

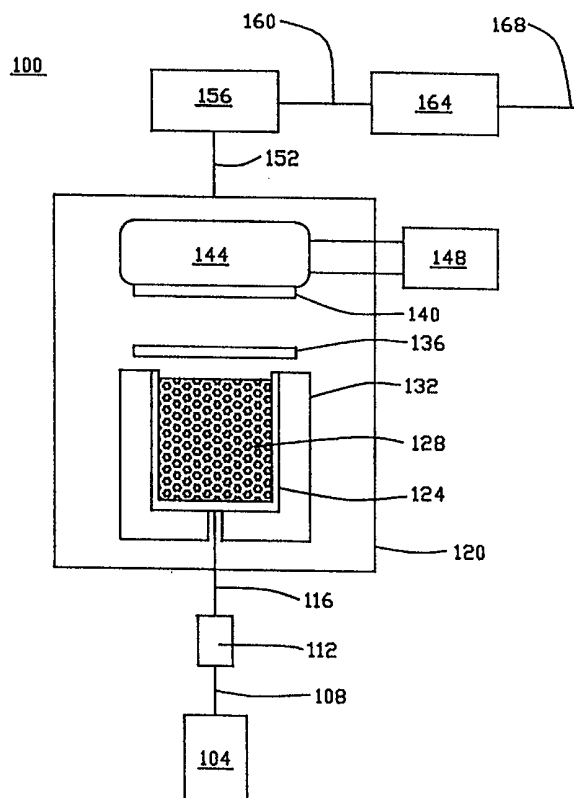


INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁶ : C09D 4/00, 4/06, 127/12, 129/10, 183/06, 201/04, B32B 7/02, C23C 4/00, 16/00, 26/00	A1	(11) International Publication Number: WO 99/21924 (43) International Publication Date: 6 May 1999 (06.05.99)
(21) International Application Number: PCT/US98/21840 (22) International Filing Date: 15 October 1998 (15.10.98) (30) Priority Data: 08/957,481 24 October 1997 (24.10.97) US (71) Applicant: QUESTER TECHNOLOGY, INC. [US/US]; 47633 Westinghouse Drive, Fremont, CA 94539 (US). (72) Inventors: LEE, Chung, J.; 1673 Oro Drive, Fremont, CA 94539 (US). WANG, Hui; 38855 Litchfield Circle, Fremont, CA 94536 (US). FOGGIATO, Giovanni, Antonio; 16340 Oakwood Lane, Morgan Hill, CA 95037 (US). (74) Agents: MEYER, Sheldon, R. et al.; Fliesler Dubb Meyer and Lovejoy LLP, Suite 400, Four Embarcadero Center, San Francisco, CA 94111-4156 (US).		(81) Designated States: AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CU, CZ, DE, DK, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, UZ, VN, YU, ZW, ARIPO patent (GH, GM, KE, LS, MW, SD, SZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG). Published <i>With international search report.</i>

(54) Title: PRECURSORS FOR MAKING LOW DIELECTRIC CONSTANT MATERIALS WITH IMPROVED THERMAL STABILITY**(57) Abstract**

Fluorinated chemical precursors, methods of manufacture, polymer thin films with low dielectric constants, and integrated circuits comprising primarily of sp^2C-F and some hyperconjugated sp^3C-F bonds are disclosed in this invention. Precursors are disclosed for creating fluorinated silanes and siloxanes, and fluorinated hydrocarbon polymers. Thermal transport polymerization (TP), chemical vapor deposition (CVD), plasma enhanced CVD (PECVD), high density PECVD (HDPECVD), photon assisted CVD (PACVD), and plasma-photon assisted (PPE)CVD and PPETP of these chemicals provides thin films with low dielectric constants and high thermal stabilities for use in the manufacture of integrated circuits.



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PRECURSORS FOR MAKING LOW DIELECTRIC CONSTANT MATERIALS WITH IMPROVED THERMAL STABILITY

5 **Cross References**

Lee et al., Chemicals and Processes for Making Fluorinated Poly(Para-Xylylenes). Attorney Docket No.: QTII 8021 SRM/DBB.

Lee et al., New Deposition Systems and Processes for Transport Polymerization and Chemical Vapor Deposition. Attorney Docket No.: QTII 8022 SRM/DBB.

10 Lee et al., Low Dielectric Constant Materials with Improved Thermal and Mechanical Properties. Attorney Docket No.: QTII 8023 SRM/DBB.

Lee et al., Low Dielectric Constant Materials Prepared from Photon or Plasma Assisted Chemical Vapor Deposition and Transport Polymerization of Selected Compounds. Attorney Docket No.: QTII 8024 SRM/DBB.

15 All of the above co-pending applications are herein incorporated fully by reference.

Field of the Invention

20 This invention relates to the precursors for manufacturing dielectric materials with low dielectric constants for use in the manufacture of semiconductor integrated circuits. The invention also relates to the polymers made from these precursors, the processes used to make polymers, and the integrated circuits made from these polymers.

Background of the Invention

25 As integrated circuits (ICs) have become progressively more microminiaturized to provide higher computing speeds, the low dielectric constant polymers used in the manufacturing of the ICs have proven to be inadequate in several ways. Specifically, they have not had sufficient thermal stability, generate toxic byproducts, are inefficient to manufacture, and the dielectric constants are too high.

30 During the past few years, several types of precursors have been used to manufacture polymers with low dielectric constants for use in manufacture of integrated

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circuits (IC). Transport Polymerization (TP) and Chemical Vapor Deposition (CVD) methods have been used to deposit low dielectric materials. The starting materials, precursors and end products fall into three groups, based on their chemical compositions. The following examples of these types of precursors and products are
5 taken from Proceedings of the Third International Dielectrics for Ultra Large Scale Integration Multilevel Interconnect Conference (DUMIC), Feb. 10-11 (1997).

I. Modification of SiO₂ by Carbon (C) and Fluorine (F)

The first method described is the modification of SiO₂ by adding carbon and/or
10 fluorine atoms. McClatchie et al., *Proc. 3d Int. DUMIC Conference*, 34-40 (1997) used methyl silane (CH₃-SiH₃) as a carbon source, and when reacted with SiH₄ and the oxidant H₂O₂ and deposited using a thermal CVD process, the dielectric constant (K) of the resulting polymer was 3.0. However, this K is too high to be suitable for the efficient miniaturization of integrated circuits.

15 Sugahara et al., *Proc. 3d Int. DUMIC Conference*, 19-25 (1997) deposited the aromatic precursor, C₆H₅-Si-(OCH₃)₃ on SiO₂ using a plasma enhanced (PE) CVD process that produced a thin film with a dielectric constant K of 3.1. The resulting polymer had only a fair thermal stability (0.9% weight loss at 450° C in 30 minutes under nitrogen). However, the 30 min heating period is shorter than the time needed to
20 manufacture complex integrated circuits. Multiple deposition steps, annealing, and metalizing steps significantly increase the time during which a wafer is exposed to high temperatures. Thus, this dielectric material is unsuitable for manufacture of multilevel integrated circuits.

Shimogaki et al., *Proc. 3d Int. DUMIC Conference*, 189-196 (1997) modified
25 SiO₂ using CF₄ and SiH₄ with NO₂ as oxidant in a PECVD process. The process resulted in a polymer with a dielectric constant of 2.6, which is lower than that of SiO₂.

However, one would expect low thermal stability due to low bonding energy of sp³C-F and sp³C-Si bonds (BE = 110 and 72 kcal/mol., respectively) in the film. The low thermal stability would result in films which could not withstand the long periods at
30 high temperatures necessary for integrated circuit manufacture.

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II. Amorphous-Carbon (α C)- and Fluorinated Amorphous Carbon (F- α C)-Containing Low Dielectric Materials

The second approach described involves the manufacture of α -carbon and α -fluorinated carbon films. Robles et al., *Proc. 3d Int. DUMIC Conference*, 26-33 (1997) used various combinations of carbon sources including methane, octafluorocyclobutane and acetylene with fluorine sources including C_2F_6 and nitrogen trifluoride (NF_3) to deposit thin films using a high density plasma (HDP) CVD process.

The fluorinated amorphous carbon products had dielectric constants as low as 2.2 but had very poor thermal stability. These materials shrank as much as 45% after annealing at 350° C for 30 minutes in nitrogen.

One theory which could account for the low thermal stability of the fluorinated amorphous carbon products is the presence of large numbers of sp^3C-F and sp^3C-sp^3C bonds in the polymers. These bonds have a bonding energy of 92 kcal/mol. Thus, the films can not withstand the long periods of high temperatures necessary for IC manufacture.

III. Fluorinated Polymers

The third approach described uses fluorinated polymers. Kudo et al., *Proc. 3d Int. DUMIC Conference*, 85-92 (1997) disclosed polymers made from C_4F_8 and C_2H_2 with a dielectric constant of 2.4. The polymers had a Tg of 450° C (Kudo et al., *Advanced Metalization and Interconnect Systems for ULSI Applications*; Japan Session, 71-75 (1996)).

However, despite its low leakage current due to presence of sp^3C-F bonds, a low thermal stability can be expected due to presence of sp^3C-F and sp^3C-sp^3C bonds in the films. Thus, like the F- α C-containing polymers discussed above, these fluorinated polymers are unable to withstand the prolonged high temperatures necessary for IC manufacture.

LaBelle et al, *Proc. 3d Int. DUMIC Conference*, 98-105 (1997) made $CF_3-CF(O)-CF_2$ polymers using a pulsed plasma CVD process, which resulted in a polymer film with a dielectric constant of 1.95. However, in spite of the low K, these polymer

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films would be expected to have low thermal stability due to presence of sp^3C-sp^3C and sp^3C-O bonds in these films which have bonding energies of 85 kcal/mol.

Therefore, none of the previously described low dielectric materials have suitably low K and high thermal stability necessary for IC manufacturing.

5 Wary et al, (*Semiconductor International*, June 1996, 211-216) used the precursor, ($\alpha, \alpha, \alpha^1, \alpha^1$) tetrafluoro-di-p-xylylene) or $\{-CF_2-C_6H_4-CF_2-\}_2$ Parylene AF-4TM, which contains a non-fluorinated aromatic moiety, and a thermal CVD process to manufacture Parylene AF-4TM which has the structural formula: $\{-CF_2-C_6H_4-CF_2-\}_n$. Films made from Parylene AF-4TM have a dielectric constant of 2.28 and have increased
10 thermal stability compared to the above-mentioned dielectric materials. Under nitrogen atmosphere, a polymer made of Parylene AF-4TM lost only 0.8 % of its weight over 3 hours at 450° C.

 However, in spite of the advantages of conventional poly(para-xylylenes), there are disadvantages of the known methods of their manufacture. First, the manufacture of
15 their precursors is inefficient because the chemical reactions have low yields, and the process is expensive and produces toxic byproducts. Further, it is difficult to eliminate redimerization of the reactive intermediates. When deposited along with polymers, these dimers decrease the thermal stability and mechanical strength of the film.

 Thus, the prior art contains no examples of dielectric material precursors for
20 semiconductor manufacture which have desired properties of low dielectric constant, high thermal stability, and low cost.

Summary of the Invention

 The present invention is directed to overcoming the disadvantages of the prior
25 art.

 Accordingly, one object of the invention is to provide precursor materials which can be used to manufacture products including polymers with low dielectric constants for IC manufacture.

 Another object of the invention is to provide precursors which can be
30 manufactured into products which have high thermal stability.

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Yet another object of the invention is to provide precursors which can be polymerized as thin layers on a substrate.

An additional object of the invention is to provide precursor materials which are inexpensive.

5 A further object is to provide materials which can be made into products with high efficiency.

An additional object of the invention is to provide precursors which can be made into dielectric materials which can be easily and accurately shaped after manufacture.

10 The invention includes novel precursors containing a fluorinated silane, a fluorinated siloxane or a fluorocarbon each containing a fluorinated aromatic moiety. The precursors are suitable for making polymers with low dielectric constants and high thermal stability. The polymers can be used for making integrated circuits.

15 Additionally, the invention includes methods for making polymers for integrated circuit manufacture using novel fluorinated silanes, fluorinated siloxanes, or fluorocarbons, each containing a fluorinated aromatic moiety.

Furthermore, the invention includes integrated circuits comprising low dielectric constant polymers made using fluorinated silanes, fluorinated siloxanes, or fluorocarbons, each containing a fluorinated aromatic moiety.

20 Accordingly, one aspect of the invention comprises precursors for using in manufacturing polymers with low dielectric constants which are useful in the manufacture of integrated circuits (ICs).

Another aspect of the invention comprises precursors for use in manufacturing polymers with high thermal stability which are useful in the manufacture of ICs.

25 Another aspect of the invention comprises methods for reacting the precursors and depositing them as thin films on substrates for IC manufacture.

Yet another aspect of the invention comprises the deposited thin film made using the novel precursors and methods for their reaction and deposition.

Another aspect of the invention is the integrated circuits comprised of thin films derived through the reaction and deposition of the novel precursors.

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Other objects, aspects and advantages of the invention can be ascertained from the review of the additional detailed disclosure, the examples, the figures and the claims.

Brief Description of the Figures

5 Fig. 1. Embodiment of an apparatus of this invention used for thermolytic transport polymerization of fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

 Fig. 2. Embodiment of an apparatus of this invention used for radio frequency plasma enhanced transport polymerization of fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

10 Fig. 3. Embodiment of an apparatus of this invention used for high density plasma enhanced chemical vapor deposition of fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

 Fig. 4. Embodiment of an apparatus of this invention used for photon assisted transport polymerization of fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

15 Fig. 5. Embodiment of a universal deposition system of this invention for deposition of fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

 Fig. 6. Schematic cross-section view of a thin film of this invention made of polymers derived from fluorinated silanes, fluorinated siloxanes, and fluorocarbons.

20 Detailed Description Of The Preferred Embodiment

I. Precursors for Making Fluorinated Materials With Low Dielectric Constant

 This invention discloses novel precursors for making three categories of
25 fluorine-containing low dielectric materials. These precursors include fluorinated silanes, fluorinated siloxanes, and fluorocarbons. The dielectric materials are useful in the manufacture of semiconductors for integrated circuits and other electronic devices. Manufacture of smaller and faster integrated circuits requires intermetal dielectric (IMD) and interlevel dielectric (ILD) materials which minimize the communication of
30 electrical signals between adjacent conductive lines, referred to as the interconnects.

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Low dielectric materials are useful to minimize this "cross-talk" within and between layers of integrated circuits.

The polymers prepared from the precursors of the present invention are rigid and contain a high degree of substitution of hydrogen atoms by fluorine atoms. In these polymers, the fluorine in the aromatic ring provides the low dielectric constant (K) and molecular rigidity. This rigidity is reflected by high glass transition temperature (T_g), high elastic modulus (E) and high shear modulus (G). The elastic modulus is above about 2, and preferably is about 3.

One theory for the thermal stability of the poly(para-xylylenes) is that the higher bonding energies of the sp²C=sp²C, sp²C-H and sp²C-sp³C bonds of 145, 111 and 102 kcal/mol. respectively. In addition, the sp³C-F bonds may also be involved in hyperconjugation with sp²C=sp²C double bonds of the adjacent phenylene groups in Parylene AF-4TM. This hyperconjugation renders a higher bond energy for the sp³C-F bonds than that found in non-hyperconjugated sp³C- bonds. (See A. Streitwieser et al., Introduction to Organic Chemistry, Appendix II, University of California Press, Berkeley, California (1992)) (Table 1).

Table 1

Energies of Carbon Bonds In Dielectric Polymers

Bond Type:	sp ³ C-Si	sp ² C-Si	sp ³ C-sp ³ C	sp ² C-H	sp ² C-F	sp ³ C-H
Bond Energy (kcal/mol)	72	92	92	111	126	86
Bond Type:	sp ³ C-O	sp ² C-sp ³ C	sp ² =sp ² C	sp ³ C-F	sp ³ Si-F	
Bond Energy (kcal/mol)	85	102	145	110	135	

Thus, carbon atoms bonded to other atoms by with sp²C=sp²C, sp²C-F and hyperconjugated sp³C-F bonds confer advantages to polymers containing them, whereas other types of bonds (such as sp³C-H and sp³C-C bonds) do not confer these advantages.

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The $sp^2C=sp^2C$ and other sp^2C bonds increase the mechanical strength and increase T_d of the polymer. The presence of fluorine atoms in the aromatic moieties of the polymers of this invention decreases the dielectric constant, and the sp^2C-F and hyperconjugated sp^3C-F bonds confer greater thermal stability to a polymer containing them. In contrast,
5 polymers which do not contain these types of bonds have lower thermal stability and higher dielectric constant.

As one can see from Table 1, although precursors containing $sp^2C=sp^2C$ bonds have high thermal stability (bond energy = 145 kcal/mol), it is not desirable to use any non-fluorinated compound that consists of these chemical bonds as starting material.
10 Using these non-fluorinated compounds as starting material for PECVD will result in high thermal stability, however, the resulting polymer will also have an high dielectric constant ($K > 2.5$). Therefore, materials with low dielectric constants, high elastic and shear moduli (E and G , respectively) and high thermal stability (T_g and T_d) can be obtained only if the fluorinated compounds of this invention are used.

15 The above conclusions are derived from calculations using an extended "Quasi-Lattice theory" (C. Lee, *JMS Rev. Macromol. Chem. Phys.*, C29(4):431-560 (1989), incorporated herein fully by reference, and a "topological approach" employed by J. Bicerano, *Prediction of Polymer Properties*, Second Edition, Marcel Dekker, Inc., New York (1996), incorporated herein fully by reference, indicated that low dielectric
20 constant materials with high thermal stability and high rigidity can be achieved with precursors containing sp^2C-F and sp^2C bonds.

Although the above calculations using cohesive energy may account for the mechanical and electrical properties of dielectric materials of the present invention, other theories may also explain the observations. Accordingly, the instant invention is not
25 limited to any particular theory which may explain the desired properties of the herein disclosed dielectric precursors.

Therefore, the precursors of this invention contain both single and conjugated double bonds. Additionally, hyperconjugation of sp^3C-F with sp^2C bonds results in increased bond strength of the sp^3C-F compared to other sp^3C-F bonds (such as in
30 polytetrafluoroethylene) which are not hyperconjugated.

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However, although precursors which contain double bonded carbon atoms without fluorine ($\text{sp}^2\text{C}=\text{sp}^2\text{C}$) have desirable Td, Tg, E, and G, the dielectric constant K is too high to be useful in the newer, smaller integrated circuits. Thus, the invention comprises incorporating fluorine atoms bonded to the double bonded carbon atoms in the precursor molecules ($\text{sp}^2\text{C-F}$) into dielectric materials.

The precursors comprise compounds containing $\text{sp}^2\text{C-F}$ bonds and functional groups which can be cleaved or oxidized using a transport polymerization (TP) or chemical vapor deposition (CVD) process to yield reactive intermediates which spontaneously polymerize on the wafer substrate. Both linearly conjugated polymers and aromatic moieties can be used, but preferred moieties are aromatic. Also preferred are multiple-ring aromatic moieties, and the most preferred are single-ring aromatic moieties such as phenyl rings (e.g., $-\text{C}_6\text{H}_4-$, $-\text{C}_6\text{H}_5$).

A single fluorine atom incorporated into an aromatic moiety decreases the dielectric constant of that moiety, and further increasing the number of fluorine atoms in the aromatic moiety further decreases K. Thus, the general formula for mono-functional aromatic precursors of the invention is: $-\text{C}_6\text{H}_{5-n}\text{F}_n$, wherein n is an integer selected from the group of 1, 2, 3, 4 or 5. Similarly, the general formula for di-functional aromatic precursors is: $-\text{C}_6\text{H}_{4-n}\text{F}_n-$, wherein n is an integer selected from the group of 1, 2, 3 or 4. The most preferred embodiments are aromatic moieties which have maximal substitution of $\text{sp}^2\text{C-H}$ hydrogen atoms by $\text{sp}^2\text{C-F}$ fluorine atoms. Thus, for a precursor containing only one functional group, the mono-fluorinated phenyl group $-\text{C}_6\text{H}_4\text{F}$ works, but multi-fluorinated phenyl groups are preferred, and $-\text{C}_6\text{F}_5$ is most preferred. For a precursor containing two functional groups, the mono-fluorinated phenyl group $-\text{C}_6\text{H}_3\text{F}-$ works, but multi-fluorinated phenyl groups are preferred, and $-\text{C}_6\text{F}_4-$ is most preferred.

A. Fluorocarbon-Modified SiO_2 : Fluorinated Silanes and Fluorinated Siloxanes

The first category of precursors consists of precursors for making fluorocarbons, fluorinated silanes and fluorinated siloxane. These precursors are useful for modifying SiO_2 by incorporating Si-C bonds, Si-F bonds, and/or fluorinated

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aromatic bonds. Incorporation of these types of bonded moieties into SiO₂ can lower the dielectric constant, but with only small decreases of the T_d, T_g, E, and G of the polymers from which they are made.

Precursors of fluorinated silanes in this invention have the general structural formula: (C₆H_{5-n}F_n)_m-SiH_{4-m}, wherein n is 1, 2, 3, 4 or 5 and n is an integer of 1, 2, 3 or 4. The preferred fluorinated silane of the present invention is C₆F₅-SiH₃.

Precursors of fluorinated siloxanes of this invention have the general structural formula: (C₆H_{5-n}F_n)_m-Si(OCH₃)_{4-m}, wherein n is 1, 2, 3, 4 or 5, and m is 1, 2, or 3. The preferred fluorinated siloxane of the present invention is (C₆F₅)-Si(OCH₃)₃.

Precursors of fluorinated hydrocarbons have the general structural formula: CH_{3-n}F_n-C₆H_{4-p}F_p-CH_{3-m}F_m, wherein n and m are 1, 2 or 3, and p is 1, 2, 3, or 4. Preferred fluorinated hydrocarbon precursors are CF₃-C₆F₄-CF₃ and CHF₂-C₆F₄-CHF₂.

There are many isomers with these above general formulas, and all are considered part of the present invention.

Table 2

**Precursors and Methods for Manufacturing
Low Dielectric SiO₂ Derivatives**

<u>C & F-source</u>	<u>Si Source</u>	<u>Oxidant</u>	<u>TP and CVD Processes</u>
C ₆ F ₅ -SiH ₃	SiH ₄	H ₂ O ₂ , NO ₂	Thermal
C ₆ F ₅ -Si(OCH ₃) ₃	-	-	Plasma Enhanced
CF ₃ -C ₆ F ₄ -CF ₃	SiH ₄	-	Plasma Enhanced

Table 2 shows the precursors, other reactants and TP and CVD processes used to manufacture fluorine-substituted aromatic SiO₂ derivatives of this invention. Although the fully fluorinated derivatives are described, mono-, di-, tri-, and tetra-fluorinated aromatic moieties, as appropriate, can also be used.

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B. Precursors for Making Fluorinated Amorphous Carbon- and Polymer-Containing Materials With Low Dielectric Constant

These precursors consist primarily of $sp^2C=sp^2C$, sp^2C-F and/or
 5 hyperconjugated sp^3C-F bonds. Unlike the precursors used for making silanes and siloxanes, no Si source is needed.

Precursors for fluorinated polymers with one aromatic ring and one sp^2C-sp^3C-F type bond have the general formula: $(C_6H_{5-n}F_n)-CH_{3-m}F_m$, where n is an integer of 1, 2, 3, 4, or 5, and m is an integer of 1, 2, or 3. A preferred precursor of a fluorinated
 10 polymer is $C_6F_5-CF_3$.

Precursors for fluorinated polymers with one aromatic ring and two sp^2C-sp^3C-F type bonds have the general formula:
 $(CH_{3-n}F_n)-(C_6H_{4-p}F_p)-(CH_{3-m}F_m)$, wherein n and m are integers selected from the group consisting of 1, 2 and 3, and p is an integer selected from the group consisting of 1, 2, 3,
 15 and 4. The precursors of fluorinated polymers which are commercially available include $CF_3-C_6F_4-CF_3$ and $CHF_2-C_6F_4-CHF_2$.

Precursors of polymers containing one fluorinated aromatic residue and an additional conjugated sp^2C carbon bond have the general structural formula: $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, where n is an integer of 1, 2, 3, 4, or 5, m is an integer of 0 or 1, and
 20 p is an integer of 0, 1, or 2. The preferred precursor of this group is $C_6F_5-CF=CF_2$.

There are many isomers with these above general formulas, and all are considered part of the present invention.

Table 3

25

Precursors for Making Low Dielectric Hydrocarbon Polymers

<u>C & F-sources</u>	<u>Primary α-C Source</u>	<u>TP or CVD Process</u>
$C_6F_5-CF_3$	CH_4	High Density Plasma
30 $C_6F_5-CF=CF_2$	CH_4	High Density Plasma
$HCF_2-C_6F_4-CF_2H$		Thermal or Photon Assisted
$CF_3-C_6F_4-CF_3$		Photon Assisted

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II. Deposition of Low Dielectric Materials

The invention comprises new methods and precursors for depositing low dielectric materials. These new methods and precursors have advantages over conventional methods.

5 During conventional manufacture of semiconductors using spin-on glass (SOG) methods, defects in the dielectric layers can form, especially in locations where there is a channel between metal lines. Such defects in the dielectric layers result in the formation of cracks or "voids" in the dielectric material. These voids and cracks result in trapping moisture, etching gas, or photoresist contaminants and ultimately lead to the degradation and loss of device reliability. Therefore, the present invention provides precursors which
10 can be deposited using chemical vapor deposition (CVD) and transport polymerization (TP) to avoid the problem.

 In CVD, a precursor is placed directly on the substrate which will have the thin film of low dielectric material applied. In his "hot chuck" method, the wafer and
15 precursor are exposed to an energy source such as heat, plasma, or electromagnetic radiation to "dissociate" or "crack" the precursor molecule directly on the wafer to form the reactive intermediate. The reactive intermediate molecules then polymerize with each other to form the thin film.

 In TP, the cracking step is performed in a chamber other than where the
20 deposition step is performed. This "cold-chuck" method confers several advantages. First, the cracking efficiency can be optimized by selecting appropriate conditions of plasma density, photon energy, or temperature. Second, the density of the intermediate molecules can be regulated to minimize the formation of unwanted side products, such as re-formed precursors. Next, because the wafer is not exposed to the harsh conditions
25 of precursor dissociation, fragile structures containing aluminum or other thermally sensitive materials are less likely to become damaged. Deposition of polymer layers of a few nanometers (nm) to >7000 nm can be achieved, along with eliminating or greatly reducing the number of voids. In principle, layers of polymer of molecular dimensions can be deposited. Thus, the invention provides the novel precursors for the deposition of

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void-free dielectric films. The type of functional group used determines the optimal type of TP or CVD process should be used. These will be discussed below.

5 **A. General Processes for the Manufacture of Fluorinated Aromatic
 Derivatives of Silanes and Siloxanes Using Thermal Transport
 Polymerization and Chemical Vapor Deposition**

10 For preparations of the low K dielectrics of this embodiment according to the
 Table 2, the concentration of the fluorinated aromatic silane is in the range of from
 about 5% to 100%, and is preferably 20%. The fluorinated aromatic siloxane is used in
15 the range of from about 5% to about 100%, and is preferably 20%. In a plasma
 enhanced chemical vapor deposition (PECVD) and photon-assisted transport
 polymerization (TP) processes, no oxidants are required, but in a thermal TP or thermal
 CVD process an oxidant such as H_2O_2 or NO_2 is added to oxidize the Si atoms of SiH_4
20 or other silane. Other suitable oxidants are oxygen containing organic or inorganic
 compounds, such as oxalic acid. When H_2O_2 is used, its concentration should be in the
 range of about 10% to about 50%, and is preferably about 30%.

 Thermal TP is carried out in a chamber 100 shown in Figure 1. The precursors
 are contained within a precursor tank 104, flow through a pipe 108 into a mass flow
25 controller 112 from which they are controllably released through another pipe 116 into
 the chamber 120 within which is a container 124 with a cracking device that may
 comprise a catalyst 128. The precursors are heated by a conventional resistive heater
 132 to generate the reactive intermediate radicals. Rates of precursor flow range from
 0.2 SCCM to 100 SCCM, preferably between 2 and 5 SCCM, and most preferably at 3
30 SCCM. Temperatures of reactions without catalysts should range from about 700° C
 to about 800° C, preferably are in the range of from about 700° C to about 750° C, and
 most preferably at about 750° C. For reactions in the presence of catalysts,
 temperatures can be as low as about 350° C. The pressure in the chamber 120 needed
 for the cracking reaction should be in the range of about 1 milliTorrr to about 500 Torr,
 and is preferably about 10 milliTorrr. After cracking, the intermediates flow to a
 diffusion plate 136 where the bulk flow is diverted away from the wafer 140. The

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diffusion plate 136 can be positioned as desired within the chamber, to optimize the pattern of flow of intermediates to the wafer. The intermediates then diffuse to the wafer 140, which is held on a cold chuck 144 which is maintained at a temperature lower than the cracking device by a chiller 148. Temperatures are maintained by any conventional cooling method including liquid nitrogen or reverse Peltier methods. Temperatures of the chuck should be in the range of about -120° C to about 300° C are useful, preferred temperatures are in the range of about -40° C to about 100° C, and the most preferred temperature is about -20° C. The chamber 120 is connected via a pipe 152, to a cold trap 156, and another pipe 160 connects the trap to a pump 156 to maintain low pressure in the chamber 120. The cold trap 156 protects the pump 160 from deposition of precursors and intermediates in the chamber 120.

1. Use of Catalysts in Thermal TP and Thermal CVD

Thermal TP and CVD processes requiring catalysts to dissociate precursor molecules into reactive intermediates can use any conventional catalyst. An ideal catalyst useful for this invention should provide high reactivity, high selectivity, long process life cycle, high recycle capability, and less severe pressure and temperature requirements. It should be inexpensive, safe for human handling, and should be environmentally friendly. The catalyst should crack or cleave the Si-H and crack or oxidize the C-H bonds. Further, the catalyst should not add any metal or metal compound, or reactive ion such as F⁻ into the dielectric film during deposition. Serious reliability problems occur when a metal contaminant resides within the dielectric materials. Other serious problems occur when a highly reactive ion such as F⁻ is introduced into the film. The ion can break interatomic bonds within the hydrocarbon moieties of the dielectric material, resulting in lowered mechanical strength.

Catalysts that are useful for this invention include dehydrogenation catalysts, reforming catalysts, and oxidative dehydration catalysts.

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a. Dehydrogenation Catalysts

The temperatures and times needed to complete pyrolysis can be reduced by employing a catalyst in the chamber. An ideal catalyst useful for this invention should provide high reactivity, high selectivity, long process life cycle, high recycle capability, and less severe pressure and temperature requirements. It should be inexpensive, safe for human handling, and should be environmentally friendly. The ideal catalyst should crack or cleave the Si-H and C-H bonds without cracking or cleaving the C-F bonds. Further, the catalyst should not add any metal or metal compound into the dielectric film during deposition. Serious reliability problems occur when a metal contaminant resides within the dielectric materials.

To assist the cracking of the SiH and C-H bonds, any commonly used dehydrogenation catalyst is suitable. These catalysts are also called "protolytic cracking catalysts", or "oxidative dehydrogenation catalysts", in petroleum processing. Additionally, most "dehydrocyclization catalysts" and some of the "aromatization catalysts" for hydrocarbon processing are also useful for this invention, because aromatization normally involves dehydrogenation.

Potassium ferrite (KFeO_2) on iron oxide is an example of a suitable catalyst which is commercially available. The ferrite commonly comprises a promoter that may contain a salt of oxide of a Group (IIA) metal, such as Mg, Ca, or Sr, and a Group VB or VIB metal, such as V, Nb, Cr, Mo, or W. [See J. I. Krochiwitz ed., *Encyclopedia of Chemical Technology, 4th edition, Catalysis and Catalysts*, Vol. 5: 320 (1991), incorporated herein fully by reference.] These catalysts can be useful at temperatures up to about 600° C. Variations of these catalysts are BASF's Lu-144FTM and Shell 105TM catalysts, and catalysts for the dehydrogenation of ethylbenzene. These include those produced by Monsanto-Combustion Engineering-Lumis, Union Carbide-Cosden-Badger, and Societe-Chimique des Charbonnages.. [See J. J. McKetta, Ed., *Encyclopedia of Chemical Processing and Designs: Dehydrogenation*, Vol. 14:276, Marcel Dekker Inc. (1992), incorporated herein fully by reference.]

Other industrial catalysts include Cu and Zn oxides on alumina and Cu, Ag or Cu-Ag alloy in the form of gauge or as metal deposited on a low surface area support

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such as kaolin, clay and active carbon. Other supports or carriers can include asbestos, pumice, kiesselguhr, bauxite, CuO, Cr₂O, MgCO₃, ZrO₂, and Zeolites. These catalysts are active by virtue of an oxide layer on the metals, and are used for hydrogen generation from methanol. Catalysts consist of copper chromite, bismuth molybdate, iron molybdate, or tin phosphate on similar supports are also useful. [See J. I. Krochiwitz ed., *Encyclopedia of Chemical Technology*, 4th edition, Catalysis and Catalysts, Vol. 5: 320 (1991), incorporated herein fully by reference; J. J. McKetta, Ed., *Encyclopedia of Chemical Processing and Designs: Dehydrogenation*, Vol. 14:276, Marcel Dekker Inc. (1992), incorporated herein fully by reference.]

b. Reforming Catalysts

In addition to dehydrogenation catalysts, reforming catalysts used in petroleum reforming processes can also be used. A first group of these include transition metal oxides, such as V₂O₅, MoO₃, WO₃ and Cr₂O₃ in bulk form or preferred on a non-acid support such as silica, neutral alumina or active carbon. [See Meriaudeau and Naccache, *Cat. Rev.-Eng. Sci.* 39(1&2):5-48 (1997), incorporated herein fully by reference. Typically useful catalysts include Shell 205TM, which consists of 62.5% Fe₂O₃, 2.2% Cr₂O₃, and 35.3% K₂CO₃, and Dow Type BTM catalyst, which consists of calcium and nickel phosphates promoted with a small amount of chromium oxide.

Another group of reforming catalysts useful for dehydrogenation include noble metals on acid supports. The most commonly used catalysts are Pt (0.3 to 0.7%) and Pt/Re on a chlorided (acidified) alumina (e.g., γ - or η -Al₂O₃). The bimetallic Pt/Re-alumina is preferred for its longer life time. In addition, Pt, Ga or An modified H-ZSM-5TM, or Pt on medium-pore zeolite support such as In-ZSM-5TM is also very effective.

Other, multimetallic reforming catalysts include Pt/Re catalysts of the above including lesser amounts of Ir, Ga, Ge, Sn or Pb supported by chlorided alumina. The catalysts typically have surface areas ranging from 170 m²/g to 300 m²/g and pore volumes ranging from 0.45 cm³/g to 0.65 cm³/g. [See J. I. Krochiwitz ed., *Encyclopedia of Chemical Technology*, 4th edition, Catalysis and Catalysts, Vol. 5: 320 (1991), incorporated herein fully by reference.] Additionally useful catalysts can

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also be found in the OJG International refining catalyst compilation-1987 (J.J. McKetta ed., *Encyclopedia of Chemical Processing and Designs: Petroleum Processing, Catalyst Usage*, Vol 35:87-89 Marcel Dekker (1992), incorporated herein fully by reference.] These catalysts comprise active agents such as Pt/ReCl, Ni, PtCl and other
5 rare earth metals on alumina and zeolites.

Other useful catalysts in this invention include (1) noble metals or metal sulfide on active carbon, (2) Ga₁₃, Cr₁₂, GaAl₁₂ & Al₁₃ on PILCs, (3) M-Al₂O₃ with M = lanthanides, (4) Al₂O₃ kneaded with M, where M is Bi & Sb compounded with periodic
10 table Group VIB & VIIB metals, (5) M-modified H-ZSM-5 and H-ZSM-11 where M is Zn, Ga, Pt-Ga, Pt-Na, Mo, Cr, K, Ca, Mg, Al, and Group VIII metals, (6) M-modified MFI (H-GalloSilicates) where M is Si/Ga, Na/Ga, Al, (7) rare earth metal exchanged Y-zeolites or ultra stable Y-zeolites, (8) Ti oxide paired with Zr oxide, (9) M plated onto
aluminum, where M is Ni, and Ni, Cr, and Al alloys.

Pure dehydrogenations are endothermic by 15 to 35 kcal/g-mol. and hence have
15 high heat requirements. The above catalysts are normally used at temperatures ranging from 300° C to 600° C depending on the residence time of the chemicals in the reactor. The effective temperature for some of these catalysts can be lowered by adding free radical initiators such as I, Br, H₂O, sulfur compounds or oxygen and their mixtures. However, special care should be taken to avoid reaction of desirable radicals with free
20 radicals generated from these initiators. This can be achieved by providing large mean free paths for these reactants in the reactor, reducing residence time and the adjustment of wafer temperatures to avoid condensation of low mass free radicals.

2. Loss Of Catalyst Function

25 With time, catalysts may lose reactivity due to changing their oxidative state or coke formation. The life time of the catalysts can be increased at high operating temperatures or high partial pressure of hydrogen. If catalysts lose activity by coke accumulation, they can be regenerated by careful oxidation followed by reduction with hydrogen before being returned to service. [See: J.J. McKetta ed., *Encyclopedia of*
30 *Chemical Processing and Designs: Catalysis and Catalysts* Vol. 6:420; *Petroleum*

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Processing, Catalyst Usage, Vol 35:89 Marcel Dekker, Inc. (1992), incorporated herein fully by reference.

After leaving the cracking device, the intermediates pass through a diffusion plate to disperse the intermediates evenly over the wafer surface. The intermediates
5 deposit upon the wafer, which is held on a chuck, which, in turn, is connected to a chiller to maintain a temperature of the chuck and wafer below the temperature of the chamber and the intermediates. The temperature of the cold chuck is maintained in the range of about -30° C to about +20° C, preferably at -20° C. Low pressure in the system is maintained by a vacuum pump with sufficient capacity to maintain the desired pressure
10 within the CVD system. Condensation of precursors and intermediates on the pump is minimized by a trap placed between the deposition chamber and the vacuum pump.

Example 1: Thermal CVD Of A Fluorinated Silane

The sp²C-F-modified SiO₂ thin films can be prepared from thermal oxidation of
15 C₆F₅-SiH₃. An admixture of 40 mole % of SiH₄ and 60 mole % of C₆F₅-SiH₃ in an aqueous solution with 30% by weight of H₂O₂ are dispensed separately through two flash evaporators onto a cold wafer inside a CVD system. The molar ratio of H₂O₂/(SiH₄+C₆F₅-SiH₃) is about 3.5. The mixtures of H₂O₂, SiH₄, and C₆F₅-SiH₃ is chilled as a thin film on the wafer. The wafer is then heated from -20° C to 500° C in
20 vacuum at a heating rate of 10° C/min. The resulting thin film has a dielectric constant of 2.7 and an initial decomposition temperature of 480° C when tested under nitrogen atmosphere.

25 B. General Methods for the Manufacture of Low Dielectric Thin Films Using Plasma TP or CVD

Plasma enhanced TP is carried out generally using methods in the art. Takai et al., *J. Appl. Phys.* 17:399-406 (1984), incorporated herein fully by reference. With low density plasma, the electron density in the plasma is in the range of about 10¹² to about
30 10¹³ electrons/cm³. Low density plasma TP and CVD can be carried out at about 100 milli Torr to about 100 Torr. High density plasma (HDP) is characterized by electron

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densities in the range of about 10^{13} to about 10^{14} electrons/cm³. High density plasma TP and CVD can be carried out at pressures of about 0.1 milliTorrr to about 100 milliTorrr.

For preparation of the materials with low dielectric constants embodying this invention, reactants such as SiH₄, or more generally siloxane such as Si(OCH₃)₄ are used. In a thermal CVD process, the presence of oxygen at high temperatures will induce oxidation, resulting in the formation of silicon dioxide (SiO₂) films having varying K based on varying the relative compositions of SiO₂ and sp²C-F bonded molecules.

1. Plasma Enhanced Transport Polymerization of Fluorinated Silanes and Fluorinated Siloxanes

To make thin films, sp²C-F-containing starting materials such as (C₆H_{5-n}F_n)_m-Si(OCH₃)_{4-m} wherein n is an integer of 1, 2, 3, 4, or 5, and m is an integer of 1, 2, or 3, (C₆H_{5-n}F_n)_m-SiH_{4-m} wherein n is an integer of 1, 2, 3, 4, or 5, and m is an integer of 1, 2, 3, or 4, C₆F₅-CF₃, or admixtures of these starting materials are delivered into a plasma enhanced transport system 200 depicted in Figure 2. Precursors are stored in a precursor holder 204. A carrier gas, typically helium, is passed through a bubbling device to volatilize the precursor. The vaporized precursors flow through a pipe 208 and to a mass flow controller 212. The mass flow controller (MFC) provides a precursor feed rate ranging from about 0.2 to about 500 standard cubic centimeter per minute (SCCM). The precursors flow from the MFC 212 through a pipe 216 and into a quartz tube 220 under low pressures in the range of from about 0.1 milliTorrr to 10 Torr, and preferably in the range from about 1 milliTorrr to about 3 milliTorrr. A carbon source such as CH₄ or C₂H₂ is provided via a second feeder (not shown) and a second MFC (not shown) with a feed rate ranging from 0 to 500 SCCM. The preferred flow rates of precursor into the plasma tube should be in the range of from about 0.2 SCCM to about 10 SCCM, and is most preferably in the range of about 0.5 SCCM to about 1 SCCM.

Pyrolyzation takes place within the quartz tube by action of a radiofrequency (RF) plasma generator 226. Effective plasma 230 can be maintained by radio

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frequencies in a range of between about 1 kHz and 2.5 GHz. A preferred range is between about 400 KHz and about 13.56 MHz. Ideally, the RF frequency is about 13.56 MHz. The RF power should be in the range of about 30 Watts to 300 Watts, preferably about 100 Watts to 250 Watts, and more preferably about 200 Watts.

5 The plasma 230 then proceeds into the deposition chamber 234 which is heated by a heater 238 to prevent deposition of precursor intermediates on the chamber walls. The pressure within the chamber 230 is maintained at a pressure between about 0.1 milliTorr and about 10 Torr. The flow of plasma is adjusted by a flow pattern adjuster 242, which can be moved within the chamber. Moving the flow pattern adjuster 242 in
10 the vertical direction adjusts the flow rate and distribution pattern of the plasma as it enters the chamber. Moving the flow pattern adjuster in the horizontal direction adjusts the distribution of plasma to different parts of the wafer 250. A gas dispersion plate 246 evens the delivery of the intermediates to the wafer 250. The wafer 250 is cooled by the chuck 254, which is cooled by liquid nitrogen, reverse Peltier effect, or other
15 conventional cooling device 258, and is maintained at a temperature in the range of about -30° C to about +20° C, and is preferably about -10° C. The deposition chamber 224 is connected via a pipe 262 to a cryogenic trap 266, which is connected via a pipe 270 to a pump 276, which maintains the pressure within the chamber at the desired level.

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2. High Density Plasma Chemical Vapor Deposition

A high density plasma deposition process can also be used to dissociate precursors. In contrast to the low density plasma process described above, in high density plasmas, the electron density is in the range of from about 10^{13} to 10^{14} electrons
25 /cm³. This process must be carried at lower pressures than conventional plasma processes. In this embodiment, a inductively coupled high density plasma apparatus 300 is shown schematically in Figure 3. A precursor delivery system 304 volatilizes or vaporizes the precursor, which flows through a pipe 308 and an anode gas injector 312 into the deposition chamber 316. The anode gas injector 312 is attached to RF
30 generators 320 which are matched by matching controllers 324. The output of the RF

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generators 320 passes through inductive coils 328 to produce an electrical field. The wafer 332 is held by a cathode electrostatic chuck 336, which is connected to the RF generator 320. IR sources 340 provide additional heating of precursors to decrease the needed plasma power and to inhibit condensation of materials on the walls of the chambers. The plasma source power needed for a wafer of 8 inch diameter is in the range of about 100 Watts to 4000 Watts, and preferably about 2000 Watts. For wafer of other sizes, the power should be adjusted accordingly. Power should range from about 1 Watt/cm² wafer surface area to about 15 Watts/cm², preferably from about 2 Watts/cm² to about 10 Watts/cm², and most preferably about 5 Watts/cm². The chamber pressure is maintained in the range of 0.01 milliTorr to 10 milliTorr, and preferably below 5 milliTorr. The wafer temperature is in the range from about 300° C to 450° C, and is preferably about 350° C.

Generally, thin films made using plasma methods contain lower carbon content than films made using thermal methods. Films with decreased carbon content are made using greater plasma power. One theory to account for this is that increasing the plasma power increases the dissociation of the precursors by more completely cracking the precursors. Decreasing the carbon content of films decreases the trapping of electrons in the film, and leads to more reliable, long lasting devices.

The deposition rate can be regulated by adjusting the flow rate of precursors into the plasma generator and thereby adjusting the flow rate of reactive intermediates over the wafer. However, as the precursor flow rate increases, the residence time within the plasma generator decreases, and this can result in less complete cracking. To overcome this problem, it is desirable to increase the plasma power as the precursor flow rate increases. This maintains the efficiency of the cracking reaction.

If desired, further increases in the efficiency of the plasma reactions can be achieved by heating the chamber. This can be done using a conventional resistive heater or using an infrared (IR) heater. Preferably, IR irradiation is used, and the wavelength of the IR radiation is chosen to maximal absorption by the precursor.

The above reaction is designed to accommodate wafers with diameters of about 200 mm. Thin films of this invention are deposited at a rate of about 1000 Å/min.

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Films deposited using this system have dielectric constants of 2.3 to 3.5, and have Td in the range of about 350° C to 500° C, depending on the fluorine content. Increasing the fluorine content decreases the dielectric constant. The thin films have no initial weight loss; Td and Tg are temperatures ranging from 450° C to 500° C under nitrogen atmosphere.

Example 2: Manufacture of a Thin Film Low Dielectric Layer Using Plasma Enhanced Transport Polymerization.

A thin film of $C_6F_5-Si(OCH_3)_3$ is made using a parallel type of plasma reactor as depicted in Figure 3. The compound is delivered into the chamber employing a bubbling device using Ar as a carrier gas. The compound is introduced at a feed rate of 50 SCCM. The operating chamber pressure is 20 milliTorr. Effective plasma is maintained by an RF frequency of 13.56 MHz at a power of 1.3 Watts/cm² applied to the upper electrode with a wafer placed on the lower electrode which is grounded. The wafer temperature is maintained at 400° C. The thin film made by this process has a dielectric constant of 2.65, and has no initial weight loss at 500° C under nitrogen atmosphere.

C. General Methods for the Manufacture of Low Dielectric Thin Films Using Photon Assisted Transport Polymerization

In addition to thermally and plasma-generated reactive intermediates, photon assisted precursor cracking is also part of this invention. Because specific chemical bonds have specific energies, and because these energies can be supplied as photons, electromagnetic radiation is a preferred method of practicing this invention.

A transport polymerization system **400** using electromagnetic radiation is shown in Figure 4. Precursors are stored in a precursor tank **404** and then flow through a pipe **408** to a mass flow controller **412**, where the precursor flow rate is regulated. Volatile precursor then passes through pipe **416** into a transparent tube **420**. For UV photolytic cracking of the precursor, tube **420** is made of quartz, preferably a single quartz crystal. For infrared (IR) cracking of the precursor, tube **420** may be made of glass. For

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vacuum ultraviolet photolytic cracking of the precursor, tube **420** is made of MgF_2 , LiF , or CaF_2 . An ultraviolet (UV) source **424** is used to photolytically dissociate the precursor. Alternatively, a vacuum ultraviolet (VUV) **426** source can be used. An infrared (IR) source **428** can be used, which heats the precursors to provide a
5 combination of thermolytic and photolytic cracking. After cracking, the intermediates pass into the deposition chamber **432**, which is heated by a resistive heater **436** to prevent deposition of intermediates on the walls of chamber **432**. The flow of intermediates is adjusted using flow pattern adjuster **440**. Moving the flow pattern
adjuster **440** in the vertical direction adjusts the flow rate of and the distribution pattern
10 of intermediates in the deposition chamber **432**. Moving the flow pattern adjuster **440** in the horizontal direction adjusts the distribution pattern of intermediates in the deposition chamber **432**. A gas dispersion plate **444** evens the delivery of the intermediates to the wafer **448**. The wafer **448** is held by a cold chuck **452**, which is maintained at low temperatures by a conventional cooling device, using liquid nitrogen, reverse Peltier
15 effect, or any other cooling apparatus known in the art. The chamber **432** is connected via a pipe **460** to a trap **464**, which is connected via another pipe **468** to a pump **472**. The pump maintains the pressure within the deposition chamber **432** at the desired level, and the trap **464** minimizes the deposition of intermediates on the pump **472**.

Using the photolytic method, the dissociation reaction can be very selective and
20 efficient if appropriate photon sources are used. The photon sources can be provided by ultraviolet (UV) light generated by mercury vapor discharge or metal halide lamps. Exemplary sources of UV radiation for transport polymerization can include (1) a mercury lamp that provides from 50 to 220 mWatts/cm² of UV ranging from 200 to 450 nm or (2) a metal halide lamp that provides from 40 to 160 mWatts/cm² of UV ranging
25 from 260 nm to 450 nm. These UV sources provide photon energies ranging from 2 to 5 eV, which are sufficient for generating many radical intermediates.

An alternative to conventional UV light is vacuum ultraviolet (VUV). Incoherent excimer radiation can provide a large number of UV and VUV wavelengths for photolytic processing of various chemicals. The preferred source is incoherent
30 excimer radiation derived from dielectric barrier discharge. UV and VUV photons that

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are in the ranges of 3 to 5 eV are especially useful. These energy levels are comparable with the bonding energies of most chemical bonds, thus are very effective for initiating photochemical reactions (see Table 4).

Table 4

Bond Energies of Selected Bonds

	Chemical Bonds	Bonding Energies (eV)
	ϕ -CH ₂ Br	2.52
10	ϕ -CH ₂ -OR	3.52
	ϕ -CH ₂ -CH ₃	3.30
	ϕ -CH ₂ -NH	3.09
	ϕ -CH ₂ -F	4.17
	ϕ -CH ₂ -SR	3.20
15	ϕ -CH ₂ -H	3.83

Table 4 shows the bonding energies in electron volts (eV) corresponding to certain bonds of this invention. This data is from Streitwieser et al., *Introduction to Organic Chemistry*, Appendix II, University of California Press, Berkeley, California (1992), incorporated herein fully by reference.

However, the energies of mercury vapor or metal halide UV radiation are too small to be useful for rapid transport polymerization. The desired residence time within the cracking chamber, which is the time available for photolysis should be in the range of a few milliseconds to several hundred milliseconds. Therefore, VUV is the most desirable form of energy for photon assisted transport polymerization.

VUV or incoherent excimer UV sources can be provided by dielectric barrier or silent discharge. For example, VUV can be generated using KrBr, Ar₂, ArCl, ArBr, Xe₂ and F₂ gases. Xe emits at 172 nm, Kr at 222 nm, and XeCl emits at 308 nm. As can be seen from Table 2, nearly all of the chemical bonds of interest in polymer manufacture can be broken using photolytic methods. Because excimer radiation is selective for the energy of the specific bonds, excimer radiation from another source or plasma may be used simultaneously if it is desired to break other bonds at the same time. Such a combination of excimer and plasma sources are useful to break bonds of precursors of

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cross-linked poly(para-xylylenes). Because the leaving groups of these precursors may be different, it is desirable to break those bonds selectively to generate tri- and tetra-functional reactive intermediates.

When IR incoherent excimer irradiation is used, the conventional stainless steel or ceramic pipe or reactor used in the pyrolyzer will have to be replaced with a quartz tube or reactor. When using UV, the transparent tube shown in Figure 4 can be made of any UV transparent material such as quartz, preferably a single quartz crystal. When using VUV, the transparent tube must be made of a material transparent to VUV wavelengths. Crystals of MgF_2 , LiF , or CaF_2 are preferred.

Table 5

Operating Conditions for Photon Assisted Dissociation of Precursors

Variable	Range	Preferred Range
Photon Wavelength	100 - 400 nm	140- 300 nm
Photon Energy	2.5 - 12 eV	4 - 9 eV
Photon Flux	5 mW/cm ² - 10 W/cm ²	40 - 100 mW/cm ²
Reaction Chamber Pressure	0.1 milliTorr - 10 Torr	1 - 100 milliTorr
Reaction Chamber Temperature	-20 - 300° C	30 - 100° C
Deposition Chamber Temperature	-30 to + 20° C	-10° C
Deposition Chamber Pressure	0.1 milliTorr - 10 Torr	1 - 100 milliTorr

In one embodiment of this invention, photon assisted dissociation occurs immediately above the wafer surface onto which the film is to be deposited. Because no transport of intermediates is needed, the efficiency of deposition is increased. Further, the photon energy or mixtures of photon energy (from mixed excimer gases) could be made to both dissociate the precursor as well as promote nucleation and adhesion of the polymer based films.

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D. General Methods for the Manufacture of Low Dielectric Thin Films Using Photon-Plasma Assisted Precursor Dissociation and Deposition

5 Additionally, a combined photon-plasma process is used for dissociation of precursors and deposition of thin films of low dielectric materials. Photon energy is used to generate a plasma in certain gases. In this embodiment of the invention, the attributes of plasma, generally high rates of dissociation, along with the cool photon energy source, provides a unique method for both precursor dissociation and even film deposition. Additionally, infrared (IR) radiation can be used to heat reactor elements and precursors. Pre-heating precursors increases the efficiency of photon and plasma dissociation of the precursors. Moreover, the photon plasma can be directed toward the surface on which films are being deposited.

10 Figure 5 depicts a schematic diagram of a TP and CVD reactor 500 embodying the elements for photon-plasma and IR dissociation and deposition. Precursors 504 are stored in a precursor container 508 which is connected via a pipe 512 to a mass flow controller 516. For TP, precursors are transported into a dissociation reactor 524 which houses the dissociation chamber 528. The wall of reactor 524 is made of crystalline materials such as LiF, MgF₂, or CaF₂, which permits light of vacuum ultraviolet wavelengths to pass. Vacuum ultraviolet and ultraviolet light is generated by a silent discharge plasma generators 532, which are placed inside infrared heaters 536. The infrared heaters 536 are placed inside DC magnets 540 and AC magnets 544. The magnets regulate the flow of plasma during dissociation, and the reactive intermediates so generated are transported to a deposition reactor 550.

20 The deposition reactor 550 contains a deposition chamber 560 containing a gas and reactant dispersion manifold 554, a gas and reactant dispersion plate 558. The walls of the deposition chamber are made of crystalline materials such as LiF, MgF₂, or CaF₂, which permits light of vacuum ultraviolet wavelengths to pass. The gas dispersion manifold 554 and the gas dispersion plate 558, are used to adjust the distribution and homogeneity of the intermediates. The intermediates are directed toward the wafer 562, which is held on a cold chuck 564. The gas dispersion manifold 554 and dispersion plate 558 are connected in parallel to a DC voltage bias anode 568, a DC voltage bias

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cathode 569, an AC voltage bias anode 570, and an AC voltage bias cathode 571. Silent discharge plasma generators 572 are placed outside the deposition chamber 560.

Infrared heaters 574 are placed outside the silent plasma discharge generators 560 and DC magnets 578 and AC magnets 580 are placed outside the infrared heaters 574.

- 5 Gases exit the deposition chamber 560 through a pipe 584, pass through a cold or reactive trap 588, pass through another pipe 592 to a vacuum pump 596. The pressure in the systems is maintained at a desired pressure using pump 596. The trap 588 protects the pump from deposition of intermediates.

For CVD, the deposition chamber can be used without the dissociation reactor.

- 10 Precursors are placed directly on wafer 562, and the chuck 564 is not cooled. IR, UV, or VUV radiation is directed toward the wafer 562. The radiation dissociates the precursor, and deposition of intermediates and polymerization takes place on the wafer.

- Table 6 shows process conditions for combined photon-plasma assisted precursor dissociation using chamber 528, and Table 7 shows process conditions for
15 combined photon-plasma precursor deposition in chamber 560.

Table 6

Process Conditions for Photon-Plasma Precursor Dissociation

	Variable	Range	Preferred Range
20	Temperature	200° C - 600° C	350° C - 500° C
	Photon Wavelength	100 nm - 400 nm	140 nm - 300 nm
	Photon Energy	2.5 eV - 12 eV	4 eV - 9 eV
	Photon Flux	10 milliW/cm ² - 5 W/cm ²	40 - 100 milliW/cm ²
	Plasma Density	10 ¹² - 10 ¹⁴ electrons/cm ³	10 ¹³ electrons/cm ³
25	Pressure	0.1 milliTorr - 10 Torr	1 milliTorr - 10 milliTorr

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Table 7

Process Conditions for Photon-Plasma Precursor Deposition

	Variable	Range	Preferred Range
	Temperature	-20° C - 300° C	-10° C
5	Photon Wavelength	100 nm - 400 nm	250 nm
	Photon Energy	2.5 eV - 12 eV	4.5 eV
	Photon Flux	10 milliW/cm ² - 5 W/cm ²	10 - 100 milliW/cm ²
	Plasma Density	10 ¹² - 10 ¹⁴ electrons/cm ³	10 ¹³ electrons/cm ³
10	Pressure	0.1 milliTorr - 10 Torr	1 milliTorr - 10 milliTorr

In Tables 6 and 7, the plasma density is reported as electron density, but it is to be noted that ion density must be the same to maintain charge neutrality of the plasma. Any non-uniformity of charge distribution can result in plasma damage to the thin film of low dielectric material, as well as imparting charge to the integrated circuit components.

Control of the plasma is by a magnetic field within the precursor chamber and in the deposition chamber. In the precursor reactor, the plasma is confined to any desired area, such as the center of the reactor. Additionally, by alternating the polarity of the magnetic field stirs the plasma, ensuring even energy distribution within the plasma, thereby increasing the efficiency of dissociation of precursor molecules into reactive intermediates. In the deposition chamber, the magnetic field is used to control the pattern of distribution of intermediates over the wafer. This would serve two purposes: (1) to direct the deposition of precursor to the desired portion of the surface, thus conserving the precursor, and (2) minimize film deposition on other parts of the reactor chamber, thus minimizing the required cleaning, minimizing particle generation, and simplifying the reactor chamber design.

Another feature comprises the placement of an electrical bias voltage within the deposition chamber. This provides a further means of controlling the flow of plasma-ionized species to the site of deposition on the wafer. A bias voltage, in the form of

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direct current (DC) or alternating current (AC) can be applied and modulated. Pulsed voltages can be used to alter the flow pattern of ions to either accelerate, decelerate, or to regulate the density of the plasma ions in the stream reaching the wafer. Optimization of ion velocity and flow, thus can be obtained using various combinations of magnetic field and bias voltage.

Table 8
Optimization of Electrical and Magnetic Field Variables

	Variable	Range	Preferred Range
	DC Bias Voltage	100 - 2000 V	500 V
10	AC Bias Voltage	10 - 200 V	50 V
	Pulsed Bias Voltage	100 - 4000 V	500 V
	Pulse Width	10 - 1000 msec	1 msec
	Pulse Frequency	10 Hz - 1000 Hz	60 Hz
	DC Magnetic Field Strength	100 - 2000 Gauss	700 Gauss
15	AC Magnetic Field Strength	100 - 1000 Gauss	500 Gauss
	AC frequency	10 Hz - 500 Hz	50 Hz - 60 Hz

Table 8 shows the ranges of the various magnetic field and bias voltage variables which are regulated in this invention.

Other reactors and reactor configurations may be used, as exemplified by the above cited co-pending applications incorporated herein fully by reference.

The low dielectric constant materials are polymerized into thin films of thicknesses of about 500 Å to about 5 µm on wafers for use in the manufacture of integrated circuits. Figure 6 depicts a diagram of a multi-level integrated circuit chip embodying the features of this invention. The substrate 604 has a source region 608, a drain region 612, and a polysilicon gate 616. A first interlevel dielectric (ILD) layer 620 overlays the substrate 600 and polysilicate gate 616. The wafer is subsequently planarized using chemical mechanical polishing another method known in the art. A floating polysilicon gate 624 is overlain by a second ILD layer 628. the

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wafer is again planarized, a first metal line 632 and a first intermetal dielectric (IMD) layer 636 are deposited. The wafer is again polished. On top of the IMD layer 636, a third metal line 640 and a second ILD layer 644 is deposited. The wafer is again planarized.

5 It should be appreciated by those of ordinary skill in the art that other embodiments may incorporate the concepts, methods, precursors, polymers, films, and devices of the above description and examples. The description and examples contained herein are not intended to limit the scope of the invention, but are included for illustration purposes only. It is to be understood that other embodiments of the invention
10 can be developed and fall within the spirit and scope of the invention and claims.

 Other features, aspects and objects of the invention can be obtained from a review of the figures and the claims.

Industrial Applicability

15 The invention includes novel precursors containing a fluorinated silane, fluorinated siloxane and fluorocarbons, each containing a fluorinated aromatic moiety. The precursors are suitable for making polymers with low dielectric constants and high thermal stability.

 Additionally, the invention includes methods for making polymers for integrated
20 circuit manufacture using novel fluorinated siloxanes, fluorocarbons, and fluorinated aromatic moieties.

 Furthermore, the invention includes integrated circuits made using fluorinated siloxanes, fluorocarbons, or fluorinated aromatic moieties, with low dielectric constants.

 According to the present invention, the polymers made from the disclosed
25 precursors have low dielectric constant and high thermal stability. Therefore, integrated circuits made from these precursors have improved electrical and mechanical properties.

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WHAT IS CLAIMED IS:

1. A precursor for making a low dielectric constant material, the precursor
5 comprising a fluorinated moiety with at least one of a sp^2 C-F bond and hyperconjugated
 sp^3 C-F bond, and further comprising at least one of (1) a fluorinated silane, (2) a
fluorinated siloxane and (3) a fluorocarbon other than $HF_2C-C_6H_4-CF_2H$.
2. The precursor of claim 1 comprising $(C_6H_{5-n}F_n)_m-SiH_{4-m}$ wherein m is an integer
10 selected from the group of 1, 2 and 3 and wherein n is an integer selected from the group
of 1, 2, 3, 4 and 5.
3. The precursor of claim 1 comprising $(C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m}$ wherein m is an
integer selected from the group of 1, 2 and 3, and wherein n is an integer selected from
15 the group of 1, 2, 3, 4 and 5.
4. The precursor of claim 1 comprising $C_6F_5-CF_3$.
5. The precursor of claim 1 comprising $C_6F_5-CHF_2$.
- 20 6. The precursor of claim 1 comprising $C_6F_5-CH_2F$.
7. The precursor of claim 1 comprising $C_6F_5-CH_3$.
- 25 8. The precursor of claim 1 comprising $C_6F_5-CF=CF_2$.
9. The precursor of claim 1 comprising $(CH_{3-n}F_n)-(C_6H_{4-p}F_p)-(CH_{3-m}F_m)$ wherein n
and m are integers selected from the group consisting of 1, 2 and 3, and p is an integer
selected from the group consisting of 1, 2, 3, and 4.

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10. The precursor of claim 1 comprising $\text{HCF}_2\text{-C}_6\text{F}_4\text{-CF}_2\text{H}$.
11. The precursor of claim 1 comprising $\text{CH}_2\text{F-C}_6\text{F}_4\text{-CH}_2\text{F}$.
- 5 12. The precursor of claim 1 comprising $\text{CF}_3\text{-C}_6\text{F}_4\text{-CF}_3$.
13. The precursor of claim 1 comprising $\text{CH}_3\text{-C}_6\text{F}_4\text{-CH}_3$.
14. A precursor for making a low dielectric material, the precursor comprising a
10 fluorinated aromatic moiety.
15. The precursor of claim 14 wherein the fluorinated aromatic moiety comprises a fluorinated phenyl moiety.
- 15 16. A precursor for making a low dielectric material, the precursor comprising a $\text{sp}^2\text{C-F}$ bond.
17. The precursor as in claim 16, consisting primarily of $\text{sp}^2\text{C-F}$ and hyperconjugated $\text{sp}^3\text{C-F}$ bonds.
- 20 18. A method for making a thin film with low dielectric constant and high thermal stability comprising the steps of:
- selecting a precursor comprising at least one of a $\text{sp}^2\text{C-F}$ bond and a hyperconjugated $\text{sp}^3\text{C-F}$ bond; and
- 25 processing the precursor in at least one of a transport polymerization (TP) process and chemical vapor deposition (CVD) process, wherein the process is selected from the group consisting of thermal process, a plasma enhanced process, a high density plasma process, a photon assisted process and a photon-plasma assisted process in order to deposit the thin film.

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19. The precursor of claim 1 wherein the fluorinated moiety is an aromatic moiety.

20. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising a fluorinated aromatic moiety.

5

21. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(C_6H_{5-n}F_n)_m-SiH_{4-m}$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5 and m is an integer selected from the group consisting of 1, 2, and 3 and 4, and the processing step includes processing the precursor in a thermal CVD process with a primary Si source of SiH_4 .

10

22. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m}$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, and m is an integer selected from the group consisting of 1, 2 and 3, and the processing step includes processing the precursor in a plasma enhanced CVD process.

15

23. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $CH_{3-n}F_n-C_6F_4-CH_{3-m}F_m$ wherein n and m are integers selected from the group consisting of 1, 2, and 3, and the processing step includes processing the precursor using a plasma enhanced CVD process with a primary Si source of SiH_4 .

20

24. The method of claim 21 wherein the processing step further comprises using an oxidant.

25

25. The method of claim 18 wherein the step of selecting includes selecting at least one of a silane and a siloxane and the processing step includes processing the precursor using a plasma enhanced CVD process.

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26. The method of claim 18 wherein the step of selecting includes selecting one of a silane and a siloxane.
27. The method of claim 18 wherein the step of selecting includes selecting a silane
5 and the processing step includes processing the precursor in a thermal CVD process.
28. The method of claim 27, wherein the processing step further comprises using an oxidant.
- 10 29. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(C_6F_5)_n-CF_{4-n}$ wherein n is an integer selected from the group consisting of 1, 2, and 3, and the processing step includes processing the precursor in a high density plasma CVD process.
- 15 30. The method of claim 27, wherein the processing step includes a hydrocarbon as a primary carbon source.
31. The method of claim 27, wherein the processing step includes using CH_4 as a primary carbon source.
- 20 32. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $C_6F_5-CF=CF_2$ and the processing step includes processing the precursor in a high density plasma CVD process.
- 25 33. The method of claim 32, wherein the processing step includes using a hydrocarbon as a primary carbon source.
34. The method of claim 32, wherein the processing step includes using CH_4 as a primary carbon source.

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35. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(\text{CH}_{3-n}\text{F}_n)-(\text{C}_6\text{H}_{4-p}\text{F}_p)-(\text{CH}_{3-m}\text{F}_m)$, wherein n and m are integers selected from the group consisting of 1, 2, and 3, p is an integer selected from the group consisting of 0, 1, 2, 3, and 4, and the processing step includes processing the precursor
5 in one of a thermal and a photon assisted CVD process.

36. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $\text{CF}_3-\text{C}_6\text{F}_4-\text{CF}_3$ and the processing step includes processing the precursor in a photon assisted CVD process.

10

37. The method of claim 18 wherein the step of selecting includes selecting a mixture of at least two different precursors at least one of which comprises a $\text{sp}^2\text{C-F}$ bond.

15 38. The method of claim 18, wherein the processing step deposits a thin film a dielectric constant below about 3.0, elastic modulus from about 2 up to about 3, and temperature of decomposition above about 300° C.

39. The precursor of claim 1, wherein the precursor comprises a mixture of a
20 fluorocarbon and a fluorinated siloxane.

40. The method of claim 24 wherein the oxidant is selected from the group consisting of NO_2 or H_2O_2 .

25 41. The method of claim 28 wherein the oxidant is selected from the group consisting of NO_2 or H_2O_2 .

42. The precursor of claim 2 comprising $\text{C}_6\text{F}_5-\text{SiH}_3$.

30 43. The precursor of claim 2 comprising $(\text{C}_6\text{F}_5)_2-\text{SiH}_2$.

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44. The precursor of claim 3 comprising $C_6F_5-Si(OCH_3)_3$.
45. The precursor of claim 3 comprising $(C_6F_5)_2-Si(OCH_3)_2$.
- 5 46. The precursor of claim 1 comprising $(C_6F_5)_n-CHF_{3-n}$, wherein n is an integer selected from the group consisting of 1, and 2.
47. The method of claim 21, wherein the step of selecting includes selecting $C_6F_5-SiH_3$.
- 10 48. The method of claim 21, wherein the step of selecting includes selecting $(C_6F_5)_2-SiH_2$.
49. The method of claim 21, wherein the step of selecting includes selecting $(C_6F_5)_3-SiH$.
- 15 50. The method of claim 22, wherein the step of selecting includes selecting $C_6F_5-Si(OCH_3)_3$.
- 20 51. The method of claim 22, wherein the step of selecting includes selecting $(C_6F_5)_2-Si(OCH_3)_2$.
52. The method of claim 22, wherein the step of selecting includes selecting $(C_6F_5)_3-SiOCH_3$.
- 25 53. The method of claim 23, wherein the step of selecting includes selecting $CHF_2-C_6F_4-CHF_2$.
54. The method of claim 23, wherein the step of selecting includes selecting $CH_2F-C_6F_4-CH_2F$.
- 30

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55. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $\text{CH}_{3-n}\text{F}_n\text{-C}_6\text{F}_4\text{-CH}_{3-m}\text{F}_m$ wherein n and m are integers selected from the group consisting of 1, 2, and 3, and the processing step includes processing the precursor using a plasma enhanced CVD process.

5

56. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(\text{CH}_{3-n}\text{F}_n)\text{-(C}_6\text{H}_{4-p}\text{F}_p)\text{-(CH}_{3-m}\text{F}_m)$ wherein n and m are integers selected from the group consisting of 1, 2, and 3, p is an integer selected from the group consisting of 0, 1, 2, 3 and 4, and the processing step includes processing the precursor using a plasma enhanced CVD process.

10

57. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(\text{C}_6\text{F}_5)_n\text{-SiH}_{3-n}$, wherein n is an integer selected from the group consisting of 1, 2, and 3, and the processing step includes processing the precursor in a thermal CVD process

15

58. The method of claim 18 wherein the step of selecting includes selecting a precursor comprising $(\text{C}_6\text{H}_{5-p}\text{F}_p)_n\text{-Si(OCH}_3)_{3-n}$, wherein n is an integer selected from the group consisting of 1, 2, and 3, p is an integer selected from the group of 1, 2, 3, 4, and 5, and the processing step includes processing the precursor in a plasma enhanced CVD process.

20

59. An integrated circuit chip with a film thickness of about 500 Å to about 5 μm, with a dielectric constant of from about 2 to about 3.0 and capable of being processed in one of a thermal process, a plasma enhanced process, a photon assisted process, and a photon-plasma enhanced process at temperatures above about 200° C, made from a precursor comprising a fluorinated aromatic moiety.

25

60. An integrated circuit chip comprising a fluorinated aromatic moiety.

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61. The chip of claim 60, wherein the fluorinated aromatic moiety is included in a thin film with a low dielectric constant.

5 62. The precursor of claim 1 comprising $(C_6H_{5-n}F_n)_m-SiH_{4-m}$ wherein n is an integer selected from the group consisting of 1, 2, 3, 4, and 5, and wherein m is an integer selected from the group of 1, 2 and 3.

10 63. The precursor of claim 1 comprising $(C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m}$ wherein m is an integer selected from the group consisting of 1, 2, 3, 4, and 5, and wherein m is an integer selected from the group of 1, 2, and 3.

15 64. The precursor of claim 1 comprising $CH_3F_n-C_6H_{4-p}F_p-CH_3F_m$ wherein n and m are integers selected from the group consisting of 1, 2 and 3, and wherein p is an integer selected from the group consisting of 1, 2, 3, and 4.

20 65. A polymer made from a precursor, the precursor comprising a fluorinated moiety with at least one of a sp^2C-F bond, and hyperconjugated sp^3C-F bond, and further comprising at least one of (1) a fluorinated silane, (2) a fluorinated siloxane and (3) a fluorocarbon.

66. The polymer of claim 65 wherein the precursor comprises $(C_6H_{5-n}F_n)_m-SiH_{4-m}$ wherein n is an integer selected from the group consisting of 1, 2, 3, 4, and 5, and wherein m is an integer selected from the group consisting of 1, 2, and 3.

25 67. The polymer of claim 65 wherein the precursor comprises $(C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m}$ wherein n is an integer selected from the group of 1, 2, 3, 4, and 5, and wherein m is an integer selected from the group of 1, 2, and 3.

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68. The polymer of claim 65 wherein the precursor comprises $\text{CH}_{3-n}\text{F}_n\text{-C}_6\text{H}_{4-p}\text{F}_p\text{-CH}_{3-m}\text{F}_m$ wherein n and m are integers selected from the group consisting of 1, 2 and 3, and wherein p is an integer selected from the group consisting of 0, 1, 2, 3, and 4.
- 5 69. The polymer of claim 65 wherein the precursor comprises $\text{C}_6\text{F}_5\text{-CF}_3$.
70. The polymer of claim 65 wherein the precursor comprises $\text{C}_6\text{F}_5\text{-CHF}_2$.
71. The polymer of claim 65 wherein the precursor comprises $\text{C}_6\text{F}_5\text{-CH}_2\text{F}$.
- 10 72. The polymer of claim 65 wherein the precursor comprises $\text{C}_6\text{F}_5\text{-CH}_3$.
73. The polymer of claim 65 wherein the precursor comprises $\text{C}_6\text{F}_5\text{-CF=CF}_2$.
- 15 74. The polymer of claim 65 wherein the precursor comprises $\text{HCF}_2\text{-C}_6\text{F}_4\text{-CF}_2\text{H}$.
75. The polymer of claim 65 wherein the precursor comprises $\text{CH}_2\text{F-C}_6\text{F}_4\text{-CH}_2\text{F}$.
76. The polymer of claim 65 wherein the precursor comprises $\text{CF}_3\text{-C}_6\text{F}_4\text{-CF}_3$.
- 20 77. The polymer of claim 65 wherein the precursor comprises $\text{CH}_3\text{-C}_6\text{F}_4\text{-CH}_3$.
78. A polymer made of a low dielectric material comprising a fluorinated aromatic moiety.
- 25 79. The polymer of claim 78 wherein the fluorinated aromatic moiety comprises a fluorinated phenyl moiety.
80. A low dielectric polymer comprising a $\text{sp}^2\text{C-F}$ bond.
- 30

- 40 -

81. The polymer as in claim 80, comprising sp^2C-F and hyperconjugated sp^3C-F bonds.
82. A film with a thickness of about 500 Å to about 5 μm made from a precursor,
 5 the precursor comprising a fluorinated moiety with at least one of a sp^2C-F bond and hyperconjugated sp^3C-F bond, and further comprising at least one of (1) a fluorinated silane, (2) a fluorinated siloxane and (3) a fluorocarbon other than $HF_2C-C_6H_4-CHF_2$.
83. The thin film of claim 82 wherein the precursor comprises $((C_6H_{5-n}F_n)_m-SiH_{4-m})$
 10 wherein n is an integer selected from the group of 1, 2, 3, 4, and 5, and wherein m is an integer selected from the group consisting of 0, 1, 2, 3, and 4.
84. The thin film of claim 82 wherein the precursor comprises $((C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m})$ wherein n is an integer selected from the group of 1, 2, 3, 4, and 5, and
 15 wherein m is an integer selected from the group consisting of 0, 1, 2, and 3.
85. The thin film of claim 82 wherein the precursor comprises $CH_{3-n}F_n-C_6H_{4-p}F_p-CH_{3-m}F_m$ wherein n and m are integers selected from the group consisting of 1, 2 and 3, and wherein p is an integer selected from the group consisting of 0, 1, 2, 3, and 4.
 20
86. The thin film of claim 82 wherein the precursor comprises $C_6F_5-CF_3$.
87. The thin film of claim 82 wherein the precursor comprises $C_6F_5-CHF_2$.
88. The thin film of claim 82 wherein the precursor comprises $C_6F_5-CH_2F$.
 25
89. The thin film of claim 82 wherein the precursor comprises $C_6F_5-CH_3$.
90. The thin film of claim 82 wherein the precursor comprises $C_6F_5-CF=CF_2$.
 30

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91. The thin film of claim 82 wherein the precursor comprises $\text{HCF}_2\text{-C}_6\text{F}_4\text{-CF}_2\text{H}$.
92. The thin film of claim 82 wherein the precursor comprises $\text{CH}_2\text{F-C}_6\text{F}_4\text{-CH}_2\text{F}$.
- 5 93. The thin film of claim 82 wherein the precursor comprises $\text{CF}_3\text{-C}_6\text{F}_4\text{-CF}_3$.
94. The thin film of claim 82 wherein the precursor comprises $\text{CH}_3\text{-C}_6\text{F}_4\text{-CH}_3$.
95. A thin film made of a low dielectric material comprising a fluorinated aromatic
10 moiety.
96. The thin film of claim 95 wherein the fluorinated aromatic moiety comprises a fluorinated phenyl moiety.
- 15 97. A low dielectric thin film comprising a $\text{sp}^2\text{C-F}$ bond.
98. The thin film as in claim 97, comprising $\text{sp}^2\text{C-F}$ and hyperconjugated $\text{sp}^3\text{C-F}$ bonds.
- 20 99. A method for making a thin film with a dielectric constant below about 3.0 and decomposition temperature above about 300°C comprising the steps of:
selecting a precursor other than $\text{HF}_2\text{C-C}_6\text{H}_4\text{-CHF}_2$, comprising at least one of a $\text{sp}^2\text{C-F}$ bond and a hyperconjugated $\text{sp}^3\text{C-F}$ bond; and
processing the precursor in one of a transport polymerization (TP) system and a
25 chemical vaporization (CVD) system, said system comprising one of a thermal
deposition process, a plasma enhanced process, a high density plasma process, a photon assisted process and a photon-plasma assisted process in order to deposit the thin film.
100. The method of claim 99, wherein the thermal process further comprises using a
30 catalyst.

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101. The catalyst of claim 100 further comprising one of a dehydrogenation catalyst, a reforming catalyst and an oxidative dehydration catalyst.
102. The catalyst of claim 101, wherein the dehydrogenation catalyst is selected from
5 the group consisting transition metal oxides and noble metals on acid supports,
103. The catalyst of claim 101 selected from the group consisting of noble metals or metal sulfide on active carbon.
104. The catalyst of claim 101 selected from the group consisting of Ga_{13} , Cr_{12} , GaAl_{12} and Al_{13} on PILCs.
10
105. The catalyst of claim 101 comprising $\text{M-Al}_2\text{O}_3$ where M is a lanthanide.
106. The catalyst of claim 101 comprising Al_2O_3 kneaded with Bi and Sb
15 compounded with a metal selected from the group consisting of periodic table Group VIB & VIIB metals.
107. The catalyst of claim 101 selected from the group consisting of M-modified H-ZSM-5 and H-ZSM-11 where M is selected from the group consisting of Zn, Ga, Pt-Ga,
20 Pt-Na, Mo, Cr, K, Ca, Mg, Al, and metals selected from periodic table Group VIII metals.
108. The catalyst of claim 101 comprising M-modified MFI (H-GalloSilicates) where
25 M is selected from the group consisting of Si/Ga, Na/Ga, and Al.
109. The catalyst of claim 101 comprising a rare earth metal exchanged Y-Zeolite.
110. The catalyst of claim 101 comprising a rare earth metal exchanged ultra stable
30 Y-Zeolite

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111. The catalyst of claim 101 comprising Ti oxide paired with Zr oxide,
112. The catalyst of claim 101 comprising M plated onto aluminum, where M is selected from the group consisting of Ni, Cr, Ni/Cr, and Ni/Cr/Al alloys.
- 5 113. The chip of claim 59, wherein the dielectric layer is substantially free of voids.
114. The thin film of claim 82 that is substantially free of voids.
- 10 115. The precursor of claim 1 comprising $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.
- 15 116. The method of claim 18, wherein the step of selecting includes selecting a precursor comprising $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.
- 20 117. The polymer of claim 65, wherein precursor comprises $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.
- 25 118. The thin film of claim 82, wherein the precursor comprises $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.
- 30

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119. The method of claim 99, wherein precursor comprises $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.

5

120. The integrated circuit chip of claim 59, wherein the precursor comprises $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.

10

121. The integrated circuit chip of claim 59, wherein the precursor comprises $(C_6H_{5-n}F_n)_m-SiH_{4-m}$ wherein m is an integer selected from the group of 1, 2 and 3 and wherein n is an integer selected from the group of 1, 2, 3, 4 and 5

15

122. The integrated circuit chip of claim 59, wherein the precursor comprises $(C_6H_{5-n}F_n)_m-Si(OCH_3)_{4-m}$ wherein m is an integer selected from the group of 1, 2 and 3, and wherein n is an integer selected from the group of 1, 2, 3, 4 and 5.

20

123. The integrated circuit chip of claim 59, wherein the precursor comprises $(CH_3)_nF_n-(C_6H_{4-p}F_p)-(CH_3)_mF_m$ wherein n and m are integers selected from the group consisting of 1, 2 and 3.

25

124. The method of claim 18, wherein the step of selecting includes selecting a precursor comprising $(C_6H_{5-n}F_n)-CH_{1-m}F_m=CH_{2-p}F_p$, wherein n is an integer selected from the group consisting of 1, 2, 3, 4 and 5, wherein m is a integer selected from the group consisting of 0 and 1, and wherein p is an integer selected from the group consisting of 0, 1 and 2.

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125. A precursor comprising a fluorinated moiety with at least one of a sp^2C-F bond and hyperconjugated sp^3C-F bond, and further comprising at least one of (1) a fluorinated silane, (2) a fluorinated siloxane and (3) a fluorocarbon other than $HF_2C-C_6H_4-CF_2H$, said precursor for making a low dielectric constant material.

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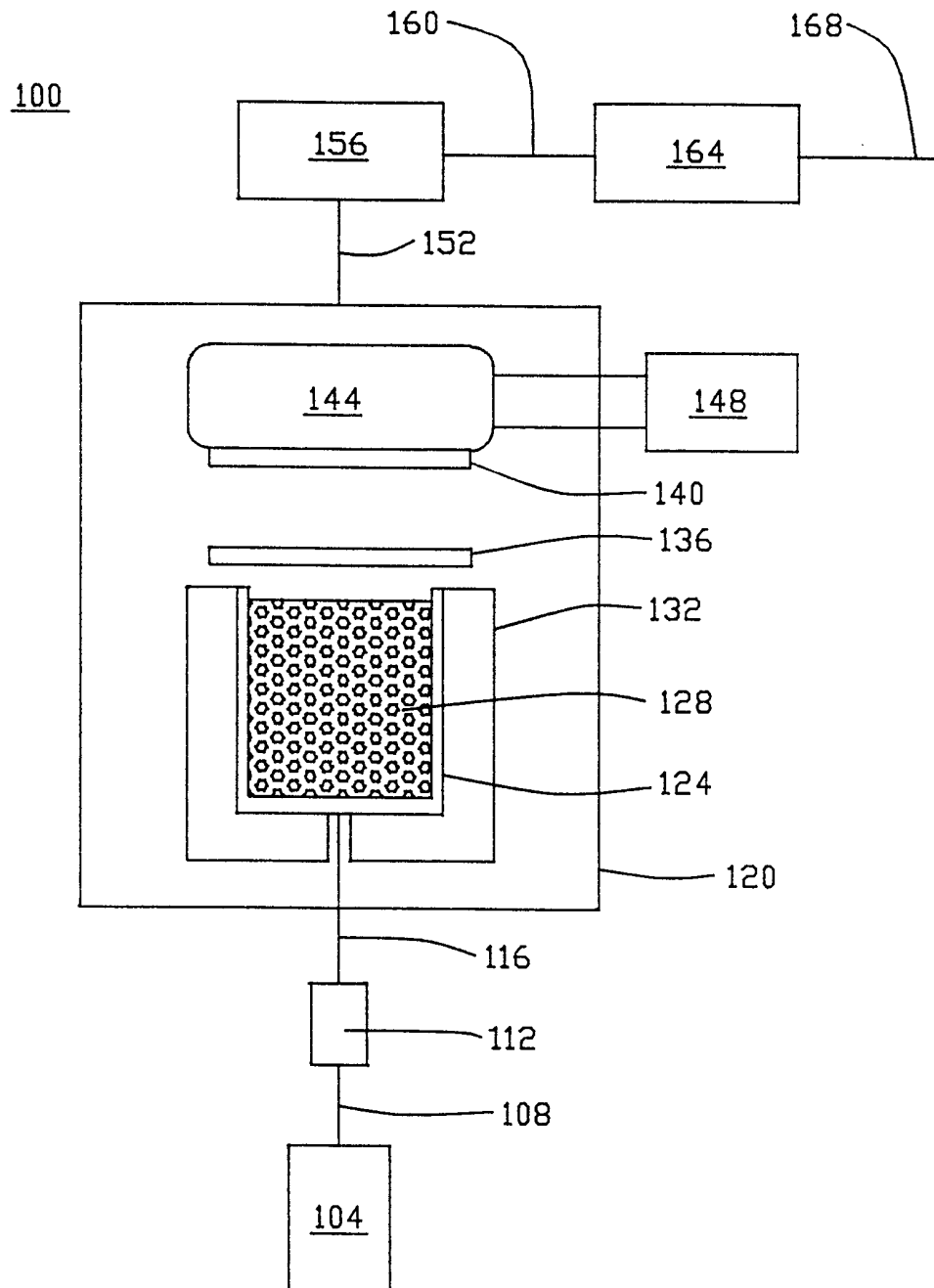


FIG. - 1

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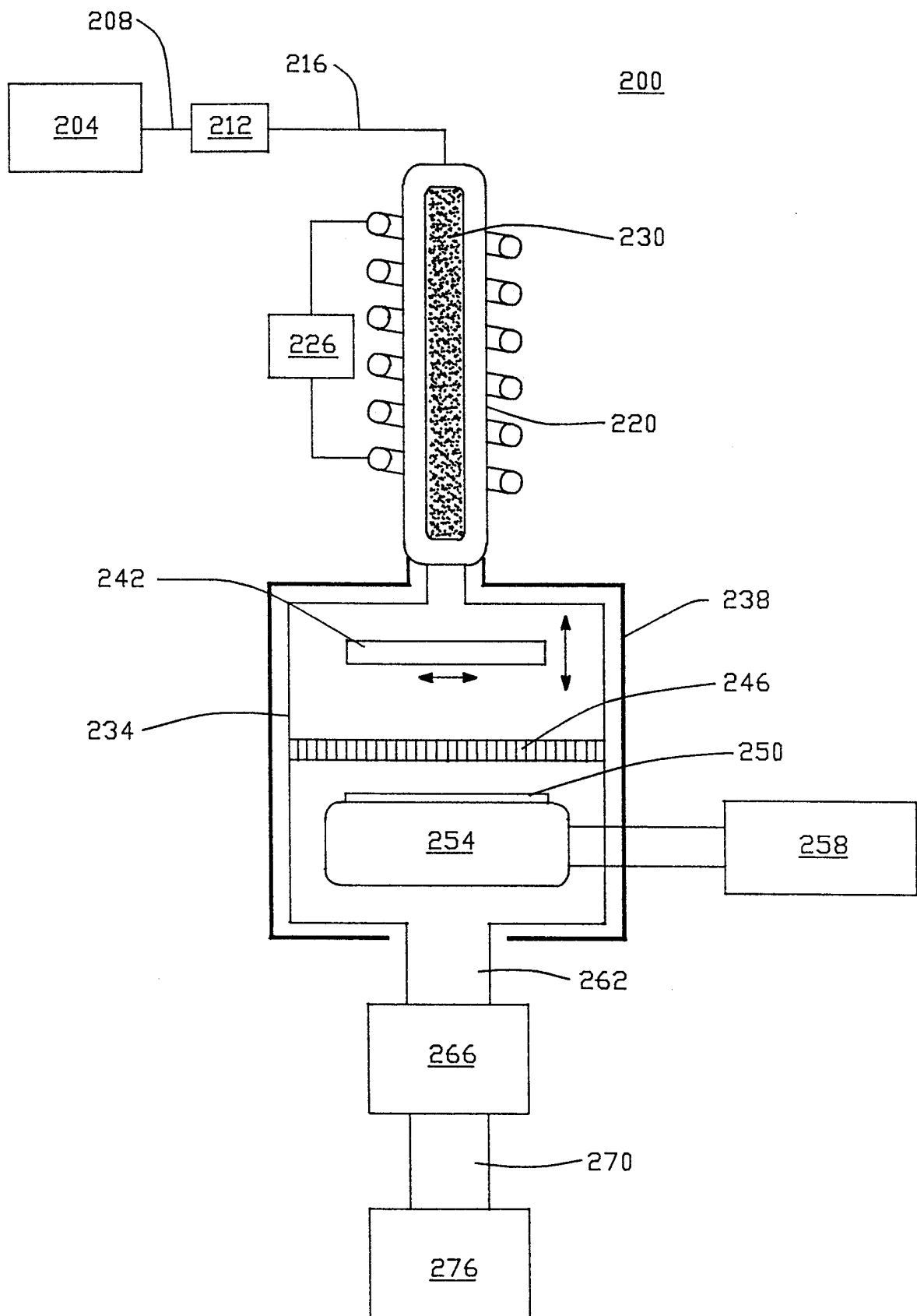
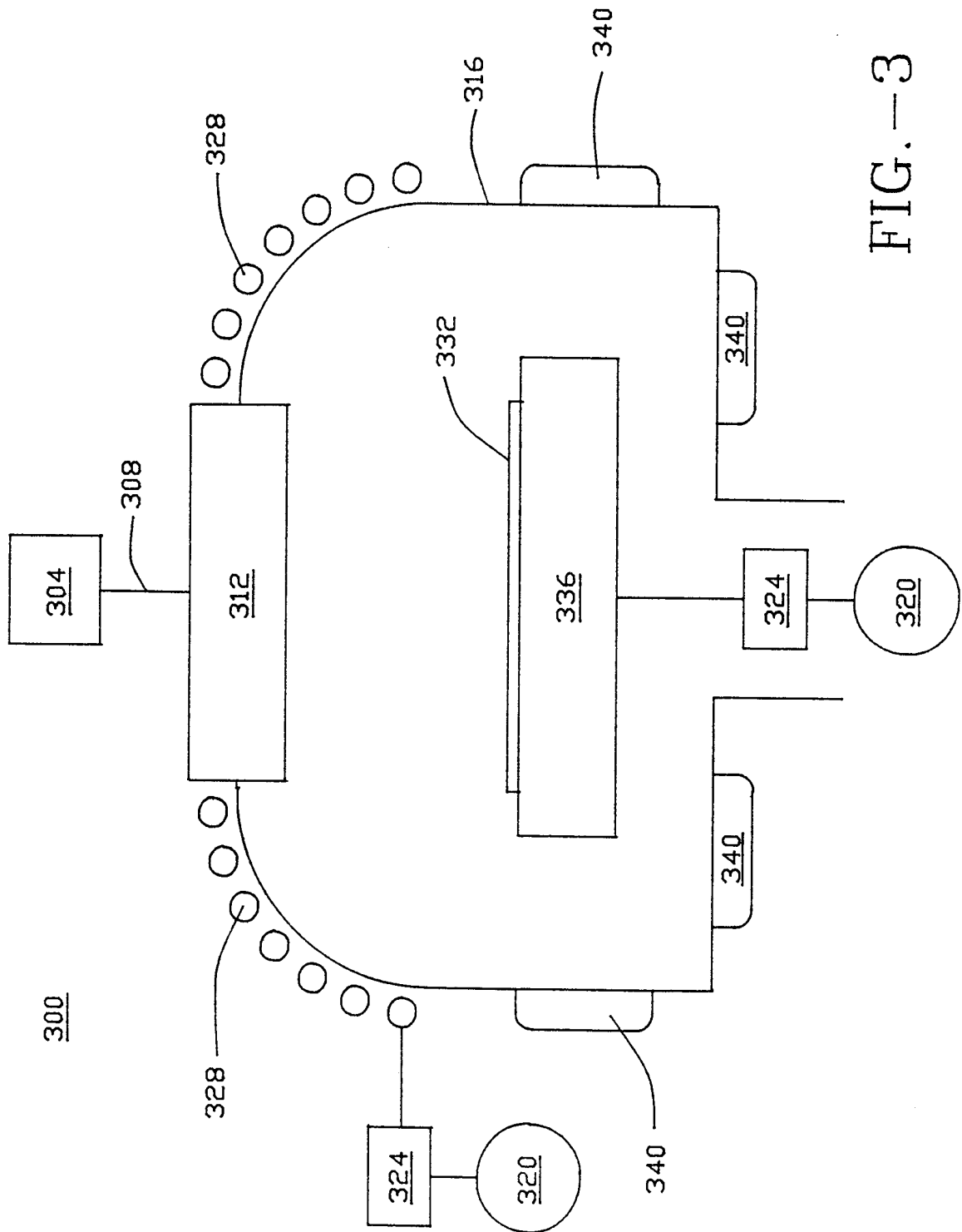


FIG.-2
SUBSTITUTE SHEET (RULE 26)

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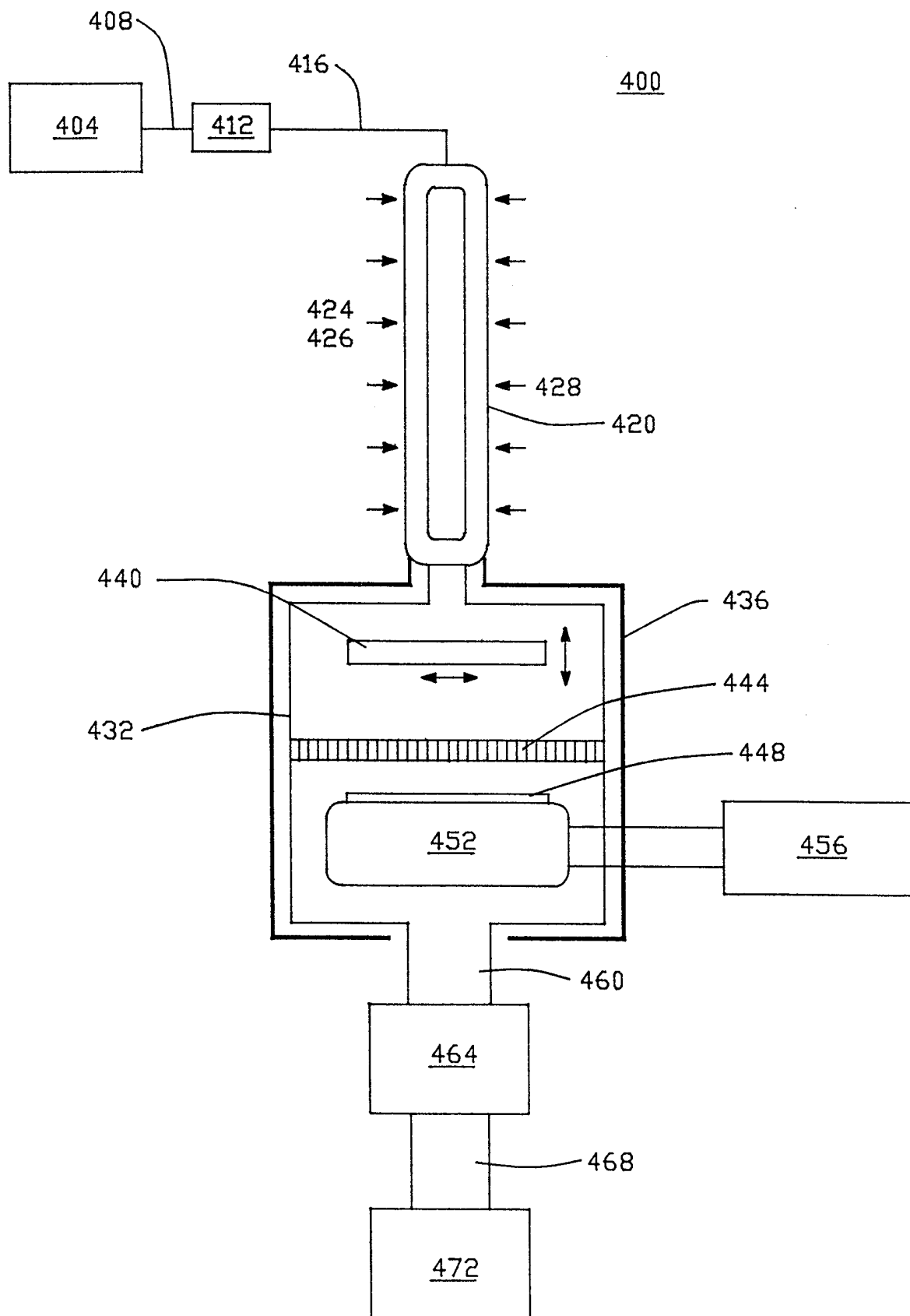


FIG. -4
SUBSTITUTE SHEET (RULE 26)

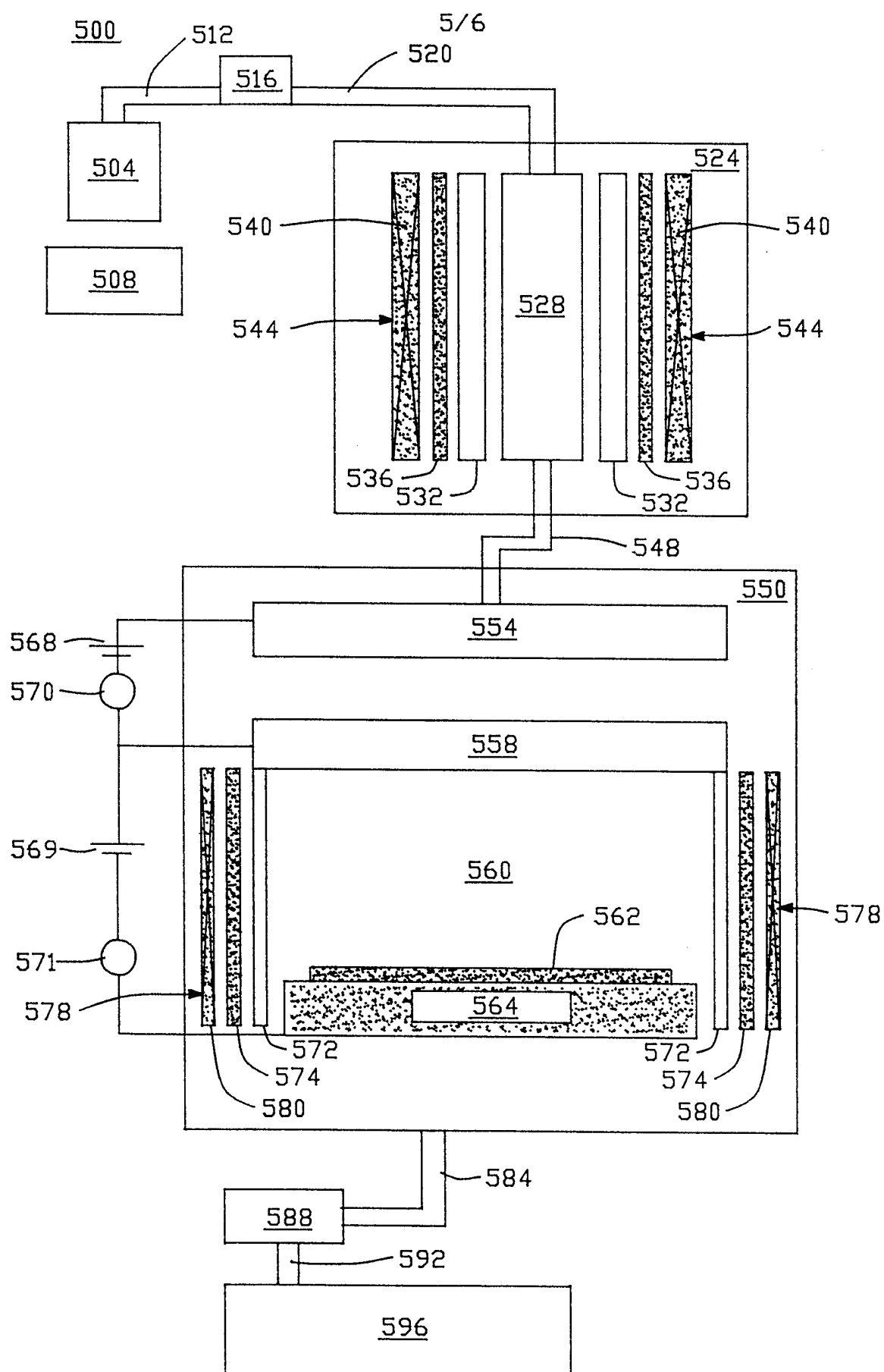


FIG.-5

SUBSTITUTE SHEET (RULE 26)

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612

604

FIG.—6

INTERNATIONAL SEARCH REPORT

International application No.
PCT/US98/21840

A. CLASSIFICATION OF SUBJECT MATTER

IPC(6) :Please See Extra Sheet.

US CL :Please See Extra Sheet.

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

U.S. :

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched
none

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

APS SEARCH TERMS:SP.SUP.2, FLUOR?, C-F

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
Y	BICERANO, JOZEF, PREDICTION OF POLYMER PROPERTIES, 1996, second edition, Marcel Dekker, Inc., pp. 1-15, 50-61, 108-111, 280-295	1-125

☐ Further documents are listed in the continuation of Box C. ☐ See patent family annex.

* Special categories of cited documents:	*T* later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
A document defining the general state of the art which is not considered to be of particular relevance	*X* document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone
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Date of the actual completion of the international search

12 JANUARY 1999

Date of mailing of the international search report

05 FEB 1999

Name and mailing address of the ISA/US
Commissioner of Patents and Trademarks
Box PCT
Washington, D.C. 20231

Facsimile No. (703) 305-3230

Authorized officer

DAVID M. BRUNSMAN

Telephone No. (703) 308-0661

INTERNATIONAL SEARCH REPORT

International application No.
PCT/US98/21840

A. CLASSIFICATION OF SUBJECT MATTER:

IPC (6):

C09D 4/00, 4/06, 127/12, 129/10, 183/06, 201/04; B32B 7/02; C23C 4/00, 16/00, 26/00

A. CLASSIFICATION OF SUBJECT MATTER:

US CL :

106/287.13, 287.15, 287.28; 427/96, 99, 447; 428/220; 438; 758, 780, 800; 522/6, 66, 172, 181, 185, 187; 526/89, 90, 95, 102, 103, 104, 106, 107, 108, 110, 112, 120, 123.1, 154, 159, 167, 168, 176, 177, 191, 242, 245, 250, 251