability and cycle life. According to an exemplary embodiment of the present disclosure, the nanocomposite cathode material 110 may comprise about 1 wt. % to about 10 wt. % graphene, more specifically about 1 wt. % to about 5 wt. % graphene, more specifically about 2 wt. % graphene. The graphene content should be sufficiently low to maintain the graphene as single sheets and avoid re-stacking. Additional information regarding the incorporation of graphene sheets 114 is disclosed in U.S. Publication No. 2015/0380732, the disclosure of which is expressly incorporated herein by reference in its entirety.

[0030] In certain embodiments, the graphene sheets 114 may be modified with one or more functional groups (e.g., $-\text{OH}$, $-\text{COOH}$, $-\text{NH}_2$). For example, the functional groups may be covalently grafted onto the surface of the graphene sheets 114 through diazonium salt via a diazonium reaction. The diazonium reaction-based functionalization may provide a simple and cost-effective way to transform the pure graphene sheets 114 into hierarchical and functional materials that can provide the desired properties (i.e. hydrophobicity, Li^+/e^- conductivity, Li^+ diffusivity, nanoparticle dispersion, local electric field, etc.) to enhance binding with the adjacent $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112.

[0031] The $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 may also include one or more optional dopants, including anion dopants (X) and/or cation dopants (Y). The dopants X, Y, may alter the crystalline structure and, consequently, the electrochemical performance of the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112. The types of dopants X, Y, the doping order, and the amount of dopants X, Y may be varied to achieve a desired specific capacity/energy and cycle life. According to an exemplary embodiment of the present disclosure, the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 may have an anion doping ratio of anion dopants X to oxygen (O) and/or a cation doping ratio of cation dopants Y to iron (Fe). The doping ratios may be about 30% or less by weight, more specifically about 2% by weight to about 20% by weight, and more specifically about 2% by weight to about 10% by weight.

[0032] Examples of suitable anion dopants X include halogen ions, such as fluorine ions (F^-), chlorine ions (Cl^-), and bromine ions (Br^-). Such anion dopants X may have a larger electronegativity than oxygen (O) and may reduce the formation of ligand holes during delithiation, which may stabilize the crystalline structure by reducing the extent of an $\text{O}^{2-} \rightarrow \text{O}_2^{2-}$ reaction during delithiation. Also, such anion dopants X may also have a high redox potential (e.g., 2.87 V for F), which may help to increase the discharge potential of the nanocomposite cathode material 110 in the Li-ion battery 100 (FIG. 1).

[0033] Examples of suitable cation dopants Y include titanium ions (Ti^{+4}), manganese ions (Mn^{+2}), copper ions (Cu^{+2}), terbium ions (Tb^{+3}), niobium ions (Nb^{+5}), and molybdenum ions (Mo^{+4}). Such cation dopants Y may improve the structure stability of the $\text{Li}_2\text{FeSiO}_4$ -based nanoparticles 112 by enhancing the coupling effect among the $\text{Li}_2\text{FeSiO}_4$ -based tetrahedra via strong *d*-orbital hybridization. In this way, the dopants Y may function like springs to contain the tetrahedra and prohibit structural fracture. Consequently, the dopants Y may achieve much improved cycle life.

[0034] The $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 may have the following Formula (F-I):



wherein:

X = anion dopant;

Y = cation dopant;

$$a \geq 0;$$

$$b = f(a)$$

$$c \geq 0; \text{ and}$$

$$d = f(c).$$

[0035] Examples of suitable $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode materials 110 are listed in **Table 2** below.

Table 2

Material	X	b	Y	d
$\text{Li}_2\text{FeSiO}_4/\text{Graphene}$	--	0	--	0
$\text{Li}_2\text{FeSiO}_{4-a/2}\text{F}_a/\text{Graphene}$	F	$a/2$	--	0
$\text{Li}_2\text{Fe}_{1-c}\text{Mn}_c\text{SiO}_4/\text{Graphene}$	--	0	Mn	c
$\text{Li}_2\text{Fe}_{1-c}\text{Cu}_c\text{SiO}_4/\text{Graphene}$	--	0	Cu	c
$\text{Li}_2\text{Fe}_{1-2.5c}\text{Nb}_c\text{SiO}_4/\text{Graphene}$	--	0	Nb	$2.5c$
$\text{Li}_2\text{Fe}_{1-2.5c}\text{Nb}_c\text{SiO}_{4-a/2}\text{F}_a/\text{Graphene}$	F	$a/2$	Nb	$2.5c$

[0036] FIG. 3 provides an exemplary method 200 for synthesizing the $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 (FIG. 2) and constructing the cathode 104 (FIG. 1). The method 200 may involve a hydrothermal synthesis process.

[0037] In step 202, a first solution is prepared including a lithium oxide, specifically lithium hydroxide (LiOH), and a silicon oxide, specifically silica (SiO_2), in a suitable solvent such as distilled water. The SiO_2 may be provided in the form of nanoparticles, also referred to herein as nano- SiO_2 .

[0038] In step 204, a second solution is prepared including an iron compound, specifically iron dichloride (FeCl_2), in a suitable solvent such as distilled water.

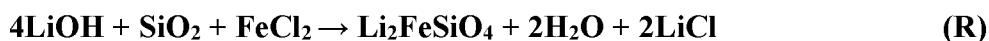
[0039] In step 206, the first and second solutions are combined. In certain embodiments, the second solution is added dropwise to the first solution. The combined solutions may be stirred together for about 30 minutes, about 1 hour, or longer before proceeding to the next step.

[0040] In step 208, a graphene oxide (GO) solution is added to the combined solution from step 206. The GO solution may be prepared using a modified Hummer's method, as disclosed in **Example 1** below, for example. In certain embodiments, the GO solution is added dropwise to the combined solution from step 206. The resulting solution may be stirred together for about 30 minutes, about 1 hour, or longer before proceeding to the next step.

[0041] In step 210, a salt of any desired dopant X, Y is added to the resulting solution from step 208. Examples of suitable salts include ammonium fluoride (NH₄F) for the F-dopant, cupric chloride (CuCl₂) for the Cu-dopant, and niobium hydroxide (Nb(OH)₅) for the Nb-dopant. The dopant X, Y may be present in a desired concentration relative to the Li₂FeSiO₄.

[0042] In step 212, the solution is reacted to produce optionally doped, Li₂FeSiO₄-based nanoparticles 112 (FIG. 2) over the GO. This reacting step 210 may be a hydrothermal step that is performed in an autoclave or another suitable heated environment. The temperature of the reacting step 210 may be about 160°C, about 180°C, about 200°C, or more, but this temperature may vary. The duration of the reacting step 210 may be about 5 hours, about 10 hours, about 15 hours, or more, but this duration may vary. After the reacting step 210 is completed, the autoclave may be allowed to return to room temperature before removing the products.

[0043] The reacting step 212 may produce Li₂FeSiO₄-based nanoparticles 112 upon the GO according to Reactions (R-I) thru (R-III), represented overall by Reaction (R). The Reaction (R) may be modified as appropriate to incorporate any desired dopants.



[0044] In step 214, the Li₂FeSiO₄-based nanoparticles 112 produced during the reacting step 212 are separated and cleaned. This step 214 may involve: removing the precipitated nanoparticles 112 from the aqueous solution, such as by filtering or drying; rinsing the nanoparticles 112 with deionized water or another suitable rinsing agent; and drying the nanoparticles 112, such as by subjecting the nanoparticles 112 to an elevated temperature and/or a vacuum environment for several hours or more.

[0045] In step 216, the Li₂FeSiO₄-based/Graphene nanocomposite cathode material 110 is formed by sintering the Li₂FeSiO₄-based nanoparticles 112 upon the GO, which reduces the GO to graphene 114 (FIG. 2). The sintering step 112 may be performed in an inert atmosphere, such as argon (Ar). The temperature of the sintering step 216 may be about 500°C, about 600°C, about 700°C, or more, but this temperature may vary. The duration of the sintering step 216 may be about 5 hours, about 10 hours, about 15 hours, or more, but this duration may vary.

[0046] In step 218, the cathode 104 (FIG. 1) is constructed using the $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110 from the sintering step 216. The constructing step 218 may involve preparing a slurry. In one example, the slurry contains about 80 wt. % $\text{Li}_2\text{FeSiO}_4$ -based/Graphene nanocomposite cathode material 110, about 10 wt. % polyvinylidene difluoride (PVDF), and about 10 wt. % carbon black. Next, the slurry may be sprayed onto or otherwise applied to the cathode current collector 105, such as a 10 μm thick aluminum (Al) foil. Then, the cathode 104 may be dried in a vacuum oven, such as at a temperature of about 90° C and a duration of about 24 hours.

[0047] While this invention has been described as having exemplary designs, the present invention can be further modified within the spirit and scope of this disclosure. This application is therefore intended to cover any variations, uses, or adaptations of the invention using its general principles. Further, this application is intended to cover such departures from the present disclosure as come within known or customary practice in the art to which this invention pertains and which fall within the limits of the appended claims.

EXAMPLES

1. Preparation of GO Solution

[0048] A GO solution was prepared using a modified Hummer's method. 2 grams of graphite flakes were mixed with 10 mL of concentrated H_2SO_4 , 2 grams of $(\text{NH}_4)_2\text{S}_2\text{O}_8$, and 2 grams of P_2O_5 . The obtained mixture was heated at 80° C for 4 hours under constant stirring. Then the mixture was filtered and washed thoroughly with DI water. After drying in an oven at 80° C overnight, this pre-oxidized graphite was then subjected to oxidation using the Hummer's method. 2 grams of pre-oxidized graphite, 1 gram of sodium nitrate and 46 mL of sulfuric acid were mixed and stirred for 15 minutes in an iced bath. Then, 6 grams of potassium permanganate was slowly added to the obtained suspension solution for another 15 minutes. After that, 92 mL DI water was slowly added to the suspension, while the temperature was kept constant at about 98° C for 15 minutes. After the suspension has been diluted by 280 mL DI water, 10 mL of 30% H_2O_2 was added to reduce the unreacted permanganate. Finally, the resulted suspension was centrifuged several times to remove the unreacted acids and salts. The purified GO were dispersed in DI water to form a 0.2 mg/mL solution by sonication for 1 hour. Then the GO

dispersion was subjected to another centrifugation in order to remove the un-exfoliated GO. The resulted GO dilute solution could remain in a very stable suspension without any precipitation for a few months.

2. Synthesis of $\text{Li}_2\text{FeSiO}_4$

[0049] First, 16 mmol LiOH and 4 mmol nano- SiO_2 were dissolved in 30 mL distilled water to produce Solution 2-A, and 4 mmol $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ was dissolved in another 20 mL distilled water to produce Solution 2-B. After 1 h stirring, the aqueous Solution 2-B was added dropwise in Solution 2-A with continued stirring for 4 h to produce Solution 2-C.

[0050] Then, Solution 2-C was transferred into a 100 mL Teflon-lined stainless-steel autoclave. After sealing, the autoclave was maintained at 180°C for 12 h to produce $\text{Li}_2\text{FeSiO}_4$.

[0051] Finally, when the reaction was completed, the autoclave was cooled to room temperature naturally. The precipitates were filtered and washed with DI water several times and finally dried at 60°C for 12 h in vacuum. The pure $\text{Li}_2\text{FeSiO}_4$ was sintered at 600°C for 10 h in argon (Ar) atmosphere.

[0052] Using transmission electron microscopy (TEM), the $\text{Li}_2\text{FeSiO}_4$ particles measured about 15 nm in diameter (FIG. 4A).

3. Synthesis of Undoped $\text{Li}_2\text{FeSiO}_4$ /Graphene

[0053] First, 16 mmol LiOH and 4 mmol nano- SiO_2 were dissolved in 30 mL distilled water to produce Solution 3-A, and 4 mmol $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ was dissolved in another 20 mL distilled water to produce Solution 3-B. After 1 h stirring, the aqueous Solution 3-B was added dropwise in Solution 3-A with continued stirring for 4 h to produce Solution 3-C.

[0054] Next, a 5 mg/mL GO gel was prepared according to **Example 1**. 20 mL of the GO gel was dropped in Solution 3-C with continued stirring for 1 h.

[0055] Then, the mixture was transferred into a 100 mL Teflon-lined stainless-steel autoclave. After sealing, the autoclave was maintained at 180°C for 12 h to produce $\text{Li}_2\text{FeSiO}_4$ upon the GO.

[0056] Finally, when the reaction was completed, the autoclave was cooled to room temperature naturally. The precipitates were filtered and washed with DI water several times and

finally dried at 60°C for 12 h in vacuum. The $\text{Li}_2\text{FeSiO}_4/\text{GO}$ was sintered at 600°C for 10 h in Ar atmosphere to reduce the GO to graphene.

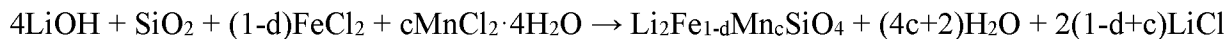
[0057] Using TEM, the undoped $\text{Li}_2\text{FeSiO}_4/\text{Graphene}$ particles appeared as nanorods and measured about 5 nm in diameter and about 10-30 nm in length (FIG. 4B).

4. Synthesis of 2% Mn-Doped $\text{Li}_2\text{Fe}_{0.98}\text{Mn}_{0.02}\text{SiO}_4/\text{Graphene}$

[0058] First, 16 mmol LiOH and 4 mmol nano- SiO_2 were dissolved in 30mL distilled water to produce Solution 4-A, and 3.92 mmol $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ and 0.08mmol $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$ as the Mn-dopant were dissolved in another 20mL distilled water to product Solution 4-B. After 1 h stirring, the aqueous Solution 4-B was added dropwise in Solution 4-A with continued stirring for 4 h to produce Solution 4-C.

[0059] Next, a 5 mg/mL GO gel was prepared according to **Example 1**. 20 mL of the GO gel was dropped in Solution 4-C with continued stirring for 1 h.

[0060] Then, the mixture was transferred into a 100 mL Teflon-lined stainless-steel autoclave. After sealing, the autoclave was maintained at 180°C for 12 h to produce $\text{Li}_2\text{Fe}_{0.98}\text{Mn}_{0.02}\text{SiO}_4$ upon the GO according to the following Mn-doped variation of Reaction (R):



wherein:

$$c = 0.02; \text{ and}$$

$$d = c = 0.02.$$

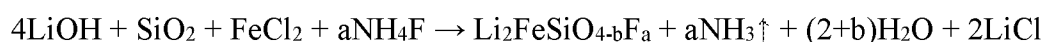
[0061] Finally, when the reaction was completed, the autoclave was cooled to room temperature naturally. The precipitates were filled and washed with DI water several times and finally dried at 60°C for 12 h in vacuum. The $\text{Li}_2\text{Fe}_{0.98}\text{Mn}_{0.02}\text{SiO}_4/\text{GO}$ was sintered at 600°C for 10 h in Ar atmosphere to reduce the GO to graphene. The doping ratio of Mn/Fe was 2% by weight, calculated as $(0.02 \cdot 54.938) / (0.98 \cdot 55.845)$.

5. Synthesis of 15% F-Doped $\text{Li}_2\text{FeSiO}_{3.76}\text{F}_{0.48}/\text{Graphene}$ and 6% F-Doped $\text{Li}_2\text{FeSiO}_{3.9}\text{F}_{0.2}/\text{Graphene}$

[0062] First, 16 mmol LiOH and 3.76 mmol nano-SiO₂ were dissolved in 30mL distilled water to produce Solution 5-A, and 4 mmol FeCl₂·4H₂O was dissolved in another 20mL distilled water to produce Solution 5-B. After 1 h stirring, the aqueous Solution 5-B was added dropwise in Solution 5-A with continued stirring for 4 h to produce Solution 5-C.

[0063] Next, a 5 mg/mL GO gel was prepared according to **Example 1**. 20 mL of the GO gel was dropped in Solution 5-C with continued stirring for 1 h. In addition to the GO, 0.48 mmol NH₄F as the F-dopant was added to Solution 5-C and stirred for 5 mins.

[0064] Then, the mixture was quickly transferred to a 100 mL Teflon-lined stainless-steel autoclave. After sealing, the autoclave was maintained at 180°C for 12 h to produce Li₂FeSiO_{3.76}F_{0.48} upon the GO according to the following F-doped variation of Reaction (R):



wherein:

$$a = 0.48; \text{ and}$$

$$b = a/2 = 0.24.$$

[0065] Finally, when the reaction was completed, the autoclave was cooled to room temperature naturally. The precipitates were filled and washed with DI water several times and finally dried at 60°C for 12 h in vacuum. The Li₂FeSiO_{3.76}F_{0.48}/GO was sintered at 600°C for 10 h in Ar atmosphere to reduce the GO to graphene. The doping ratio of F/O was 15% by weight, calculated as (0.48*18.998)/(3.76*15.999).

[0066] Using TEM, the F-doped Li₂FeSiO_{3.76}F_{0.48}/Graphene particles appeared as nanorods (FIG. 4C) and measured even smaller in particle size than the undoped Li₂FeSiO₄/Graphene particles (FIG. 4B).

[0067] A similar process was performed to produce F-doped particles with other doping ratios from 2% by weight to 10% by weight. For example, F-doped Li₂FeSiO_{3.9}F_{0.2}/Graphene particles were produced having a doping ratio of F/O of 6% by weight, calculated as (0.2*18.998)/(3.9*15.999).

6. Preparation of Li₂FeSiO₄-Based Electrodes

[0068] Cathodes were prepared using the various Li₂FeSiO₄-based nanocomposite cathode materials from **Examples 2-5**. Each material was slurried with 10% carbon black

(Super P™ Conductive Carbon Black, TIMCAL) and 10% PVDF, sprayed onto a 10 μm thick Al foil, placed in a vacuum oven, and allowed to dry at 90° C for 24 hours. The resulting cathodes were assembled into R2016 coin cells using Li metal anodes and dielectric separators with electrolytes including 1.0 M LiPF₆ in a 3:7 by weight solvent mixture of EC and EMC.

7. Electrochemical Performance of F-Doped Electrodes

[0069] The F-doped Li₂FeSiO₄-based electrodes were subjected to electrochemical testing. The 6% F-doped Li₂FeSiO_{3.9}F_{0.2}/Graphene electrodes from **Example 5** (labeled F-LFSO-G) exhibited better overall performance than the undoped Li₂FeSiO₄/Graphene electrodes from **Example 3** (labeled LFSO-G), which exhibited better overall performance than the pure Li₂FeSiO₄ electrodes from **Example 2** (labeled LFSO-blank). The electrochemical test results are presented in FIGS. 5A-5D and are summarized in **Table 3** below.

Table 3

Electrode Material	Doping Ratio (to O)	Discharge Capacity (mAh/g)	Cycle Life (at 600 Wh/kg)	Specific Energy (Wh/kg; at 1/3C)	Diffusion (cm ² /s)
Li ₂ FeSiO ₄	0%	188	<10	<250	
Li ₂ FeSiO ₄ /Graphene	0%	247	42	<550	2.90×10 ⁻¹⁶
Li ₂ FeSiO _{3.9} F _{0.2} /Graphene	6%	305	63	720	2.21×10 ⁻¹²

[0070] A surprising phenomenon was seen in the specific capacity/energy of the Li₂FeSiO_{3.9}F_{0.2}/Graphene electrodes, where the specific energy increased during the first 25 cycles from 830 Wh/kg to 1020 Wh/kg before decreasing (FIG. 5B).

[0071] The capacity results for the other F-doped Li₂FeSiO₄-based electrodes referenced in **Example 5** are presented in FIG. 6. The 6% F-doped electrode showed the highest capacity. However, the additional F-dopant had a negative impact on cycle life, as shown by comparing the 4% F-doped results to the 6% F-doped results in FIG. 7.

[0072] The 6% F-doped electrode was subjected to further electrochemical testing at 0.1C, and the results are presented in FIGS. 8A-8C.

8. Electrochemical Performance of Mn-Doped Electrodes

[0073] The 2% Mn-doped Li₂FeSiO₄-based electrodes from **Example 4** were subjected to electrochemical testing at 0.1C, and the results are presented in FIGS. 9A-9C.

[0074] Similar Mn-doped $\text{Li}_2\text{FeSiO}_4$ -based electrodes were prepared with different Mn-doping ratios from 0.5% by weight to 30% by weight, and the comparative capacity results are presented in FIG. 10. The 2% Mn-doped electrode showed the highest capacity.

9. Electrochemical Performance of Other Cation-Doped Electrodes

[0075] Other $\text{Li}_2\text{FeSiO}_4$ -based electrodes were prepared with cations other than Mn at different doping ratios. The comparative capacity results are presented in **Table 4**.

Table 4

Doping Ratio (to Fe)	Specific Capacity (mAh/g)			
	Ti	Cu	Nb	Mo
2%	259	258	271	169
5%	142	269	282	151
8%	248	286	307	236

10. Electrochemical Performance of Hybrid-Doped Electrodes

[0076] Other $\text{Li}_2\text{FeSiO}_4$ -based electrodes were prepared with both anion and cation dopants, specifically 6% F-dopant and from 2% to 10% cation dopant. The initial cycle life data for 6%F-2%Ti hybrids and 6%F-2%Cu hybrids is presented in **Table 5**. The cycle life data for 6%F-2%Nb, 6%F-8%Nb, and 6%F-10%Nb hybrids is presented in FIG. 11. The cycle life data for 6%F-8%Ti hybrids is presented in FIG. 12. In general, the cation dopants may improve cycle life compared to F-dopants alone.

Table 5

Dopants	Specific Capacity (mAh/g)		
	1	2	3
6%F-2%Ti	259	240	246
6%F-2%Cu	258	235	243

WHAT IS CLAIMED IS:

1. An electrode material comprising:
a conductive matrix comprising a plurality of graphene sheets;
a plurality of lithium iron silicate based nanoparticles coupled to the conductive matrix.
2. The material of claim 1, wherein the nanoparticles include at least one dopant, the nanoparticles having the formula:



wherein:

X = an anion dopant;

Y = a cation dopant;

$a \geq 0$;

$b = f(a)$

$c \geq 0$; and

$d = f(c)$.

3. The material of claim 2, wherein X is selected from the group consisting of: fluorine, chlorine, and bromine.
4. The material of claim 2, wherein Y is selected from the group consisting of: titanium, manganese, copper, terbium, niobium, and molybdenum.
5. The material of claim 2, wherein $a > 0$ and $c > 0$.
6. The material of claim 2, wherein $a > c$.
7. The material of claim 2, wherein:
an anion doping ratio of X to oxygen is about 0% to about 10% by weight; and
a cation doping ratio of Y to iron is about 0% to about 10% by weight.
8. The material of claim 2, wherein:

X is fluorine; and

$$b = a/2.$$

9. The material of claim 2, wherein:

Y is manganese or copper; and

$$d = c.$$

10. The material of claim 2, wherein:

Y is niobium; and

$$d = 2.5c.$$

11. The material of claim 1, wherein the material comprises about 1 wt. % to about 10 wt. % of the graphene sheets.

12. The material of claim 11, wherein the material comprises about 2 wt. % of the graphene sheets.

13. A battery comprising an electrode with the material of claim 1.

14. The battery of claim 13, wherein the battery is configured for use in a portable electronic device, an electric vehicle, or an energy storage device.

15. The battery of claim 13, wherein the electrode lacks cobalt, nickel, and manganese.

16. A method of making an electrode material comprising:

preparing a solution containing a lithium oxide, a silicon oxide, and iron;

adding graphene oxide to the solution;

heating the solution to produce a plurality of lithium iron silicate based nanoparticles over the graphene oxide; and

sintering the nanoparticles to reduce the graphene oxide to graphene and produce a nanocomposite material.

17. The method of claim 16, wherein the lithium oxide of the preparing step is lithium hydroxide, the silicon oxide is silica, and the iron compound is iron dichloride.
18. The method of claim 16, further comprising adding at least one dopant to the solution before the heating step.
19. The method of claim 16, wherein the temperature of the heating step is about 160°C or more.
20. The method of claim 16, further comprising constructing a cathode with the nanocomposite material.

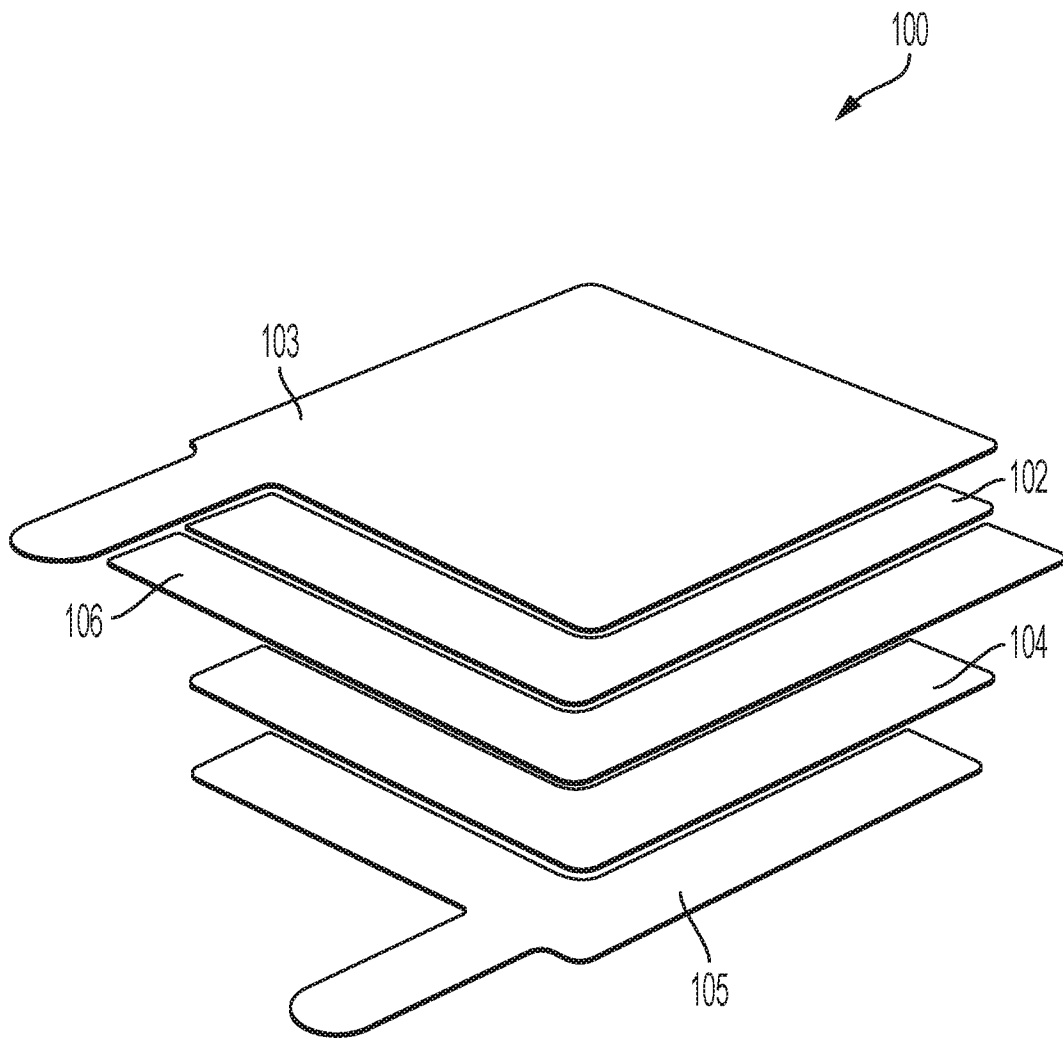


FIG. 1

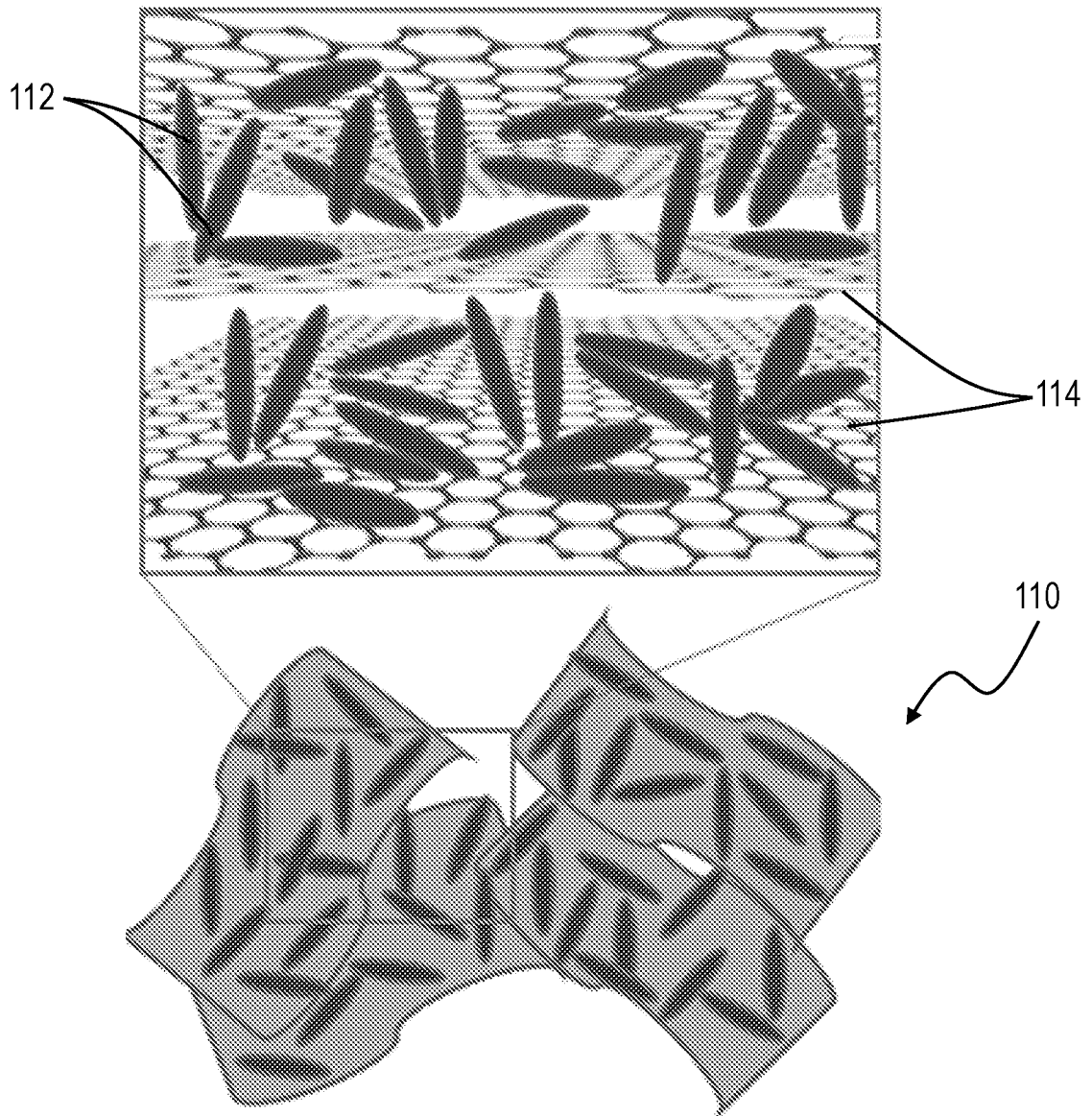
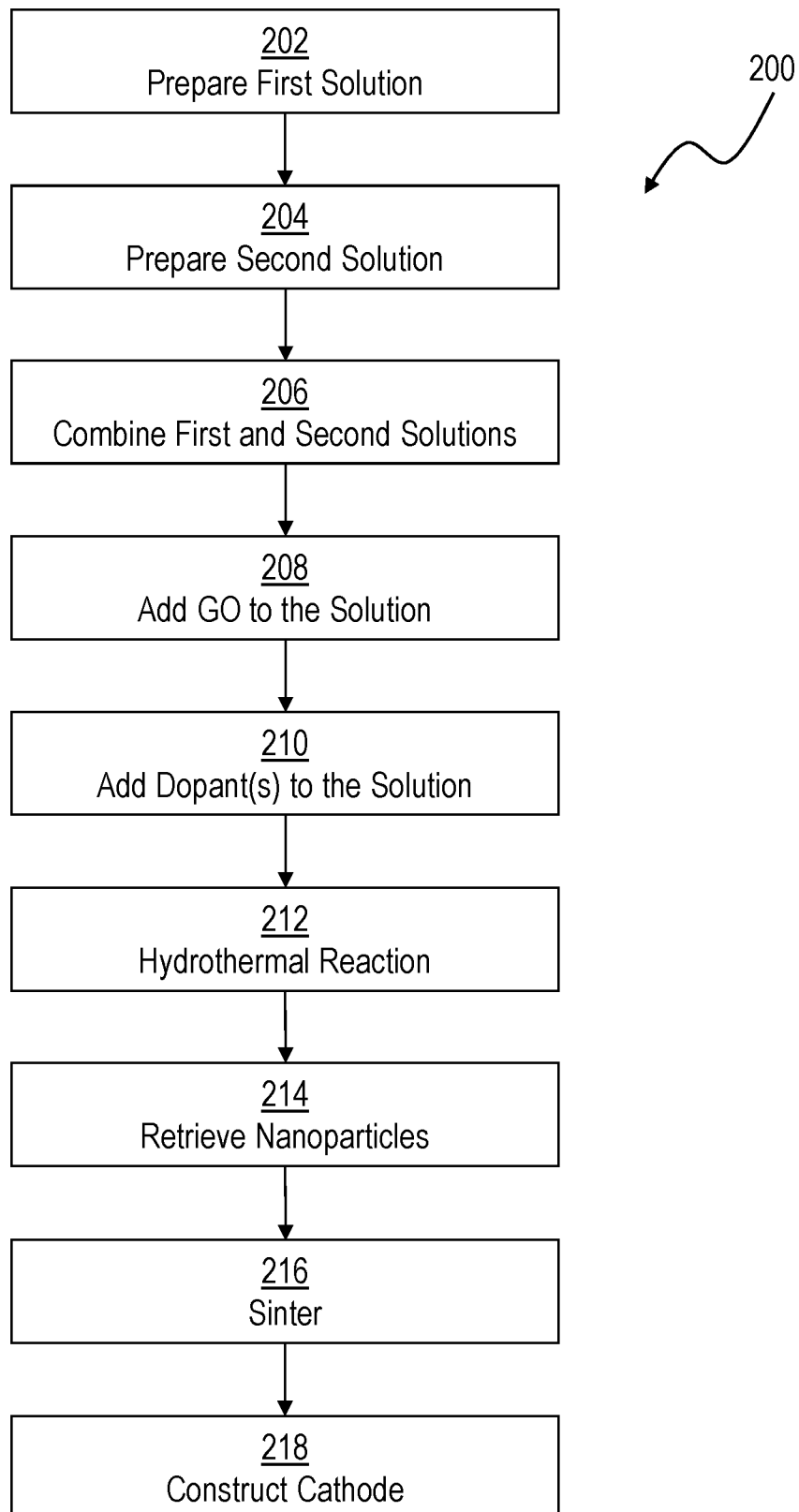


FIG. 2

**FIG. 3**

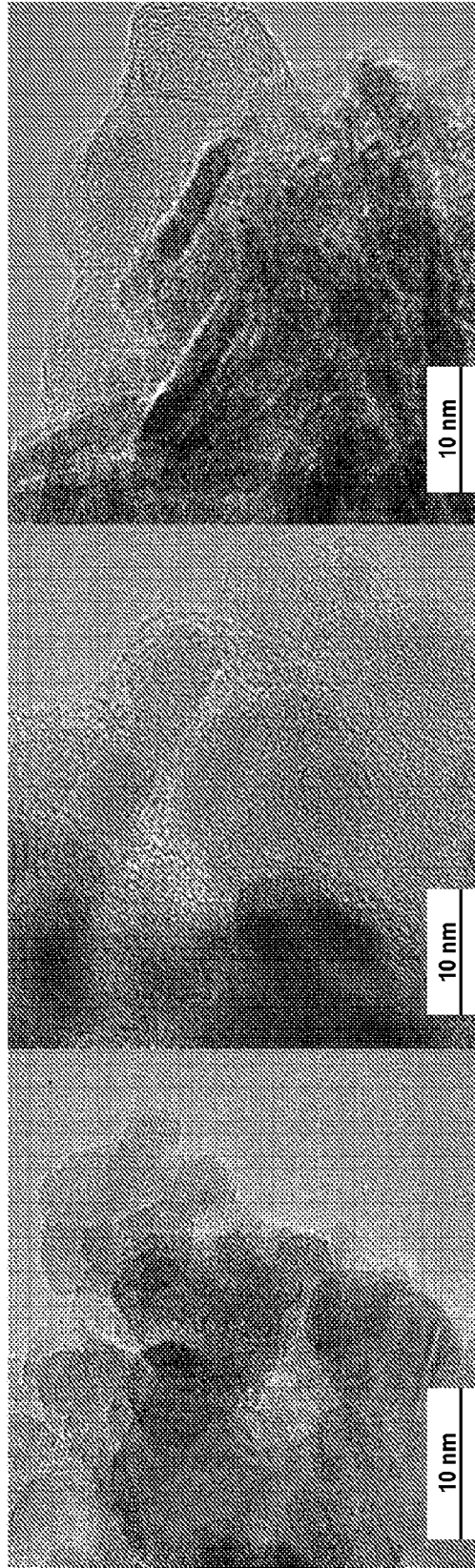


FIG. 4C

FIG. 4B

FIG. 4A

FIG. 5A

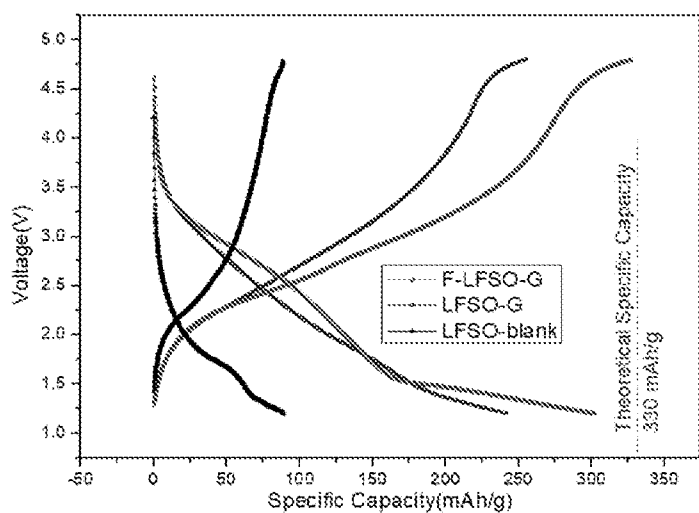


FIG. 5B

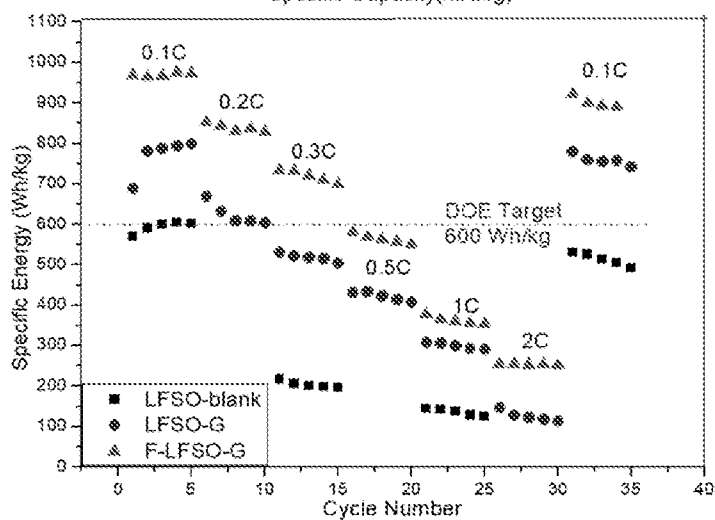
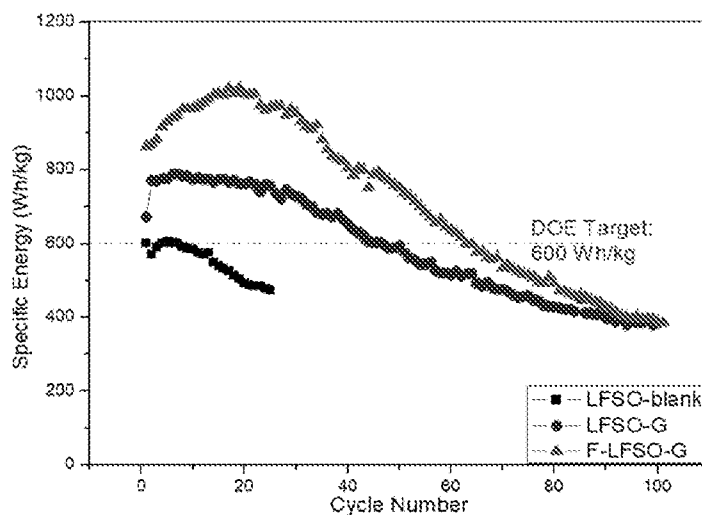


FIG. 5C

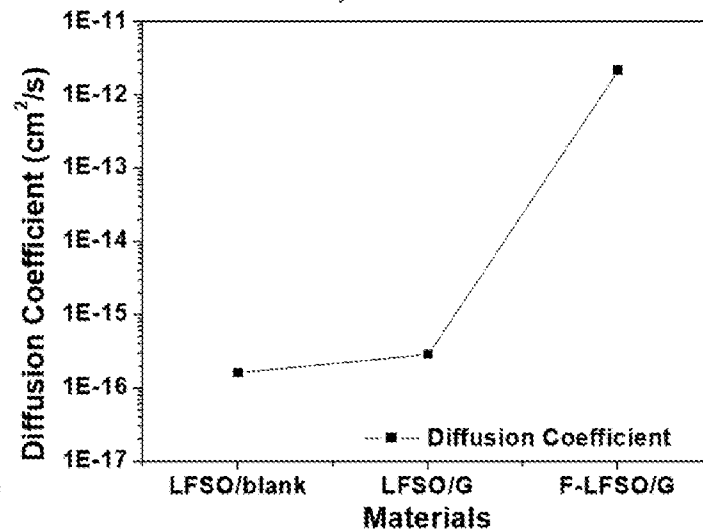
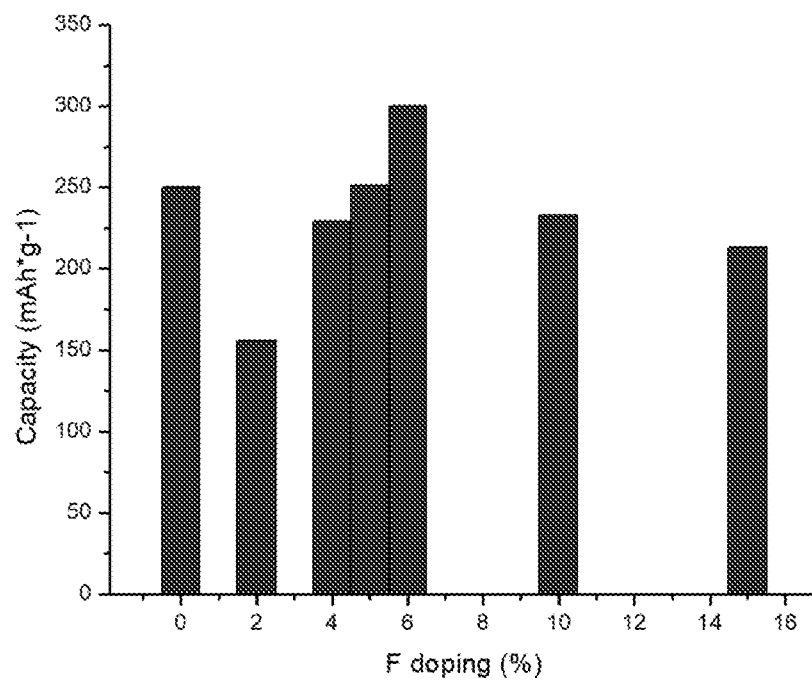


FIG. 5D

**FIG. 6**

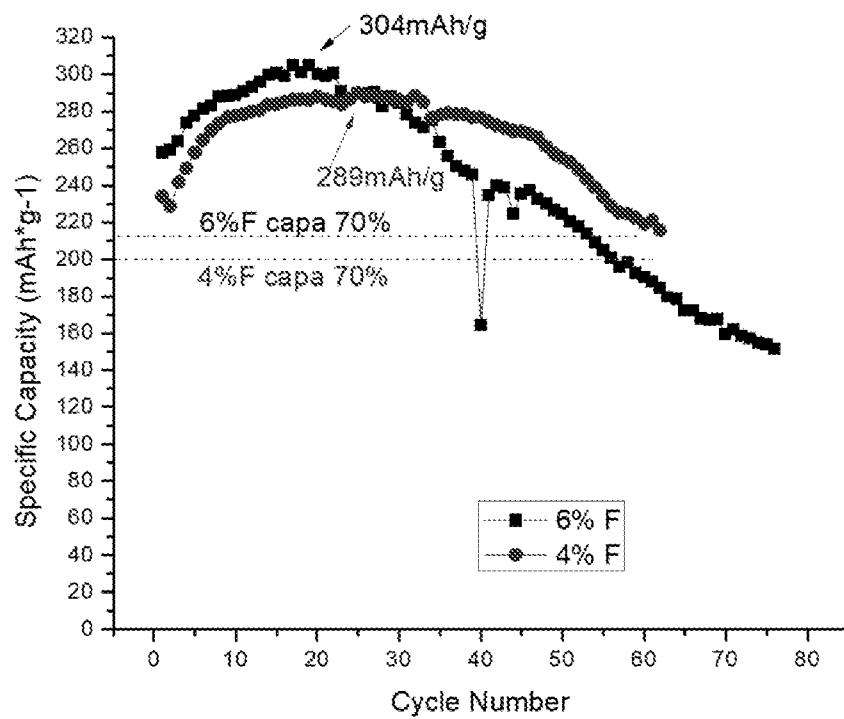


FIG. 7

FIG. 8A

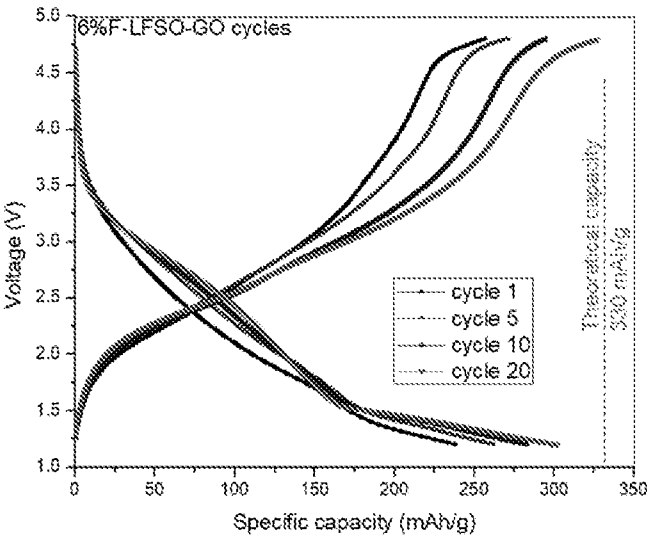


FIG. 8B

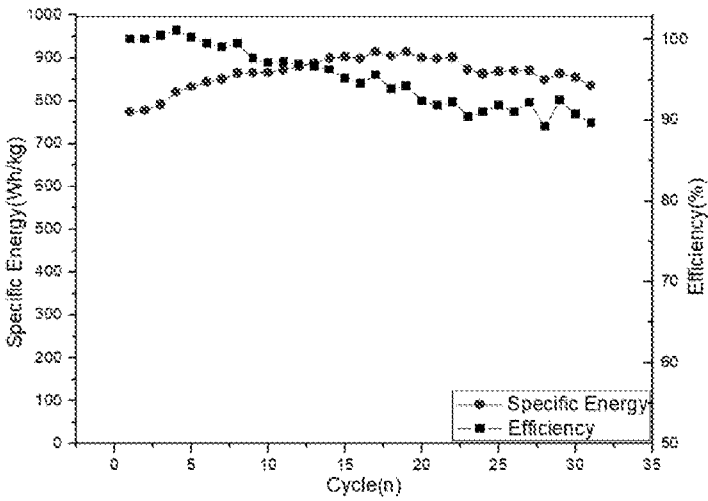
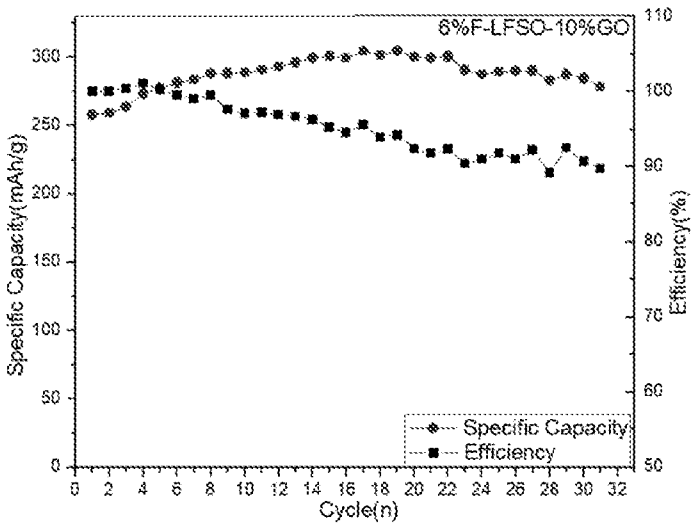


FIG. 8C

FIG. 9A

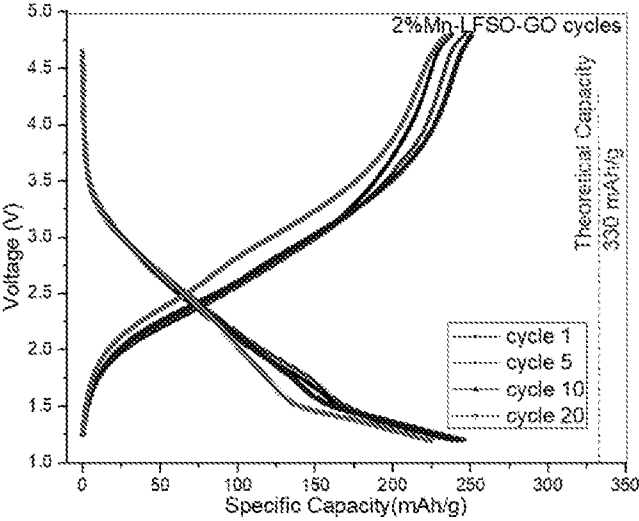


FIG. 9B

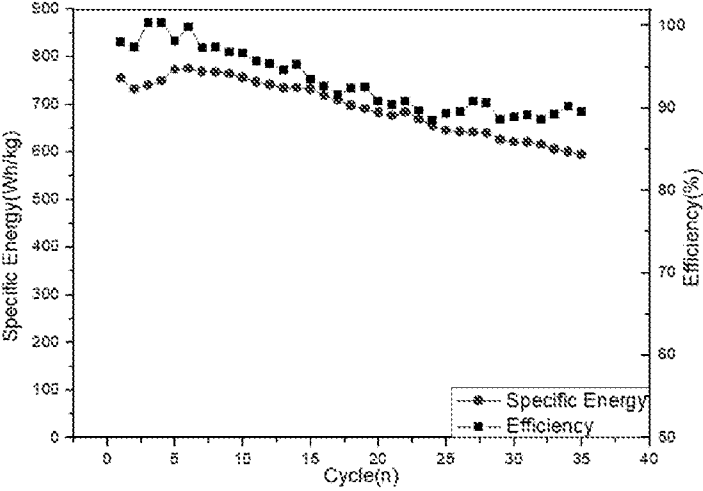
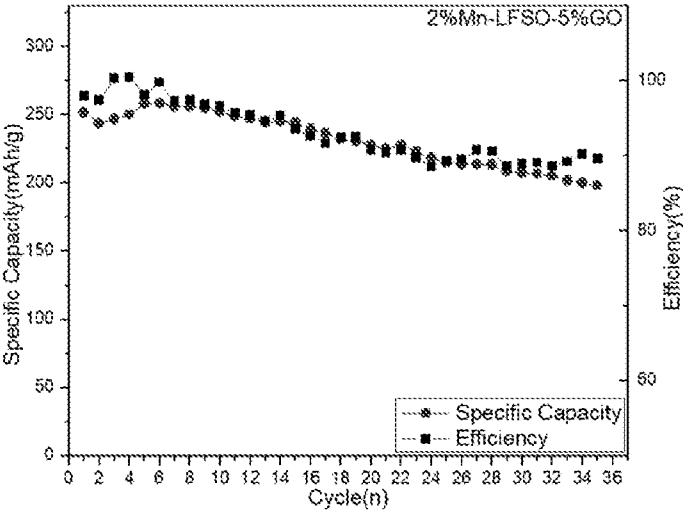
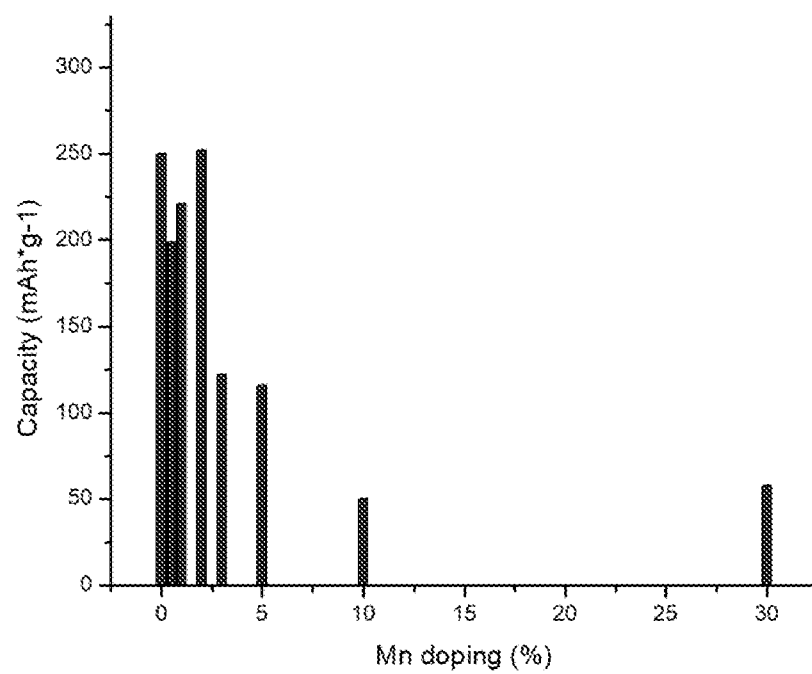


FIG. 9C

**FIG. 10**

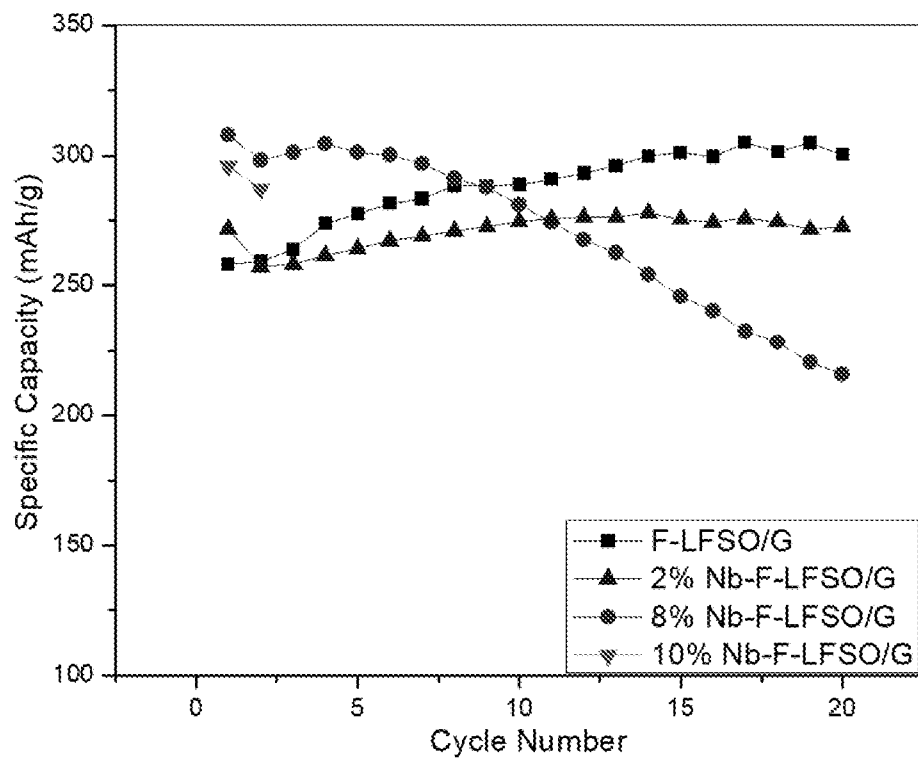


FIG. 11

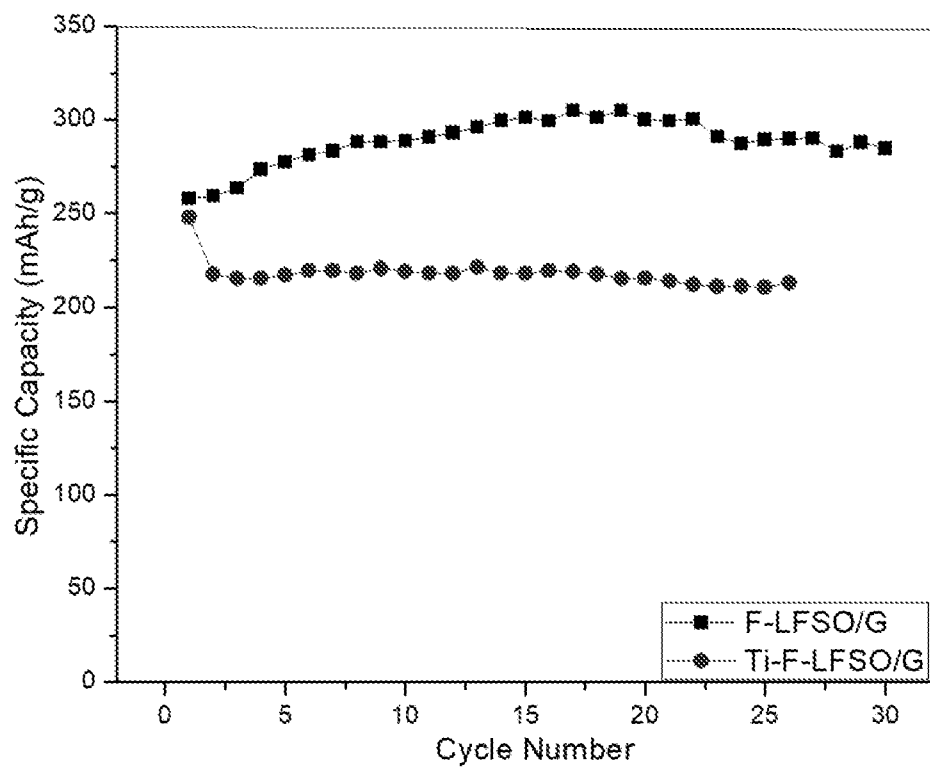


FIG. 12

INTERNATIONAL SEARCH REPORT

International application No.

PCT/US2020/021107

A. CLASSIFICATION OF SUBJECT MATTER

IPC(8) - H01M 4/52; C01B 33/20; H01M 4/134; H01M 4/64; H01M 10/052 (2020.01)

CPC - H01M 4/48; C01B 33/20; H01M 4/5825; H01M 4/587; H01M 10/052 (2020.05)

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

See Search History document

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

USPC - 429/122; 429/209; 429/221; 429/231.8 (keyword delimited)

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

See Search History document

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	US 2012/0321953 A1 (CHEN et al) 20 December 2012 (20.12.2012) entire document	1, 2, 4, 7, 9, 11-14
Y	WO 2015/146423 A1 (FURUKAWA ELECTRIC CO LTD) 01 October 2015 (01.10.2015) see machine translation	1, 2, 4, 7, 9, 11-14
A	US 2014/0333264 A1 (THE BOARD OF TRUSTEES OF THE LELAND STANFORD JUNIOR UNIVERSITY) 13 November 2014 (13.11.2014) entire document	1-20

☐ Further documents are listed in the continuation of Box C.☐ See patent family annex.

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"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention

"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone

"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art

"&" document member of the same patent family

Date of the actual completion of the international search

05 May 2020

Date of mailing of the international search report

05 JUN 2020

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