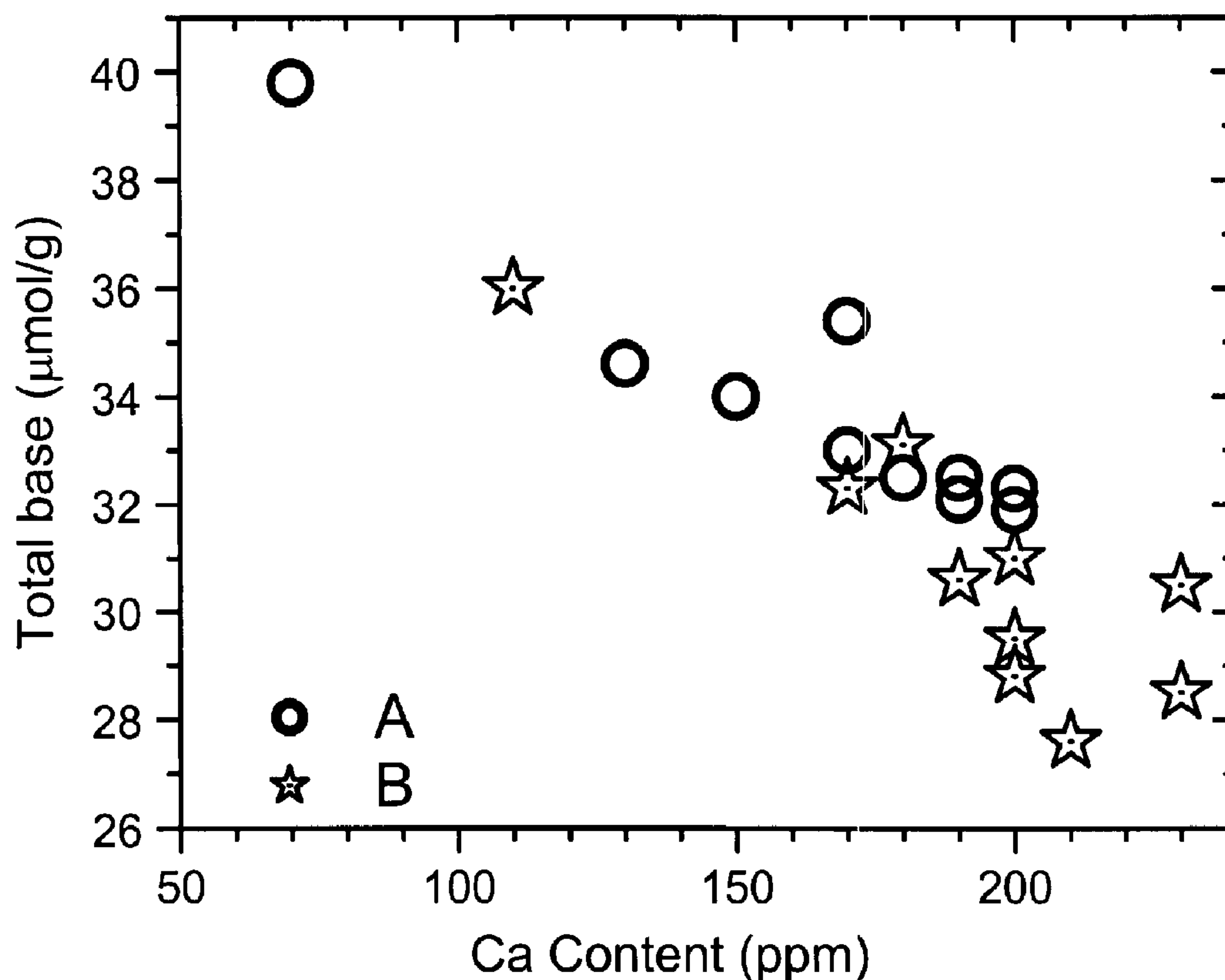




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(54) Titre : OXYDES DE METAUX DE TRANSITION LITHIES DOPES CONTENANT DU SOUFRE
(54) Title: DOPED LITHIUM TRANSITION METAL OXIDES CONTAINING SULFUR



(57) Abrégé/Abstract:

The invention covers a powderous lithium transition metal oxide having a layered crystal structure $\text{Li}_{1+a}\text{M}_{1-a}\text{O}_{2\pm b}\text{M}'_k\text{Sm}$ with $-0.03 < a < 0.06$, $b \cong 0$, M being a transition metal compound, consisting of at least 95% of either one or more elements of the group Ni,



(57) Abrégé(suite)/Abstract(continued):

Mn, Co and Ti; M' being present on the surface of the powderous oxide, and consisting of either one or more elements from (IUPAC) of the Periodic Table, each of said Group 2, 3, or 4 elements having an ionic radius between 0.7 and 1.2 Angstrom, M' however not comprising Ti, with $0.015 < k < 0.15$, k being expressed in wt%, and $0.15 < m \leq 0.6$, m being expressed in mol%. The addition M' (like Y, Sr, Ca, Zr,...) improves the performance as cathode in rechargeable lithium batteries. In a preferred embodiment a content of 250-400 ppm calcium and 0.2-0.6 mol% of sulfur is used. Particularly, a significantly lower content of soluble base and a dramatically reduced content of fine particles are achieved. Especially preferred performance is achieved if 11.5-13.5% of the metal atoms of the cathodes are divalent nickel.

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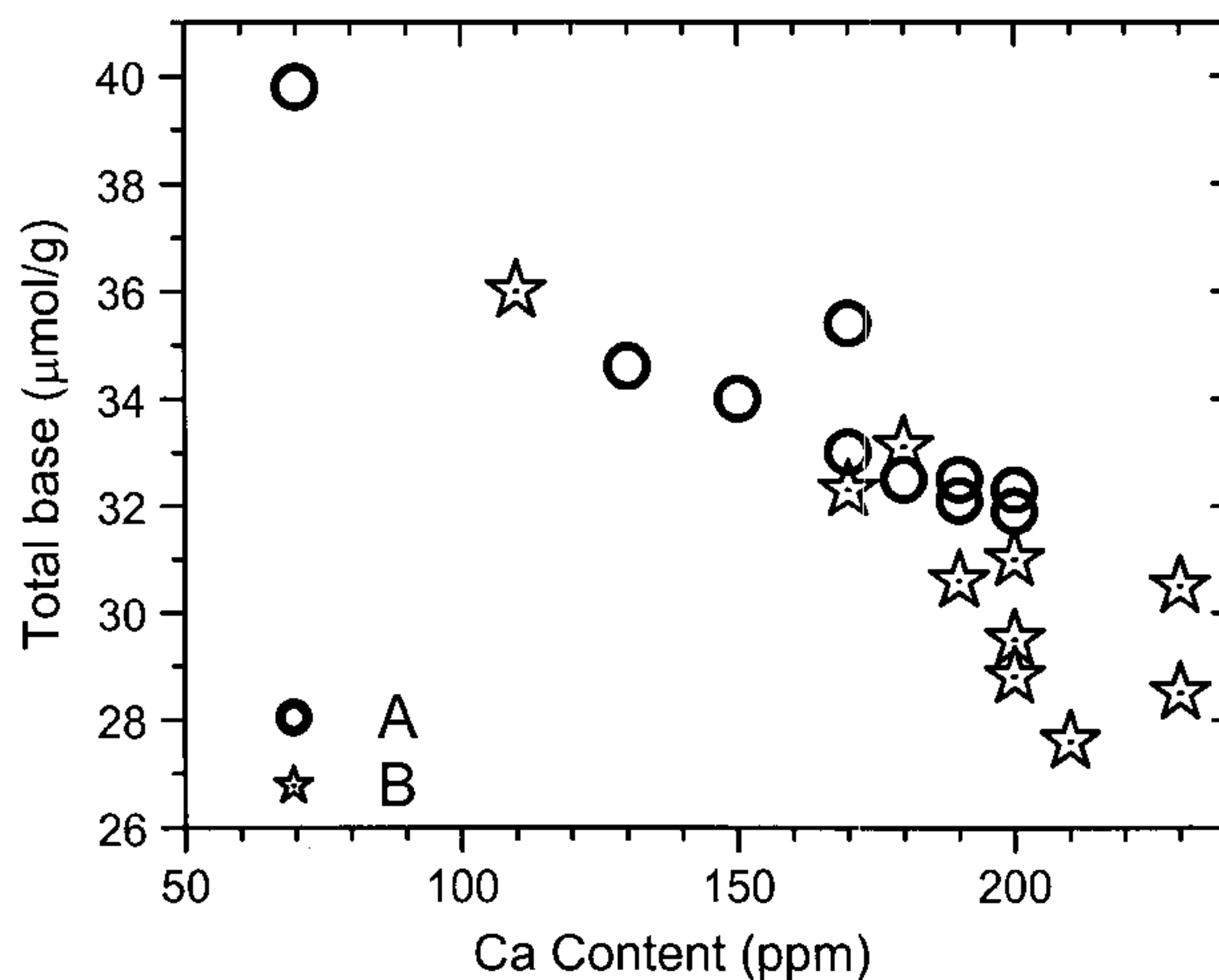


Fig. 1

(57) Abstract: The invention covers a powdery lithium transition metal oxide having a layered crystal structure $\text{Li}_{1+a}\text{M}_{1-a}\text{O}_{2\pm b}\text{M}'_k$ Sm with $-0.03 < a < 0.06$, $b \cong 0$, M being a transition metal compound, consisting of at least 95% of either one or more elements of the group Ni, Mn, Co and Ti; M' being present on the surface of the powdery oxide, and consisting of either one or more elements from (IUPAC) of the Periodic Table, each of said Group 2, 3, or 4 elements having an ionic radius between 0.7 and 1.2 Angstrom, M' however not comprising Ti, with $0.015 < k < 0.15$, k being expressed in wt%, and $0.15 < m \leq 0.6$, m being expressed in mol%. The addition M' (like Y, Sr, Ca, Zr,...) improves the performance as cathode in rechargeable lithium batteries. In a preferred embodiment a content of 250-400 ppm calcium and 0.2-0.6 mol% of sulfur is used. Particularly, a significantly lower content of soluble base and a dramatically reduced content of fine particles are achieved. Especially preferred performance is achieved if 11.5-13.5% of the metal atoms of the cathodes are divalent nickel.

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Doped Lithium transition metal Oxides containing Sulfur

The invention relates to a powdery lithium transition metal oxide, used as active cathode material in rechargeable lithium batteries. More particularly, in $\text{Li}(\text{Mn-Ni-Co})\text{O}_2$ type
 5 compounds containing sulfur the addition of certain amounts of elements like Ca, La, Y, Sr, Ce or Zr optimizes the electrochemical and safety characteristics of the cathode material.

LiCoO_2 is the most widely applied cathode material for rechargeable batteries. However, there exists a strong pressure to replace it by other materials for particular reasons.
 10 Currently, scarce resources of cobalt and fear of high prices accelerate this trend. Besides LiFePO_4 and Li-Mn-spinel, which both suffer from much lower energy density, LiNiO_2 based layered cathode materials and $\text{Li}(\text{Mn-Ni-Co})\text{O}_2$ based layered cathode materials are the most likely candidates to replace LiCoO_2 in commercial battery applications. Today it is basically known that any composition $\text{Li}[\text{Li}_x\text{M}_{1-x}]\text{O}_2$ with $\text{M}=\text{Mn, Ni, Co}$ within the
 15 quaternary system $\text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2 - \text{LiCoO}_2 - \text{LiNiO}_2 - \text{LiNi}_{0.5}\text{Mn}_{0.5}\text{O}_2$ exists as a layered phase, and in most cases is electrochemically active.

Even this quaternary system is to be seen as a simplified model because it does not take into account further phenomena like the possibility of cation mixing. One type of cation mixing is known from LiNiO_2 where some nickel is misplaced on lithium sites of the r-3m
 20 layered crystal structure, a more realistic formula is approximated as $\{\text{Li}_{1-x}\text{Ni}_x\}[\text{Ni}]\text{O}_2$. It is also known that $\text{Li}_{1+x}\text{M}_{1-x}\text{O}_2$ with $\text{M}=\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$ is better written as $\{\text{Li}_{1+y}\text{M}_y\}[\text{Li}_z\text{M}_{1-z}]\text{O}_2$.

As a result, layered $\text{Li}(\text{Mn-Ni-Co})\text{O}_2$ phases which are of interest for battery cathode materials belong to the quaternary (according Gibbs phases rule) subspace of the 5
 25 dimensional thermodynamic system $\text{LiNiO}_2 - \{\text{Li}_{1-a}\text{Ni}_a\}\text{NiO}_2 - \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2 - \text{LiCoO}_2 - \text{LiNiO}_2$. Most of the phases within this triangle are electrochemically active.

By very basic thermodynamic reasons, if further parameters are included (like oxygen particle pressure or temperature) the numbers of dimensions might increase further to 5 or 6, to explain phenomena like dependence of cation mixing of a given composition as

function of temperature, or the existence of vacancies (oxygen or cationic) as function of temperature and oxygen pressure as observed for LiCoO_2 .

This has not even taken account for further dopants, which might fit into the crystal structure, like Mg, Al, Cr, Ti; such doping introducing further degrees of freedom, adding
5 more dimensions to the already complex thermodynamic system.

Since many years it is known that the layered structure of LiNiO_2 can be stabilized, and electrochemical properties can be improved if Ni is replaced by Mn or Co, resulting in $\text{LiNi}_{1-x}\text{Mn}_x\text{O}_2$ and $\text{LiNi}_{1-x}\text{Co}_x\text{O}_2$. Quite soon it was discovered that Mn and Co can be co-doped, resulting in layered $\text{Li}(\text{Ni-Mn-Co})\text{O}_2$ phases $\text{LiNi}_{1-x-y}\text{Mn}_x\text{Co}_y\text{O}_2$. So JP3244314
10 (Sanyo) claims $\text{Li}_a\text{M}_b\text{Ni}_c\text{Co}_d\text{O}_e$ covering a wide range of metal compositions.

It was also discovered quite early that Al can replace Ni. So, already in the early and middle nineties there exist many patent with claims like $\text{Li}_x\text{Ni}_{1-a-b}\text{M1}_a\text{M2}_b\text{O}_2$ where generally x is near to unity, M1 is transition metal and M2 a further dopant like aluminum. Examples typically focus on LiNiO_2 based materials (say $a+b < 0.4$), and can be found in
15 JP3897387, JP3362583, JP 3653409 or JP3561607, the latter disclosing
 $\text{Li}_a\text{Co}_b\text{Mn}_c\text{M}_d\text{Ni}_{1-(b+c+d)}\text{O}_2$ with $0 < a < 1.2$, $0.1 \leq b \leq 0.5$, $0.05 \leq c \leq 0.4$, $0.01 \leq d \leq 0.4$, and $0.15 \leq b+c+d \leq 0.5$.

It can be summarized that at the mid 90ties prior art were compositions within the Ni rich corner of the solid state solution between $\text{LiCoO}_2 - \text{LiMn}_{1/2}\text{Ni}_{1/2}\text{O}_2 - \{\text{Li}_{1-x}\text{Ni}_x\}\text{NiO}_2$,
20 including further dopants (like Al). The other corners (LiCoO_2 , in US4302518, US4357215) and $\text{LiNi}_{1/2}\text{Mn}_{1/2}\text{O}_2$ were also known.

During the 90ties there was put little focus on the Li stoichiometry. So the patents above just claim LiMO_2 , or a range of Li stoichiometries, but it has generally not been understood that the Li:M ratio is an important variable needing optimization. Li_1M_1 was
25 typically seen as a desired stoichiometry which only can be obtained if a small lithium excess is used.

In the late 90ties slowly understanding of the role of excess Lithium evolved. The first document which conclusively shows that additional lithium can be doped into LiMO_2 is JP2000-200607, claiming $\text{Li}[\text{Co}_{1-x}\text{M}_x]\text{O}_2$ and $\text{Li}[\text{Ni}_{1-x}\text{M}_x]\text{O}_2$ where M is at least 2 metals

which have an average valence state of 3. Metals M include lithium, Mn, Co, Ni. Not surprisingly, within the next years several more publications regarding lithium rich ($=\text{Li}[\text{Li}_x\text{M}_{1-x}]\text{O}_2$) materials were published. To our knowledge, the first disclosure of the possibility of excess lithium, doped into the crystal structure of LiMO_2 ($\text{M}=\text{Mn}, \text{Ni}, \text{Co}$) was JP11-307097, claiming $\text{Li}_{(1-a)}\text{Ni}_{1-b-c-d}\text{Mn}_b\text{Co}_c\text{M}_d\text{O}_2$ where $-0.15 < a < 0.1$, $0.02 < b < 0.45$, $0 < c < 0.5$ and $0 \leq d < 0.2$. The formula of claim 1 Li_xMO_2 (if $x=1.05$ $\text{Li}_{1.05}\text{MO}_2$) at first glance contradicts today's consent that it be better written as $\text{Li}_{1.025}\text{M}_{0.975}\text{O}_2$, i.e. there is a slight discrepancy between the oxygen stoichiometry, the first formula having a slightly lower (Li+M):O ratio. Both formulas describe the same material, and furthermore, none of them, describes the material completely accurate, simply because any "real" material possibly has a certain number of other disorder parameters like oxygen or cationic vacancies or interstitials, different composition on the surface etc. .

Thus <1998 prior art can be defined as all solid solutions within the ternary system $\text{LiNiO}_2 - \text{LiCoO}_2 - \text{LiNi}_{1/2}\text{Mn}_{1/2}\text{O}_2 - \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$.

Most of the hundreds of recent publications focus on compositions $\text{Li}[\text{Li}_x\text{M}_{1-x}]\text{O}_2$ with $\text{M}=\text{Mn-Ni-Co}$, almost exclusively the Ni:Mn ratio is 1, and in many cases the compositions is either $\text{M}=\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$ or $(\text{Mn}_{1/2}\text{Ni}_{1/2})_{1-x}\text{Co}_x$ with $0.1 < x < 0.2$. It can be argued that there is a common consent that an excess of lithium ($\text{Li:M} > 1$) is desired to obtain high rate capabilities.

Another issue is doping to alter the cathode materials. Above mentioned JP3561607 claims lithium nickel-cobalt-manganese oxide doped with at least 1% of a further dopant, chosen from Al, B, Si, Fe, V, Cr, Cu, Zn, Ga, and W. The patent does not show or explain why these particular dopants were chosen. JP3141858 disclosed fluorine doped cathode materials, whereas JP3355102 discloses doped (Mn, Co, B, Al, P, Mg or Ti) LiNiO_2 with a BET surface area of $0.01 - 0.5 \text{ m}^2/\text{g}$, containing less than 0.5 % SO_4 .

Another issue is the shape of X-ray diffraction peaks. Sharp peaks with narrow FWHM (full width at half maximum) are related to high crystallinity. JP3653409 (Sanyo) claims a doped LiNiO_2 with FWHM of the main peak at 003 of $0.15-0.22 \text{ deg}$ of 2θ , using Cu – K alpha radiations.

JP3301931 (Sanyo) claims a doped ($> 1\%$) LiNi-Mn-Co oxide where the main 003 peak (at 18.71 ± 0.25) has a FWHM < 0.22 degree.

Despite of the impressive numbers of prior art – it is still not fully clear which compositions within the ternary triangle $\text{LiNiO}_2 - \text{LiCoO}_2 - \text{LiNi}_{1/2}\text{Mn}_{1/2}\text{O}_2 -$
5 $\text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$ gives the best performance in terms of capacity and rate performance.

The overall development of cathode materials involves improving parameters which matter in the batteries. Some of the parameters are relatively easy to measure, like capacity, voltage profile and rate performance, which can be measured by making and testing coin cells. Other parameters are less obvious. So it is not fully clear how safety or
10 swelling properties (e.g. of charged polymer batteries during storage at elevated temperature) can be measured, without assembling real batteries. There exists a strong indication that these safety and storage parameters are not only determined by the chemical composition of the cathode but also by surface properties. However, reliable previous art in this area is rare.

15 In this respect, the authors observed a problem that resides in the reaction of the surface of the active lithium transition metal oxide cathode material and the electrolyte in the battery, leading to poor storage properties and a decreased safety of the battery. The authors argue that lithium located near to the surface thermodynamically is less stable and goes into solution, but lithium in the bulk is thermodynamically stable and cannot go to dissolution.
20 Thus a gradient of Li stability exists, between lower stability at the surface and higher stability in the bulk. By determining the "soluble base" content, based on the ion exchange reaction ($\text{LiMO}_2 + \delta \text{H}^+ \leftrightarrow \text{Li}_{1-\delta}\text{H}_\delta\text{MO}_2 + \delta \text{Li}^+$), the Li gradient can be established. The extent of this reaction is a surface property.

25 To improve safety, aluminum doping of LiNiO_2 based cathodes, as well as Al, Mg-Ti or Ni-Ti doping of LiCoO_2 has been frequently disclosed, for example in JP2002-151154 (Al+Co doped LiNiO_2) or JP2000-200607 (doped LiCoO_2). Typical for doping is that the doped element fits to the host crystal structure, which limits doping of LiMO_2 more or less to transition metals, Li, Mg, Ti, Al, and maybe B. Several disclosures show anionic

doping, like fluorine doping, phosphor doping or sulfur doping. It is however very questionable if these anions can replace oxygen because they differ significantly in size or valence. It is more likely that they instead are present at the surface and grain boundaries as lithium salts. The lithium salts LiF, Li₃PO₄ and Li₂SO₄ all have high
5 thermal stability which promotes a thermodynamic co-existence with the LiMO₂ phase.

In general doping is the modification of the bulk structure, whereas, for safety and storage properties, the surface chemistry is more important. Unfortunately, in many cases, the improvement of surface properties is more than outweighed by the deterioration of bulk properties. Typical examples are the doping by aluminum, where better thermal stability
10 often is accompanied by a dramatic decrease of power (rate performance).

An alternative approach, widely disclosed in the literature is coating. An early disclosure of a coated cathode was KR20010002784, where a LiMO₂ cathode (M=Ni_{1-x}Co_x) (or the sulfur or fluorine “doped” LiMO₂ cathode is coated with a metal oxide with metal selected from Al, Al, Mg, Sr, La, Ce, V and Ti and the stoichiometric amount of metal is at least
15 1%.

An alternative approach is the creation of core-shell cathode materials, or gradient type cathode materials. Here a thick and dense shell of a more robust cathode material protects a core of a more sensitive cathode material. Depending on sintering temperature and chemical composition, the final cathode has either a core-shell morphology or a gradient
20 morphology. Typically both the shell and the core are electrochemically active (have reversible capacity). Examples are found in US2006105239 A1, US2007122705 A1 or US2002192552 A1.

Sulphate is an impurity of concern in layered lithium transition metal oxides. Sulphate typically originates from the mixed hydroxide precursors. This is because the mixed
25 hydroxide preferably is precipitated from transition metal sulphate solution, which is the cheapest water soluble transition metal precursor. Complete removal of sulfur is difficult and increases the cost of the precursor. The sulphate impurity is suspected to cause (a) poor overcharge stability and (b) contribute to the highly undesired low Open Circuit Voltage (OCV) phenomena, where a certain fraction of batteries show a slow deterioration

of OCV after initial charge. Sulphate impurities normally measured when using transition metal sulphate solutions in the manufacturing process can be up to 5 wt%. Finally, manufacturers are frequently confronted with the presence of very fine particles in the cathode materials. This is highly undesired because very fine particles - in the final battery - might electromigrate across the separator, depositing on the anode and causing so-called "soft shorts". These "soft shorts" are highly undesired because they might cause field failure of batteries.

It is an object of this invention to develop lithium transition metal oxide cathode materials having improved electrochemical properties, like capacity, voltage profile and rate performance; besides offering solutions to safety and storage problems that are not only determined by the chemical composition of the cathode but also by surface properties. Also the presence of "soft shorts" can be eliminated.

The invention discloses a powderous lithium transition metal oxide having a layered crystal structure $\text{Li}_{1+a}\text{M}_{1-a}\text{O}_{2\pm b}\text{M}^1_k\text{S}_m$ with $-0.03 < a < 0.06$, $b \approx 0$, M being a transition metal compound, consisting of at least 95% of either one or more elements of the group Ni, Mn, Co and Ti; M^1 being present on the surface of the powderous oxide, and consisting of either one or more elements of the group Ca, Sr, Y, La, Ce and Zr, with $0.0250 < k \leq 0.1$ in wt% and $0.15 < m \leq 0.6$, m being expressed in mol%. Preferably $0.25 \leq m \leq 0.6$. Preferably also M is consisting of at least 99% of either one or more elements of the group Ni, Mn, Co, Al, Mg and Ti. Preferably M^1 is Ca, with $0.0250 \leq k < 0.0500$, and preferably $k \leq 0.0400$.

In another preferred embodiment $M = \text{Ni}_x\text{Mn}_y\text{Co}_z$ with $0.1 \leq x \leq 0.7$, $0.1 \leq y \leq 0.7$, $0.1 \leq z \leq 0.7$, and $x+y+z=1$. In a special embodiment $1.0 \leq x/y \leq 1.3$ and $0.1 < z < 0.4$, and M comprises 10-15 at.% of Ni^{2+} , and preferably 11.5 -13.5 at.% per total metal $\text{Li}_{1+a}\text{M}_{1-a}$. For M, most preferred is $x=y=z=0.33$.

5

This invention demonstrates that the surface properties, determining the safety and stability of cathodes in real batteries – the surface properties being measured as base content by pH titration –are strongly determined by the sulfur and the content of elements like Ca, Sr, Y, La, Ce and Zr.

- 10 At least 150 ppm M' (preferably Ca, Sr, Y, La, Ce and Zr) is needed to achieve the beneficial effect, if the M' addition level is too high (> 1500 ppm), the electrochemical properties suffer, particularly the rate performance decreases and the irreversible capacity increases. In a preferred embodiment sulfur levels of 0.15 – 0.6 mol% can be tolerated if 150-1500 ppm of Ca impurity is present. It was found that 0.15-0.6 mol% of sulfur is
- 15 harmful to the cathode performance if the Ca doping is lower than 150 ppm.

It is not known and has not been published that Li-Ni-Mn-Co cathode materials, over a wide stoichiometric range, show a better performance if they contain a certain concentration of divalent nickel. There is no prior art that teaches that there exists an optimum Li:M stoichiometric ratio, corresponding to a content of 11.5 -13.5 % of divalent

20 nickel per metal in the cathode. The actual invention discloses that, surprisingly, the requirement of 11.5-13.5% of divalent nickel relates lithium excess and Ni:Mn ratio in a simple manner. This involves that in some cases, surprisingly, a certain lithium deficiency is preferred.

The invention also covers an electrochemical cell comprising a cathode comprising as

25 active material the powderous lithium transition metal oxide as described above.

The lithium transition metal oxide can be prepared by a cheap process, for example by a single firing of a mixture of a suitable precursor and lithium carbonate in air. Preferably

the precursor is a mixed metal precursor like mixed hydroxide, oxyhydroxide or carbonate, already containing adequate amounts of sulfur and calcium. Hence, the invention further covers a method for preparing the powderous lithium transition metal oxide described above, comprising the steps of:

- providing for a mixture of M-sulphate, a precipitation agent, preferably NaOH or Na_2CO_3 , and a complexing agent, hereby
- precipitating a M-hydroxide, -oxyhydroxide or -carbonate precursor from said mixture having a given sulfur content,
- ageing said precursor whilst adding a base, thereby obtaining a certain base:precursor ratio, followed by washing with water, and drying,
- mixing said aged M-hydroxide or -oxyhydroxide precursor with a Li precursor,
- sintering said mixture at a temperature T of at least 900°C , and preferably at least 950°C , for a time t between 1 and 48 hrs, thereby obtaining a sintered product; where either:
- a salt of M^1 is added to said M-sulphate containing mixture, or
- M^1 is added to said base during ageing, or
- M^1 is added to the water used in said washing step, or
- a M^1 salt solution is added to a slurry prepared by suspending said sintered product in water, followed by drying.

Where a salt of M^1 is added to the M-sulphate containing mixture, this can be to the M-sulphate itself, to the hydroxide (NaOH) or the complexing agent.

In the method, preferably $\text{M}^1=\text{Ca}$ and the salt is either one of $\text{Ca}(\text{NO}_3)_2$ and CaCl_2 . It is preferred that the sulfur content is controlled during the ageing step by selecting a given base:precursor ratio.

The actual invention discloses that the application of less than one monolayer of a suitable element, particularly Ca, dramatically changes the surface properties of layered lithium

transition metal oxides $\text{Li}_{1+x}\text{M}_{1-x}\text{O}_2$, $\text{M} = \text{Ni-Mn-Co}$, with $-0.03 < x < 0.06$. Calcium is a suitable element but it is very likely that other elements can be added, typical candidates being rare earths and earth alkali metals, as well as Zr, Pb, Sn.

Surface modified cathode materials are prepared in a single step. Precursors can be
5 enriched by e.g. Ca to reach a concentration of 150-1500 ppm. These precursors are used to prepare surface modified LMO by a single cook. If the Ca level of the precursors is lower, then Ca can be added to the precursor, preferably in liquid form, by a technique which the authors call slurry doping. High surface area precursor (for example mixed
10 hydroxide) is dispersed in as little as possible water (or any other solvent) to form a paste of high viscosity. During rigid stirring a dissolved calcium salt like CaCl_2 or $\text{Ca}(\text{NO}_3)_2$ is slowly added until the desired concentration is reached. During addition, and during the following drying, calcium precipitates and is well-dispersed onto the surface of the mixed hydroxide.

Alternatively the calcium can be added during the precursor preparation process. This is
15 possible by adding a small concentration of calcium (typically less than 100 ppm) to the water used to dissolve the metal salt (for example MSO_4) precursor or base (NaOH) precursor. Alternatively Ca can be added in higher concentration to the water used to wash the precursor after finished precipitation.

The surface modification by calcium is possibly a catalytic de-activation of active surface
20 sites, because (a) Calcium has a much larger ionic radius and cannot be doped into the bulk structure and (b) up to 1500 ppm Ca is simply not enough to form a coating layer, as shown below. Here the word coating is used in the conventional sense as a layer consisting of at least 10-100 atomic layers, corresponding to a few nm to about 100 nm thickness. The authors speculate that the mechanism of de-activation is related to a
25 phenomenon known from catalyst technology, called catalyst poisoning. During operation of a catalyst (for example platinum in a gas containing traces of sulfur species) trace amounts can de-activate the catalyst by covering catalytically active sites.

The complex layered lithium transition metal oxides are solid state solutions within the ternary system $\text{LiNiO}_2 - \text{LiCoO}_2 - \text{LiNi}_{1/2}\text{Mn}_{1/2}\text{O}_2 - \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$ additionally

including the possibility of lithium deficient cathodes $\{\text{Li}_{1-x}\text{M}_x\}\text{MO}_2$ and not excluding the possibility of cation mixing



The authors discovered that the optimum Li:M ratio depends on the metal composition.

- 5 The authors investigated several metal compositions $\text{M}=\text{Ni}_{(1-a-b)}\text{Mn}_a\text{Co}_b$ by measuring the electrochemical performance and base content of test samples as a function of Li:M ratio. Typically “ $\cap$ ” shaped curves (similar as figure 2 of Japanese Patent application 10-109746) are obtained. The capacity is typically a relatively flat maximum, deteriorating fast with lower Li:M and more slowly with higher Li:M. The authors discovered that the
- 10 maximum (=optimum) of these “ $\cap$ ” shaped curves appears at different Li:M ratio's, where the optimum Li:M ratio depends on the metal composition. Particularly, the optimum depends on the Ni composition and the Ni:Mn stoichiometric ratio. The authors discovered that the optimum region is related to the content of divalent nickel as described below:

- 15 $\text{M}''\text{O}_2$ is a layered ordered rock salt compound with $\text{M}''=\text{Li}_{1+k}\text{M}_{1-k}$ where M contains a mixture of manganese, cobalt and nickel, $-0.03 < k < 0.06$. If $k > 0$ then the formula corresponds to a solid state solution of the ternary system $\text{LiNiO}_2 - \text{LiCoO}_2 - \text{LiNi}_{1/2}\text{Mn}_{1/2}\text{O}_2 - \text{Li}[\text{Li}_{1/3}\text{Mn}_{2/3}]\text{O}_2$ and can be rewritten as $\text{Li}[\text{Li}_{x/3}\text{Mn}_{2x/3}\text{Mn}_{y/2}\text{Ni}_{y/2}\text{Co}_z\text{Ni}_{1-x-y-z}]\text{O}_2$. In this formula all Mn is tetravalent, all cobalt is trivalent and the $y/2$ Ni is
- 20 divalent whereas the $1-x-y-z$ Ni is trivalent. If $k < 0$, furthermore assuming that divalent nickel substitutes for lithium sites, the formula can be rewritten as $\{\text{Li}_{1-x/3}\text{Ni}_{x/3}\}[\text{Mn}_{y/2-2x/3}\text{Ni}_{y/2-x/3}\text{Co}_z\text{Ni}_{1-x-y-z}]\text{O}_2$. In this formula all Mn is tetravalent, all cobalt is trivalent and the $y/2$ Ni is divalent whereas the $1-x-y-z$ Ni is trivalent.

- The authors observed that the optimum Li:M ratio $(= (1+x/3) / (1-x/3))$ sensitively
- 25 depends on the transition metal composition, and corresponding to a quite narrow stoichiometric range of Ni^{II} , which again leads to optimized electrochemical properties. It is preferred that the Li:M is chosen so that divalent nickel comprises not less than 10% and not more than 15% of the total metal $\text{M}'' (= \text{Li}-\text{M})$. More preferred, divalent nickel comprises not less than 11.5 and not more than 13.5 at% of the total metal.

This requirement is strictly valid for layered Li-M-O₂ within a certain transition metal stoichiometric range. The requirement becomes less accurate if the sample is “high Ni”, i.e. Ni_{1-x-y}Co_xMn_y with $1-x-y > 0.6$, especially if $y < 0.3$. The requirement is also less valid if the sample is “low Ni & low Co”, i.e. Ni_{1-x-y}Co_xMn_y with Ni:Mn < 1.3 and $x < 0.2$.

5 The requirement of course makes no sense for samples which do not contain enough nickel, i.e. Ni_{1-x-y}Co_xMn_y with $1-x-y < 0.2$. In the first case (high Ni) there is a trend that more Ni²⁺ is required to obtain good electrochemical performance. In the latter case (low Ni & Co) there is a trend that less Ni²⁺ is needed. In a medium stoichiometric range Ni_{1-x-y}Co_xMn_y (i.e. with $0.1 < x < 0.4$ and $1.0 \leq 1-x-y/y \leq 1.3$) the best electrochemical
10 properties are obtained if the Ni²⁺ comprises between 10-15 at%, more preferred 11.5 – 13.5 at% of the total M".

The amount of base which goes to dissolution (soluble base content) is directly related to the surface properties of the cathode. Since surface properties of the cathode dominate the stability (i.e. safety and overcharge/high T storage properties of the real battery) there will
15 be a correlation between base content and stability. The present invention shows that there is a surprising correlation between base content and Ca content (ppm range) and sulfur content (0.1% range). Certainly, to obtain highly stable cathodes, the optimization of the Ca and sulfur content is important.

The amount of base dissolving is a function of BET surface area, composition of the bulk
20 and dopants, particularly Ca, on the surface. Somehow Ca stabilizes the lithium in the surface region and causes less lithium to dissolve. The increased stability of lithium on the surface causes beneficial properties of the cathode in the battery, like improved storage properties and better safety.

Sulfur also contributes to the amount of specific base (base per surface). The authors
25 believe that this is mostly due to a closure of pores of Li-M-oxide by sulfur salts, which cause low BET surface areas to be measured. In the presence of sulfur the “real surface area” of the Li-M-Oxide is much larger, then measured by BET, so the base content increases. Therefore the present invention teaches, that if sulfur is present, then also elements like Ca must be present to effectively lower the base content to an acceptable
30 level.

The invention is further explained by the following Examples and Figures. The Figures are summarized as follows:

- 5 Fig. 1: Correlation of Ca concentration and soluble base content
- Fig. 2: Correlation of Ca concentration and irreversible capacity and rate performance
- Fig. 3: Correlation of Ca and S concentration with soluble base content
- Fig. 4: Correlation of Ca and S concentration with first cycle discharge capacity
- Fig. 5: Settling down kinetics of Ca treated LiMO_2
- 10 Fig. 6: Settling down kinetics of Ca free versus Ca treated LiMO_2

Example 1: Improved safety and lower base content of Ca containing cathode

2 cathode materials MP1 and MP2 with composition $\text{Li}_{1+a}\text{M}_{1-a}\text{O}_{2\pm b}\text{Ca}_k\text{S}_m$ were produced at large scale (several tons) from mixed transition metal hydroxide, which contained
 15 different amounts of Ca and sulfur. In both cases the stoichiometry was very similar ($a=0.05$, $\text{M}=\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$, $m\approx 0.4$ mol%) but the level of Ca was different : MP1 had 393 ppm Ca, whereas MP2 had a normal impurity level of 120 ppm Ca (normally more than 50 but less than 150 ppm is found in non-doped cathode material). Other properties (lithium stoichiometry, particle size, BET surface area, X-ray diffraction pattern) were
 20 basically similar.

The content of soluble base was measured as follows: 100 ml of de-ionized water is added to 7.5g of cathode, followed by stirring for 8 minutes. Settling-down is allowed for typically 3 minutes, then the solution is removed and passed through a 1 μm syringe filter, thereby achieving > 90g of a clear solution which contains the soluble base.

25 The content of soluble base is titrated by logging the pH profile during addition of 0.1 M HCl at a rate of 0.5 ml/min until the pH reaches 3 under stirring. A reference voltage profile is obtained by titrating suitable mixtures of LiOH and Li_2CO_3 dissolved in low concentration in DI water. In almost all cases two distinct plateaus are observed. The upper plateau is $\text{OH}^-/\text{H}_2\text{O}$ followed by $\text{CO}_3^{2-}/\text{HCO}_3^-$, the lower plateau is $\text{HCO}_3^-/\text{H}_2\text{CO}_3$.

The inflection point between the first and second plateau as well as the inflection point after the second plateau is obtained from the corresponding minima of the derivative $d pH / d Vol$ of the pH profile. The second inflection point generally is near to pH 4.7. Results are listed as micromole of base per g of cathode.

- 5 The amount of base which goes into solution is very reproducible, and is directly related to surface properties of the cathode. Since these have a significant influence on the stability (i.e. safety and overcharge/high T storage properties of the final battery) there is a correlation between base content and stability.
- 10 Table 1A and 1B summarize the results:

Table 1A: Properties of sample MP1 and MP2

Sample	Li % wt	Ni % wt	Mn % wt	Ca ppm	S % wt	Co % wt	PSD (μm)			TapD g/cm^3	BET m^2/g
							D10	D50	D90		
MP1	7.568	19.573	18.625	393	0.087	19.441	4.3	6.9	10.8	2.07	0.42
MP2	7.523	19.733	18.439	120	0.148	19.707	3.7	6.4	10.5	2.09	0.44

TapD: tap density; 0.087 and 0.148 wt % S corresponds to approx. 0.3 and 0.5 mol% S.

- 15 Table 1B: Properties of sample MP1 and MP2

Sample	Unit cell (X-ray)			Soluble Base $\mu\text{mol/g}$	Qrev 4.3- 3.0V mAh/g	Qirr %	Rate versus 0.1C %			Safety Over charge
	a ($\text{\AA}$)	c ($\text{\AA}$)	Vol, $\text{\AA}^3$				3C	2C	3C	
MP1	2.8590	14.2327	33.584	25.9	155.0	11.8	86.0	89.1	86.0	Pass
MP2	2.8594	14.2337	33.595	51.2	156.3	10.9	86.6	89.1	86.6	Fail

The samples are very similar, with one exception: the soluble base content of sample MP1 (with high Ca) was significantly lower than for MP2. Other properties are very similar,

and although MP2 (with low Ca) shows slightly higher capacity, slightly lower irreversible capacity and slightly higher rate performance, the results for MP1 are still acceptable. More important, the samples MP1 and MP2 were sent to battery producer for safety testing. Whereas MP1 passed the safety test, MP2 did not pass.

5

The "Safety overcharge test" used here is a safety test where a battery is charged at a very high rate (for example with 1C charge rate) until a much higher voltage than the normal operating voltage (for example 20V) is reached. In many cases during such a test more lithium is extracted from the cathode than can be inserted to the anode, so the dangerous effect of lithium plating occurs. At the same time the highly delithiated cathode is in a highly reactive state, and ohmic (resistive) heat is generated. The heat can initiate the dramatic thermal run-away reaction, ultimately leading to the explosion of the battery. If a battery passes such a test (i.e. does not explode) or not is strongly dependent on the choice of cathode material, its morphology, impurity levels and its surface chemistry. Very little fundamental scientific understanding exists, but the presence of fine particles definitively contributes to poor safety. (see also below)

Conclusion: the higher content of Ca caused lower soluble base content and higher safety.

Example 1 showed that a Ca content of approx. 250-400 ppm effectively lowered the base content and improved the safety of the cathode. If we now estimate the number of atomic layers on top of the surface of the cathode, assuming that

- a) all of the calcium is located at the surface of the cathode particles,
- b) the surface area of the cathode is reliably obtained by 5 point BET measurement using nitrogen,
- c) Calcium is evenly distributed on the surface,
- d) the average distance between Ca atoms is the same as in CaO;

then it can be concluded that the effect of Ca is rather a catalytic effect (less than a few one atomic layer) and not caused by a conventional coating effect (many layers of atoms). This is shown in

5 Example 2: calculation of the "thickness" of the Ca surface layer.

The estimation, based on the data of Example 1, goes as follows:

CaO has an fcc crystal structure with $4.8108 \text{ \AA}$ lattice constant; thus nearest neighbors form tetrahedrons with $3.401 \text{ \AA}$ side length. Thus a one-atom monolayer of Ca (having a hexagonal 2-dim lattice with $3.401 \text{ \AA}$ lattice constant) corresponds to a density of 0.664
 10 mg / m^2 . The cathode material MP1 (MP2) of Example 1 has a BET area of $0.42 \text{ m}^2/\text{g}$ ($0.44 \text{ m}^2/\text{g}$). A monolayer covering this BET area corresponds to 280 ppm (292 ppm) Ca. Therefore sample MP1 has a surface coverage of approx. 1.4 monolayers and MP2 has a coverage of only 0.41 monolayer of calcium. This is much thinner than conventional coating.

15 It can be concluded that the observed effect of calcium is not a protection by a coating layer but rather a catalytic effect (de-activation of active surface sites)

Example 3: theoretical background: Base content / Ca chemistry

It might be argued that a possible dissolution of Ca somehow interferes with the solubility
 20 of lithium or base, thus causing the observation of lower base content for samples with higher Ca. This argumentation is wrong.

First, Lithium compounds have higher solubility than corresponding Calcium compounds. Secondly, this example shows that the amount of Calcium is negligible, thus it cannot change the solubility of Li or base during the pH titration measurement.

25 We use samples MP1 and MP2 of Example 1 to make the following estimations:

25.9 μmol of base per g of cathode are titrated for sample MP1.

51.2 μmol are titrated for the lower Ca sample MP2.

Thus the content of soluble base differs by 25.2 $\mu\text{mol} / \text{g}$.

MP1 has 393 ppm Ca, MP2 has 120 ppm Ca. This is a difference of Ca content of 271 ppm.

The molar weight of Ca is 40.1 g/mol.

A simple calculation yields that the difference in Calcium is $271/40.1=6.76 \mu\text{mol/g}$.

- 5 We conclude that an increase of Ca by only $6.76 \mu\text{mol/g}$ causes a much larger decrease of base by $25.2 \mu\text{mol}$. The large decrease can only be explained if we accept that Ca stabilizes the surface so that less Li goes into solution.

Example 4: Base content as a function of Ca content – different precursors

- 10 Example 1 demonstrated that the low Ca sample MP2 had higher base content than high Ca sample MP1. This is confirmed in Example 4 by detecting a good correlation between lower base content and higher Ca content for a larger series of samples with similar morphology and composition (Li, Mn, Co, Ni, S).

- Ten transition metal hydroxide precursors from a mass production batch were received,
15 denominated MOOH1-10. The hydroxides have a metal composition of $M=\text{Ni}_{1/3}\text{Mn}_{1/3}\text{Co}_{1/3}$. Ten samples - lithium transition metal oxide samples S1a-S10a (each approx. 250g) - were prepared with a Li:M=1.1 blend ratio (according chemical analysis of the precursor) at a temperature of 960°C in air. The lithium content was checked (by comparing the unit cell volume) and the BET surface area was measured. All samples had
20 a very similar morphology (particle size distribution, tap density, particle shape, SEM micrographs, crystallite size).

- The Ca content of all precursors was obtained by chemical analysis. The content of Ca in the final product is the same as in the precursor. The crucibles do not contain Ca, evaporation is not observed, and Ca practically does not diffuse into the crucible. The
25 soluble base content of the Li-M-oxide samples was measured by pH titration.

Tables 2A and Figure 1 summarize the results.

Table 2A Properties of samples S1a-S10a prepared from transition metal hydroxide precursors.

Precursor	Sample	Ca ppm	SO ₄ wt%	a (Å)	c (Å)	Vol, Å ³	BET m ² /g	Base μmol/g
MOOH1	S1a	170	0.489	2.8597	14.2353	33.605	0.43	77.0
MOOH2	S2a	180	0.441	2.8599	14.2353	33.610	0.42	78.1
MOOH3	S3a	150	0.456	2.8602	14.2363	33.620	0.41	83.4
MOOH4	S4a	130	0.465	2.8597	14.2348	33.604	0.41	85.9
MOOH5	S5a	65	0.489	2.8595	14.2338	33.599	0.4	99.2
MOOH6	S6a	190	0.486	2.8599	14.2353	33.610	0.43	74.7
MOOH7	S7a	170	0.51	2.8594	14.2341	33.596	0.42	82.9
MOOH8	S8a	200	0.525	2.8596	14.2357	33.605	0.42	78.1
MOOH9	S9a	190	0.319	2.8595	14.2347	33.600	0.37	88.4
MOOH10	S10a	200	0.525	2.8595	14.2344	33.599	0.38	85.0

- 5 Then a second series of test samples S1b-S10b (each approx. 700g) was prepared. The temperature and Li:M blend ratio was corrected slightly to achieve samples with a more narrow distribution of BET and identical final Li:M ratio. Table 2B and Figure 1 summarize the results. It contains the data of Table 2A (A: bullets ●) and data of some further samples (mass production samples), indicated as stars (B: ☆)

Table 2B Properties of samples S1b-S10b prepared from transition metal hydroxide precursors

Sample	Ca ppm	SO ₄ wt%	a (Å)	c (Å)	Vol, Å ³	BET m ² /g	Base μmol/g	Qrev 4.3- 3.0V mAh/g	Qirr %	Rate versus 0.1C %		
										1C	2C	3C
S1b	200	0.500	2.8597	14.2331	33.601	0.42	29.5	156.5	11.3	92.3	88.9	86.7
S2b	190	0.440	2.8599	14.2320	33.603	0.37	30.6	157.4	13.5	92.4	89.1	86.9
S3b	180	0.460	2.8598	14.2317	33.600	0.39	33.1	156.8	10.6	92.6	89.1	86.3
S4b	170	0.480	2.8597	14.2322	33.599	0.39	32.3	157.4	10.1	92.6	89.1	86.9
S5b	110	0.500	2.8599	14.2310	33.601	0.42	36.0	155.6	10.6	92.7	89.2	86.9
S6b	200	0.500	2.8592	14.2319	33.586	0.42	31.0	156.6	11.1	92.5	89.0	86.7
S7b	200	0.530	2.8601	14.2333	33.611	0.39	28.8	155.2	11.2	92.4	88.7	86.2
S8b	210	0.530	2.8595	14.2322	33.594	0.40	27.6	154.9	11.5	92.5	88.6	85.0
S9b	230	0.580	2.8595	14.2329	33.596	0.37	28.5	155.5	11.6	92.4	88.6	85.3
S10b	230	0.560	2.8593	14.2331	33.591	0.35	30.5	155.3	11.5	92.47	88.70	85.34

Apparently, there exists a clear correlation between increasing Ca content and lower
5 soluble base content. The example confirms that a small amount of Ca dramatically
decreases the amount of soluble base, without much deteriorating the electrochemical
performance: a slight increase of irreversible capacity and a slight deterioration of rate
performance are observed. As expected, the normal impurity level of Ca (<150 ppm) gives
the worst results for base content. Figure 2 summarizes the measured electrochemical
10 properties as function of calcium content, taken from Tables 2A and B (indicated by
bullets **○**). The left figure plots the irreversible capacity (%) vs. Ca content, the right
figure the rate performance at 2C (%) vs. Ca content. Data for irreversible capacity of
some further samples (mass production samples) were added to the Figure as stars (☆).

In practice it is worth to accept the slight deterioration of rate performance if this allows to dramatically lower the base content, thus achieving improved high temperature stability and safety of real cells.

5 Example 5: Soluble base content and Electrochemical performance as a function of effective S-Ca content

Most samples of Example 4 had a similar level of sulfur. Example 5 will show that the content of Ca and the content of sulfur completely determines the soluble base content as well as other properties (electrochemically performance) for a larger series of mass scale
10 production samples (> 500 kg sample size). The samples had the same composition (Li, Mn, Ni, Co) but differed in Ca and Sulfur content.

Data analysis showed that Ca has a negative regression coefficient versus the soluble base content, whereas the SO₄ content has a positive regression coefficient. This allowed to define a statistical variable k being the “effective S-Ca” content by $k = 0.84 * S - Ca$
15 where S and Ca are the ppm results of the ICP analysis for S and Ca. The formula can be interpreted as the statistical proof that a higher content of sulfur can be neutralized by addition of Ca.

Figure 3 shows that there is a very good correlation between effective S-Ca content and soluble base content. Both Ca and Sulfur correlate reasonable well with base content. The
20 top left figure gives the soluble base content (μmol/g) vs. Ca content, the bottom left figure gives the same against the SO₄ content. A statistical variable k (a linear combination of $0.84 * S$ (ppm) – Ca (ppm)) shows an almost perfect positive correlation. The correlation coefficient is + 0.95. This is shown on the right figure.

Surprisingly, there is also a very good correlation between soluble base content (μmol/g)
25 and electrochemical performance, as shown in Figure 4. Here the electrochemical performance is given by the discharge capacity of the first cycle (1st cycle DC Q - in mAh/g). The correlation factor is 0.94.

Figure 3 and Figure 4 are important examples showing the need to control very well the Ca level and S levels. Note that the base content varies by almost 100%, and the discharge capacity by 5%, these are comparably huge numbers considering that the Ca content varies by less than 600 ppm and the sulfur content by about 0.25 mol%

5

Example 6: Optimization of Ca and Sulfur additions.

This Example serves to demonstrate 2 aspects of the invention:

- (1) it confirms the observation of Example 5 that Ca “neutralizes” the negative effect of a
10 high soluble base content of sulfur containing cathodes, and
(2) it demonstrates that only samples which contain both sulfur and calcium according to the invention show good overall performance.

The Example uses a mixed transition metal hydroxide precursor with metal composition $M = \text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$. The precursors naturally are low in Ca but contain some sulfur. The
15 sulfur is removed after preparation of a preliminary Li-M-Oxide sample (Li:M = 1.1) by washing. Then the preliminary sample is used as precursor, and the following material matrix is prepared:

- (6a): no addition of sulfur or calcium
(6b): addition of 400 ppm Ca
20 (6c): addition of 0.5 wt% SO_4
(6d): addition of both 400 ppm Ca and 0.5 wt% SO_4 ,

This is followed by a re-sintering. Final samples with the same morphology but different Ca, S composition are obtained. The addition of Ca and S is performed by slurry doping of the Li-M-oxide preliminary sample (also described below in example 7). Slurry doping is
25 the drop-wise addition of a Li_2SO_4 solution or of a $\text{Ca}(\text{NO}_3)_2$ solution during stirring of a preliminary sample powder-in-water slurry of high viscosity, followed by drying in air. A total of 400 ppm Ca and/or 5000 ppm (SO_4) sulfur was added. Note that 1000 ppm of sulfate generally corresponds to approx. 0.1 mol% of sulfur, more accurate – for $\text{Li}_{1.04}\text{M}_{0.96}\text{O}_2$ 1000 ppm correspond to 0.105 mol %.

The experiment was repeated for a precursor with $M=\text{Ni}_{0.53}\text{Mn}_{0.27}\text{Co}_{0.2}$ composition, where the preliminary sample – the precursor during slurry doping - was prepared using a Li:M=1.02 blend ratio. The conclusions of Example 5 (neutralization of sulfur by Ca) are confirmed: if the sample contains sulfur, the addition of Ca neutralizes the high soluble
 5 base content caused by the sulfur.

Electrochemical properties are tested, and settling down kinetics is measured (see also Example 8 for more details). The sample without added Ca showed the highly undesired fine particles which do not settle down. All samples with Ca settled down very fast. Of all samples – only the sample which contains Ca *and* sulfur show overall good
 10 performances, as can be seen in Tables 3A and 3B.

Samples situated outside the claimed concentrations (either too high or too low) show the following disadvantages:

Low Ca & low $\text{SO}_4 \rightarrow$ unacceptable level of fine particles

Low Ca and high $\text{SO}_4 \rightarrow$ high soluble base content, fine particles

15 High Ca and low $\text{SO}_4 \rightarrow$ relatively poor electrochemical performances.

(see also below Table 4A)

Table 3A: Slurry doped Li $\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}\text{O}_2$

Slurry doping : Li-M- O_2 M= $\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$ Addition of	BET m^2/g	Ca ppm	SO_4 (wt %)	Base $\mu\text{mol/g}$	Q DC 3.0- 4.3V mAh/g	Q irr (%)	Rate @ 2C (%)
(6a) Nothing	0.41	150	0.180	26.9	157.5	10.95	89.08
(6b) 400 ppm Ca	0.41	500	0.182	20.8	156.3	11.53	88.24
(6c) 0.5 wt% SO_4	0.44	150	0.620	31.0	157.8	10.65	88.84
(6d) 400 ppm Ca, 0.5 wt% SO_4	0.45	510	0.630	23.3	156.4	11.02	88.80

Table 3B: Slurry doped Li Ni_{0.53}Mn_{0.27}Co_{0.2}O₂

Slurry doping : Li-M-O ₂ M= Ni _{0.53} Mn _{0.27} Co _{0.2} Addition of	BET m ² /g	Ca ppm	SO ₄ (wt %)	Base μmol/g	Q DC 3.0- 4.3V mAh/g	Q irr (%)	Rate @ 2C (%)
(6a) Nothing	0.3	120	0.095	37.8	169.1	12.76	87.10
(6b) 400 ppm Ca	0.32	430	0.087	27.0	166.6	13.79	86.83
(6c) 0.5 wt% SO ₄	0.36	110	0.25	58.1	169.6	11.84	87.39
(6d) 400 ppm Ca, 0.5 wt% SO ₄	0.33	440	0.28	49.4	168.1	12.75	87.82

Note that in this test (3B) some of the added SO₄ was lost due to crystallization.

Example 7: Comparing Ca and Mg with same precursor material

5

This example shows data of different samples prepared from one single hydroxide precursor, with varying Ca concentration by addition of different amounts of Ca to the precursor during preparation. As reference Mg was added to confirm the role of Ca .

A hydroxide with low content of Ca (60 ppm) was received. The transition metal composition was approx. Ni_{0.37}Co_{0.32}Mn_{0.31}. Sulfur content was approx. 0.4 wt% SO₄. The hydroxide was divided into smaller samples (each approx. 500g). A water-based slurry of high viscosity was prepared from each sample. The water used to slurry the precursor contained appropriate additions of dissolved CaCl₂ . The slurry was continuously stirred. Thus a Ca doped slurry was achieved which was dried in a convection oven without filtering, resulting in a Ca treated mixed hydroxide. In the same way Mg doped (dissolved Mg(NO₃)₂ was added to the water) and Mg + Ca doped mixed hydroxide was prepared from the same precursor.

15

Six samples (CaAdd1 – CaAdd6) were prepared from the Ca doped mixed hydroxide by mixing with Li_2CO_3 , the Li:M blend ratio was 1.07, followed by a heating at 960°C .

Table 4 gives an overview of the prepared samples. The Ca concentration of the undoped sample is slightly higher than expected (120 ppm), possibly caused by a slight Ca

- 5 dissolution from the baker used during the slurry preparation. The table shows that (a) the content of base decreases with increasing content of Ca and (b) the addition of Mg does not alter the base content at all, (c) the BET surface area decreases with increasing calcium content. (c) indicates that the sintering kinetics speeds up with higher Ca (or Cl contamination) content.

- 10 The base content decreases by 33%, whereas the BET area only decreases by 18%, proving that part of the decreased base content is caused by a different surface chemistry, and not, as could be assumed, by a decrease of the surface area itself. Note that the reduction of base is slightly less than expected, possibly caused by a less than perfect dispersion of Ca on the surface of the precursor during slurry doping.

- 15 Table 4 also shows that the magnesium does not influence the base content at all. The base content however depends on the Ca content, independently of how much Mg is added. The soluble base decreases with increasing calcium level. It is believed that the ionic radius of Mg is too small (0.66 Angstrom) compared to Ca (0.99 Angstrom), the latter having a size that fits very well to the surface of Li-M-oxide - see Example 11 below.

20

25

Table 4: Properties of samples prepared from a single MOOH modified by adding Ca and/or Mg

Sample	Ca added ppm	Mg added ppm	Ca ppm	BET m ² /g	Base μ mol/g	Qrev 4.3-3.0V mAh/g	Qirr %	Rate versus 0.1C %		
								1C	2C	3C
CaAdd1	0	0	120	0.66	67.9	158.5	11.5	90.8	86.7	83.7
CaAdd2	100	0	190	0.64	63.1	157.9	11.8	90.9	86.7	83.7
CaAdd3	400	0	420	0.57	50.8	156.1	12.8	90.9	87.8	85.0
CaAdd4	1000	0	900	0.54	43.5	155.1	12.9	91.1	87.0	84.0
CaAdd5	400	300	0		51.7					
CaAdd6	0	300	0		67.6					

Example 8: Ca level and fine particles

5

As said above, the presence of very fine particles is highly undesired because very fine particles – in the final battery - might electromigrate across the separator, depositing on the anode and causing so-called "soft shorts", leading to field failure of batteries. These particles are normally finer than 1 μ m. It is believed that the decrease of these fine particles is responsible for better safety.

10

Example 8 shows that the addition of Calcium eliminates fine particles, although the mechanism causing this beneficial effect is not fully understood by the authors.

The samples CaAdd1, CaAdd2, CaAdd3 and CaAdd4 (of Example 7) were investigated in a settling experiment. After disposing a cathode material in water it is desired that the particles settle down fast, and that a clear solution remains on top. A slow settling indicates the presence of fine particles.

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Figure 5 shows photographs of a settling down experiment. Ca content: from left to right: (1) 120 ppm (2) 190 ppm, (3) 420 ppm, (4) 900 ppm. After a settling time of 1 minute of 5 g of suspended particles in a 50 ml measuring (graduated) cylinder, the height of the separation line between clear solution and the particle suspension layer was situated at (1) 50, (2) 30, (3) 22, and (4) 13 ml, after 5 min: (1) 49, (2) 11, (3) 9, and (4) 8 ml. Obviously, an increase of Ca impurity causes a dramatic increase of settling kinetics – proving that Ca addition eliminates the presence of fine particles.

As a result of Examples 4 to 8 the following Table 4A gives an overview of the addition of Ca and S.

Table 4A: Overview

	Low Sulfur	Sulfur: 0.15-0.6 mol%	High Sulfur
Low Calcium	too many fine particles	high soluble base content - too many fine particles	very high soluble base content - too many fine particles
Calcium: 150-1500 ppm	poor electrochemical properties	good electrochemical properties - low soluble base content	high soluble base content
High Calcium	very poor electrochemical properties	poor electrochemical properties	poor electrochemical properties

Example 9: Soluble base content is a thermodynamic materials property

This example discusses that the base content is a thermodynamic equilibrium materials property. It can be changed by well-designing the calcium and sulfur content. It can, however not be changed by altering the preparation conditions after being sufficiently
5 equilibrated.

A few kg of the mass production sample MP2 of Example 1 (having a "natural" Ca content of 120 ppm) is used to investigate if the soluble base content can be lowered, depending on heating temperature, air flow, or washing in water followed by reheating. For the washing – the amount of water is limited, and the Li lost is monitored. This figure
10 is negligible, consisting of approx. 0.1% of the total Li in the sample. Reheating temperatures are lower than the initial sintering temperature, thus the morphology does not change during reheating.

The soluble base content of the initially received sample can be slightly lowered by a heat treatment (equilibration), indicating that the lithiation of the MP2 sample is not 100%
15 completed. However, after reheating, independently of heating conditions, the same soluble base content is always achieved. This base content is the equilibrium content, depending of surface area, metal composition and Ca and sulfur level. Washing removes a large fraction of Sulfur - as soluble Li_2SO_4 - but does not remove Ca (this was checked by ICP), resulting in a low Sulfur - low Ca sample. The low sulfur – low calcium sample has
20 a lower soluble base content. After washing, already at low drying temperature (150°C) the same equilibrium value is re-established which is achieved after washing and reheating at 750°C. All these observations are summarized in Table 5.

Table 5:

Sample	Re-heating treatment	Airflow during re-heating	Base $\mu\text{mol/g}$
MP2	As received		51.0
MP2A	Heated to 600°C, 5h	yes	45.5
MP2B	Heated to 750°C, 5h	Yes	45.4
MP2C	Heated to 750°C, 5h	No	46.5
MP2G	Heated to 750°C, 5h extremely low bed-depth	Yes	46.0
MP2E	Washed, dried at 150°C	Yes	25.8
MP2F	Washed, dried at 150°, heated to 750°C	Yes	27.3

Example 10: Comparison of identical morphology with high/ low Ca content

A sample EX10A (1 kg size) is prepared from a mass scale production precursor mixed
 5 hydroxide with metal composition $\text{Mn}_{1/3}\text{Ni}_{1/3}\text{Co}_{1/3}$ by mixing the precursor with Li_2CO_3
 (blend ratio 1.1) followed by heating to 960°C. EX10B is prepared in the same way, with
 the exception that the precursor was modified by the previously described slurry doping: a
 total of 400 ppm Ca was slowly (drop wise) added in the form of $\text{Ca}(\text{NO}_3)_2$ to a water
 based slurry of the precursor, followed by drying (no filtering).

Table 6A and 6B summarize the results

Table 6A

Sample	Ca ppm	BET m ² /g	PSD D10	PSD D50	PSD D90	a hex Å	c hex Å	Vol Å ³	size nm
EX10A	140	0.43	3.79	5.925	9.08	2.8590	14.2259	33.567	281
EX10B	420	0.42	3.78	5.914	9.07	2.8593	14.2316	33.588	254

5 Table 6B

Sample	Qrev 4.3-3.0V mAh/g	Qirr %	Rate versus 0.1C (%)			Base μmol/g
			1C	2C	3C	
EX10A	154.9	10.39	92.85	89.14	85.33	39.6
EX10B	153.8	11.71	92.61	88.99	86.05	26.5

As Tables 6A and 6B show, besides of the Ca impurity level, all 3 samples are, as expected for samples prepared under similar conditions from the same precursor, very similar. The PSD, obtained by laser diffraction are identical. Similar as observed in previous examples – the sample with Ca addition shows the smallest content of soluble base.

Despite that the particle size distribution of sample EX10A and EX10C is identical – the settling down kinetics after dispersing the cathode in water is dramatically different. Figure 6 shows photographs of a settling down experiment of Ca treated LiMO₂ :
 Addition of Ca: (1: left) 0 ppm, (2: right) 400 ppm: after a settling time of 1 minute of 5 g of suspended particles in a 50 ml measuring (graduated) cylinder, the height of the separation line between "clear" solution and the particle suspension layer was situated at (1) 27, (2) 18 ml, only after 5 min the suspended particles in both cylinders have nearly all

settled down. Obviously, an increase of Ca impurity causes a dramatic decrease of fine particles – as a result Ca rich cathodes settle down much faster.

Example 11: Alternative elements besides Ca.

This example uses a mixed transition metal hydroxide precursor with metal composition
 5 M1= $\text{Mn}_{0.33}\text{Ni}_{0.38}\text{Co}_{0.29}$ as precursor. The precursor is low in Ca but, as expected, contains some sulfur. A similar experiment is done with a mixed hydroxide precursor with
 M2= $\text{Ni}_{0.53}\text{Ni}_{0.27}\text{Co}_{0.2}$ composition.

The precursors are doped by slurry doping: 1000 ppm of nitrate solutions of Ca, Y, Sr, La, Ba, Fe are added, respectively. A reference was slurry doped but no metal was added.

10 After slurry doping the precursors were mixed with Li_2CO_3 and cooked. Besides of the doping, final composition (Li, Mn, Ni, Co) was very similar.

To compare the efficiency to lower the base content the following parameters are considered:

- (a) Soluble base content (= soluble base / mass of cathode)
- 15 (b) Specific surface base (= soluble base content / surface area of cathode)
- (c) Molar efficiency of dopant (μmol) versus gravimetric efficiency of dopant (ppm)

The results are summarized in Tables 7A (M1) and 7B (M2) below.

Table 7A: Efficiency of Ca, Y, Ba, Sr, La for Li-M-oxide with M=Mn_{0.33}Ni_{0.38}Co_{0.29}

Mn _{0.33} Ni _{0.38} Co _{0.29}	Slurry doping by	BET m ² /g	Base μmol/g	Spec Surf Base μmol/m ²	Rel % spec base	Unit cell X ray
LN MnCD0548	Ref. 0 ppm	0.52	51.3	98.7	100.0	33.702
LN MnCD0549	Ba, 1000 ppm	0.45	44.1	98.0	99.3	33.705
LN MnCD0550	Ca, 1000 ppm	0.50	30.5	61.0	61.8	33.718
LN MnCD0551	La, 1000 ppm	0.54	50.5	93.5	94.8	33.703
LN MnCD0552	Sr, 1000 ppm	0.42	35.1	83.6	84.7	33.713
LN MnCD0553	Y, 1000 ppm	0.54	41.8	77.4	78.5	33.713

5 Table 7B: Comparison of the efficiency of Ca, Y, Ba, Sr, La for Li-M-oxide with M=Ni_{0.53}Ni_{0.27}Co_{0.2}

Ni _{0.53} Mn _{0.27} Co _{0.2}	Slurry doping by	BET m ² /g	Base μmol/g	Spec Surf Base μmol/m ²	Rel % spec base	Unit cell X ray
MLMOx0132	Ref. 0 ppm	0.38	74.7	196.6	100.0	33.889
MLMOx0127	Ca, 1000 ppm	0.34	51.0	150.0	76.3	33.880
MLMOx0128	Sr, 1000 ppm	0.27	47.9	177.4	90.2	33.869
MLMOx0129	Ba, 1000 ppm	0.32	64.1	200.3	101.9	33.861
MLMOx0130	La, 1000 ppm	0.37	70.4	190.3	96.8	33.862
MLMOx0131	Y, 1000 ppm	0.38	64.5	169.7	86.3	33.874

The conclusions are as follows:

- (a) Base content: Sr and Ca, and to a lesser degree Y and Ba are most efficient to lower the soluble base content.
- 5 (b) The final samples have different BET area, hence the "Specific Surface Base Content" is observed: Ca, Sr and Y, and to a lesser degree La lower the specific surface base content of the cathode.
- (c) Gravimetric efficiency: Sr and Ca are the most efficient. Molar efficiency: Considering the high molecular weight of Y (more than twice that of Ca) we conclude that both Ca and
- 10 Y are most efficient to neutralize high base caused by sulfur. Sr is somewhat less effective and La shows noticeable, but small efficiency. Ba is not effective, as can be seen in the "Specific Surface Base Content". Fe is inert (not reported).

The authors speculate that the effective elements have an ionic radius of 0.7 to 1.2

15 Angstrom. Especially Ca and Y – which have almost similar and quite small ionic radius (in 6 coordination Ca: 0.99, Y: 0.893 Å) - have a size that fits very well to the surface of Li-M-oxide. The more preferred range for ionic radii is 0.85-1.15 Angstrom.

Example 12: Strontium versus Calcium

20 Example 11 compared the efficiency of Ca, Sr, La, Ba, Y to lower the content of soluble base.

However, Example 11 did not take into account that the sintering kinetics change with different additives - yielding very different BET values. Example 12 compares the effect of Ca and Sr more carefully.

25 A reference without addition of additive (Ca or Sr) was prepared from a mixture of mixed transition metal hydroxide ($M = \text{Ni}_{0.38}\text{Mn}_{0.33}\text{Co}_{0.28}$) and Li_2CO_3 at 980°C. Further samples with addition of 400 and 1000 ppm Sr and 400 ppm Ca were prepared. Each sample used 1 kg of MOOH + Li_2CO_3 . The additive (Ca, Sr) was added by the previously described

“slurry doping” process. Appropriate amounts of solution of $\text{Sr}(\text{NO}_3)_2$ and $\text{Ca}(\text{NO}_3)_2$ were added to a high viscous slurry of the precursor hydroxide during rigid stirring.

The sintering temperature was adjusted to achieve a similar sintering. Base content was measured, unit cell volume and crystallite size was obtained from X-ray diffraction and
5 electrochemical properties were tested by coin cells. Table 8A and 8B summarizes the preparation conditions results

Table 8A: Preparation and morphology of samples with Sr, Ca addition

Li-M-oxide, $\text{Ni}_{0.38}\text{Mn}_{0.33}\text{Co}_{0.28}$	Slurry doping by	BET m^2/g	ICP Ca, Sr ppm	Base $\mu\text{mol/g}$	Vol $\text{\AA}^3$	Size nm	D5 μm	D50 μm	D95 μm
LN MnCD0555	Reference	0.50	< 100	$\cong 50$	33.7348	336			
MLMOx0149	Sr, 400 ppm	0.52	349	42.9	33.6910	332	5.14	8.99	15.3
MLMOx0150	Sr, 1000 ppm	0.50	832	37.6	33.6891	320	5.13	8.97	15.2
MLMOx0151	Ca, 400 ppm	0.49	425	34.3	33.6078	319	5.14	8.98	15.3

10 Table 8B Electrochemical performance (capacity, irreversible capacity and rate (versus 0.1C) of samples with Sr, Ca addition

$\text{Ni}_{0.38}\text{Mn}_{0.33}\text{Co}_{0.28}$	Q rev 4.3-3.0V	Q irr %	1C %	2C %	3C %
LN MnCD0555	161.9	11.1	91.8	88.4	85.1
MLMOx0149	161.5	11.3	92.7	89.1	86.4
MLMOx0150	159.8	11.8	92.4	88.7	85.8
MLMOx0151	159.3	12.1	92.2	88.4	85.5

The morphology (BET, particle size) of all samples was basically identical. Ca addition is most effective to lower the base content. 1000 ppm Sr reduces the base content about the
15 same, but less than for 400 ppm Ca. However, Sr is interesting because it reduces the base

and at the same time the electrochemical properties deteriorate less than for 400 ppm Ca addition.

Example 13: What is the optimum Li-Mn-Ni-Co composition ?

So far, this invention demonstrated that the surface properties, determining the safety and
5 stability of cathodes in real batteries – the surface properties being measured as base
content by pH titration – are strongly determined by the sulfur and Ca (amongst others)
content. The authors also analyzed large amounts of data to understand what else
determines the base content. The analysis shows clearly that the base content furthermore
depends on BET surface area of Li-M-O₂, it also varies strong with Li:M ratio and Ni:Mn
10 ratio.

The base content increases linearly with BET, it increases with increasing Li:M ratio and
with increasing Ni:Mn ratio. Table 9 shows a typical example for Li-M-oxide where M
contains 33% Co.

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Table 9: Base content as function of Ni:Mn ratio and Li:M ratio for samples prepared at different temperature

Sinter T	Li:M blend ratio	Base content ($\mu\text{mol/g}$)				Spec. Base content ($\mu\text{mol/m}^2$)			
		Ni:Mn 0.95	Ni:Mn 1.05	Ni:Mn 1.2	Ni:Mn 1.3	Ni:Mn 0.95	Ni:Mn 1.05	Ni:Mn 1.2	Ni:Mn 1.3
930	1.05	20.0	43.8	71.5	76.6	37.8	59.2	96.6	111.0
	1.15	65.1	88.0	115.5	121.8	135.5	127.5	172.4	213.6
960	1.0	13.3	23.0	53.1	58.6	28.9	35.4	80.5	93.0
	1.05	20.5	42.7	71.2	77.4	50.0	62.8	111.3	126.9
	1.1	42.6	60.6	89.1	89.4	109.2	116.5	174.7	194.3
	1.15	60.7	81.8	104.2	106.1	151.8	146.1	221.7	235.8
990	1.05	22.2	42.0	67.0	69.8	63.4	85.7	148.9	166.2
	1.15	55.1	67.0	88.5	90.6	166.8	163.4	252.7	323.5

The authors intended to optimize the BET and Li:M as well as Ni:Mn composition in order to achieve the optimum of high electrochemical performance, but keeping base content low. It was shown that a similar electrochemical performance can be achieved by high BET but lower Li:M, or lower BET and higher Li:M. By trying to optimize the composition, BET and crystallinity - it was recognized that within the region of interest only samples with a certain content of divalent Ni, high crystallinity allows to achieve overall optimized cathodes.

Table 10 below lists the preferred upper and lower Li:M stoichiometric range for Li-M-O₂ with different transition metal composition. The columns in the table refer to the following formulas

(a) $\text{Li}_{1+k}\text{M}_{1-k}\text{O}_2$ with $\text{Ni}_{1-a-b}\text{Mn}_a\text{Co}_b$ and (b) $\text{Li}[\text{Li}_{x/3}\text{Mn}_{2x/3}\text{Ni}_{y/2}\text{Mn}_{2/y}\text{Co}_z\text{Ni}_{1-x-y-z}]\text{O}_2$ as follows :

35

Ni, Mn, Co are the mol fractions $1-a-b$, a , b in the transition metal M

"Ni:Mn" is the molar ratio of Ni to Mn ($= (1-a-b)/a$) in the transition metal M

"Li:M" is the molar ratio of Li:M ($= (1+k)/(1-k) = (1+x/3)/(1-x/3)$)

The column " Ni^{2+} " gives twice the fraction of divalent nickel ($= 2 * y/4$).

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Table 10: Preferred upper and lower Li:M stoichiometric range for Li-M-O₂ with different transition metal composition.

5	Li/M	Ni/Mn	M			Ni ²⁺
	(1+k)/(1-k)	(1-a-b)/a	Ni(1-a-b)	Mn(a)	Co(b)	y/2
	1.028	1.00	0.3333	0.3333	0.3333	0.3011
	1.055					0.2709
	1.092					0.2307
	1.121					0.2002
10	1.036	0.95	0.3248	0.3419	0.3333	0.3005
	1.063					0.2704
	1.100					0.2304
	1.128					0.2010
	1.021	1.05	0.3415	0.3252	0.3333	0.3010
	1.048					0.2707
15	1.085					0.2304
	1.113					0.2009
	1.002	1.20	0.3636	0.3030	0.3333	0.3007
	1.029					0.2701
	1.065					0.2305
	1.093					0.2007
20	0.991	1.30	0.3768	0.2899	0.3333	0.3002
	1.017					0.2706
	1.053					0.2307
	1.081					0.2007

25 An analysis of the data reveals:

(1) It is difficult to obtain a good overall performance if Ni:Mn=1. Ni:Mn > 1 allows for

better electrochemical performance.

(2) The optimum Li:M stoichiometric region depends on the transition metal composition.

The optimum Li:M is achieved if the cathode $\text{Li}_{1+a}\text{M}_{1-a}\text{O}_2$ contains 11.5-13.5% of divalent nickel per 2 mol metal (Li+M).

5 The optimum Li:M decreases with increasing Ni:M.

(a) Ni:Mn=0.95 : Li:M = 1.07

(b) Ni:Mn=1.05 : Li:M = 1.06

(c) Ni:Mn=1.2 : Li:M = 1.05

(d) Ni:Mn=1.3 : Li:M = 1.04

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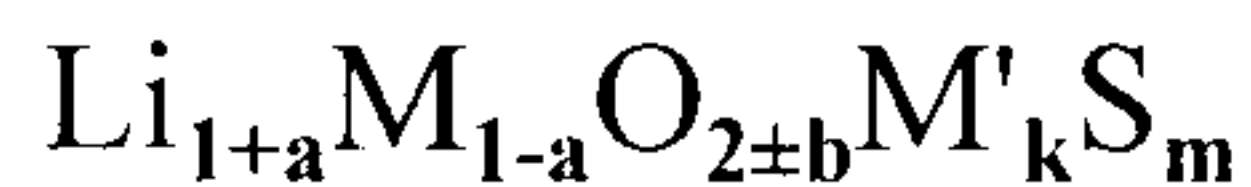
Similar experiments were repeated for different metal compositions, including

$\text{M}=\text{Mn}_{0.45}\text{Ni}_{0.45}\text{Co}_{0.1}$, $\text{M}=\text{Ni}_{0.67}\text{Mn}_{0.22}\text{Co}_{0.11}$, $\text{M}=\text{Ni}_{0.53}\text{Mn}_{0.26}\text{Co}_{0.2}$, $\text{M}=\text{Ni}_{0.5}\text{Mn}_{0.3}\text{Co}_{0.2}$,
 $\text{M}=\text{Ni}_{0.55}\text{Mn}_{0.3}\text{Co}_{0.15}$, $\text{M}=\text{Mn}_{0.4}\text{Ni}_{0.5}\text{Co}_{0.1}$, $\text{Mn}_{0.33}\text{Ni}_{0.39}\text{Co}_{0.28}$, $\text{Mn}_{0.33}\text{Ni}_{0.37}\text{Co}_{0.3}$.

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CLAIMS:

1. A powderous lithium transition metal oxide having a layered crystal structure



wherein:

$$-0.03 < a < 0.06,$$

$$b \cong 0,$$

M is one or more elements wherein at least 95% of M consists of one or more elements selected from the group consisting of Ni, Mn, Co and Ti;

M' is present on the surface of the powderous oxide and consists of one or more elements selected from the group consisting of Ca, Sr, Y, La, Ce and Zr;

$$0.0250 \leq k \leq 0.1 \text{ in wt\%}; \text{ and}$$

$$0.15 < m \leq 0.6 \text{ in mol\%}.$$

2. The powderous lithium transition metal oxide of claim 1, wherein M is one or more elements and wherein at least 99% of M consists of one or more elements selected from the group consisting of Ni, Mn, Co, Al, Mg and Ti.
3. The powderous lithium transition metal oxide of claim 1 or 2, wherein M' is Ca and $0.0250 \leq k < 0.0500$ in wt%.
4. The powderous lithium transition metal oxide of claim 3, wherein $0.0250 \leq k < 0.0400$, in wt%.
5. The powderous lithium transition metal oxide according to any one of claims 1 to 4, wherein $0.25 \leq m \leq 0.6$ in mol%.
6. The powderous lithium transition metal oxide according to any one of claims 1 to 5, wherein M is:



wherein:

$$0.1 \leq x \leq 0.7;$$

$$0.1 \leq y \leq 0.7;$$

$$0.1 \leq z \leq 0.7; \text{ and}$$

$$x+y+z=1.$$

7. The powderous lithium transition metal oxide according to claim 6, wherein:

$$1.0 \leq x/y \leq 1.3; \text{ and}$$

$$0.1 < z < 0.4, \text{ and}$$

wherein the powderous lithium transition metal oxide comprises 10-15 at.% of Ni^{2+} per total metal $\text{Li}_{1+a}\text{M}_{1-a}$.

8. The powderous lithium transition metal oxide according to claim 7 wherein the powderous lithium transition metal oxide comprises 11.5 -13.5 at.% of Ni^{2+} per total metal $\text{Li}_{1+a}\text{M}_{1-a}$.

9. The powderous lithium transition metal oxide according to claim 7 or 8, wherein $x=y=z=0.33$.

10. An electrochemical cell comprising a cathode comprising as active material the powderous lithium transition metal oxide according to any one of claims 1 to 9.

11. A method for preparing a powderous lithium transition metal oxide according to any one of claims 1 to 10, comprising the steps of:

providing for a mixture of M-sulphate, a precipitation agent, and a complexing agent,

precipitating a M-hydroxide, a M-oxyhydroxide or a M-carbonate precursor from said mixture having a given sulfur content,

ageing said precursor whilst adding a base, thereby obtaining a certain base:precursor ratio,
 washing with water,
 drying,
 mixing said aged M-hydroxide, said aged M-oxyhydroxide or said M-carbonate precursor with a Li precursor,
 sintering said mixture at a temperature T of at least 900°C for a time t between 1 and 48 hrs, thereby obtaining a sintered product;
 wherein:

- a salt of M' is added to said M-sulphate containing mixture,
 - M' is added to said base during ageing,
 - M' is added to the water used in said washing step, or
 - a M' salt solution is added to a slurry prepared by suspending said sintered product in water, followed by drying.
12. The method of claim 11 wherein the precipitation agent is NaOH or Na₂CO₃.
13. The method of claim 11 or 12 wherein the temperature T is at least 950°C.
14. The method for preparing a powderous lithium transition metal oxide according to any one of claims 11 to 13 wherein M' is Ca and said salt is either Ca(NO₃)₂ and CaCl₂.
15. The method for preparing a powderous Lithium transition metal oxide according to any one of claims 11 to 14, wherein said given sulfur content is controlled during said ageing step by selecting a given base : precursor ratio.

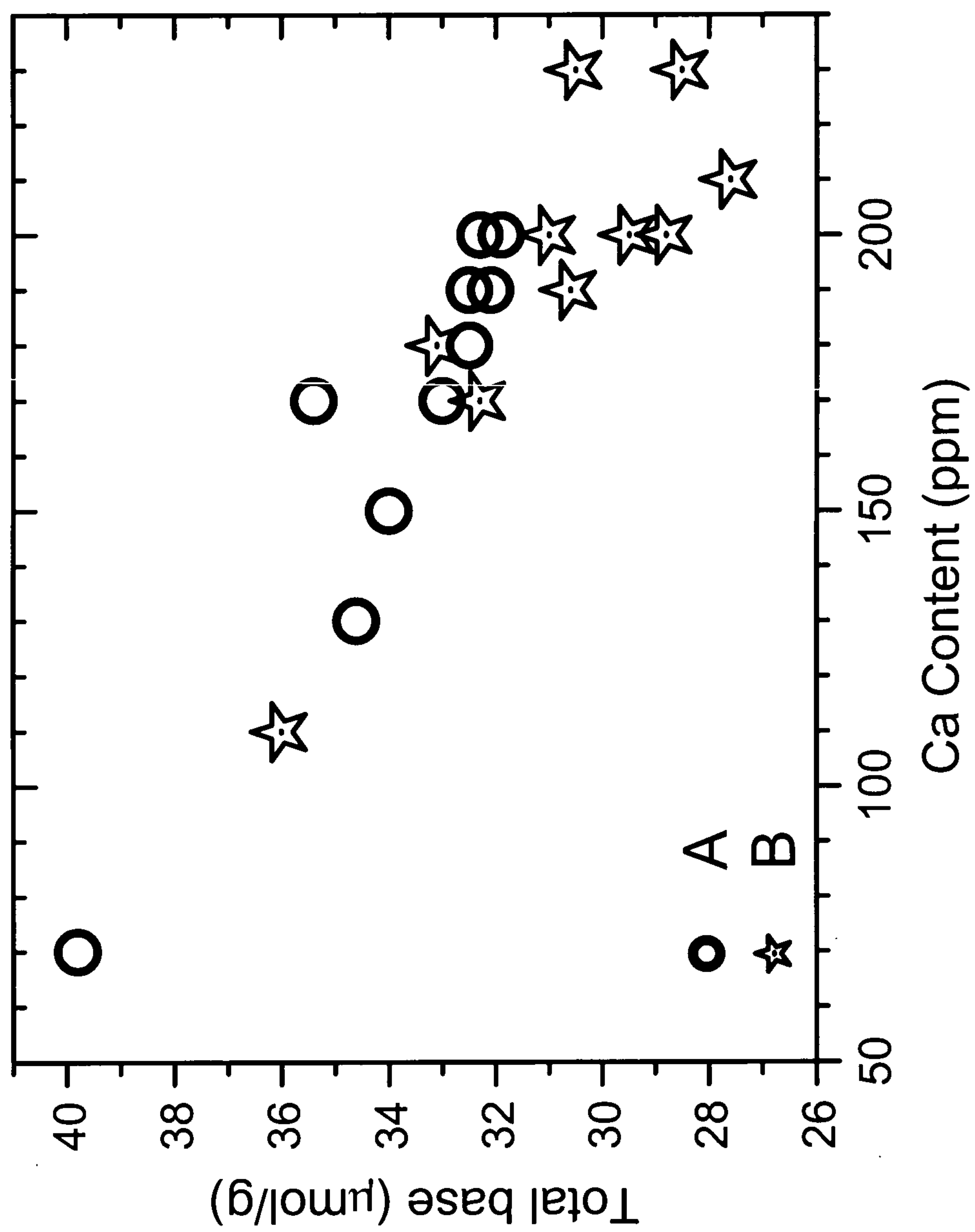


Fig. 1

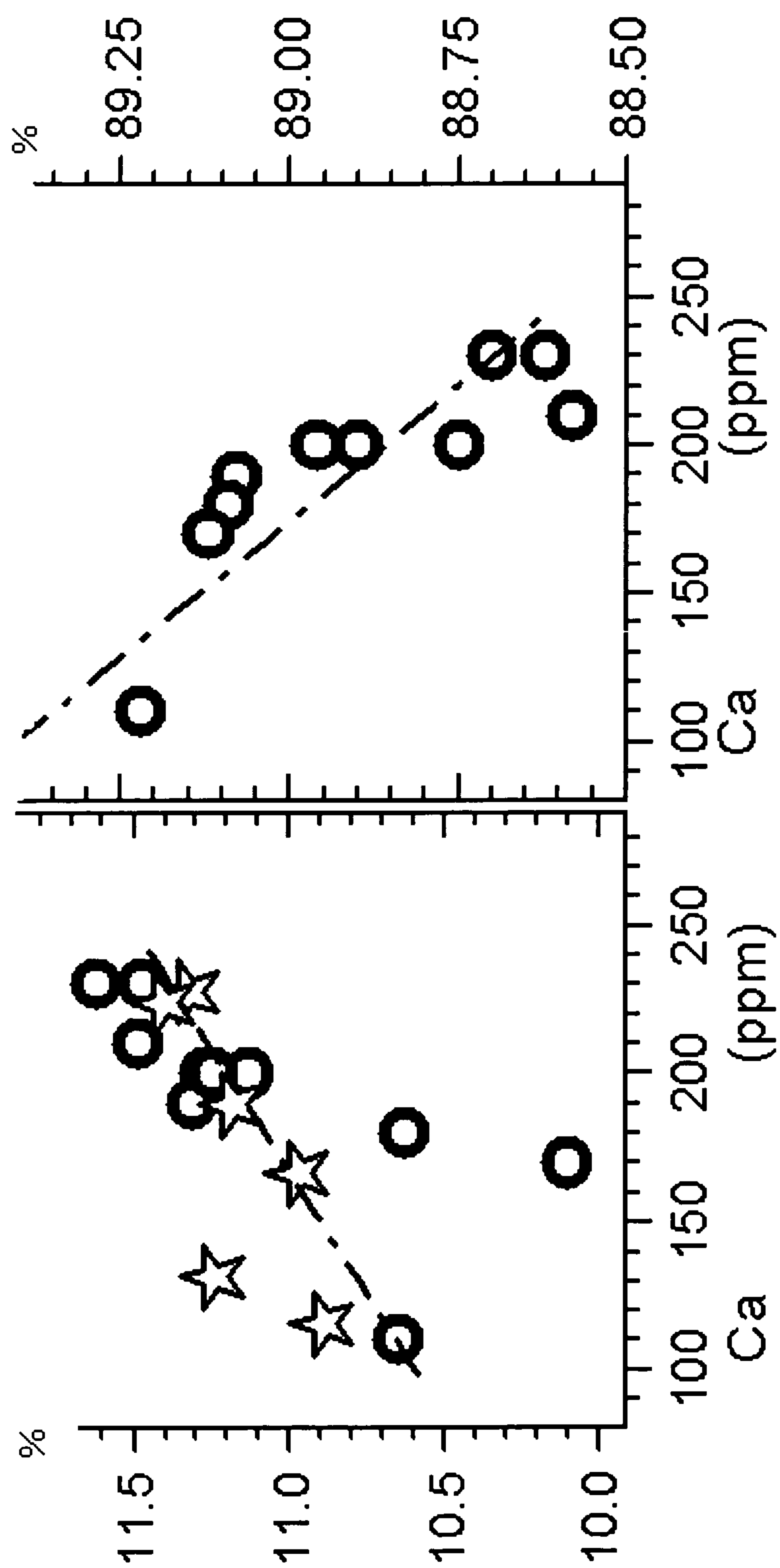


Fig. 2

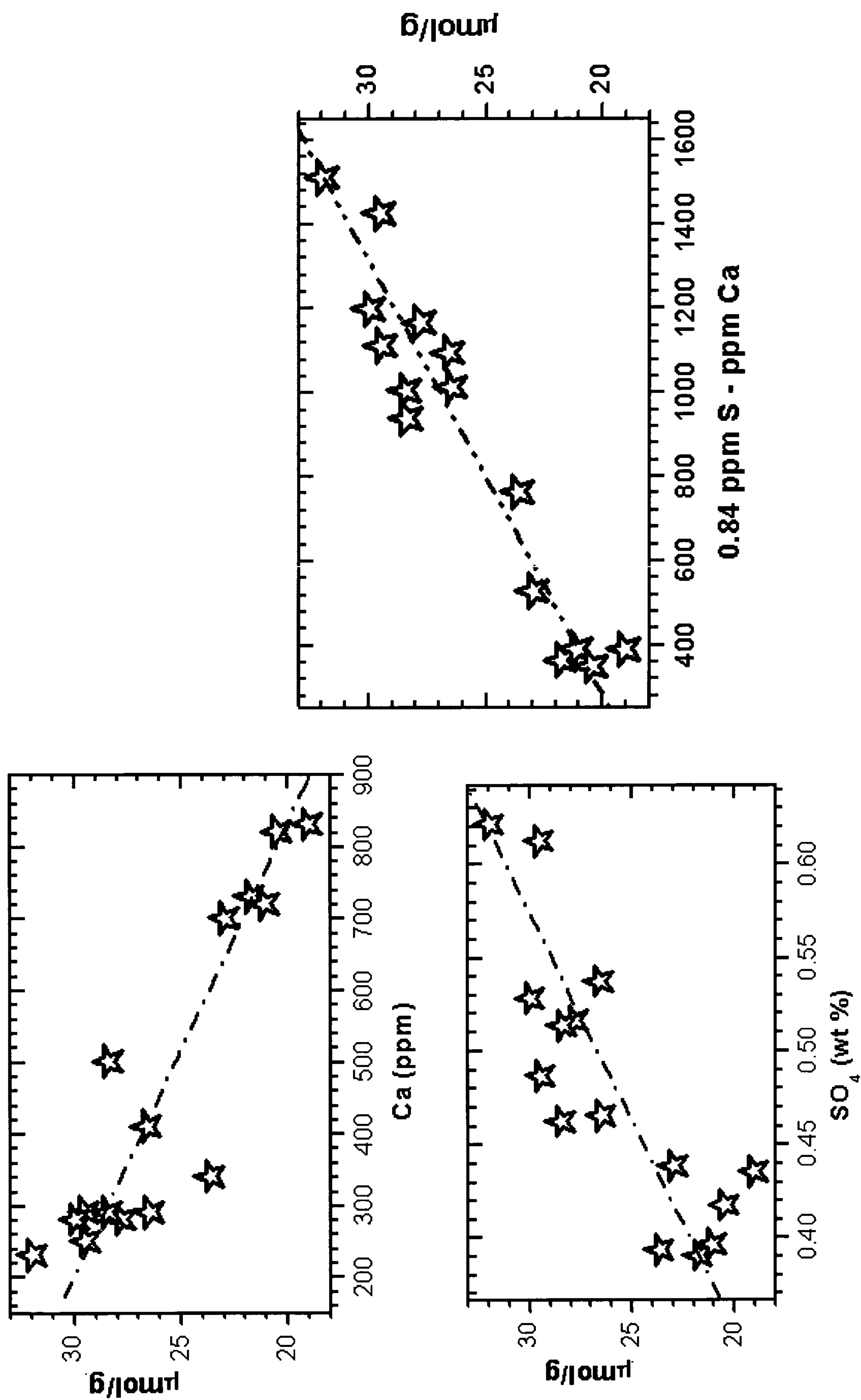


Fig. 3

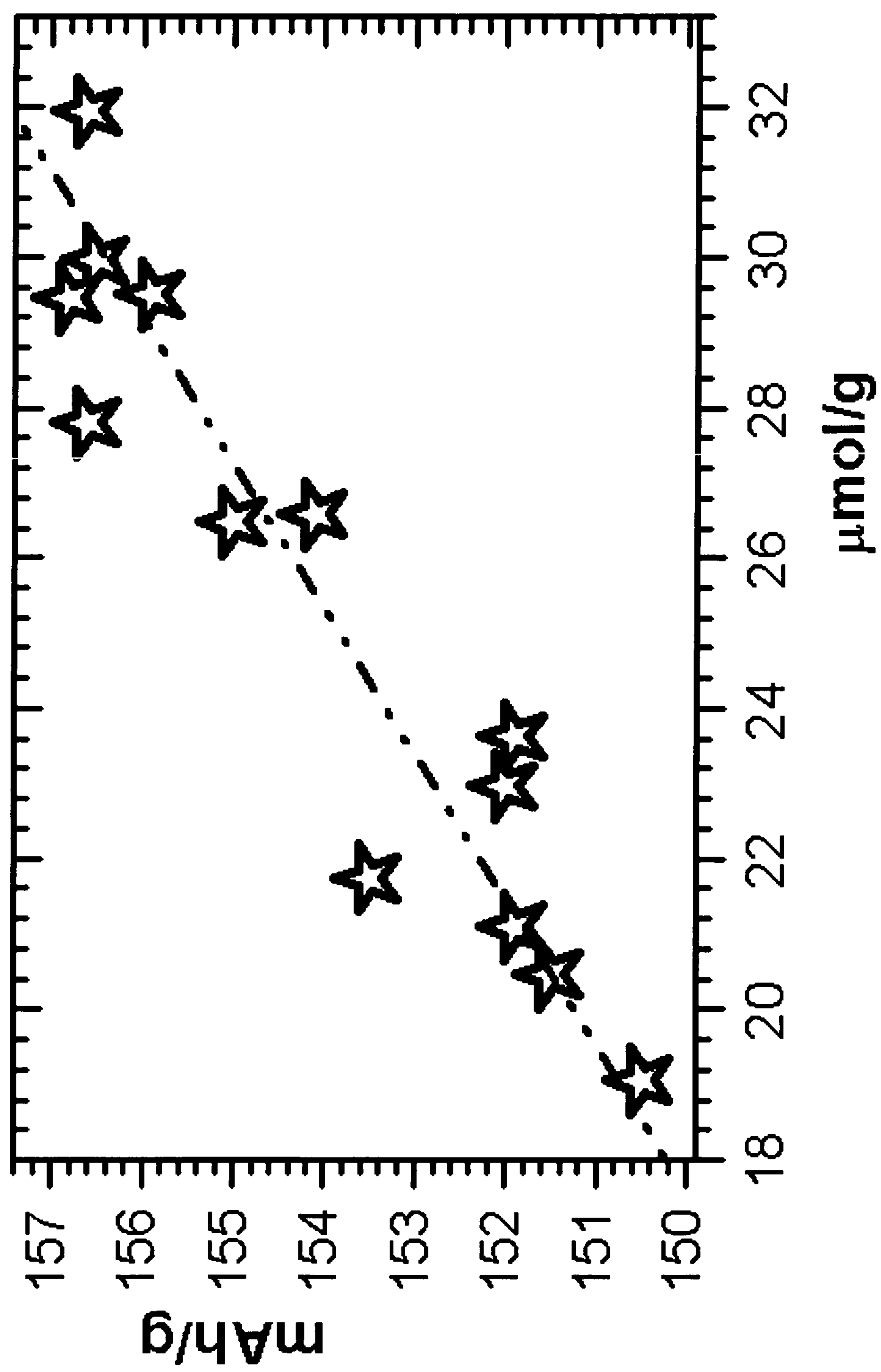


Fig. 4

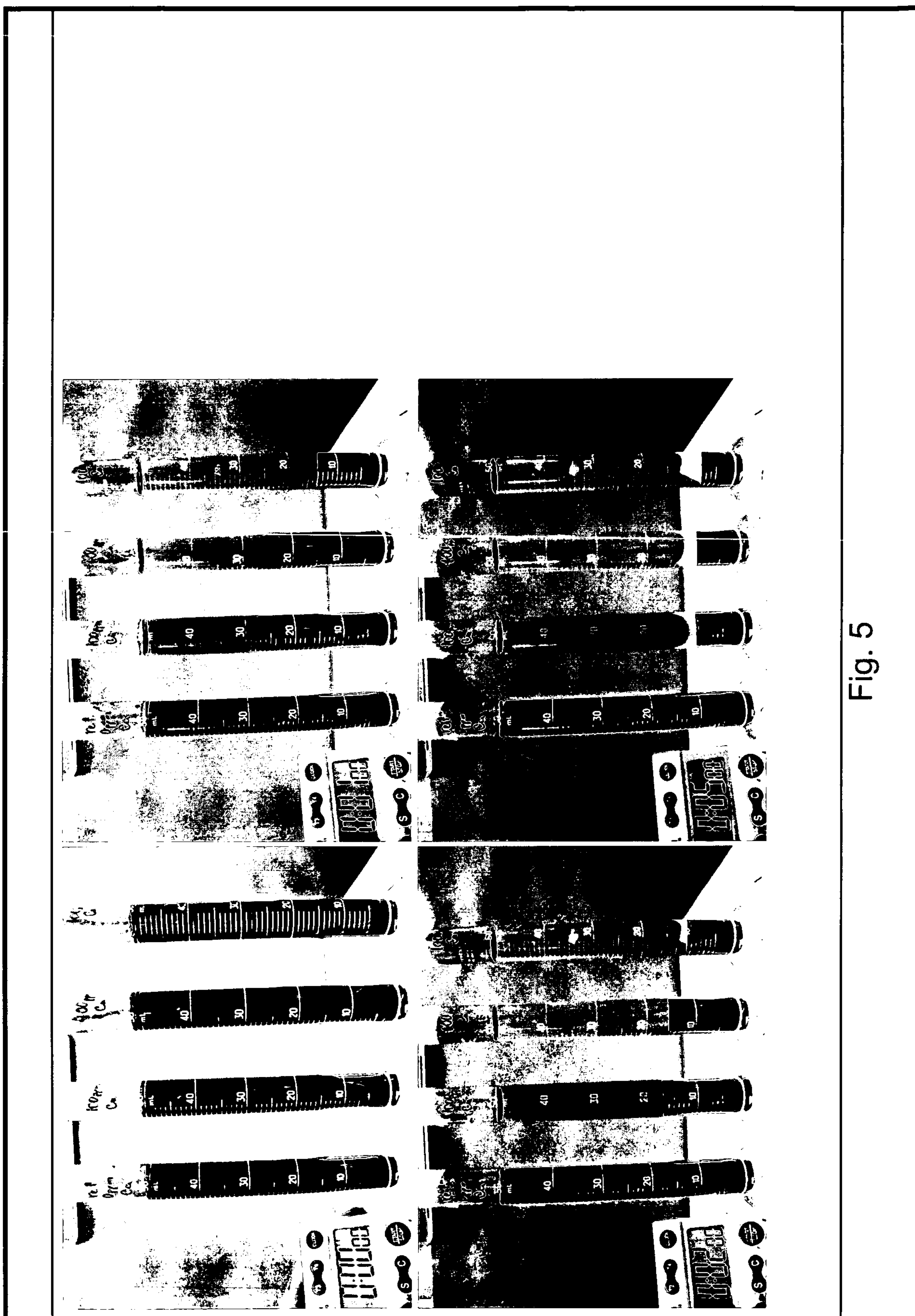


Fig. 5

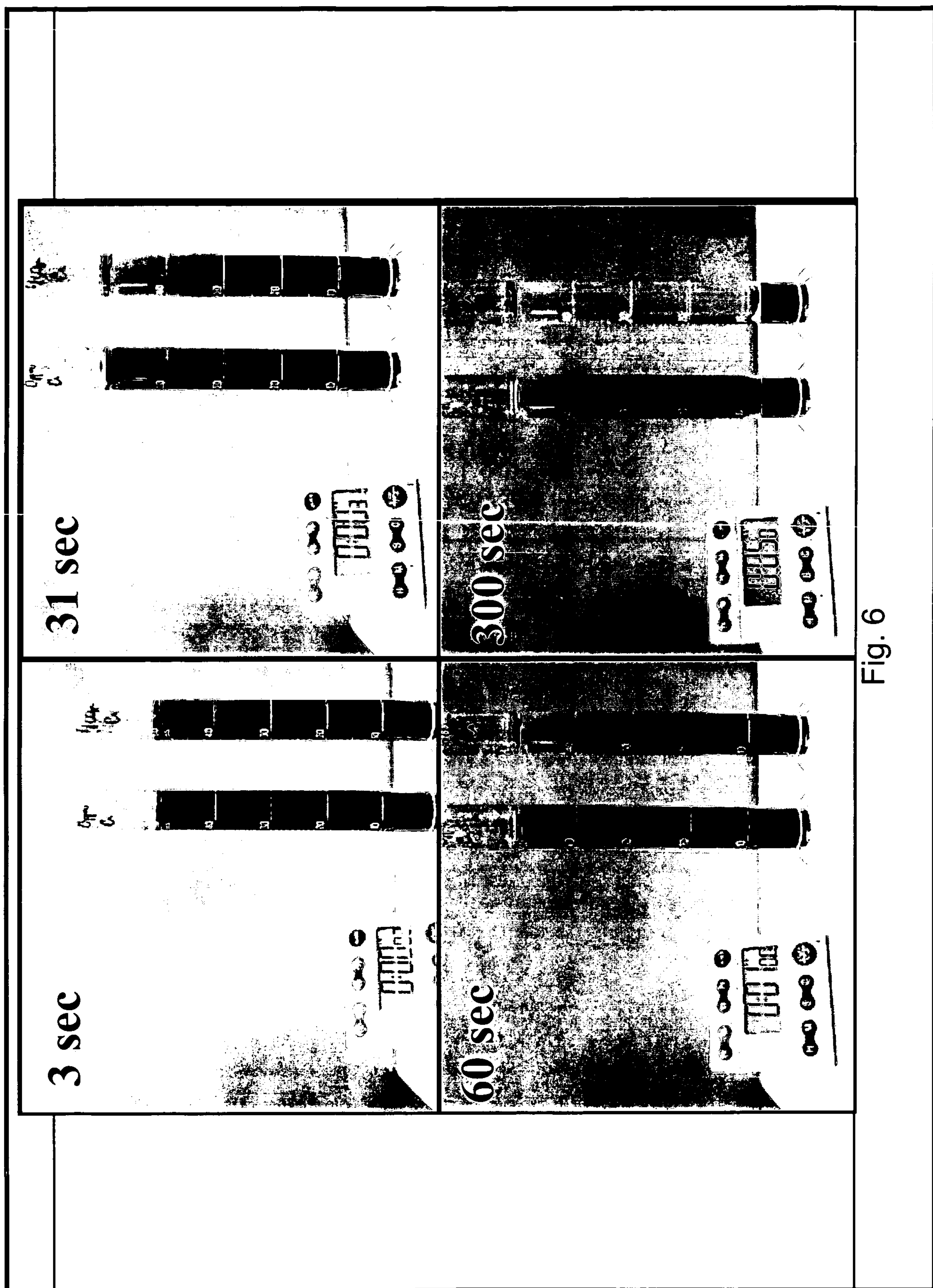


Fig. 6

