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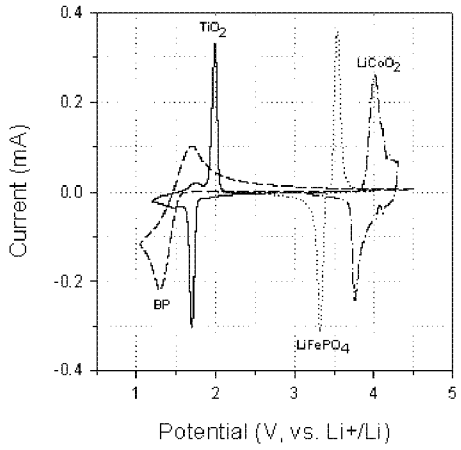
(54) **Title:** OVERCHARGE OR DISCHARGE PROTECTION FOR LITHIUM-ION BATTERIES



(57) **Abstract:** A lithium-ion battery comprising a first electrode made of cathodic material, a second electrode made of anodic material and an electrolyte, said lithium-ion battery containing an overcharge protection material consisting of redox molecules, characterized by the fact that said redox molecules have a reduction potential which is lower than said anodic material.

WO 2008/032240 A2

Overcharge or discharge protection for lithium-ion batteries

Field of invention

The present invention relates to lithium-ion batteries including an overcharge/discharge protection. It more precisely relates to an overcharge/discharge protection consisting of redox molecules.

State of the art

Lithium-ion are known since more than 15 years. Today they are used in different fields, in particular as power sources for portable consumer electronics.[1] When using lithium intercalation compounds as electrode material, lithium ion batteries show excellent performances of high energy density, long cycle life, and slow self-discharge rate, etc.

Despite its success, the safety of single cell and its pack under aggressive conditions is still one big issue for using as “drugstore” or consumer-assembled batteries, where the hazard caused by overcharge or overdischarge is one of the critical reasons.[2] One approach for overcharge protection is using redox shuttle additives in the electrolyte[3] or embedded in a separator[4]. When the cell is overcharged, the redox shuttles start to be either oxidized at the positive electrode or reduced at the negative electrode when a characteristic cell potential is obtained. The mobile reaction products diffuse through the electrolyte to the opposite side of the cell where a second electron-transfer reaction returns it to its original state. The net effect is an internal shunt which prevents deterioration of the cell by imposing a limit on cell potential.

There have been many studies focusing on redox shuttles with potentials higher than those of the cathodic material (we call it “p-type shuttle” here).[3,4] As indicated in Figure 1(a), upon overcharge with a fully delithiated cathodic material, the redox molecules (**S**) are oxidized at the positive electrode, and then the oxidized species ($\mathbf{S}^+$) diffuse to the negative electrode, where they are reduced to the original state. Consequently, the upper potential of the cell is kept by the redox molecules, and the possible damage to the electrolyte and electrode material is prohibited. Numerous p-type redox molecules have been tested so far, such as metallocenes[3a-3c], metal complexes[3d], aromatic molecules[3e-3j,3m,3n], phenothiazine derivatives[3k,3l], 2,2,6,6-

tetramethyl-1-piperidinyloxy (TEMPO)-like molecules[3l,3o], conducting polymer[4], perfluoroborane cluster salts[5] etc., where only 2,5-di-tert-butyl-1,4-dimethoxybenzene shows promise in $LiFePO_4$ based lithium ion batteries during long term cycling.[2,3h] While for 4 V class cathodic material, no practically acceptable shuttle molecule has been proposed so far because of the difficulties in achieving high potential and stability.

General description of the invention

The present invention offers an alternative to existing overcharge/discharge protection systems. It is based on the surprising effect that the use of redox molecules having reduction potential which is below than the reduction potential of the anode offers an efficient level of safety, against either overcharge or discharge.

More precisely the invention relates to a lithium-ion battery comprising a first electrode made of cathodic material, a second electrode made of anodic material and an electrolyte, said lithium-ion battery containing an overcharge/discharge protection material consisting of redox molecules, characterized by the fact that said redox molecules have a reduction potential which is lower than said anodic material.

Preferred embodiments of the invention are defined in the dependent claims.

Compared to the conventional state-of-the-art p-type shuttles, the shuttles according to the invention, called n-type shuttles in the present text, provide several advantages, in particular : As the oxidation potential of the shuttle molecules is high enough, any cathodic material can be used in the battery. In addition. The use of n-type shuttle molecules also opens new opportunity for having molecules with much better stability.

Detailed description of the invention

The invention is discussed below in a more detailed way with examples illustrated by the following figures :

Figure 1A represents a state-of-the-art schematic model showing the working principle of molecular shuttles for lithium ion batteries upon overcharge protection wherein p-type shuttle molecules are oxidized at positive electrode as $E_{S+/S}>E_{+}$, and reduced at negative electrode.

Figure 1A represent a model according to the invention wherein n-type shuttle molecules are reduced at negative electrode as $E_{S/S-}<E_{-}$, and oxidized at positive electrode.

Figure 2 shows cyclic voltammograms of various electrode materials: TiO_2 (full line), $LiFePO_4$ (point line), and $LiCoO_2$ (dashed line) obtained in the electrolyte of 0.66 M $LiClO_4$/PC+DME (2:1), the scan rate being 0.1 mV/s. The dashed curve was obtained with a Pt disk electrode in the presence of 10 mM benzophenone in the electrolyte, the scan rate being 0.1 V/s.

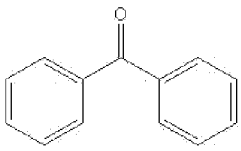
Figure 3A shows galvanostatic voltage profiles of $TiO_2/LiCoO_2$ (full line) and $TiO_2/LiFePO_4$ (dashed line) full cells during charge/discharge in the absence of 0.4 M benzophenone in the electrolyte. The electrolyte is 0.66 M $LiClO_4$/PC+DME (2:1). The current density is 0.05 mA/cm^2.

Figure 3B shows galvanostatic voltage profiles of $TiO_2/LiCoO_2$ (full line) and $TiO_2/LiFePO_4$ (dashed line) full cells during charge/discharge in the presence of 0.4 M benzophenone in the electrolyte. The electrolyte is 0.66 M $LiClO_4$/PC+DME (2:1). The current density is 0.05 mA/cm^2.

As indicated in Figure 1(b), upon overcharge with a fully lithiated anodic material, its potential further decreases and the redox molecules (**S**) start to reduce at the negative electrode. Then the reduced species ($\mathbf{S}^-$) diffuse to the positive electrode, where **S** is regenerated. In this strategy, except that the onset reduction potential of the redox shuttle molecule must be slightly lower than that of the anodic material to prevent self-discharge during storage and to allow for some overpotential during

charging, requirements for such additives are similar to their p-type counterparts: firstly the redox shuttle molecule must be unreactive toward all cell components in both its oxidized and reduced forms; secondly it must be present in sufficient concentration and have a high enough diffusion coefficient to give a shuttle current at least as great as that at which the cell is charged; and very importantly the molecule must be stable enough or highly reversible to sustain long term cycling. En embodiment of the invention is discussed below wherein benzophenone (BP) is used as n-type shuttle in TiO_2/$LiFePO_4$ and TiO_2/$LiCoO_2$ cells.

The structure of a benzophenone (BP) molecule is represented as follows :



The mesoscopic anatase TiO_2 films (particle size ~20 nm) were prepared on F-doped conducting glass (FTO) by reported method.[6] $LiCoO_2$ (Degussa) and $LiFePO_4$ (ZSW) films were prepared by mixing the powders with 5 *wt.* % PVDF, 10~15 *wt.*% acetylene black, and stirring with *N*-methyl-2-pyrrolidone (NMP). The resulting homogeneous slurry was then doctor-bladed onto FTO. The electrolyte was 0.66 M $LiClO_4$/PC+DME (2:1, w/w). Benzophenone (BP) was purchased from Fluka (>99.0%), and was dissolved in above electrolyte. Voltammetric and galvanostatic measurements employed a PC-controlled AutoLab® PSTA30 electrochemical workstation (Eco Chimie).

Mesoscopic anatase TiO_2 has been extensively studied as anodic material for lithium ion batteries in many literatures.[7] Li^+ is inserted into the lattice of anatase TiO_2 at ~1.75 V (vs. Li^+/Li) during charging. As the cyclic voltammograms shown in Figure 2, the onset reduction potential of BP is ~1.60 V, which is lower than that of TiO_2. It is expected that as the fully lithiated TiO_2 electrode is overcharged, it will surpass the lithium insertion potential of TiO_2 and reaches the redox potential of BP. As a consequence, BP will be reduced at the anodic current collector and the reduced species BP^- will diffuse to the positive electrode. The cathodic material and

electrolyte are thus stabilized, since the total flux of current flow is carried by the shuttle molecule. By using voltammetric method, the diffusion coefficient of BP in 0.66 M $LiClO_4$/PC+DME (2:1) electrolyte is estimated to be ~1.1×10^{-6} cm^2/s, [8] which is in the same order to the other shuttle molecules[3j,3k].

In addition, because of the efficient charge delocalization of the molecule (see the BP structure above), the oxidation potential of BP is very high, being far beyond the measured potential range, 4.50 V (vs. Li^+/Li). In other words, it renders the validity of cathodic material with potential lower than this value when using BP as redox shuttle molecule. As the cyclic voltammograms shown in Figure 2, the charging potentials of the typical cathodic material $LiFePO_4$ and $LiCoO_2$ are perfectly located in this potential window, being 3.45 and 3.90 V, respectively.

Figure 3A shows the galvanostatic voltage profiles of TiO_2/$LiFePO_4$ and TiO_2/$LiCoO_2$ full cells during continuous charge/discharge cycling, the cutoff voltage being 1.90 and 2.30 V, respectively. Consistent with the potentials of the electrode material, the voltage plateaus during charging process are around 1.70 and 2.15 V for TiO_2/$LiFePO_4$ and TiO_2/$LiCoO_2$ cells, respectively. When the cells reach full charge, the terminal voltage increases steeply. Figure 3B shows the voltage profiles of the cells in the presence of 0.4 M BP in the electrolyte as redox shuttle. As the cells are overcharged, the voltage increases rapidly at first. When the negative electrodes reach the shuttle potential, the cell voltage is clamped while the constant current continues to flow. The second voltage plateaus of the charging process, being around 1.80 and 2.26 V for TiO_2/$LiFePO_4$ and TiO_2/$LiCoO_2$ cells, respectively, just indicate the effective operation of BP. It also proves that the utilization of BP is independent of the cathodic material, as it is stable enough for oxidation. In addition, it is clear that the presence of BP in the electrolyte almost has no influence on the cell voltage and capacity, indicating the validity of the shuttle molecule during overcharge protection. However, because BP is not electrochemically reversible, here we are not trying to test its stability during long time turnover cycling.

Similar to the p-type approach, the n-type shuttle can also be used for overdischarge protection. For a lithium ion battery, when the cell reaches end of discharge, the terminal voltage decreases rapidly as a consequence of steep dropping of the positive electrode potential. As the potential of the fully lithiated positive electrode reaches that of the n-type shuttle molecule, the molecule will start

to reduce, then the electrode potential is clamped there just as the case of overcharge.

Several types of anodic material can be used in the present invention, in particular, anodic material with high lithium insertion potentials, such as intermetallic alloys, 3D-nanostructured metal oxides (nitrides, sulfides, phosphides, fluorides)[9], TiO_2, and $Li_4Ti_5O_{12}$.

In summary, a novel overcharge protection strategy of using n-type redox shuttle molecules has been successfully demonstrated. By using TiO_2 as anodic material, $LiFePO_4$ and $LiCoO_2$ as cathodic material, the benzophenone molecule has shown quite nice shuttle functions during overcharge of the cells. The invention constitutes therefore a very promising alternative strategy for achieving overcharge (overdischarge) protection in lithium ion batteries.

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Claims

1. A lithium-ion battery comprising a first electrode made of cathodic material, a second electrode made of anodic material and an electrolyte, said lithium-ion battery containing an overcharge/discharge protection material consisting of redox molecules, characterized by the fact that said redox molecules have a reduction potential which is lower than said anodic material.

2. A lithium-ion battery according to claim 1 wherein said redox molecules are benzophenone molecules.

3. A lithium-ion battery according to claim 1 or 2 wherein said anodic material is TiO_2.

4. A lithium-ion battery according to claim 1 or 2 wherein said anodic material is selected from the group of intermetallic alloys, 3D-nanostructured metal oxides (nitrides, sulfides, phosphides, fluorides)[9] and $Li_4Ti_5O_{12}$.

5. A lithium-ion battery according to claim 3 wherein said cathodic material is $LiFePO_4$.

6. A lithium-ion battery according to claim 3 wherein said cathodic material is $LiCoO_2$.

7. A lithium-ion battery according to anyone of the previous claims wherein said redox molecules are in the form of an additive in the electrolyte.

8. A lithium-ion battery according to anyone of the previous claims 1 to 6 wherein said redox molecules are embedded in a separator.

FIG. 1A

FIG. 1B

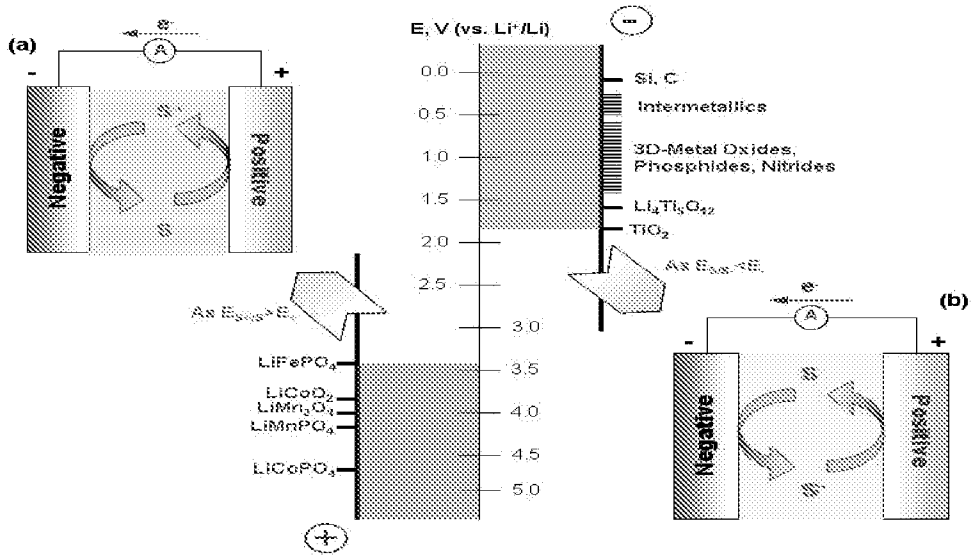
(a)
e⁻
A
-
+
Negative
Positive
E, V (vs. Li⁺/Li)
0.0
0.5
1.0
1.5
2.0
2.5
3.0
3.5
4.0
4.5
5.0
Si, C
Intermetallics
3D-Metal Oxides, Phosphides, Nitrides
$Li_4Ti_5O_{12}$
TiO_2
$LiFePO_4$
$LiCoO_2$
$LiMn_2O_4$
$LiMnPO_4$
$LiCoPO_4$
(b)

FIG. 2

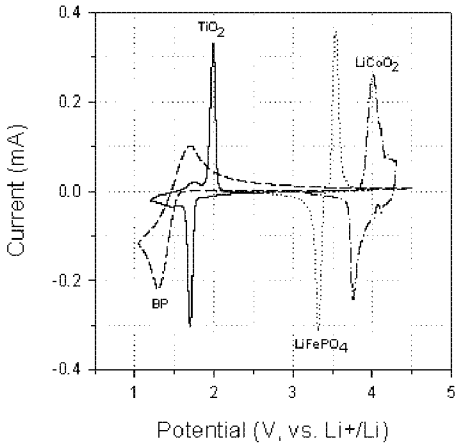

FIG. 3A

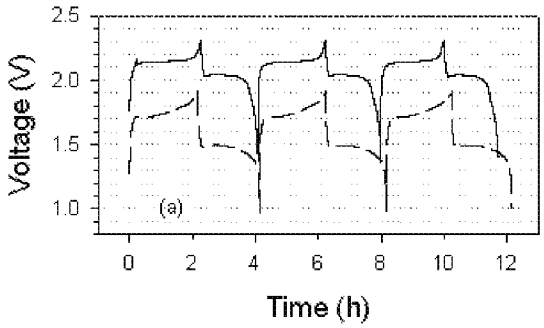

FIG. 3B

