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(71) Applicant: MINNESOTA MINING AND MANUFACTURING COMPANY [US/US]; 3M Center, P.O. Box 33427, Saint Paul, MN 55133-3427 (US).

(72) Inventors: CHENG, Hwa ; DEPUYDT, James, M. ; HASE, Michael, A. ; QIU, Jun ; P.O. Box 33427, Saint Paul, MN 55133-3427 (US).

(74) Agents: SHERMAN, Lorraine, R. et al.; Minnesota Mining and Manufacturing Company, Office of Intellectual Property Counsel, Post Office Box 33427, Saint Paul, MN 55133-3427 (US).

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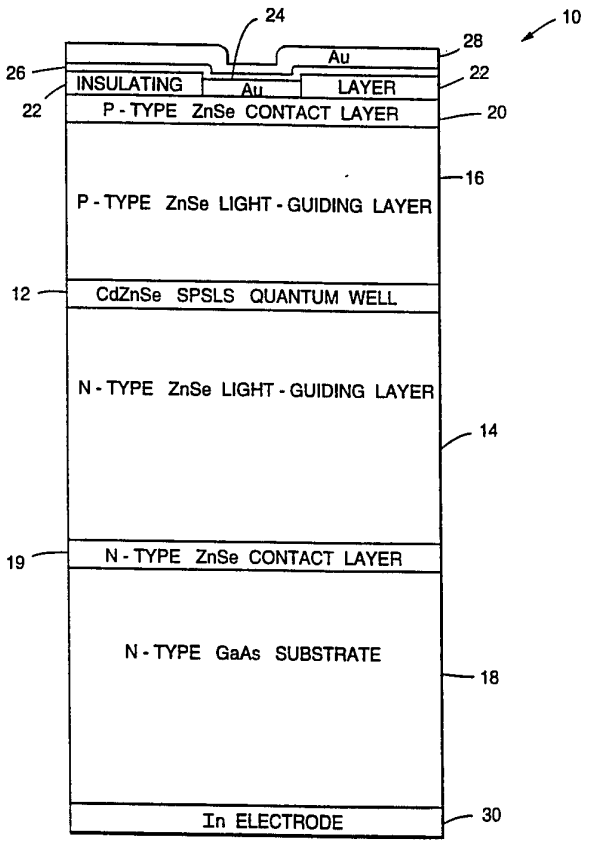
(54) Title: SINGLE QUANTUM WELL II-VI LASER DIODE WITHOUT CLADDING

(57) Abstract

A single quantum well II-VI laser diode without semiconductor cladding layers includes a pn junction formed by overlaying light-guiding layers (14, 16) of p-type and n-type ZnSe on an n-type GaAs substrate. A CdSe/ZnSe short-period strained-layer superlattice single quantum well active layer (12) is positioned between the guiding layers. An Au electrode (24) overlays the p-type guiding layer opposite the single quantum well active layer. The guiding layers have thicknesses which enable the substrate and Au electrode to confine the light beam generated by the device within the active layer and the guiding layers.



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SINGLE QUANTUM WELL II-VI LASER DIODE WITHOUT CLADDING

BACKGROUND OF THE INVENTION

Research undertaken by 3M in St. Paul, Minnesota, culminated in the demonstration of the world's first laser diodes fabricated from II-VI semiconductor materials. These devices emit coherent radiation at 490 nm in the blue-green portion of the spectrum. They are disclosed generally in the Haase et al. article *Short Wavelength II-VI Laser Diodes*, Conference Proceedings for Gallium Arsenide and Related Compounds, 1991 Institute of Physics Conference Series, No. 120, pp 9-16.

There remains a continuing need for improved II-VI laser diode technology. For widespread commercial viability, the laser diodes must be relatively simple in construction and inexpensive to manufacture. The devices must also be capable of operation at relatively low voltages.

SUMMARY OF THE INVENTION

The present invention is a II-VI laser diode capable of efficiently generating high intensity beams of light at relatively low operating voltages. The device is relatively simple in construction and inexpensive to manufacture since it does not include semiconductor cladding layers for the primary purpose of confining the beam.

In one embodiment, the laser diode includes a single crystal semiconductor substrate of a first conductivity type. A first guiding layer of II-VI semiconductor of the first conductivity type overlays the substrate. A single quantum well active layer of II-VI semiconductor overlays the first guiding layer. A second guiding layer of II-VI semiconductor of a second conductivity type overlays the active layer and forms a pn junction with the first guiding layer. A

first electrode layer overlays the substrate opposite the first guiding layer, and a second electrode layer overlays the second guiding layer opposite the active layer. The first and second guiding layers have thicknesses enabling the substrate and second electrode to confine the beam of light within the active layer and the first and second guiding layers.

In another embodiment, the substrate includes a GaAs substrate. The first and second guiding layers include ZnSe layers having combined thicknesses of about 3.5 μm. The active layer includes a CdSe/ZnSe short-period strained-layer superlattice layer. The second electrode is a Au electrode.

BRIEF DESCRIPTION OF THE DRAWINGS

Figure 1 is a cross sectional view (not to scale) illustrating the structure of a II-VI semiconductor laser diode in accordance with the present invention.

Figure 2 is a graph illustrating the product of the loss coefficient (α) and the full width at half maximum (FWHM) intensity of the optical mode as a function of the thickness of the light-guiding layers, for laser diodes of the type shown in Figure 1.

Figure 3 is an illustration of a molecular beam epitaxy (MBE) system that can be used to fabricate laser diodes in accordance with the present invention.

Figure 4 is a detailed cross sectional view of the quantum well layer shown in Figure 1.

Figure 5 is graph of the shutter sequence by which the MBE system, shown in Figure 3, is operated to fabricate the active layer of laser diodes in accordance with the present invention.

Figure 6 is a high resolution transmission electron micrograph of a cross-section of the quantum well of a laser diode fabricated in accordance with the present invention.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

A II-VI compound semiconductor laser diode 10 (i.e., an electroluminescent device) in accordance with the present invention is illustrated generally in Figure 1. Laser diode 10 includes a short-period strained-layer superlattice (SPSLS) quantum well layer 12 surrounded by a ZnSe pn junction formed by n-type ZnSe light-guiding layer 14 and p-type ZnSe light-guiding layer 16. As described in greater detail below, quantum well layer 12 is a high efficiency active layer grown by atomic layer epitaxy (ALE) and/or migration enhanced epitaxy (MEE). Laser diode 10 is fabricated on an n-type GaAs substrate 18 and includes an n-type ZnSe ohmic contact layer 19 between the substrate and guiding layer 14. A p-type ZnSe ohmic contact layer 20 overlays p-type guiding layer 16. A polyimide insulating layer 22 covers the surface of ohmic contact layer 20 opposite light-guiding layer 16.

Electrical contact to p-type ohmic contact layer 20 is made by Au electrode 24, which is formed in an open stripe in insulating layer 22. A thin Ti layer 26 and a final Au layer 28 are applied over insulating layer 22 to facilitate lead bonding. Electrical contact to the lower side of laser diode 10 is made by an In electrode 30 on the surface of substrate 18 opposite n-type ohmic contact layer 19.

Light-guiding layer 14 and contact layer 19 are both doped n-type with Cl in prototypes of laser diode 10. Light-guiding layer 16 and ohmic contact layer 20 are doped p-type with N in these prototypes. The net donor concentration to which the lower light-guiding layer 14 is doped is $1x10^{17}cm^{-3}$, while the upper light-guiding layer 16 is doped to a net acceptor concentration of $2x10^{17}cm^{-3}$. Ohmic contact layers 19 and 20 are both deposited to a thickness of 0.1 μm in the prototype devices. The lower contact layer 19 is doped n-type to a net donor concentration of $1x10^{18}cm^{-3}$.

The upper contact layer 20 is doped p-type to a net acceptor concentration of $1x10^{18}cm^{-3}$.

Light generated in quantum well active layer 12 is guided within light-guiding layers 14 and 16, clad only by the GaAs substrate 18 and the Au electrode 24. Good optical confinement and sufficiently low loss are obtained in laser diode 10 without the need for II-VI semiconductor cladding layers. Computer modeling is used to select appropriate thicknesses for light-guiding layers 14 and 16. This modeling approach takes into account the ZnSe waveguide formed by light-guiding layers 14 and 16, as well as the complex indices of refraction of GaAs substrate 18 and Au electrode 24. Modeling methods of this type are generally known and disclosed, for example, in M.R. Ramdas et al., *Analysis of Absorbing and Leaky Planar Waveguides: A Novel Method*, Optics Letters, Vol. 14, p. 376 (1989) and the references cited therein.

Figure 2 is a graph illustrating the product of the loss coefficient (α) and the full width at half maximum intensity (FWHM) of the desired optical mode (TE polarized for the prototypes described herein) as a function of the thickness of the ZnSe layers 14 and 16 (Fig. 1). To minimize the threshold current density of the device, this product should be minimized. Using this design criterion and the information shown in Figure 2, the thickness of the waveguide (i.e., the combined thicknesses of light-guiding layers 14 and 16) is approximately 3.5 μm in the prototype laser diode 10. In this embodiment, n-type light-guiding layer 14 has a thickness of 2.0 μm, while p-type light-guiding layer 16 has a thickness of 1.5 μm. The loss due to free-carrier absorption and scattering is estimated to be 8 cm^{-1} in this embodiment. Quantum well layer 12 has only a relatively small effect on the loss and optical confinement characteristics of the device, and its presence is neglected during the design procedure

described above. Theory suggests that total waveguide thicknesses less than 2.0 μm result in excessive absorption losses in substrate 18 and electrode 24. At a thickness of 2.5 μm, the substrate and electrode absorption losses are 11.7 cm^{-1}. On the other hand, the FWHM of the optical mode is found to be almost exactly half of the waveguide thickness. Therefore, for thicknesses greater than about 6 μm the optical confinement is so poor that the single quantum well layer 12 cannot practically supply enough gain to overcome the losses. The maximum modal gain is inversely proportional to the FWHM of the waveguide mode. For a waveguide that is 6 μm thick, the FWHM is about 3 μm, and the modal gain from a single quantum well can be estimated to be 12 cm^{-1}. See, e.g., N.K. Dutta, Applied Physics Letters, vol. 53, p. 72 (Nov 1982).

Figure 3 is an illustration of a molecular beam epitaxy (MBE) system used to fabricate the prototypes of laser diode 10 described above. The MBE system includes a chamber 54 with a high energy electron gun 58, a phosphorus screen 60, a substrate heater 90 and a flux monitor 62. MBE chambers such as 54 are generally known and commercially available.

Prototypes of laser diode 10 are fabricated on Si-doped n^+-type GaAs substrates 18 having a (100) crystal orientation. Substrates of this type are commercially available. Substrate 12 is cleaned and prepared using conventional or otherwise known techniques, and mounted to a Molybdenum sample block (not shown in Figure 3) by In solder before being positioned within chamber 54.

Chamber 54 includes a Zn effusion cell 70, cracked-Se effusion cell 72, Cd effusion cell 76 and a standard Se (i.e., primarily Se_6) effusion cell 79. As shown, cracked-Se effusion cell 72 includes a bulk evaporator 84 and high temperature cracking zone 82, and provides a source of cracked Se (including Se^2 and

other Se molecules with less than 6 atoms). The bulk evaporator 84 and high temperature cracking zone 82 used to produce the prototype laser diodes 10 are of a custom design, the details and capabilities of which are described in the Cheng et al. article, *Molecular-Beam Epitaxy Growth of ZnSe Using a Cracked Selenium Source*, J. Vac. Sci. Technol., B8, 181 (1990). Cl effusion cell 78 which utilizes $ZnCl_2$ source material provides the Cl n-type dopant. The p-type dopant is provided by N free-radical source 80. Free-radical source 80 is connected to a source 86 of ultra-pure N_2 through leak-valve 88. The free-radical source 80 used in the fabrication of laser diodes 10 is commercially available from Oxford Applied Research Ltd. of Oxfordshire, England (Model No. MPD21). The beam exit plate at the end of the source is made of pyrolyric boron nitride (PBN) and has nine 0.2 mm diameter holes through it. This source is mounted on a standard port for an effusion cell through a 10" extension tube. N_2 source 86 used to fabricate laser diodes 10 is of research purity grade. The pressure at the inlet of the leak-valve of source 86 is 5 psi.

MBE chamber 54 is operated in a manner described in the Cheng et al. article *Growth of p- and n- type ZnSe by Molecular Beam Epitaxy*, J. Crystal Growth 95, 512 (1989) using the Se_6 source 79 as the source of Se to grow the n-type contact and light-guiding layers 19 and 14, respectively, of the prototype laser diode 10. SPSLS quantum well layer 12 is grown on the light-guiding layer 14 of laser diode 10 using atomic layer epitaxy (ALE) and/or migration enhanced epitaxy (MEE). Using these techniques, which are generally known, quantum well layer 12 is formed as a series of overlaying single crystal thickness layers (i.e., monolayers) of Cd, Zn and Se. A detailed illustration of quantum well layer 12 is shown in Figure 4. In this embodiment, quantum well layer 12 includes adjacent

monolayers of Cd and Se between a pair of adjacent monolayers of Zn and Se. This structure can be described generally by the following notation:

$$[(CdSe)_m(ZnSe)_a]_p$$

where: m, n and p are integers.

In the embodiment illustrated in Figure 4, m=1, n=2, and p=1. In other embodiments (not shown) m=1, n=1-5 and p=1-5. The equivalent Cd concentration in quantum well layer 12 is determined by the ratio of the number of CdSe layers to the total number of layers (including both ZnSe and CdSe) in the quantum well layer. Total thickness of quantum well layer 12 is given by the number of monolayers grown times the thickness of each monolayer. Figure 6 is a high resolution transmission electron micrograph analysis of a cross section of a prototype laser diode 10 having a SPSLS quantum well with a period (p) equal to 6, clearly representing the monolayer structure of the quantum well.

Control over the composition and thickness of the Cd, Zn and Se monolayers of quantum well layer 12 are accurately achieved by ALE and/or MEE. Through use of these techniques, monolayer growth is controlled primarily by the sequence and timing by which the shutters (not separately shown) of the Cd, Zn and Se effusion cells 76, 70, and 72, respectively, are opened and closed. A portion of the effusion cell shutter sequence used to grow the quantum well layer 12 illustrated in Figure 4 is shown in Figure 5. A characteristic delay time is introduced between the sequential pulses of the reactant species to allow for the reevaporation of excess reactant.

Prototype laser diodes 10 including SPSLS quantum well layers such as that shown in Figure 4 have been grown at temperatures of 150°C and using the thermally cracked Se (Se_2) effusion cell 76. The shutter

sequence begins with the Se shutter open. The Se shutter is closed after depositing at least one monolayer (about 5 seconds). The Zn shutter is then opened after a slight delay (about 2 seconds) to allow excess Se to evaporate. Next, the Zn shutter is closed after depositing at least one monolayer of Zn (about 4 seconds). A slight delay (about 1 second) is included between the closing of the Zn shutter and the reopening of the Se shutter to allow time for the evaporation of any excess Zn. Growth continues by alternately opening and closing the shutters to sequentially deposit overlaying layers of Cd, Se and Zn. The Cd shutter is opened for about 4 seconds, followed by a delay of about 1 second before the Se shutter is reopened. The sequence beginning with the open Se shutter is then repeated to complete quantum well layer 12. Other operating parameters of MBE chamber 54 used to produce the quantum well layer 12 of the prototype laser diodes 10 are as follows:

Cd beam equivalent pressure: 1.0×10^{-7} Torr *
Zn beam equivalent pressure: 1.0×10^{-7} Torr *
Se cracking zone temperature: 600°C *
Se bulk evaporator temperature: 250°C *
*parameters dependent upon specific MBE system configuration and plasma source.

Prototype laser diodes 10 having SPSLS quantum well layers 12 grown in the manner described above at 150°C have exhibited the highest quantum efficiency. However, quantum well test wafers with SPSLS quantum well layers grown at temperatures up to 235°C have exhibited good characteristics. Acceptable characteristics have been observed in test wafers with SPSLS quantum well layers grown at temperatures as high as 300°C. It is also anticipated that devices having desirable characteristics can be grown at temperatures less than 150°C.

MBE chamber 54 is operated in a manner described in the Park et al. article *P-type ZnSe by Nitrogen Atom Beam Doping During Molecular Beam Epitaxial Growth*, Appl. Phys. Lett. 57, 2127 (1990), using the Se_6 source 79 to grow the p-type light-guiding layer 16.

Low resistivity p-type ZnSe ohmic contact layer 20 has been achieved by growing the contact layer at low temperature within MBE chamber 54 utilizing the cracked Se source 72 (i.e., cracking zone 82 and evaporator 84), while at the same time doping the semiconductor material of the contact layer p-type in accordance with the disclosure in Park et al., *P-Type ZnSe by Nitrogen Atom Beam Doping During Molecular Beam Epitaxial Growth*, Appl. Phys. Lett. 57 (20) 12 November 1990. The low temperature growth technique used to produce the contact layer 20 of the prototype laser diode 10 is described generally in the Cheng et al. article *Low Temperature Growth of ZnSe By Molecular Beam Epitaxy Using Cracked Selenium*, Appl. Phys. Lett. (Feb. 1990). The semiconductor body with layers 19, 14, 12, and 16 on substrate 18 is heated to a temperature less than 250°C but high enough to promote crystalline growth of the ZnSe doped with the N p-type dopants to a net acceptor concentration of at least $1x10^{17}cm^{-3}$. A net acceptor concentration of $1x10^{18}cm^{-3}$ was achieved in the ohmic contact layer 20 of prototype laser diodes 10, when grown at a substrate temperature of about 150°C. However, it is anticipated that ohmic contact layers 20 with acceptable characteristics can be achieved at other growth temperatures down to at least 130°C. Other operating parameters of MBE chamber 54 used to produce the ohmic contact layer 20 of the prototype laser diodes 10 are as follows:

Zn beam equivalent pressure:	$1.0x10^{-7}$ Torr*
Se cracking zone temperature:	600° C*
Se bulk evaporator temperature:	250° C*

Growth rate:	0.3-0.6 μm/hr
Surface reconstruction:	Zn-stabilized
Nitrogen pressure in chamber:	$>3.5x10^{-7}$ Torr*
rf power:	150-250 W*

* parameters dependent upon specific MBE system configuration and plasma source.

Following the deposition of contact layer 20, the as yet incomplete laser diode 10 is removed from MBE chamber 54. Electrode 24 includes Au which is vacuum evaporated onto contact layer 20 and patterned into a stripe (typically about 20 μm wide) using conventional photolithography and lift-off. An insulating layer 22 is then applied over electrode 24 and the exposed surface of contact layer 20. For an insulator that can be applied at low temperatures, polyimide photoresist is preferred. Probimide 408 from Ciba-Geigy Corp. was used to produce the prototype laser diodes 10. A stripe (about 20 μm wide) of the polyimide layer 22 directly above electrode 24 is removed by UV exposure through a photomask and development using the manufacturer's recommended processing recipe, except for the post-development cure. To cure the developed polyimide, the device was flood exposed to 1 J/cm^2 of UV light from a mask aligner, and baked at 125°C on a hot plate in air for 3 minutes. Ti-Au layer 26 was then evaporated on the exposed surface of the Au electrode 24 and polyimide layer 22 to facilitate lead-bonding. The in used for MBE substrate bonding also served as electrode 30 on substrate 18. Opposite ends of the device were cleaved along (110) planes to form facet mirrors. The facets were coated with a total of four alternating quarter wavelength stacks of MgF_2 and ZnSe to provide a 90% reflectivity. Cavity length of the prototype devices 10 is about 1000 μm. Laser diodes 10 were then bonded p-side up to ceramic sample holders with silver-filled epoxy.

Laser diodes 10 in accordance with the present invention offer considerable advantages. The ALE/MEE techniques for growing the quantum well layer enables better composition control, better control of the quantum well thickness and luminescence efficiency than random alloy quantum wells grown by conventional MBE. These characteristics are evidenced by increased room temperature photoluminescence and electroluminescence intensities, and lower laser threshold currents. Although the room temperature threshold currents in these prototype devices (as low as 1030 A/cm^3) are not as low as in devices with cladding layers, these devices offer the advantages of simpler construction and lower operating voltages (about 13 V).

The embodiments described above include light guiding layers that are not lattice matched to the substrate. It is anticipated that increased device lifetime and improved performance will be exhibited by devices in which the lattice parameters of the light guiding layers are matched to that of the substrate. The first and second light guiding layer may include $ZnS_{0.06}Se_{0.94}$ or $Cd_{0.43}Zn_{0.57}S$ on GaAs substrates or ZnSe light guiding layers on $In_{0.04}Ga_{0.96}As$ or $In_{0.52}Ga_{0.48}P$ substrates. The light guiding layers may also be comprised of ZnS_xSe_{1-x}, $Cd_xZn_{1-x}S$, $ZnS_{1-x}Te_x$, $Zn_{1-x}Cd_xSe$, $Zn_{1-x}Mg_xS_ySe_{1-y}$ or $Cd_xZn_{1-x-y}Mg_yS$ layers lattice matched to substrates such as GaAs, AlAs, GaP, $Al_xGa_{1-x}As$, $In_xGa_{1-x}As$, $In_xAl_{1-x}As$, $In_xGa_{1-x}P$, $In_xAl_{1-x}P$, $GaAs_{1-x}P_x$, $In_xGa_{1-x-y}Al_yAs$, $In_xGa_{1-x-y}Al_yP$, ZnSe or $Zn_{1-x}Cd_xS$.

Although the present invention has been described with reference to preferred embodiments, those skilled in the art will recognize that changes may be made in form and detail without departing from the spirit and scope of the invention. For example, it is expected that the inventive concepts used to fabricate the prototype laser diodes disclosed herein are well suited to the fabrication of laser diodes with ALE/MEE active

layers and/or guiding layers from a wide variety of other II-VI semiconductors and alloys. These include ZnSe, ZnTe, ZnSeTe, CdS, CdZnSeTe, MgZnSe, ZnSSe, CdZnS, ZnSTe and CdZnTe.

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WHAT IS CLAIMED IS:

1. A II-VI compound semiconductor laser diode for generating a beam of light, including:

a single crystal semiconductor substrate of a first conductivity type;

a first guiding layer of II-VI semiconductor of the first conductivity type overlaying the substrate;

a single quantum well active layer of II-VI semiconductor overlaying the first guiding layer;

a second guiding layer of II-VI semiconductor of a second conductivity type overlaying the active layer, the first and second guiding layers forming a pn junction;

a first electrode layer overlaying the substrate opposite the first guiding layer;

a second electrode layer overlaying the second guiding layer opposite the active layer; and

the first and second guiding layers having thicknesses enabling the substrate and second electrode to confine the beam of light within the active layer and the first and second guiding layers.

2. The laser diode of claim 1 wherein the single quantum well active layer includes a CdZnSe layer.

3. The laser diode of claim 2 wherein the single quantum well active layer includes a CdSe/ZnSe short-period strained-layer superlattice layer.

4. The laser diode of claim 2 wherein the first and second guiding layers include ZnSe layers.

5. The laser diode of claim 4 wherein the substrate includes a GaAs substrate.

6. The laser diode of claim 4 wherein the second electrode is an Au electrode layer.

7. The laser diode of claim 4 wherein the combined thickness of the first and second guiding layers is between about 2 μm and 6 μm.

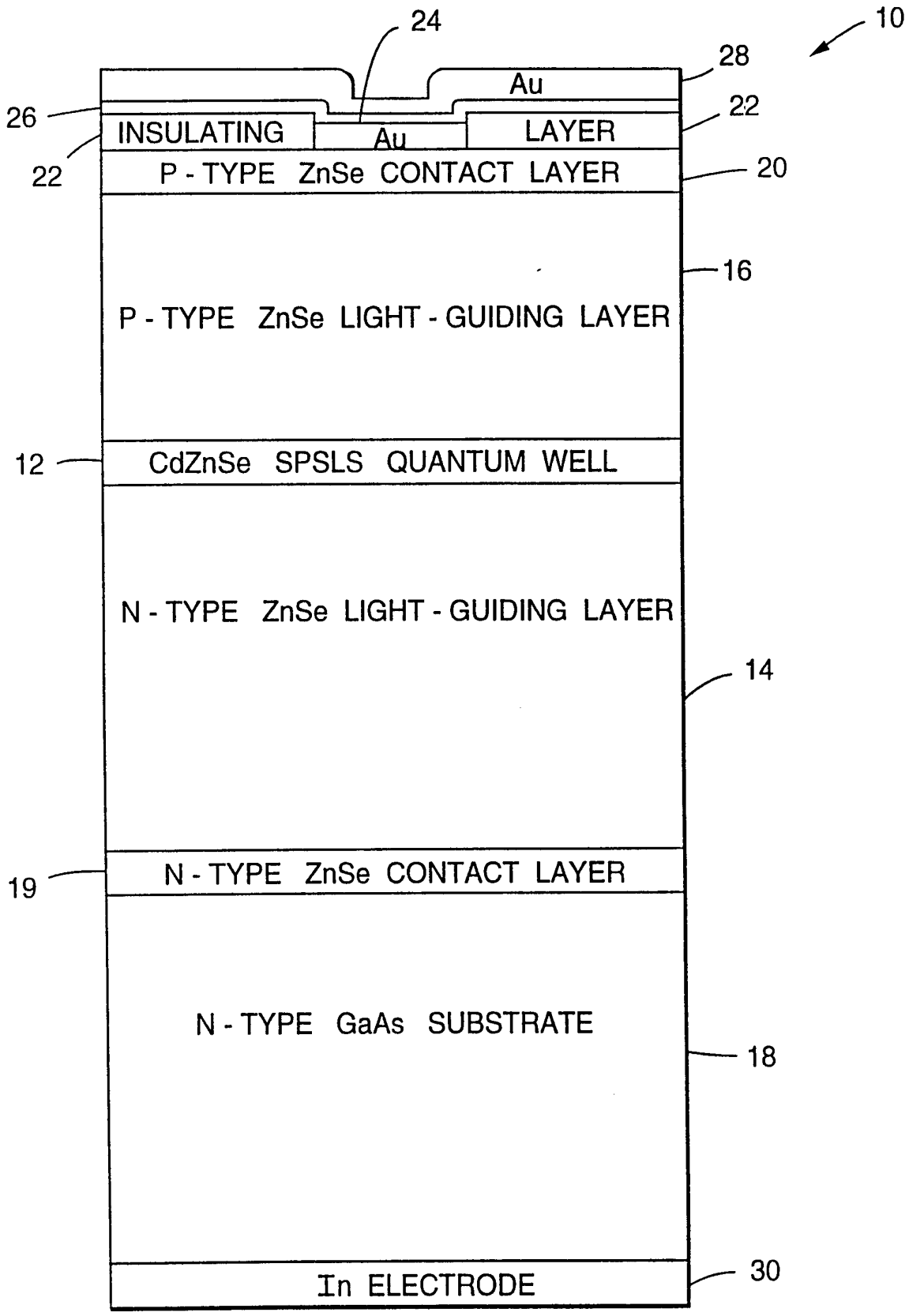
8. The laser diode of claim 1 wherein:
the substrate includes a GaAs substrate;
the first and second guiding layers include ZnSe layers having a combined thickness between about 2 μm and 6 μm;
the single quantum well active layer includes a CdSe/ZnSe short-period strained-layer superlattice layer; and
the second electrode includes a Au electrode.

9. The laser diode of claim 8 wherein the first and second guiding layers have a combined thickness of about 3.5 μm.

10. The laser diode of claim 1 and further including:
a first ohmic contact layer of the first conductivity type between the substrate and the first guiding layer; and
a second ohmic contact layer of the second conductivity type between the second guiding layer and the second electrode layer.

11. The laser diode of claim 1 wherein the first and second guiding layers include layers of having a lattice constant which is matched to the lattice constant of the substrate.

12. The laser diode of claim 1 wherein the first and second guiding layers include layers of II-VI semiconductor having a lattice constant which is mismatched with the lattice constant of the substrate.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16
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12
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19
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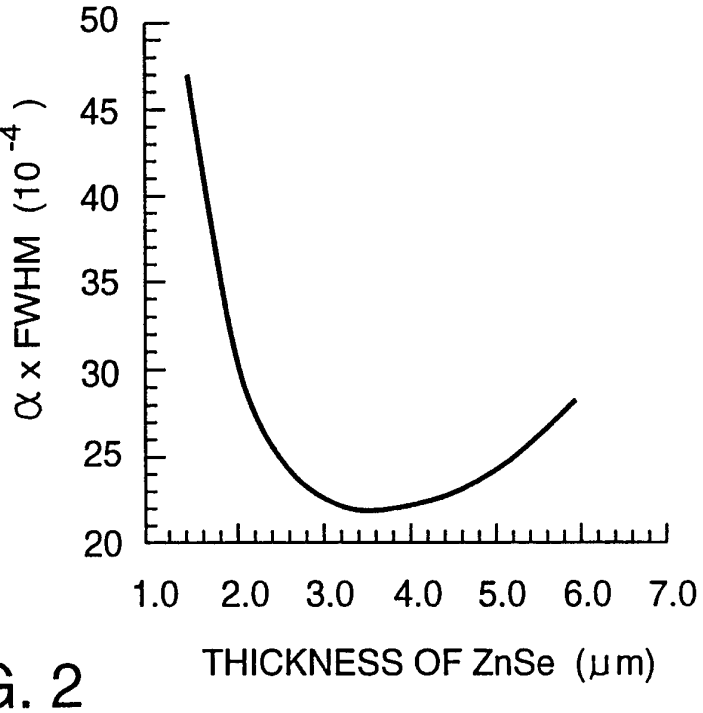
N - TYPE GaAs SUBSTRATE
18
In ELECTRODE
30

FIG. 1


50
45
40
35
30
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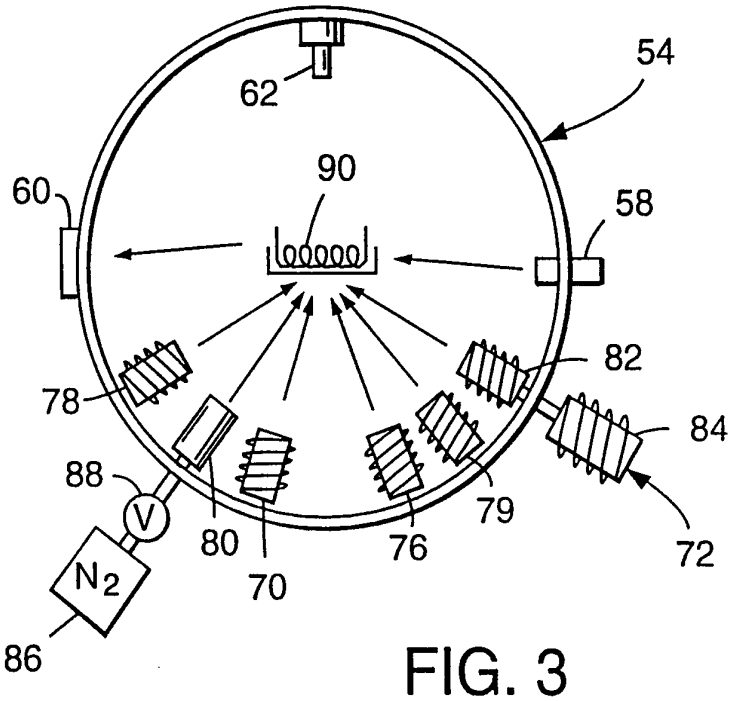
20
α x FWHM (10 -4)
1.0 2.0 3.0 4.0 5.0 6.0 7.0
THICKNESS OF ZnSe (μm)

FIG. 2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N2
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FIG. 3

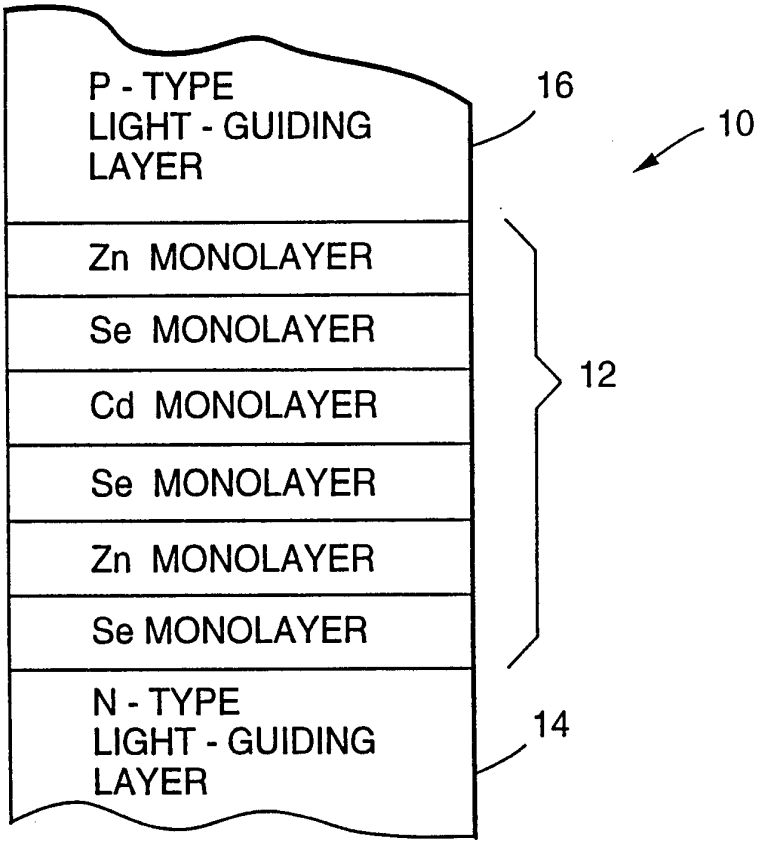
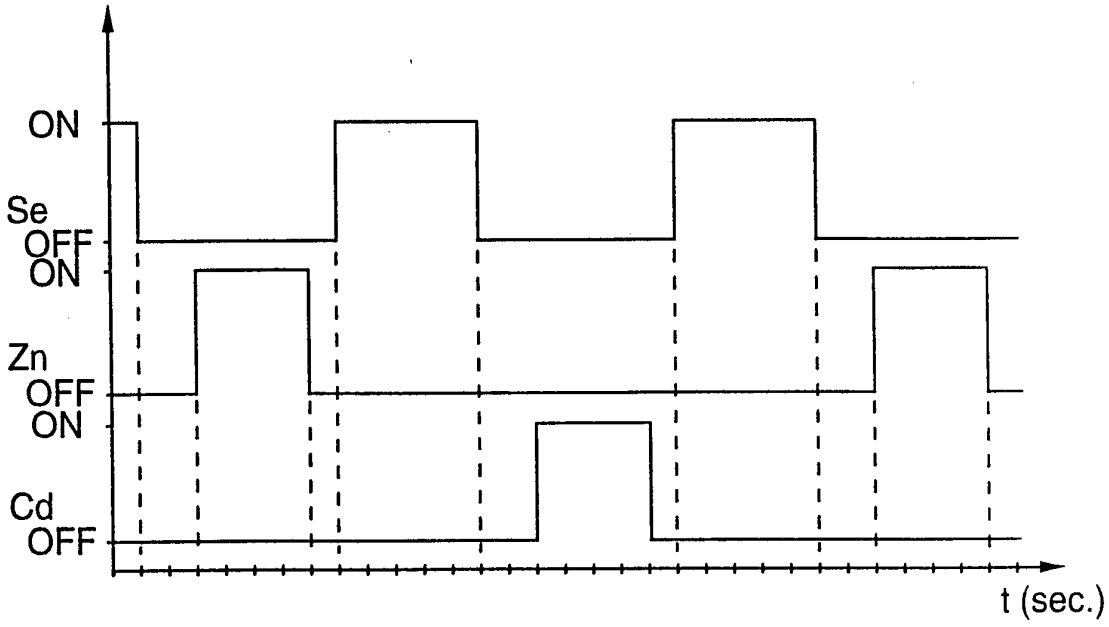

P - TYPE
LIGHT - GUIDING
LAYER
16
10
Zn MONOLAYER
Se MONOLAYER
Cd MONOLAYER
Se MONOLAYER
Zn MONOLAYER
Se MONOLAYER
12
N - TYPE
LIGHT - GUIDING
LAYER
14

FIG. 4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ON
Zn
OFF
ON
Cd
OFF
t (sec.)

FIG. 5

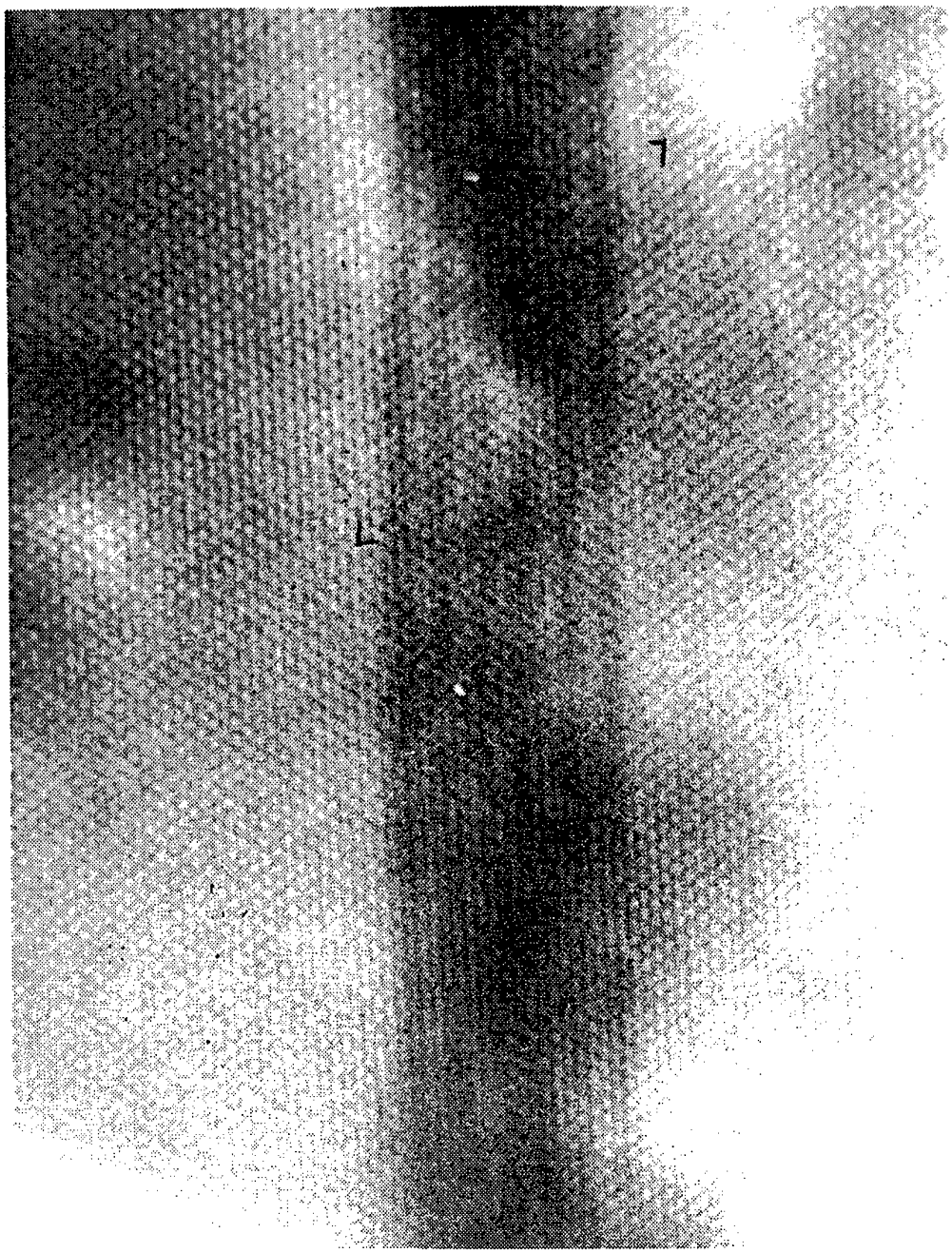

FIG. 6

INTERNATIONAL SEARCH REPORT

International Application No PCT/US 93/04670

I. CLASSIFICATION OF SUBJECT MATTER (if several classification symbols apply, indicate all)[6]

According to International Patent Classification (IPC) or to both National Classification and IPC

Int.Cl. 5 H01S3/19

II. FIELDS SEARCHED

Minimum Documentation Searched[7]

Classification System	Classification Symbols
Int.Cl. 5	H01S

Documentation Searched other than Minimum Documentation to the Extent that such Documents are Included in the Fields Searched[8]

III. DOCUMENTS CONSIDERED TO BE RELEVANT[9]

Category °	Citation of Document,[11] with indication, where appropriate, of the relevant passages[12]	Relevant to Claim No.[13]
X	APPLIED PHYSICS LETTERS. vol. 60, no. 17, 27 April 1992, NEW YORK US pages 2045 - 2047 H. JEON ET AL. 'Blue and green diode lasers in ZnSE-based quantum wells' see the whole document	1,2,4-7, 10,11
A	---	3,8-10, 12
A	JOURNAL OF THE OPTICAL SOCIETY OF AMERICA - B. vol. 8, no. 9, September 1991, NEW YORK US pages 1799 - 1801 W.J. WALECKI ET AL 'Characteristics of thin film ZnSe/(Zn,Cd)Se/ZnSe heterostructure waveguides in the visible and near infrared' see the whole document	1

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° Special categories of cited documents :[10]

"A" document defining the general state of the art which is not considered to be of particular relevance

"E" earlier document but published on or after the international filing date

"L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)

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"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step

"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.

"&" document member of the same patent family

IV. CERTIFICATION

Date of the Actual Completion of the International Search	Date of Mailing of this International Search Report
09 SEPTEMBER 1993	22 SEP. 1993
International Searching Authority	**Signature of Authorized Officer**
EUROPEAN PATENT OFFICE	CLAESSEN L.M.

Form PCT/ISA/210 (second sheet) (January 1985)

III. DOCUMENTS CONSIDERED TO BE RELEVANT (CONTINUED FROM THE SECOND SHEET)

Category °	Citation of Document, with indication, where appropriate, of the relevant passages	Relevant to Claim No.
A	JOURNAL OF CRYSTAL GROWTH. vol. 117, no. 1/4, February 1992, AMSTERDAM NL page 1077 K.P. O'DONNELL ET AL 'Are localised excitons responsible for lasing in disordered short period II-VI strained layer superlattices?' see the whole document	1,3,8
A	APPLIED PHYSICS LETTERS. vol. 59, no. 11, 9 September 1991, NEW YORK US pages 1293 - 1295 H.JEON ET AL. 'Low threshold pulsed and continuous-wave laser action in optically pumped (Zn,Cd)Se/ZnSe mwq lasers in the blue-green' see the whole document	1,12
A	APPLIED PHYSICS LETTERS. vol. 59, no. 11, 9 September 1991, NEW YORK US pages 1272 - 1274 M.A HAASE ET AL 'Blue green laser diodes' see the whole document	1-12