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(74) Agent: **HAM, Hyun-Kyung**; 14F., Kukdong Building,  
60-1, Chungmuro3-ka, Chung-ku, Seoul 100-705 (KR).

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(71) Applicant (for all designated States except US): **LG CHEM, LTD.** [KR/KR]; 20, Yoido-dong, Young-dungpo-gu, Seoul 150-721 (KR).

(72) Inventors; and

(75) Inventors/Applicants (for US only): **SEO, Dae-Jong** [KR/KR]; 103-1904, Yangjimaoul 1Danji Sambu Apartment, Banseok-dong, Yuseong-gu, Daejeon 305-750 (KR). **LEE, Sang-Young** [KR/KR]; 103-904, Expo Apartment, Jeonmin-dong, Yuseong-gu, Daejeon 305-761 (KR). **KIM, Seok-Koo** [KR/KR]; 317-8, Jangdae-dong, Yuseong-gu, Daejeon 305-308 (KR). **HONG, Jang-Hyuk** [KR/KR]; 105-606, Gukhwa Apartment, 991, Samcheon-dong, Seo-gu, Daejeon 302-782 (KR).

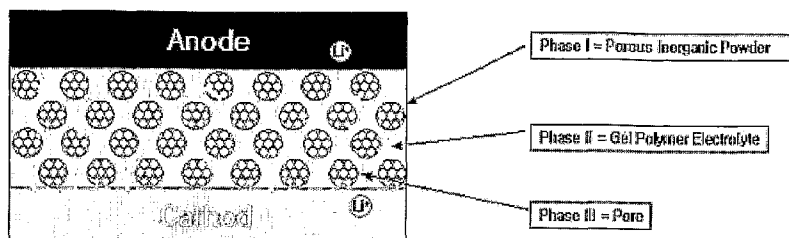
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(54) Title: ORGANIC/INORGANIC COMPOSITE ELECTROLYTE AND ELECTROCHEMICAL DEVICE PREPARED THEREBY



(57) Abstract: Disclosed is an electrode whose surface includes an organic/inorganic composite porous coating layer comprising porous inorganic particles and a binder polymer, wherein the porous inorganic particles have pores having such a size that lithium ions (Li<sup>+</sup>) solvated in an electrolyte solvent can pass therethrough. A method for manufacturing the electrode and an electrochemical device using the electrode are also disclosed. The organic/inorganic composite porous coating layer formed on the electrode according to the present invention provides an additional pathway for lithium ion conduction due to a plurality of pore structures present in the porous inorganic particles. Thus, when the organic/inorganic composite porous coating layer is used instead of a conventional polymer-based separator in a battery, the battery can provide improved quality and an increased energy density per unit weight due to a reduced weight of the organic/inorganic composite porous coating layer.



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**ORGANIC/INORGANIC COMPOSITE ELECTROLYTE AND ELECTROCHEMICAL  
DEVICE PREPARED THEREBY**

**Technical Field**

5       The present invention relates to an electrode that comprises a coating layer capable of substituting for a separator and ensures improved quality and safety of an electrochemical device, a method for manufacturing the same, and an electrochemical device comprising the same.

**Background Art**

10       Recently, there is increasing interest in energy storage technology. Batteries have been widely used as energy sources in portable phones, camcorders, notebook computers, PCs and  
15       electric cars, resulting in intensive research and development into them. In this regard, electrochemical devices are subjects of great interest. Particularly, development of rechargeable secondary batteries has been the focus of attention.

20       Among the currently used secondary batteries, lithium secondary batteries, developed in early 1990's, have drive voltage and energy density higher than those of conventional batteries using aqueous electrolytes (such as Ni-MH batteries, Ni-Cd batteries and H<sub>2</sub>SO<sub>4</sub>-Pb batteries), and thus they are  
25       spotlighted in the field of secondary batteries. However, lithium secondary batteries have problems related to safety, caused by ignition and explosion due to the use of organic electrolytes, and are manufactured through a complicated process. Lithium ion polymer batteries, appearing more  
30       recently, solve the above-mentioned disadvantages of secondary lithium ion batteries, and thus become one of the most potent candidates of the next generation batteries. However, such secondary lithium ion polymer batteries still have low

capacity compared to secondary lithium ion batteries. Particularly, they show insufficient discharge capacity at low temperature. Hence, there is an imminent need for the improvement of secondary lithium ion batteries.

5        Such batteries have been produced by many battery producers. However, most lithium secondary batteries have different safety characteristics depending on several factors. Evaluation of and security in safety of batteries are very important matters to be considered. Particularly, users should  
10 be protected from being injured by malfunctioning batteries. Therefore, safety of batteries is strictly restricted in terms of ignition and combustion in batteries by safety standards. Many attempts have been made to solve the problem related to the safety of a battery. However, ignition of a battery,  
15 caused by a forced internal short circuit due to external impacts (particularly, in the case of a customer-abused battery) cannot be solved yet.

More fundamentally, currently available lithium ion batteries and lithium ion polymer batteries use polyolefin-  
20 based separators in order to prevent short circuit between a cathode and an anode. However, because such polyolefin-based separators use a polymer component having a melting point of 200°C or less and are subjected to a stretching step for controlling their pore sizes and porosities so as to be used  
25 as separators, they have a disadvantage in that they show high heat shrinking property upon exposure to high temperature. In other words, such separators can be shrunk or molten when the temperature of a battery increases due to internal and/or external factors. Therefore, there is a great possibility of  
30 short-circuit between a cathode and an anode that are in direct contact with each other due to shrinking or melting of separators, resulting in accidents such as ignition and explosion of a battery caused by rapid emission of electric

energy.

To solve the above problems related with polyolefin-based separators, many attempts are made to develop an electrolyte using an inorganic material capable of substituting for a conventional separator.

US Patent No. 6,432,586 discloses a polyolefin-based separator coated with an inorganic layer, such as calcium carbonate, silica, etc. However, the separator merely using the polyolefin-based polymer cannot provide a significant improvement in the safety of a battery, particularly in terms of the prevention of heat shrinking at high temperature.

Additionally, Creavis Co. (Germany) have developed an organic/inorganic composite separator comprising a non-woven polyester support coated with silica ( $\text{SiO}_2$ ) or alumina ( $\text{Al}_2\text{O}_3$ ). However, in the case of the above separator, the non-woven polyester support cannot provide excellent mechanical and physical properties by nature, and the chemical structure of polyester is liable to electrochemical reactions. Thus, it is thought that the above separator shows many problems in practical use ("Desalination", vol. 146, p.23 (2002)).

Accordingly, there is an imminent need for developing a separator that can improve the quality and safety of an electrochemical device, or a composite electrolyte that also serves as such a separator.

### **Brief Description of the Drawings**

The foregoing and other objects, features and advantages of the present invention will become more apparent from the following detailed description when taken in conjunction with the accompanying drawings in which:

FIG. 1 is a schematic view showing the electrode comprising an organic/inorganic composite porous coating layer according to the present invention and the function thereof in

a battery;

FIG. 2 is a photographic view showing the porous inorganic particles used in Example 1, taken by SEM (scanning electron microscopy);

5 FIG. 3 is a photographic view showing the non-porous inorganic particles used in Comparative Example 1, taken by SEM;

FIG. 4 is a photographic view showing the inorganic particles (zeolite) having micropores (as defined by IUPAC)  
10 used in Comparative Example 3, taken by SEM;

FIG. 5 is a photographic view showing the surface (5a) and the section (5b) of a conventional polyolefin-based separator, taken by SEM;

FIG. 6 is a photographic view showing the electrode  
15 comprising an organic/inorganic composite porous coating layer according to the present invention;

FIG. 7 is photographic view showing the surface of the electrode comprising an organic/inorganic composite porous coating layer according to the present invention, taken by  
20 SEM;

FIG. 8 is a schematic view showing a process for manufacturing porous inorganic particles and the structure of the porous inorganic particles obtained thereby;

FIG. 9 is a graph showing variation in loading amount of  
25 organic/inorganic composite porous coating layer on the content ratio of porous inorganic particles; and

FIG. 10 is a graph showing variation in permeability / porosity of an organic/inorganic composite porous coating layer depending on the content ratio of porous inorganic  
30 particles.

## **Disclosure of the Invention**

### **Technical Problem**

The inventors of the present invention have found that an organic/inorganic composite porous coating layer, which is formed by coating a mixture of inorganic particles with a polymer capable of being swelled with an electrolyte directly  
5 onto the surface of an electrode, can substitute for a conventional separator, and is firmly bound to an interface between the coating layer and an electrode so as to prevent heat shrinking, resulting in improvement in the safety of an electrochemical device.

10 However, such inorganic particles should be used in a high proportion in the organic/inorganic composite porous coating layer in order to ensure sufficient safety of the coating layer. The inorganic particles may serve as a resistance layer to lithium ion ( $\text{Li}^+$ ) movements, resulting in  
15 degradation of the quality of a battery due to a drop in lithium ion conductivity. Additionally, such an increase in the weight of the layer of inorganic particles causes a drop in energy density of a battery per unit weight.

Therefore, the inventors of the present invention have  
20 conducted many studies to solve the above problems. According to the present invention, an organic/inorganic composite porous coating layer capable of functioning as a separator is formed on a preliminarily formed electrode, wherein the coating layer comprises porous inorganic particles having a  
25 plurality of pores, each pore having such a size that lithium ions ( $\text{Li}^+$ ) can pass therethrough.

#### **Technical solution**

An aspect of the present invention provides an electrode  
30 whose surface includes an organic/inorganic composite porous coating layer comprising porous inorganic particles and a binder polymer, wherein the porous inorganic particles include pores having such a size that lithium ions ( $\text{Li}^+$ ) solvated in an

electrolyte solvent can pass therethrough; and an electrochemical device, preferably a lithium secondary battery comprising the same.

Another aspect of the present invention provides a method for manufacturing the electrode having an organic/inorganic composite porous coating layer, the method comprising the steps of: (a) dispersing inorganic precursors and heat-decomposable compounds in a dispersion medium, misting the inorganic precursor solution, and performing a thermal decomposition and a crystallization processes, to thereby prepare porous inorganic particles; (b) adding and mixing the porous inorganic particles to a polymer solution in which a binder polymer is dissolved; and (c) coating the mixture of step (b) onto a preliminarily formed electrode and drying the coating layer.

Hereinafter, the present invention will be explained in more detail.

The present invention is characterized by forming an organic/inorganic composite porous coating layer that also serves as a separator on the substrate of an electrode and by using porous inorganic particles in the coating layer. By doing so, a novel monolithic composite electrode serving not only as a "separator" (that prevents a cathode and an anode from being in direct contact with each other, and provides a path through which lithium ions move) but also as an "electrode" (that is capable of reversible lithium intercalation/ deintercalation) is provided according to the present invention.

A separator is interposed between a cathode and an anode to prevent both electrodes from being in direct contact with each other, and provides not only a pathway, through which lithium ions ( $\text{Li}^+$ ) move as active components for the electrochemical reactions in a battery, but also a space to be

impregnated with an electrolyte capable of conducting lithium ions. The pathway for lithium ions and the space to be impregnated with an electrolyte depend on the pores in a separator. Therefore, pore size and porosity are important factors in controlling ion conductivity in a battery and relate directly to the quality of a battery.

In other words, when lithium ions move between a cathode and an anode, pores in a separator interposed between both electrodes may serve sufficiently as a pathway for the lithium ions as long as they have a diameter equal to or greater than the diameter of a lithium ion. For reference, a lithium ion has a diameter of several angstroms (Å). However, in fact, a lithium ion does not move alone but move in a form solvated with a plurality of molecules of electrolyte, including a carbonate-based compound, functioning as a lithium ion conduction medium. Thus, if a separator merely has a pore size or porosity similar to the diameter of a lithium ion, lithium ion movements and ion conductivity in a battery are degraded, resulting in deterioration of the quality of a battery. For example, when ethylene carbonate (EC), dimethyl carbonate (DMC), etc. is used as an electrolyte, lithium ions move between both electrodes in solvated forms surrounded with four molecules of EC or DMC. In this case, such solvated lithium ions may have a size of about 2 nm or more. Therefore, it is necessary to consider the size of a lithium ion in combination with that of an electrolyte molecule so as to improve the quality of a battery.

However, most inorganic particles introduced into a conventional organic/inorganic composite layer are non-porous inorganic particles (see FIG. 3) having no pores or, if any, having non-uniform micropores with a diameter less than 2 nm (micropores as defined by IUPAC) (see FIG. 4).

The organic/inorganic composite porous coating layer



according to the present invention is differentiated from the conventional organic/inorganic composite layer in that porous inorganic particles having a uniform pore size and porosity are used and the pore size of the particles is controlled so  
5 that lithium ions ( $\text{Li}^+$ ) solvated with an electrolyte solvent can pass through the pores (see FIG. 2).

Since the pores present in the porous inorganic particles used according to the present invention have such a size that lithium ions solvated with electrolyte solvent  
10 molecules can sufficiently pass therethrough, they can serve as an additional pathway for lithium ions in cooperation with the organic/inorganic composite coating layer serving as a separator. Therefore, lithium ion conductivity in a battery is improved and electrochemical reactions in the battery are  
15 activated, so as to obtain the same quality as a battery using a conventional polyolefin-based separator (see the following Table 3). Contrary to a conventional polyolefin-based separator, the organic/inorganic composite porous coating layer does not cause heat shrinking at high temperature (see  
20 the following Table 2).

Additionally, while the inorganic particles used according to the prior art serve as a resistance layer interrupting lithium ion movements, the porous inorganic particles show a high degree of swelling with an electrolyte  
25 due to a plurality of pores present in the porous inorganic particles and filled with a large amount of electrolyte. Thus, the porous inorganic particles have electrolyte ion conductivity, or increases ion conductivity of the organic/inorganic composite porous coating layer, resulting in  
30 improvement in the quality of a battery.

Further, because the organic/inorganic composite porous coating layer uses porous inorganic particles having a plurality of pores, it has a reduced weight, and thus

increases the energy density per unit weight of an electrochemical device (see the following Table 1).

<Porous Inorganic Particles>

There are no particular limitations in composition, porosity or form of the porous inorganic particles that may be used in the present invention, as long as the porous inorganic particles have such a pore size that lithium ions solvated with electrolyte molecules can sufficiently pass therethrough.

The pores preferably include mesopores having an average diameter of 2 nm or more or macropores having an average diameter of 50 nm or more. More preferably, the pores include macropores having an average diameter of between 50 nm and 1  $\mu\text{m}$ . If the pores have a size less than 0.002  $\mu\text{m}$ , the pores are too small to be impregnated with an electrolyte. On the other hand, if the pores have a size greater than 1  $\mu\text{m}$ , the porous inorganic particles having such an increased size may result in an increase in thickness of a finally formed organic/inorganic composite porous coating layer.

For reference, mesopores and macropores mean pores having a pore size of 2~50 nm and 50 nm or more, respectively, as defined by IUPAC (International Union of Pure and Applied Chemistry). Pores having a diameter of less than 2 nm are defined as micropores.

Preferably, the pores are interconnected among themselves so that lithium ions can move through a pore structure present in the inorganic particles.

There is no particular limitation in the porosity of the porous inorganic particles and the porosity may be controlled in a range of 10~95%. Preferably, the porous inorganic particles have a porosity of 50~90%. If the porous particles have a porosity less than 30%, it is not possible to allow the pores present in the porous particles to be sufficiently impregnated with an electrolyte and to improve the quality of

a battery. If the porous particles have a porosity greater than 95%, the porous particles may have low mechanical strength.

Additionally, the porous inorganic particles have a significantly increased surface area due to the pores present in the particles, resulting in a drop in density. In fact, when using inorganic particles having a high density, it is difficult to disperse the particles during a coating step. Also, in this case, there is an additional problem in that such high-density particles may result in a battery having an increased weight. Thus, it is preferable that the porous particles have as low a density as possible. For example, the porous inorganic particles have a density of 1~4 g/cc and a surface area of 10~50 m<sup>2</sup>/g.

<Organic/Inorganic Composite Porous Coating Layer>

One component of the organic/inorganic composite porous coating layer formed on the surface of the electrode according to the present invention is a binder polymer currently used in the art.

The binder polymers preferably have a glass transition temperature ( $T_g$ ) as low as possible, more preferably  $T_g$  of between -200°C and 200°C, because they can improve mechanical properties such as flexibility and elasticity of a final coating layer.

When the binder polymer has ion conductivity, it can further improve the performance of an electrochemical device. Therefore, the binder polymer preferably has a dielectric constant as high as possible. In practice, because the dissociation degree of a salt in an electrolyte depends on the dielectric constant of a solvent used in the electrolyte, the polymer having a higher dielectric constant can increase the dissociation degree of a salt in the electrolyte used in the present invention. The dielectric constant of the polymer may

range from 1.0 to 100 (as measured at a frequency of 1 kHz), and is preferably 10 or more.

Further, when using a binder polymer having a high degree of swelling with an electrolyte, the binder polymer may absorb the electrolyte infiltrated, and thus the binder polymer can impart electrolyte ion conductivity to the organic/inorganic composite porous coating layer or can improve the electrolyte ion conductivity. More particularly, the surface of the inorganic particles used according to the prior art serves as a resistance layer interrupting lithium ion movements. However, when a binder polymer has a high degree of swelling with an electrolyte on the surface of the porous inorganic particles and/or in the pores formed on the interstitial volumes among the inorganic particles, interfacial resistance generated between the inorganic particles and the electrolyte decreases so that solvated lithium ions can be drawn and moved toward the inside of the pores. Such improved lithium ion conduction can activate electrochemical reactions in a battery and can improve the quality of a battery. Additionally, when binder polymers that can be gelled and swelled with a liquid electrolyte are used, such binder polymers can form a gel type organic/inorganic composite electrolyte through a reaction between the infiltrate electrolyte and the polymer. Such electrolyte demonstrates high ion conductivity and a high degree of swelling, thereby contributing to the improvement of battery performance. Therefore, it is preferable to use a polymer having a solubility parameter between 15 and 45 MPa<sup>1/2</sup>, more preferably between 15 and 25 MPa<sup>1/2</sup>, and between 30 and 45 MPa<sup>1/2</sup>. If the binder polymer has a solubility parameter of less than 15 MPa<sup>1/2</sup> or greater than 45 MPa<sup>1/2</sup>, it is difficult for the binder polymer to be swelled with a conventional liquid electrolyte for a battery.

Non-limiting examples of the binder polymer that may be used in the present invention include polyvinylidene fluoride-co-hexafluoropropylene, polyvinylidene fluoride-co-trichloroethylene, polymethylmethacrylate, polyacrylonitrile, 5 polyvinyl pyrrolidone, polyvinyl acetate, polyethylene-co-vinyl acetate, polyethylene oxide, cellulose acetate, cellulose acetate butyrate, cellulose acetate propionate, cyanoethyl pullulan, cyanoethyl polyvinyl alcohol, cyanoethyl cellulose, cyanoethyl sucrose, pullulan, carboxymethyl 10 cellulose, acrylonitrile-styrene-butadiene copolymer, polyimide or mixtures thereof. Other materials may be used alone or in combination, as long as they satisfy the above characteristics.

The other component of the organic/inorganic composite 15 porous coating layer formed on the surface of an electrode according to the present invention is the porous inorganic particle as described above.

There is no particular limitation in the inorganic particles, as long as they are electrochemically stable. In 20 other words, there is no particular limitation in the porous inorganic particles that may be used in the present invention, as long as they are not subjected to oxidation and/or reduction at the range of drive voltages (for example, 0-5 V vs.  $\text{Li/Li}^+$ ) of a battery, to which they are applied. 25 Particularly, it is preferable to use inorganic particles having as high ion conductivity as possible, because such inorganic particles can improve ion conductivity and quality in an electrochemical device. Further, it is preferable to use inorganic particles having a high dielectric constant, because 30 they can contribute to increase the dissociation degree of an electrolyte salt in a liquid electrolyte, such as a lithium salt, thereby improving the ion conductivity of the electrolyte. Therefore, it is preferable to use inorganic

particles having a high dielectric constant of 5 or more, preferably of 10 or more, inorganic particles having lithium ion conductivity, or mixtures thereof.

Particular non-limiting examples of inorganic particles  
 5 having a dielectric constant of 5 or more include:  $\text{BaTiO}_3$ ,  $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$  (PZT),  $\text{Pb}_{1-x}\text{La}_x\text{Zr}_{1-y}\text{Ti}_y\text{O}_3$  (PLZT),  $\text{PB}(\text{Mg}_3\text{Nb}_{2/3})\text{O}_3$ - $\text{PbTiO}_3$  (PMN-PT), hafnia ( $\text{HfO}_2$ ),  $\text{SrTiO}_3$ ,  $\text{SnO}_2$ ,  $\text{CeO}_2$ ,  $\text{MgO}$ ,  $\text{NiO}$ ,  $\text{CaO}$ ,  $\text{ZnO}$ ,  $\text{ZrO}_2$ ,  $\text{Y}_2\text{O}_3$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{TiO}_2$ ,  $\text{SiC}$  or mixtures thereof.

As used herein, "inorganic particles having lithium ion  
 10 conductivity" refer to inorganic particles containing lithium elements and having a capability of conducting lithium ions without storing lithium. Inorganic particles having lithium ion conductivity can conduct and move lithium ions due to defects present in their structures, and thus can improve  
 15 lithium ion conductivity. Non-limiting examples of such inorganic particles having lithium ion conductivity include: lithium phosphate ( $\text{Li}_3\text{PO}_4$ ), lithium titanium phosphate ( $\text{Li}_x\text{Ti}_y(\text{PO}_4)_3$ ,  $0 < x < 2$ ,  $0 < y < 3$ ), lithium aluminum titanium phosphate ( $\text{Li}_x\text{Al}_y\text{Ti}_z(\text{PO}_4)_3$ ,  $0 < x < 2$ ,  $0 < y < 1$ ,  $0 < z < 3$ ),  $(\text{LiAlTiP})_x\text{O}_y$   
 20 type glass ( $0 < x < 4$ ,  $0 < y < 13$ ) such as  $14\text{Li}_2\text{O}-9\text{Al}_2\text{O}_3-38\text{TiO}_2-39\text{P}_2\text{O}_5$ , lithium lanthanum titanate ( $\text{Li}_x\text{La}_y\text{TiO}_3$ ,  $0 < x < 2$ ,  $0 < y < 3$ ), lithium germanium thiophosphate ( $\text{Li}_x\text{Ge}_y\text{P}_z\text{S}_w$ ,  $0 < x < 4$ ,  $0 < y < 1$ ,  $0 < z < 1$ ,  $0 < w < 5$ ), such as  $\text{Li}_{3.25}\text{Ge}_{0.25}\text{P}_{0.75}\text{S}_4$ , lithium nitrides ( $\text{Li}_x\text{N}_y$ ,  $0 < x < 4$ ,  $0 < y < 2$ ) such as  $\text{Li}_3\text{N}$ ,  $\text{SiS}_2$  type glass ( $\text{Li}_x\text{Si}_y\text{S}_z$ ,  $0 < x < 3$ ,  $0 < y < 2$ ,  
 25  $0 < z < 4$ ) such as  $\text{Li}_3\text{PO}_4$ - $\text{Li}_2\text{S}$ - $\text{SiS}_2$ ,  $\text{P}_2\text{S}_5$  type glass ( $\text{Li}_x\text{P}_y\text{S}_z$ ,  $0 < x < 3$ ,  $0 < y < 3$ ,  $0 < z < 7$ ) such as  $\text{LiI}$ - $\text{Li}_2\text{S}$ - $\text{P}_2\text{S}_5$ , or mixtures thereof. Particular examples of the inorganic particles having lithium ion conductivity include  $(\text{Li}_{0.5}\text{La}_{0.5})\text{TiO}_3$ ,  $\text{Li}_{2x}\text{Ca}_{0.5-x}\text{TaO}_3$ ,  $\text{Li}_{0.2}[\text{Ca}_{1-y}\text{Sr}_y]_{0.4}\text{TaO}_3$ ,  $\text{Li}_6\text{BaLa}_2\text{Ta}_2\text{O}_{12}$ ,  $\text{Li}_3\text{VO}_4$ ,  $\text{Li}_3\text{PO}_4/\text{Li}_4\text{SiO}_4$ ,  $\text{Li}_2\text{S}$ -  
 30  $\text{GeS}_2$ - $\text{P}_2\text{S}_5$ ,  $\text{Li}_2\text{S}$ - $\text{P}_2\text{S}_5$ ,  $\text{Li}_2\text{S}$ - $\text{GeS}_2$ - $\text{Ga}_2\text{S}_3$ ,  $\text{Li}_2\text{S}$ - $\text{SiS}_2$ ,  $\text{Li}_2\text{S}$ - $\text{P}_2\text{S}_5$ ,  $\text{Li}_2\text{S}$ - $\text{P}_2\text{S}_5$ - $\text{SiS}_2$ , or the like.

Particularly, among the above-mentioned inorganic particles,  $\text{BaTiO}_3$ ,  $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$  (PZT),  $\text{Pb}_{1-x}\text{La}_x\text{Zr}_{1-y}\text{Ti}_y\text{O}_3$  (PLZT),

PB(Mg<sub>3</sub>Nb<sub>2/3</sub>)O<sub>3</sub>-PbTiO<sub>3</sub> (PMN-PT) or hafnia (HfO<sub>2</sub>) has a high dielectric constant of 100 or more and shows piezoelectricity due to electric charges generated upon elongation or compression caused by the application of a predetermined pressure leading to a potential difference between both surfaces. Therefore, such particles having piezoelectricity can prevent an internal short circuit between both electrodes, caused by external impacts, and thus can contribute to improvement in the safety of a battery. Additionally, combination of the inorganic particles having a high dielectric constant with the inorganic particles having lithium ion conductivity can provide a synergic effect.

Although there is no particular limitation in the size of inorganic particles, the inorganic particles preferably have a size of 0.01-10  $\mu\text{m}$ . If the size is less than 0.01  $\mu\text{m}$ , the inorganic particles have poor dispersibility so that the structure and physical properties of the organic/inorganic composite porous coating layer cannot be controlled with ease. If the size is greater than 10  $\mu\text{m}$ , the resultant organic/inorganic composite porous coating layer has an increased thickness under the same solid content, resulting in degradation in mechanical properties. Furthermore, such excessively large pores may increase a possibility of internal short circuit being generated during repeated charge/discharge cycles.

The inorganic particles are present in the mixture of the inorganic particles with the binder polymer forming the organic/inorganic composite porous coating layer, preferably in an amount of 10~99 parts by weight, and more particularly in an amount of 50~95 parts by weight based on 100 parts by weight of the total weight of the mixture. If the content of the inorganic particles is less than 10 parts by weight, the binder polymer is present in such a large amount that the

interstitial volumes formed among the inorganic particles are decreased, and thus the pore size and porosity are reduced, resulting in degradation in the quality of a finished battery. If the content of the inorganic particles is greater than 99  
5 parts by weight, the polymer content is too low to provide sufficient adhesion among the inorganic particles, resulting in degradation in mechanical properties of a finally formed organic/inorganic composite porous coating layer.

The organic/inorganic composite porous coating layer  
10 according to the present invention may further comprise conventional additives known to those skilled in the art, other than the inorganic particles and the binder.

Although there is no particular limitation in the structure of the organic/inorganic composite porous coating  
15 layer formed by coating a mixture of the inorganic particles and the binder polymer onto an electrode substrate, the porous coating layer may have any one of the following two types of structures.

The first embodiment of the porous coating layer  
20 according to the present invention comprises the binder polymer layer and the inorganic particles having pores through which lithium ions can pass, the inorganic particles being individually scattered in the binder polymer layer while not being interconnected among themselves. The second embodiment  
25 of the porous coating layer according to the present invention comprises the porous inorganic particles interconnected and fixed among themselves by the binder polymer, wherein interstitial volumes among the porous inorganic particles form a porous structure. In a particularly preferred embodiment,  
30 porous structures are formed both in the organic/inorganic composite porous coating layer and in the porous inorganic particles.



In the second embodiment, the porous inorganic particles serve not only to form pores by the interstitial volumes among such interconnected inorganic particles but also to maintain the physical shape of the organic/inorganic composite layer as a kind of spacer.

There is no particular limitation in the thickness of the organic/inorganic composite porous coating layer. Also, it is possible to independently control the thickness of each coating layer on a cathode and on an anode. According to the present invention, it is preferable to control the thickness of the coating layer in a range of 1~100  $\mu\text{m}$ , more preferably of 1~30  $\mu\text{m}$ , in order to reduce the internal resistance of a battery.

Additionally, the organic/inorganic composite porous coating layer preferably has a pore size of 0.002~10  $\mu\text{m}$  and a porosity of 5~95%, but is not limited thereto.

According to the present invention, the surface of an electrode may be coated with the mixture of porous inorganic particles and a binder polymer by using a conventional method known to those skilled in the art. In a preferred embodiment of the coating method, the porous inorganic particles are added to and mixed with a polymer solution in which a binder is dissolved, and then the mixture is coated and dried on a preliminarily formed electrode substrate.

Herein, the porous inorganic particles may be prepared according to a typical method used in the art, for example, a self-assembly process, a sol-gel method, a condensation-drying method, a spray pyrolysis method, or a combination thereof. Especially, the spray pyrolysis method is preferred.

According to a preferred embodiment, the porous inorganic particles can be prepared by dispersing inorganic precursors and heat-decomposable compounds in a dispersion medium, misting the inorganic precursor solution, and

performing a thermal decomposition and a crystallization processes.

There is no particular limitation in the inorganic precursors, as long as they contain at least one of the  
5 typical inorganic components. For instance, in case of preparing alumina, aluminum containing salts such as aluminum nitrate, aluminum chloride, aluminum acetate, aluminum sulfate and the like may be used. Moreover, a nano alumina such as fumed alumina having a stable dispersed phase may also be used  
10 as the precursor material.

There is no particular limitation in the heat-decomposable compounds, as long as they are decomposed or pyrolyzed at a temperature lower than the melting temperature of the inorganic particles. Such examples include polymers or  
15 foaming agents. In particular, polystyrene is preferred. Although there is no particular limitation in shape of the heat-decomposable compounds either, bead shaped compounds are desired for the formation of uniform pores.

In order to prepare porous particles by the spray  
20 pyrolysis method, it is necessary to disperse uniform-sized, heat-decomposable beads (e.g., polystyrene in the present invention) having been prepared in an inorganic precursor solution. Such a precursor solution is misted through a liquid aerosol generator, and droplets thusly obtained then pass  
25 through a high-temperature reactor, being subjected to the drying and thermal decomposition processes, and are crystallized to produce inorganic porous particles.

There is no particular limitation in the temperature for the thermal treatment, as long as it is lower than the melting  
30 point of the inorganic compound and higher than a decomposition temperature of the heat-decomposable compounds. The heat-decomposable compounds after the thermal treatment are all decomposed in a reactor, and parts filled with the

compounds become pores. The resulting inorganic porous particles may have various pore sizes and porosities, depending on size and mixing ratio of the heat-decomposable compounds.

5        Later, the inorganic porous particles are added and dissolved in a polymer solution. Here, non-limiting examples of the solvent for dissolving polymers include acetone, tetrahydrofuran, methylene chloride, chloroform, dimethylformamide, N-methyl-2-pyrrolidone, cyclohexane, water  
10       or mixtures thereof.

      It is preferable to perform a step of pulverizing inorganic particles after adding the inorganic porous particles to the prepared polymer solution. Conventional pulverization methods, preferably a method using a ball mill,  
15       may be used.

      In order to control the pore size, porosity and thickness of the organic/inorganic composite porous coating layer to be formed finally, factors for controlling pores of the separator, such as, the pore size, porosity, size  
20       (particle diameter), content of the inorganic porous particle, and a composition (mixing) ratio of the inorganic porous particles to the binder polymer can be properly controlled. For instance, as the weight ratio (I/P) of the inorganic particles (I) to the polymer (P) increases, porosity of the  
25       organic/inorganic composite porous coating layer according to the present invention increases. Therefore, the thickness of the organic/inorganic composite porous coating layer increases under the same solid content (weight of the inorganic particles + weight of the binder polymer). Additionally, the  
30       pore size increases in proportion to the pore formation among the inorganic particles. When the size (particle diameter) of the inorganic particles increases, interstitial distance among the inorganic particles increases, thereby increasing the pore

size.

The mixture of the prepared inorganic porous particles and the binder polymer is coated onto the prepared porous substrate having pores, followed by drying to obtain the organic/inorganic composite porous coating layer of the present invention. Here, as for the coating step, any methods known to one skilled in the art including dip coating, die coating, roll coating, comma coating or combinations thereof may be used.

Since the electrode comprising the organic/ inorganic composite porous coating layer according to the present invention is formed by coating the layer directly onto the surface of an electrode substrate including electrode active material particles bound to a collector while forming a pore structure, the electrode active material layer and the organic/inorganic composite porous coating layer can be anchored to each other and firmly bound to each other physically and organically. Therefore, problems related with mechanical properties, such as brittleness, can be improved by virtue of an excellent interfacial adhesion between the electrode substrate and the organic/inorganic composite porous coating layer.

Additionally, the composite electrode according to the present invention comprises: (a) an electrode substrate including electrode active material particles bound to a collector while forming a pore structure; (b) an organic/inorganic composite porous coating layer formed on the electrode substrate and having a pore structure; and porous inorganic particles functioning as a factor for forming the pores of the coating layer and having a plurality of pores therein. Each of the above constitutional elements has a uniform pore structure formed by pores having such a size that lithium ions solvated with electrolyte solvent molecules can

pass therethrough (see FIG. 7), and the pore structures are maintained independently from each other, thereby forming a unique triple pore structure. Therefore, when the electrode is swelled with an electrolyte, interfacial resistance can be significantly decreased. Also, the above triple pore structure facilitates lithium ion conduction, thereby minimizing degradation of the quality of a battery (see FIG. 1).

The electrode comprising the organic/inorganic composite porous coating layer according to the present invention may be used not only as an electrode but also as a separator in an electrochemical device, preferably a lithium secondary battery. Particularly, when a polymer capable of being gelled when swelled with a liquid electrolyte is used in the organic/inorganic composite coating layer, the polymer may react with the electrolyte upon the injection of the electrolyte to form a gel type organic/inorganic composite electrolyte.

Further, the present invention provides an electrochemical device comprising a cathode, an anode and an electrolyte, wherein either or both of the cathode and the anode are the electrode comprising the organic/inorganic composite porous coating layer according to the present invention.

Such electrochemical devices include any devices in which electrochemical reactions occur, and particular examples thereof include all kinds of primary batteries, secondary batteries, fuel cells, solar cells or capacitors. Particularly, the electrochemical device is a lithium secondary battery including a lithium metal secondary battery, lithium ion secondary battery, lithium polymer secondary battery or lithium ion polymer secondary battery.

The electrochemical device may be manufactured by a conventional method known to one skilled in the art. In one

embodiment of the method for manufacturing the electrochemical device, an electrode assembly having a cathode and an anode is formed, and an electrolyte is injected into the electrode assembly. While the conventional electrochemical device is assembled by inserting a separator between a cathode and an anode, only the electrodes comprising the organic/inorganic composite porous coating layer are used according to the present invention. Therefore, it is possible to simplify the process for manufacturing an electrochemical device according to the present invention.

There are no particular limitations in the cathode, anode and electrolyte that may be applied in combination with the organic/inorganic composite porous coating layer according to the present invention. Any cathodes, anodes and electrolytes generally used in conventional electrochemical devices may be used.

Particularly, because the electrode according to the present invention is a monolithic electrode serving also as a separator, it is not necessary to use a conventional microporous separator. However, the electrode according to the present invention may be applied in combination with a microporous separator depending on the particular use and characteristics of a finished electrochemical device.

### **Best Mode for Carrying Out the Invention**

Reference will now be made in detail to the preferred embodiments of the present invention. It is to be understood that the following examples are illustrative only and the present invention is not limited thereto.

#### **[Examples 1~3]**

##### Example 1

##### 1-1. Preparation of Alumina Porous Inorganic Particles

Aluminum nitrate, an alumina precursor compound, was

dissolved in distilled water as a solvent to a concentration of 0.2M. After aluminum nitrate was completely dissolved in distilled water, preliminarily formed polystyrene beads having a diameter of 100 nm was added to the solution in an amount of 3 parts by weight based on the weight of alumina to be prepared, and the mixture was thoroughly agitated. FIG. 8 is a photographic view of the polystyrene beads taken by SEM. The precursor solution obtained as described above was introduced into a spray pyrolysis system to provide porous alumina inorganic particles. FIGs. 2 and 8 are photographic views of the porous alumina particles taken by SEM.

1-2. Manufacture of Electrode Comprising Organic/Inorganic Composite Porous Coating Layer

To N-methyl-2-pyrrolidone (NMP) as a solvent, 96 wt% of carbon powder as an anode active material, 3 wt% of PVdF (polyvinylidene fluoride) as a binder and 1 wt% of carbon black as a conductive agent were added to form mixed slurry for an anode. The slurry was coated on Cu foil having a thickness of about 10  $\mu\text{m}$  as an anode collector, and then dried to form an anode. Then, the anode was subjected to roll press.

To N-methyl-2-pyrrolidone (NMP) as a solvent, 92 wt% of lithium cobalt composite oxide as a cathode active material, 4 wt% of carbon black as a conductive agent and 4 wt% of PVDF as a binder were added to form slurry for a cathode. The slurry was coated on Al foil having a thickness of about 20  $\mu\text{m}$  as a cathode collector, and then dried to form a cathode. Then, the cathode was subjected to roll press.

(Electrode Surface Coating)

About 5 wt% of PVdF-CTFE polymer (polyvinylidene fluoride-chlorotrifluoroethylene copolymer) having a solubility parameter of 20~25  $\text{MPa}^{1/2}$  was added to acetone and dissolved therein at 50°C for about 12 hours or more to provide a polymer solution. To the preformed polymer solution, the

porous  $\text{Al}_2\text{O}_3$  powder having an average pore size of 100 nm and a porosity of 50% (see FIG. 2) and obtained as described in Example 1-1 was added in an amount of 20 wt% on the solid content basis, and the porous  $\text{Al}_2\text{O}_3$  powder was pulverized and dispersed by using a ball mill for 12 hours or more to provide slurry. In the slurry, particle diameter of the porous  $\text{Al}_2\text{O}_3$  particles may be controlled according to the size (particle size) of the beads and the ball milling time. In this example, the porous  $\text{Al}_2\text{O}_3$  particles are pulverized into a size of about 1  $\mu\text{m}$  to provide slurry. Then, the slurry was coated onto surfaces of the preliminarily formed cathode and anode to a thickness of about 15  $\mu\text{m}$  via a dip coating process and the coating layer was dried to provide electrodes comprising an organic/inorganic composite porous coating layer.

FIGs. 6 and 7 are a photograph of the electrode comprising the organic/inorganic composite porous coating layer using porous particles and a photographic view of the surface of the electrode, respectively, taken by SEM. After measuring the pore size and porosity by using a porosimeter, the organic/inorganic composite porous coating layer formed on the electrode had a pore size of 0.49  $\mu\text{m}$ , a porosity of 60% and the structure as shown in FIG. 7.

### 1-3. Manufacture of Lithium Secondary Battery

The anode and the cathode obtained as described in Example 1-2 were stacked to provide an electrode assembly, and an electrolyte comprising 1M lithium hexafluorophosphate ( $\text{LiPF}_6$ ) in EC, PC and DEC in a weight ratio of 30/20/50 (EC/PC/DEC) was injected thereto to provide a lithium secondary battery.

### Example 2

Example 1 was repeated to provide an electrode comprising an organic/inorganic composite porous coating layer (PVdF-HFP/porous  $\text{Al}_2\text{O}_3$ ) and a lithium secondary battery using



the same, except that PVDF-HFP having a solubility parameter of 22~30 MPa<sup>1/2</sup> was used instead of PVDF-CTFE. After measuring the pore size and porosity by using a porosimeter, the organic/inorganic composite porous coating layer formed on the electrode had a pore size of 0.51  $\mu\text{m}$  and a porosity of 62%.

#### Example 3

Example 1 was repeated to provide an electrode comprising an organic/inorganic composite porous coating layer (PVdF-CTFE/TiO<sub>2</sub>) and a lithium secondary battery using the same, except that porous Al<sub>2</sub>O<sub>3</sub> powder was substituted with porous TiO<sub>2</sub> powder having the same pore size and porosity as porous Al<sub>2</sub>O<sub>3</sub> powder. After measuring the pore size and porosity by using a porosimeter, the organic/inorganic composite porous coating layer formed on the electrode had a pore size of 0.37  $\mu\text{m}$  and a porosity of 65%.

#### **[Comparative Examples 1~3]**

##### Comparative Example 1

An electrode and a lithium secondary battery were provided in the same manner as described in Example 1, except that porous particles were not used but non-porous particles (see FIG. 3) having the same composition were used. After measuring the pore size and porosity by using a porosimeter, the organic/inorganic composite porous coating layer formed on the electrode had a pore size of 0.43  $\mu\text{m}$  and a porosity of 53%.

##### Comparative Example 2

A lithium secondary battery was provided in the same manner as described in Example 1, except that conventional electrodes and a PP/PE/PP separator were used instead of the organic/inorganic composite porous coating layer.

FIGs. 5a and 5b are photographic views showing the surface and section of the PP/PE/PP separator, taken by SEM, and the separator had a porosity of about 40%.

##### Comparative Example 3

Example 1 was repeated to provide an electrode comprising an organic/inorganic composite porous coating layer and a lithium secondary battery using the same, except that zeolite having a pore size of 1 nm and a porosity of 33% was used instead of the porous inorganic particles ( $\text{Al}_2\text{O}_3$ ). FIG. 4 is a photographic view of the zeolite having a pore size of 1 nm, taken by SEM.

After measuring the pore size and porosity by using a porosimeter, the organic/inorganic composite porous coating layer formed on the electrode had a pore size of  $0.37\ \mu\text{m}$  and a porosity of 68%.

#### Experimental Example 1: Characterization of Porous Inorganic Particles

The following test was carried out to characterize the porous inorganic particles ( $\text{Al}_2\text{O}_3$ ) used in Examples 1 and 2.

As controls, non-porous  $\text{Al}_2\text{O}_3$  having the same composition of the porous particles and used in Comparative Example 1, and zeolite of Comparative Example 3 were used.

First, the particles were observed for their shapes via SEM (scanning electron microscopy). After the observation, the non-porous  $\text{Al}_2\text{O}_3$  particles used in Comparative Example 1 had no pores and were present in an irregular form (see FIG. 3). Even if the zeolite used in Comparative Example 3 had pores, it could be seen that the pores were too small to be observed by SEM (see FIG. 4). On the contrary, the porous  $\text{Al}_2\text{O}_3$  particles used in the examples according to the present invention had a spherical particle shape and included pores both on the surfaces thereof and inside thereof (see FIGs. 2 and 8).

The above morphological characteristics can be determined by surface area analysis. After determining the surface area of each particle via a nitrogen adsorption method, the non-porous  $\text{Al}_2\text{O}_3$  particles used in Comparative Example 1 had a surface area as low as  $6.4\ \text{m}^2/\text{g}$ , while the

porous Al<sub>2</sub>O<sub>3</sub> particles used in Examples 1 and 2 had a surface area of 33.9 m<sup>2</sup>/g (see Table 1), which was increased by about 5 times as compared to the surface area of the non-porous particles. It can be estimated that such an increase in surface area results from the pores present in the porous Al<sub>2</sub>O<sub>3</sub> particles.

#### Experimental Example 2: Characterization of Organic/Inorganic Composite Porous Coating Layer

The following test was performed to carry out surface analysis of the organic/inorganic composite porous coating layer formed on the electrode according to the present invention.

##### 2-1. Surface Analysis (SEM)

The composite electrode comprising a porous Al<sub>2</sub>O<sub>3</sub>/PVdF-CTFE coating layer according to Example 1 was subjected to surface analysis.

When analyzed by using a Scanning Electron Microscope (SEM), the composite electrode according to Example 1 showed uniform pore structures formed of pores having a size of 1 μm or less both in the electrode substrate and in the organic/inorganic composite porous coating layer comprising Al<sub>2</sub>O<sub>3</sub> particles (see FIG. 7). Also, it could be seen that the porous particles had pores therein.

##### 2-2. Analysis of Physical Properties

The composite electrode comprising a porous Al<sub>2</sub>O<sub>3</sub>/PVdF-CTFE coating layer according to Example 1 was used as a sample. As controls, used were the composite electrode comprising a non-porous Al<sub>2</sub>O<sub>3</sub>/PVdF-CTFE coating layer according to Comparative Example 1, the conventional polyethylene separator according to Comparative Example 2, and the electrode comprising an organic/inorganic composite porous coating layer using zeolite having micropores (as defined by IUPAC) with a pore size of about 1 nm according to Comparative

## Example 3.

The following Table 1 shows the characteristics of the organic/inorganic composite porous coating layer of Example 1 as compared to those of the coating layers of Comparative Examples 1~3 having the same thickness. As can be seen from Table 1, the organic/inorganic composite porous coating layer using porous inorganic particles according to Example 1 has a weight per unit area corresponding to a half or less of the weight per unit area of the organic/inorganic composite porous coating layer using non-porous inorganic particles according to Comparative Example 1. Also, as compared to the coating layer of Comparative Example 1, the composite porous coating layer of Example 1 has a higher porosity and higher ion conductivity (see Table 1). Particularly, the organic/inorganic composite porous coating layer shows a higher pore size and porosity as compared to Comparative Example 3 including a porous coating layer having micropores with a size of 1 nm.

Further, each of the sample and controls was measured for ion conductivity. Herein, ion conductivity was measured via AC impedance determination by disposing the composite electrodes or separator according to Example 1 and Comparative Examples 1~3 between SUS (stainless steel) electrodes having a unit area, allowing the separator or composite electrodes to be sufficiently swelled with an electrolyte, and by sealing the separator or composite electrodes with SUS electrodes while they were in close contact with each other. In fact, it can be seen that the organic/inorganic composite porous coating layer using porous inorganic particles according to Example 1 has excellent ion conductivity (see Table 1). This demonstrates that pores formed by interstitial volumes among the inorganic particles in the organic/inorganic composite porous coating layer, as well as a pore structure present in

the porous inorganic particles facilitates movements and conduction of solvated lithium ions, resulting in a significant improvement in ion conductivity. Therefore, it can be expected that such improved ion conductivity results in improvement in the quality of a battery.

[Table 1]

	Ex. 1	Comp. Ex. 1	Comp. Ex. 2	Comp. Ex. 3
BET surface area (inorganic particles)	33.9	6.4		90
Density (inorganic particles)	2.71	4.08		2.2
Coating layer thickness ( $\mu\text{m}$ )	5	5	-	5
Coating layer weight ( $\text{g}/\text{m}^2$ )	3.94	8.4	-	5.4
Porosity (%)	77.5	46.8	35	65
Ion conductivity ( $\text{S}/\text{cm}$ )	$4.8 \times 10^{-4}$	$2.5 \times 10^{-4}$	$2.0 \times 10^{-4}$	$2.8 \times 10^{-4}$

### 2-3. Analysis of Interrelation between Porous Inorganic Particles and Organic/Inorganic Composite Porous Coating Layer

The composite electrode comprising a porous  $\text{Al}_2\text{O}_3$ /PVdF-CTFE coating layer according to Example 1 was used as a sample.

The organic/inorganic composite porous coating layer was determined for physical properties, for example, loading amount, air permeability and porosity, as a function of the content of porous inorganic particles. After the determination, it can be seen that the weight of the organic/inorganic composite coating layer decreases as the content of the porous inorganic particles increases based on 100 wt% of the mixture of the binder polymer and the porous inorganic particles forming the coating layer (see FIG. 9). Additionally, it can be seen that the porosity of the organic/inorganic composite porous coating layer increases while the air permeability thereof decreases as the content of the porous inorganic particles increases (see FIGs. 9 and 10).

### Experimental Example 3: Evaluation for Shrinkage of Electrode

To evaluate the shrinkage of an electrode at high temperature, the electrode according to Example 1 was used as a sample, and the electrode according to Comparative Example 1 and the PP/PE/PP separator according to Comparative Example 2 were used as controls.

Each of the sample and controls was checked for its heat shrinkage after stored at a high temperature of 150°C for 1 hour. The polyolefin-based separator of Comparative Example 2 provided a heat shrinkage of about 60% and showed severe shrinkage along the direction to which a tensile force is applied during the manufacture of the separator (see Table 2). On the contrary, the electrodes comprising an organic/inorganic composite porous coating layer according to Example 1 and Comparative Example 1 showed no shrinkage before and after the storage at high temperature.

[Table 2]

	Ex. 1	Comp. Ex. 1	Comp. Ex. 2
Shrinkage	0 %	0%	60%

Experimental Example 4: Evaluation for quality of lithium secondary battery

The following test was performed in order to evaluate the quality of each lithium secondary battery according to Example 1 and Comparative Examples 1 and 2 by measuring the capacity and C-rate characteristics of each battery.

Each battery having a capacity of 760 mAh was subjected to cycling at a discharge rate of 0.2C, 0.5C, 1C and 2C. The following Table 3 shows the discharge capacity of each battery, the capacity being expressed on the basis of C-rate characteristics.

After the test, it can be seen that the lithium secondary battery using porous inorganic particles according to Example 1 shows excellent C-rate characteristics as

compared to the battery using conventional non-porous particles according to Comparative Example 1. Also, the battery according to Example 1 is comparable to the battery using a conventional separator according to Comparative Example 2 in terms of C-rate characteristics (see Table 3).

[Table 3]

	Ex. 1 (mAh)	Comp. Ex. 1 (mAh)	Comp. Ex. 2 (mAh)
0.2C	760	755	759
0.5C	757	751	757
1C	744	739	745
2C	694	686	695

#### Experimental Example 5: Evaluation for safety of lithium secondary battery

The following test was performed to evaluate the safety of each lithium secondary battery according to Example 1 and Comparative Examples 1 and 2. Each battery was stored at a high temperature of 150°C and 160°C for 1 hour. Table 4 shows the conditions of each battery after the storage.

After the test, each battery using an electrode comprising an organic/inorganic composite porous coating layer according to Example 1 and Comparative Example 1 causes no ignition or combustion under high-temperature storage conditions regardless of the porosity of the inorganic particles forming the organic/inorganic composite porous coating layer, and thus shows excellent safety. However, the battery using a conventional separator according to Comparative Example 2 causes ignition and combustion under the same conditions (see Table 4).

[Table 4]

	Hot Box (°C / 1 hr)	
	150°C	160°C

Ex. 1	Pass	Pass
Comp. Ex. 1	Pass	Pass
Comp. Ex. 2	Failure	Failure

### **Industrial Applicability**

As can be seen from the foregoing, the organic/inorganic composite porous coating layer according to the present invention capable of substituting for a conventional separator uses porous inorganic particles having pores through which lithium ions can pass. Thus, the organic/inorganic composite porous coating layer according to the present invention has an additional pathway for lithium ion conduction and an additional space to be impregnated with an electrolyte, resulting in improvement in the quality of a battery. Also, the organic/inorganic composite porous coating layer according to the present invention has a decreased weight, and thus can increase energy density per unit weight. Further, the organic/inorganic composite porous coating layer according to the present invention causes no heat shrinkage under high-temperature storage conditions and prevents an internal short circuit between a cathode and an anode even under high-temperature storage conditions, and thus can improve the safety of an electrochemical device.

Although several preferred embodiments of the present invention have been described for illustrative purposes, those skilled in the art will appreciate that various modifications, additions and substitutions are possible, without departing from the scope and spirit of the invention as disclosed in the accompanying claims.



### Claims

1. An electrode whose surface includes an organic/inorganic composite porous coating layer comprising  
5 porous inorganic particles and a binder polymer, wherein the porous inorganic particles have pores having such a size that lithium ions ( $\text{Li}^+$ ) solvated in an electrolyte solvent can pass therethrough.

10 2. The electrode as claimed in claim 1, wherein the porous inorganic particles have pores with an average diameter of 2 nm or more.

15 3. The electrode as claimed in claim 1, wherein the porous inorganic particles have pores with an average diameter of 50 nm~1  $\mu\text{m}$ .

20 4. The electrode as claimed in claim 1, wherein the pores in the porous inorganic particles are interconnected among themselves.

5. The electrode as claimed in claim 1, wherein the porous inorganic particles have a porosity of 30~95%.

25 6. The electrode as claimed in claim 1, wherein the porous inorganic particles have a density of 1~4 g/cc and a surface area of 10~50  $\text{m}^2/\text{g}$ .

30 7. The electrode as claimed in claim 1, wherein the porous inorganic particles have the pore structure by dispersing inorganic precursors and heat-decomposable compounds that are pyrolyzed at a temperature lower than a melting temperature of the inorganic particles in a dispersion

medium, misting the inorganic precursor solution, and performing thermal decomposition and crystallization processes.

5           8.     The electrode as claimed in claim 1, which includes an organic/inorganic composite porous coating layer formed by coating a mixture of the porous inorganic particles and the binder polymer onto a surface of an electrode comprising electrode active material particles bound to a collector while forming a pore structure, wherein the porous  
10           inorganic particles are interconnected and fixed among themselves by the binder polymer, and interstitial volumes among the porous inorganic particles form a pore structure.

15           9.     The electrode as claimed in claim 1, wherein the organic/inorganic composite porous coating layer serves as a separator preventing a cathode and an anode from being in direct contact with each other and permitting lithium ions ( $\text{Li}^+$ ) to pass therethrough.

20           10.    The electrode as claimed in claim 1, wherein the porous inorganic particle is at least one particle selected from the group consisting of: (a) inorganic particles having a dielectric constant of 5 or more; and (b) inorganic particles  
25           having lithium ion conductivity.

          11.    The electrode as claimed in claim 1, wherein the porous inorganic particles have a size of 0.01~10  $\mu\text{m}$ .

30           12.    The electrode as claimed in claim 1, wherein the binder polymer has a solubility parameter of 15~45  $\text{MPa}^{1/2}$ .

          13.    The electrode as claimed in claim 1, wherein the

porous inorganic particles and the binder polymer are used in a weight ratio of 10:90~99:1.

14. The electrode as claimed in claim 1, wherein the  
5 organic/inorganic composite porous coating layer has a thickness of 1~100  $\mu\text{m}$ .

15. An electrochemical device comprising a cathode, an anode and an electrolyte, wherein either or both of the  
10 cathode and the anode are the electrode as defined in any one of claims 1 to 14.

16. The electrochemical device as claimed in claim 15, which further comprises a microporous separator.

15

17. The electrochemical device as claimed in claim 15, which is a lithium secondary battery.

18. A method for manufacturing the electrode having an  
20 organic/inorganic composite porous coating layer as defined in any one of claims 1 to 14, the method comprising the steps of:

(a) dispersing inorganic precursors and heat-decomposable compounds in a dispersion medium, misting the inorganic precursor solution, and performing a thermal  
25 decomposition and a crystallization processes, to thereby prepare porous inorganic particles;

(b) adding and mixing the porous inorganic particles to a polymer solution in which a binder polymer is dissolved; and

(c) coating the mixture of step (b) onto a preliminarily  
30 formed electrode and drying the coating layer.

19. The method as claimed in claim 18, wherein heat-decomposable compounds are polymer or foaming agent decomposed

or pyrolyzed at a temperature lower than a melting point of the inorganic compound.

20. The method as claimed in claim 18, wherein the  
5 thermal decomposition of step (a) is performed at a temperature lower than the melting point of the inorganic compound and higher than a decomposition temperature of the heat-decomposable compounds.

10

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FIG. 1

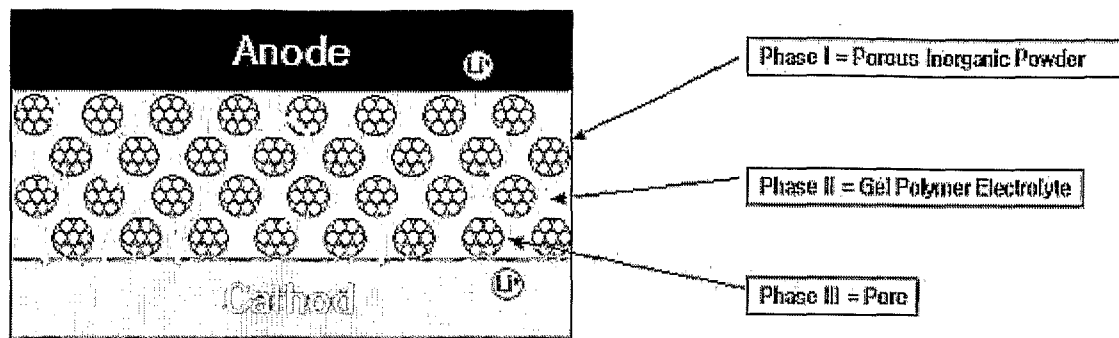


FIG. 2

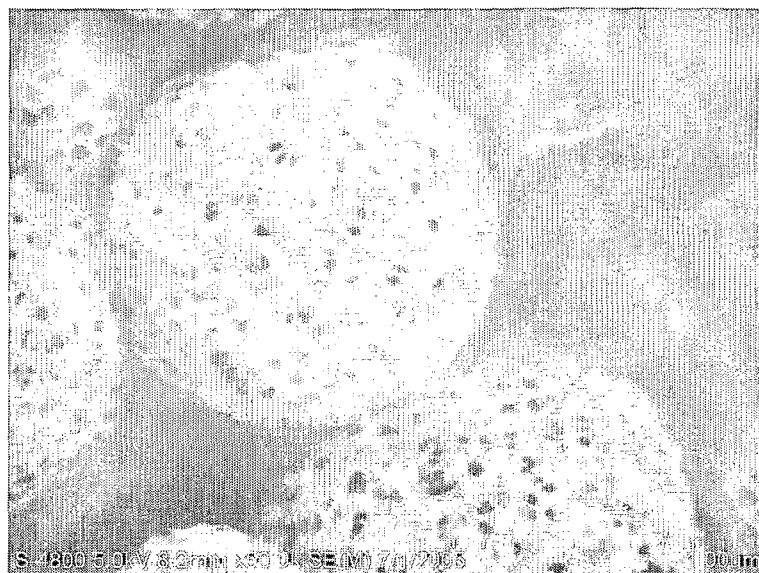


FIG. 3



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FIG. 4



FIG. 5

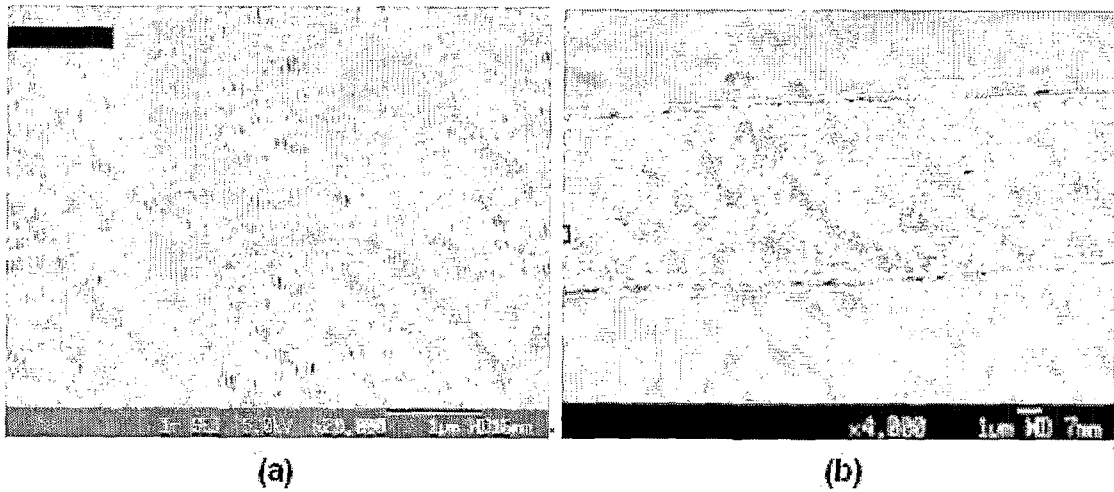


FIG. 6

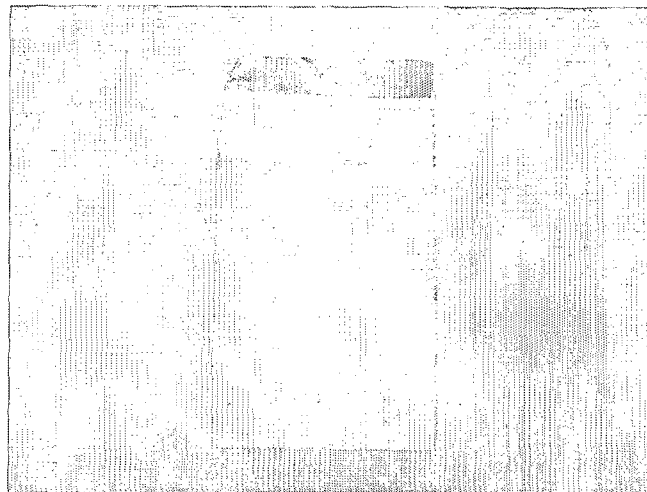


FIG. 7

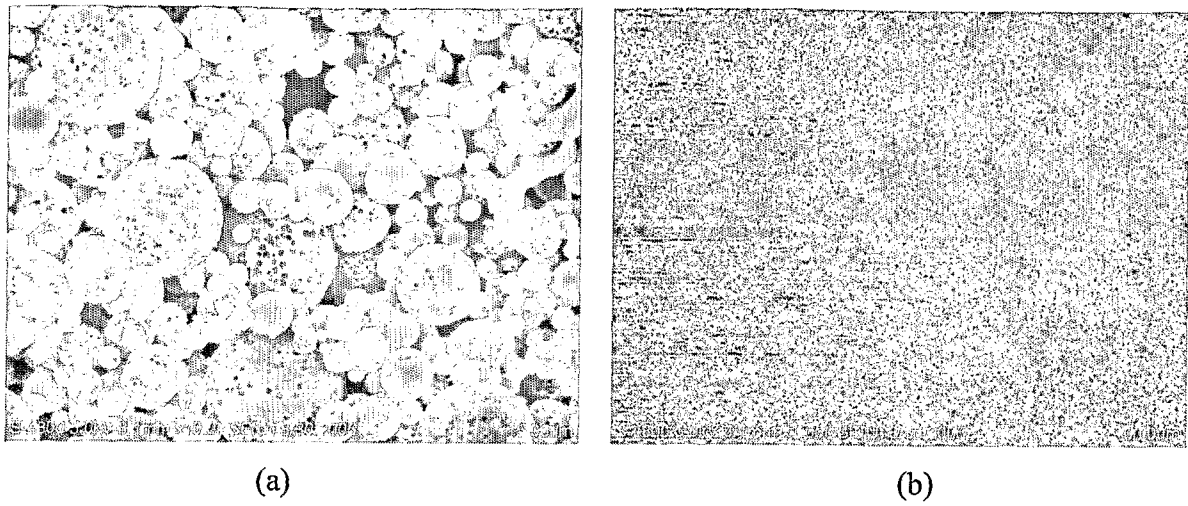
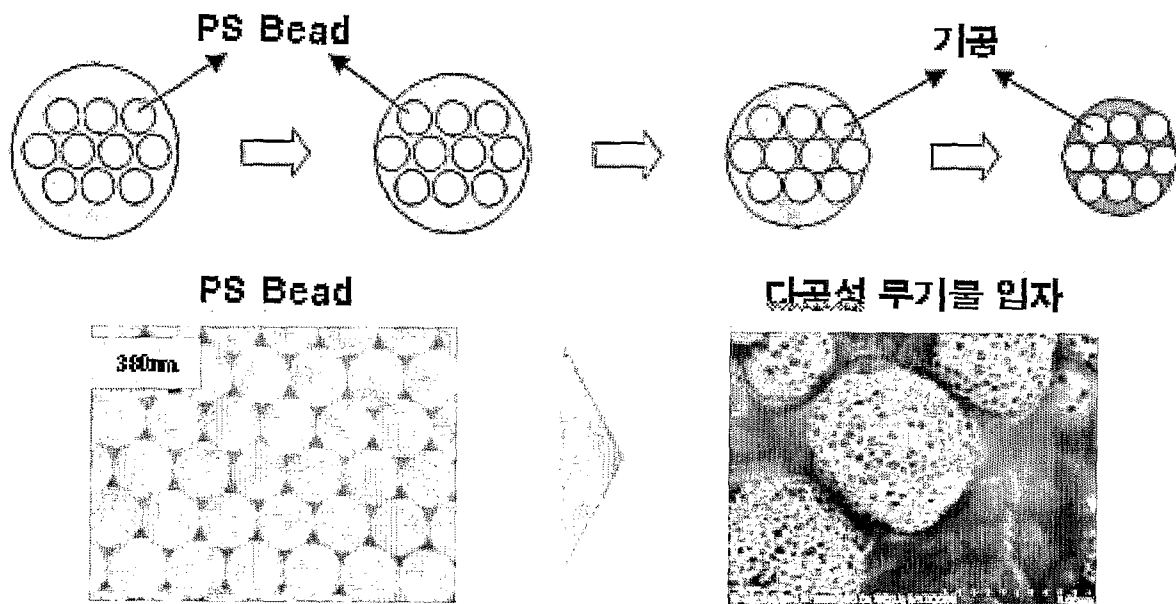


FIG. 8



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FIG. 9

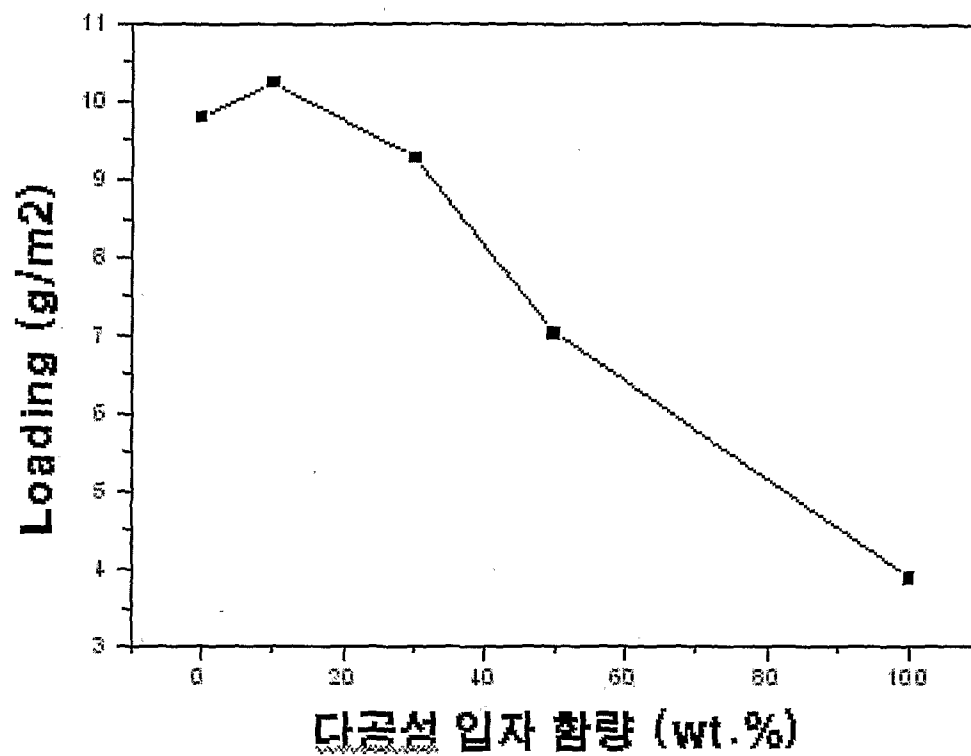
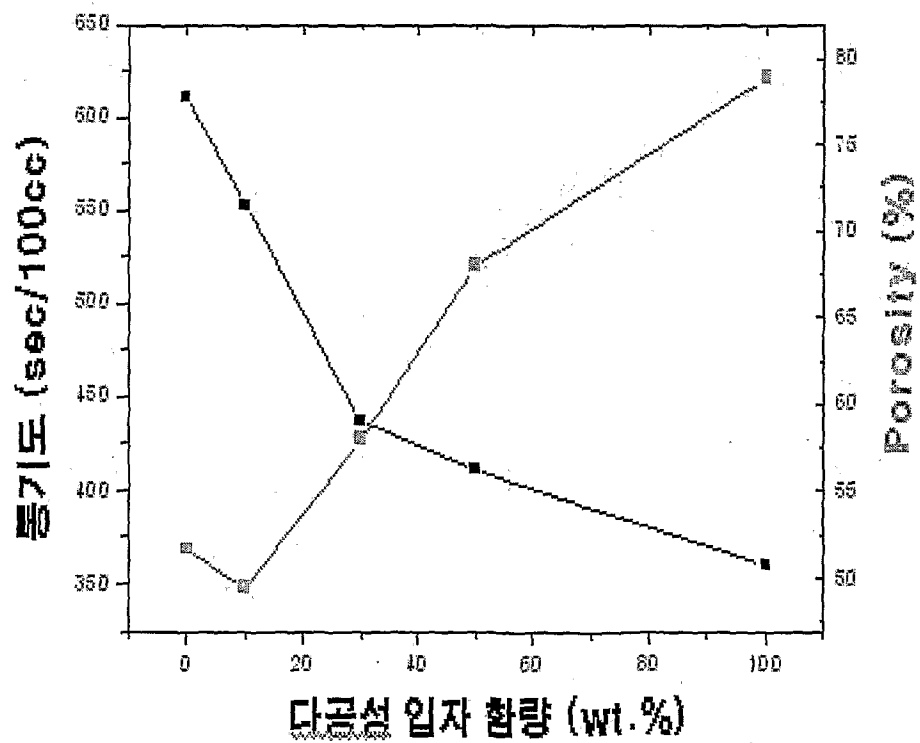


FIG. 10





## INTERNATIONAL SEARCH REPORT

International application No.  
**PCT/KR2007/000848****A. CLASSIFICATION OF SUBJECT MATTER****H01M 4/02(2006.01)i**

According to International Patent Classification (IPC) or to both national classification and IPC

**B. FIELDS SEARCHED**

Minimum documentation searched (classification system followed by classification symbols)

IPC 8: H01M 4/02

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Korean Utility Models and Applications for Utility Models: IPC as above

Japan Utility Models and Applications for Utility Models: IPC as above

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

e-KIPASS (KIPO internal) "porous particle", "lithium ion", "mobility", "electrode", "solvat\*"

**C. DOCUMENTS CONSIDERED TO BE RELEVANT**

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	US6432586B1 (Celgard Inc.) 13 Aug 2002 See abstract and claims 1-3, 10-12	1-20
A	US6287728B1 (Kao Corporation) 11 Sep 2001 See abstract and claims 1-3	1-20
A	US06096453A (Adven Polymers, Inc.) 1 Aug 2000 See abstract and claims 1-5	1-20



Further documents are listed in the continuation of Box C.



See patent family annex.

\* Special categories of cited documents:

"A" document defining the general state of the art which is not considered to be of particular relevance

"E" earlier application or patent but published on or after the international filing date

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"O" document referring to an oral disclosure, use, exhibition or other means

"P" document published prior to the international filing date but later than the priority date claimed

"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention

"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone

"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art

"&amp;" document member of the same patent family

Date of the actual completion of the international search

25 MAY 2007 (25.05.2007)

Date of mailing of the international search report

**25 MAY 2007 (25.05.2007)**

Name and mailing address of the ISA/KR

Korean Intellectual Property Office  
920 Dunsan-dong, Seo-gu, Daejeon 302-701,  
Republic of Korea

Facsimile No. 82-42-472-7140

Authorized officer

PARK, Hyung Dal

Telephone No. 82-42-481-5581



**INTERNATIONAL SEARCH REPORT**

Information on patent family members

International application No.

**PCT/KR2007/000848**

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