



(86) Date de dépôt PCT/PCT Filing Date: 1998/10/16

(87) Date publication PCT/PCT Publication Date: 1999/05/14

(45) Date de délivrance/Issue Date: 2008/04/22

(85) Entrée phase nationale/National Entry: 2000/05/02

(86) N° demande PCT/PCT Application No.: US 1998/021892

(87) N° publication PCT/PCT Publication No.: 1999/023287

(30) Priorité/Priority: 1997/11/03 (US08/963,406)

(51) Cl.Int./Int.Cl. *D02G 3/00* (2006.01),
C08K 7/06 (2006.01), *C08K 9/08* (2006.01),
C08L 23/00 (2006.01), *C08L 25/06* (2006.01)

(72) Inventeurs/Inventors:
TENNENT, HOWARD G., US;
NIU, CHUN MING, US;
HOCH, ROBERT, US;
FISCHER, ALAN B., US

(73) Propriétaire/Owner:
HYPERION CATALYSIS INTERNATIONAL, INC., US

(74) Agent: SMART & BIGGAR

(54) Titre : RESEAUX A INTERPENETRATION TRIDIMENSIONNELLE D'ASSEMBLAGES MACROSCOPIQUES DE
FIBRILLES DE CARBONE A ORIENTATION ALEATOIRE ET DE POLYMERES ORGANIQUES

(54) Title: THREE-DIMENSIONAL INTERPENETRATING NETWORKS OF MACROSCOPIC ASSEMBLAGES OF
RANDOMLY ORIENTED CARBON FIBRILS AND ORGANIC POLYMERS

(57) **Abrégé/Abstract:**

The invention relates to an interpenetrating network of carbon fibrils and a polymer, which comprises a rigidized, three-dimensional macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils and an interpenetrating mass of an organic polymer. The invention also relates to methods of making and using such interpenetrating networks.

**PCT**WORLD INTELLECTUAL PROPERTY ORGANIZATION
International Bureau

INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁶ : D02G 3/00	A1	(11) International Publication Number: WO 99/23287 (43) International Publication Date: 14 May 1999 (14.05.99)
(21) International Application Number: PCT/US98/21892 (22) International Filing Date: 16 October 1998 (16.10.98) (30) Priority Data: 08/963,406 3 November 1997 (03.11.97) US (71) Applicant: HYPERION CATALYSIS INTERNATIONAL, INC. [US/US]; 38 Smith Place, Cambridge, MA 02138 (US). (72) Inventors: TENNNENT, Howard, G.; 301 Chandler Mill Road, Kennett Square, PA 19348 (US). NIU, Chun, Ming; 81 Fottler Avenue, Lexington, MA 02173 (US). HOCH, Robert; R.R. 1, Box 422, Hensonville, NY 12439 (US). FISCHER, Alan, B.; 80 Antrim Street, Cambridge, MA 02139 (US). (74) Agents: EVANS, Barry et al.; Whitman Breed Abbott & Morgan LLP, 200 Park Avenue, New York, NY 10166 (US).		(81) Designated States: AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CU, CZ, DE, DK, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, UZ, VN, YU, ZW, ARIPO patent (GH, GM, KE, LS, MW, SD, SZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG). Published <i>With international search report.</i>
(54) Title: THREE-DIMENSIONAL INTERPENETRATING NETWORKS OF MACROSCOPIC ASSEMBLAGES OF RANDOMLY ORIENTED CARBON FIBRILS AND ORGANIC POLYMERS (57) Abstract The invention relates to an interpenetrating network of carbon fibrils and a polymer, which comprises a rigidized, three-dimensional macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils and an interpenetrating mass of an organic polymer. The invention also relates to methods of making and using such interpenetrating networks.		

**THREE DIMENSIONAL INTERPENETRATING NETWORKS
OF MACROSCOPIC ASSEMBLAGES OF RANDOMLY
ORIENTED CARBON FIBRILS AND ORGANIC POLYMERS**

5 **FIELD OF THE INVENTION**

 The invention relates generally to
interpenetrating networks of carbon fibrils and polymers.
More specifically, the invention relates to an
interpenetrating network of carbon fibrils and a polymer,
10 which comprises a rigidized, three dimensional
macroscopic assemblage of a multiplicity of randomly
oriented carbon fibrils and an interpenetrating mass of a
polymer. Even more specifically, the invention relates
to methods for making such interpenetrating networks by
15 forming porous rigidized structures and forming an
organic polymer therein.

BACKGROUND OF THE INVENTION

 Carbon fibrils are vermicular carbon deposits
having diameters less than 500 nanometers (nm). They
20 exist in a variety of forms, and have been prepared
through the catalytic decomposition of various carbon-
containing gases at metal surfaces.

 Tennent, U.S. 4,663,230, describes carbon
fibrils that are free of a continuous thermal carbon
25 overcoat and have multiple graphitic outer layers that
are substantially parallel to the fibril axis. As such
they may be characterized as having their c-axes, the
axes which are perpendicular to the tangents of the
curved layers of graphite, substantially perpendicular to
30 their cylindrical axes. They generally have diameters no
greater than 0.1 micron and length to diameter ratios of
at least 5. Desirably they are substantially free of a
continuous thermal carbon overcoat, i.e., pyrolytically
deposited carbon resulting from thermal cracking of the
35 gas feed used to prepare them.

 Fibrils are useful in a variety of
applications. For example, they can be used as

51096-6

2

reinforcements in fiber-reinforced network structures or hybrid network structures, i.e., networks containing reinforcements such as continuous fibers in addition to fibrils.

5 In recent years, much interest has been expressed in the formation of molecular composites, i.e. composites in which individual molecules of rigid rod polymers are dispersed in more flexible matrix polymers to form mutually interpenetrating networks. It is
10 generally believed that such composites will be able to withstand stresses substantially greater than conventional composites because stress will be distributed throughout the interpenetrating molecular system. Also, such composites will be less likely to
15 suffer localized stress and will be able to withstand higher stress and/or strain before failure.

WO 94/25268 discloses methods for forming
20 three dimensional macroscopic assemblages of randomly oriented carbon fibrils. Broadly, dispersions of fibrils in a compatible liquid are prepared and the liquid is then removed to form a low-density porous plug or mat. In a preferred method, a low-density porous fibril plug
25 is prepared by dispersing the fibrils in solvent, e.g., n-pentane, the dispersion is charged to a pressure vessel, the vessel is heated above the critical temperature of the solvent, and the supercritical vapor is bled out of the vessel. In this manner a solid plug
30 having the shape of the vessel interior is obtained.

WO 97/43116 describes rigid, porous carbon structures comprised of carbon fibrils. The fibrils are bonded or glued to one another at their intersections. Bonding can be induced
35 by chemical modification of the surface of the fibers to promote bonding, by adding "gluing" agents and/or by

51096-6

3

pyrolyzing the fibrils to cause fusion or bonding at the points of interconnection.

OBJECTS OF THE INVENTION

It is a primary object of the invention to
5 produce a molecular composite based on a network of carbon fibrils interpenetrated with a polymer.

SUMMARY OF THE INVENTION

These and other objects of the invention are achieved by interpenetrating networks of carbon fibrils
10 and polymer, comprising a rigidized, three-dimensional, macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils and an interpenetrating mass of a polymer. Such interpenetrating networks can be obtained by forming a rigidized, three-dimensional, porous,
15 macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils, introducing a liquid or gas phase organic monomer into the porous interior of the assemblage, together with an appropriate catalyst or

51096-6

4

free-radical initiator, and causing the monomer to polymerize under polymerization conditions within the assemblage.

DETAILED DESCRIPTION OF THE INVENTION

5 Definitions

The term "fibril" refers to carbon fibers having a small diameter and includes fibrils, whiskers, nanotubes, buckytubes, etc.

10 The term "assemblage" refers to any configuration of a mass of individual fibrils and embraces intertwined as well as discrete fibril embodiments.

The term "macroscopic" means that the assemblages may be of any suitable size to achieve an industrial or scientific purpose.

Fibrils

20 The fibrils used in the invention have a diameter less than 1000 nm, preferably less than about 200 nm, even more preferably less than 100 nm and most preferably less than 50 nm. According to one embodiment of the invention, carbon fibrils having diameters in the range of 3.5 to 70 nm are used to form the rigid assemblage.

U.S. Patent No. 4,663,230 to Tennent describes carbon fibrils that are free of a continuous thermal carbon overcoat and have multiple ordered graphitic outer layers that are substantially parallel to the fibril axis. As such they may be characterized as having their c-axes, the axes which are perpendicular to the tangents of the curved layers of graphite, substantially perpendicular to their cylindrical axes. They generally have diameters no greater than 100 nm and length to diameter ratios of at least 5. Desirably they are substantially free of a continuous thermal carbon overcoat, i.e., pyrolytically deposited carbon resulting from thermal cracking of the gas feed used to prepare them. The Tennent invention

51096-6

5

provided access to smaller diameter fibrils, typically 3.5 to 70 nm and to an ordered, "as grown" graphitic surface. Fibrillar carbons of less perfect structure, but also without a pyrolytic carbon outer layer have also
5 been grown.

U.S. Patent No. 5,171,560 to Tennent et al., describes carbon fibrils free of thermal overcoat and having graphitic layers substantially parallel to the fibril axes such that the projection of the layers on the fibril axes
10 extends for a distance of at least two fibril diameters. Typically, such fibrils are substantially cylindrical, graphitic nanotubes of substantially constant diameter and comprise cylindrical graphitic sheets whose c-axes
15 are substantially perpendicular to their cylindrical axis. They are substantially free of pyrolytically deposited carbon, have a diameter less than 100 nm and a length to diameter ratio of greater than 5. These fibrils are of primary interest in the invention.

20 When the projection of the graphitic layers on the fibril axis extends for a distance of less than two fibril diameters, the carbon planes of the graphitic nanofiber, in cross section, take on a herring bone appearance. These are termed fishbone fibrils. Geus,
25 U.S. Patent No. 4,855,091, hereby incorporated by reference, provides a procedure for preparation of fishbone fibrils substantially free of a pyrolytic overcoat. These fibrils are also useful in the practice of the invention.

30 Oxidized nanofibers may be used to form the rigid porous assemblage. McCarthy et al., U.S. Patent No. 5,965,470, describes processes for oxidizing the surface of carbon fibrils that include contacting the fibrils with an oxidizing agent that
35 includes sulfuric acid (H_2SO_4) and potassium chlorate (KClO_3) under reaction conditions (e.g., time,

51096-6

6

temperature, and pressure) sufficient to oxidize the surface of the fibril. The fibrils oxidized according to the processes of McCarthy, et al. are non-uniformly oxidized, that is, the carbon atoms are substituted with
5 a mixture of carboxyl, aldehyde, ketone, phenolic and other carbonyl groups. Fibrils have also been oxidized non-uniformly by treatment with nitric acid.

WO 95/07316 discloses the formation of oxidized fibrils containing a mixture of functional groups.
10

In published work, McCarthy and Bening (Polymer Preprints ACS Div. of Polymer Chem. 30 (1)420(1990)) prepared derivatives of oxidized fibrils in order to demonstrate that the surface comprised a variety of
15 oxidized groups. Fibrils may also be oxidized using hydrogen peroxide, chlorate, nitric acid and other suitable reagents.

The fibrils may be further functionalized as set forth in WO 96/18059 entitled "Functionalized
20 Fibrils".

Carbon fibrils of a morphology similar to the catalytically grown fibrils described above have been
25 grown in a high temperature carbon arc (Iijima, Nature 354 56 1991). It is now generally accepted (Weaver, Science 265 1994) that these arc-grown fibrils have the same morphology as the earlier catalytically grown fibrils of Tennent. Arc grown carbon fibrils are
30 also useful in the invention.

The fibrils may also be high surface area fibrils disclosed in WO 97/43473 entitled "High Surface Area Nanofibers, Methods of Making, Methods

51096-6

7

of Using and Products Containing Same" .

Fibril Aggregates and Assemblages

The "unbonded" precursor fibrils may be in the
5 form of discrete fibers, aggregates of fibers or both.
Aggregates, when present, are generally of the bird's
nest, combed yarn or open net morphologies. The more
"entangled" the aggregates are, the more processing will
be required to achieve a suitable composition if a high
10 porosity is desired. This means that the selection of
combed yarn or open net aggregates is most preferable for
the majority of applications. However, bird's nest
aggregates will generally suffice.

The fibril mats or assemblages have been
15 prepared by dispersing fibrils in aqueous or organic
media and then filtering them to form a mat or
assemblage.

Rigidized assemblages are prepared by
intimately mixing fibrils with a glue, e.g. sugar,
20 glycerin, polyethylene oxide, polyethylene glycol,
polyacrylamide or polyacrylic acid, or, with carbonizable
resins, such as phenolic resins, in a kneader, followed
by extruding or pelletizing and pyrolyzing. Rigidized
assemblages have also been prepared by forming a gel or
25 paste of fibrils in a fluid, e.g. a solvent such as
carbon dioxide, acetone, or a C₂-C₇ alkane or alkene,
heating the gel or paste to a temperature above the
critical temperature of the medium, removing
supercritical fluid and finally removing the resultant
30 porous mat or plug from the vessel in which the process
has been carried out. See, WO 94/25268 entitled Three-
Dimensional Macroscopic Assemblages of Randomly Oriented
Carbon Fibrils and Composites Containing Same and
WO 97/43116 entitled Rigid Porous Carbon Structures, Methods
35 of Making, Methods of Using and Products Containing Same
referred to above.

Polymerization of Monomers in Rigid Assemblages

An interpenetrating network of the rigidized, three-dimensional macroscopic assemblage of randomly oriented fibrils and a polymer can be formed by
5 introducing a monomer into the rigidized assemblage together with a suitable polymerization catalyst and causing polymerization to take place under suitable conditions. The monomer may be in the liquid or the gas phase. Suitable monomers include vinyl compounds, i.e.
10 compounds having a terminal double bond, e.g. styrene, substituted styrene, methyl methacrylate, alpha olefins, substituted alpha olefins, etc. The monomer can be any compound that can be polymerized by either free radical mechanisms or by Ziegler-Natta catalysis. The
15 catalyst location and the polymerization rate must be adjusted by those skilled in the art to avoid having most of the polymer formed on the outside edges of the fibril network. In such event, the interior of the porous assemblage will have void spaces and the composition will
20 be nonuniform and therefore unsatisfactory. Accordingly it is preferred to evenly and thoroughly distribute the catalyst throughout the interior of the assemblage.

Fibrils have free radical traps and oxidized fibrils have even more free radical traps and accordingly
25 it is advantageous to treat the rigidized assemblages in order to reduce the number of those free radical traps, particularly where the mechanism of the contemplated polymerization is based on free radicals. Such mechanisms are advantageous in the methods of the
30 invention because initiators can be distributed throughout the fibril network before monomer is introduced and polymerization can be triggered uniformly by a temperature change.

Ziegler-Natta catalysts such as those derived
35 from TiCl_4 and $\text{ZrCl}_2(\text{Cp})_2$ can also be adsorbed on the fibrils and thus can uniformly catalyze a polymerization reaction, e.g. of polypropylene, throughout the

assemblage. Styrene can be polymerized, in situ, by both free radical initiation and Ziegler catalysis. Styrene is a suitable monomer because of its low cost and because it typically forms brittle matrices that can be toughened
5 with carbon fibrils.

Propylene, the lowest cost monomer, can be polymerized with a Ziegler catalyst distributed in the rigidized assemblage, and can be polymerized to isotactic (crystalline) polypropylene, useful for its higher
10 melting point, hardness, stiffness and toughness relative to other polyolefins.

EXAMPLES**EXAMPLE I****Preparation of Carboxylic Acid-Functionalized****5 Fibrils Using Nitric Acid**

A weighed sample of fibrils was slurried with nitric acid of the appropriate strength in a bound bottom multi-neck indented reactor flask equipped with an overhead stirrer and a water condenser. With constant
10 stirring, the temperature was adjusted and the reaction carried out for the specified time. Brown fumes were liberated shortly after the temperature exceeded 70°C, regardless of acid strength. After the reaction, the slurry was poured onto cracked ice and diluted with DI
15 water. The slurry was filtered and excess acid removed by washing in a Soxhlet extractor, replacing the reservoir with fresh DI water every several hours, until a slurried sample gave no change in pH from DI water. The fibrils were dried at 100°C at 5" vacuum overnight.

20 A weighed portion of fibrils was reacted with standard 0.100 N NaOH and the carboxylic acid content determined by back-titration with 0.100 N HCl. Surface oxygen content was determined by XPS. Dispersibility in water was tested at 0.1 wt% by mixing in a Waring Blender at
25 high for 2 min. Results are summarized in the Table I below.

Table 1
Summary of Direct Oxidation Runs

COMPONENTS										
Disp.	Ex. H ₂ O	Gms. Fibrils	cc Acid	Acid Conc.	Temp. °C	Time	Wgt. Loss	COOH meg/g	ESCA, C at% O	
12A P		1 (BN)	300	70%	RT	24 hr	0	<0.1	98	2
12B P		1 (BN)	300	70	rflx	48	<5%	<0.1	not analyzed	
12C G		20 (BN)	1.01	70	rflx	7	25%	0.8	not analyzed	
12D G		48 (BN)	1.01	70	rflx	7	20%	0.9	not analyzed	

P=Poor; G=Good

5

10

15

EXAMPLE II**Preparation of a Rigidized Low-Density Porous Fibril Plug**

Supercritical fluid removal from a well dispersed-fibril paste is used to prepare low density shapes. 50 cc of a 0.5% dispersion of the fibrils from Example 1 above in n-pentane is charged to a pressure vessel of slightly larger capacity which is equipped with a needle valve to enable slow release of pressure. After the vessel is heated above the critical temperature of pentane ($T_c = 196.6^\circ$), the needle valve is cracked open slightly to bleed the supercritical pentane over a period of about an hour.

The resultant solid plug of Fibrils, which has the shape of the vessel interior, has a density of 0.005 g/cc, corresponding to a pore volume fraction of 0.997%. The resistivity is isotropic and about 20 ohm/cm.

The preform is rigidized by heating to 650°C in argon for 1 hr. Alternatively the preform is rigidized by heating to 300°C in air for one hour.

EXAMPLE III**Preparation of a Rigidized Low-Density Porous Fibril Plug**

0.333g resorcinol, (Aldrich) is dissolved in 5.3 cc H_2O . After 0.491g formaldehyde solution (37% in H_2O , Aldrich) is added, the solution is mixed thoroughly with 8.879g fibril slurry (5.8%). After the addition of 7.4cc 0.2M Na_2CO_3 , the mixture is transferred to a glass vial. The sealed vial is placed in an oven at 80°C for four days. The gel formed is washed with water. Finally, the water in the gel is exchanged with acetone.

After the supercritical acetone in the gel is removed, the product may be heated under Ar at 400°C for 2 hr., 800°C for 4 hr. and 1200°C for 4 hr. to carbonize the resoreinol-formaldehyde polymer.

EXAMPLE IV**Preparation of a Rigidized Low-Density Porous Fibril Plug**

Supercritical fluid removal from a well dispersed-fibril paste is used to prepare low density

shapes. 50 cc of a 0.5% dispersion of "as made" fibrils in n-pentane containing a mixture of phenol formaldehyde/PEG/Glycerin as a glue is charged to a pressure vessel of slightly larger capacity which is
5 equipped with a needle valve to enable slow release of pressure. After the vessel is heated above the critical temperature of pentane ($T_c = 196.6^\circ$), the needle valve is cracked open slightly to bleed the supercritical pentane over a period of about an hour.

10 The resultant solid plug of fibrils, which has the shape of the vessel interior is air dried and heated to 350°C to remove PEG/Glycerin and pyrolyze the phenolic resin. The plug has a density of 0.005 g/cc, corresponding to a pore volume fraction of 0.997%.

15

EXAMPLE V

Porous preforms prepared as described in Examples II, III and IV are hydrogenated at $800\text{--}1400^\circ\text{C}$ in an aluminum tube using 99.999% (high purity) hydrogen and cooled to room temperature under argon.

20

The preforms are infiltrated with a mixture of styrene monomer/benzyl peroxide. The preform is then heated to 80° for 1 hr. to polymerize the styrene monomer. A solid polystyrene/fibril network is obtained.

EXAMPLE VI

25

Preforms prepared as in Examples II, III or IV are placed in a serum-capped pressure bottle containing a magnetic stirring bar. The bottle is evacuated and filled with dry nitrogen several times. It is then chilled to 0°C and sparged with nitrogen which passes
30 through a serum-capped bottle containing ca. 0.01 gm TiCl_4 per gm of preform at 50°C until all the TiCl_4 has evaporated. The bottle containing the fibrils was then held at 50°C until all the TiCl_4 has evaporated. The bottle containing the fibrils is then held at 50°C for an
35 hour to distribute the TiCl_4 on the fibrils. Triethyl aluminum in the amount of 0.03 gm per gm of preform is similarly introduced.

With the bottle at 25°C, a stream of ethylene is introduced at atmospheric pressure. A strong exotherm indicates that the ethylene is polymerizing. The black fibrils turn grey, then nearly white. Polymerization is continued until ca. 150 gm per gm of preform of polyethylene has formed. An interpenetrating network of fibrils and polyethylene is obtained.

EXAMPLE VII

A mixture of 1 m mol methyl alomoxane and 10 μ mol biscyclopenta dienyl zirconium dichloride in benzene is prepared and added to a rigidized preform prepared according to Example II, III or IV to the point of incipient wetness. The system is cooled to below the freezing point of benzene and the benzene is removed by sublimation. The treated preform is taken to dry ice temperature and infiltrated with liquid propylene at 100 psi. The system is slowly warmed until polymerization initiates and maintained at that temperature for one hour. An interpenetrating network of fibrils and polypropylene is obtained.

51096-6

15

CLAIMS:

1. An interpenetrating network of carbon fibrils and a polymer comprising
 - (a) a rigidized, three dimensional macroscopic
5 assemblage of a multiplicity of randomly oriented carbon fibrils, and
 - (b) an interpenetrating mass of a polymer.
2. An interpenetrating network of carbon fibrils and a polymer comprising:
 - 10 (a) a rigidized, three-dimensional, macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils, the fibrils being substantially cylindrical with a substantially constant diameter, having c-axes substantially perpendicular to their
15 cylindrical axis, being substantially free of pyrolytically deposited carbon and having a diameter between about 3.5 and 70 nanometers, and
 - (b) an interpenetrating mass of an organic polymer.
- 20 3. A method for producing an interpenetrating network of carbon fibrils and a polymer comprising the steps of:
 - (a) forming a rigidized, three dimensional, porous, macroscopic assemblage of a multiplicity of
25 randomly oriented carbon fibrils, and
 - (b) introducing a liquid or gas phase monomer into the interior of the assemblage together with a polymerization catalyst or initiator; and
 - 30 (c) causing the monomer to polymerize within the assemblage.
4. A method for producing an interpenetrating network of carbon fibrils and a polymer comprising the steps of
 - 35 (a) forming a rigidized, three-dimensional, porous, macroscopic assemblage of a multiplicity of randomly oriented carbon fibrils, the fibrils being

51096-6

16

substantially cylindrical with a substantially constant diameter, having c-axes substantially perpendicular to their cylindrical axis, being substantially free of pyrolytically deposited carbon and having a diameter
5 between about 3.5 and 70 nanometers;

(b) introducing a liquid or gas phase organic monomer into the porous interior of the assemblage together with a Ziegler catalyst or a free-radical initiator; and

10 (c) causing the monomer to polymerize under polymerization conditions within the assemblage.

51096-6

17

5. The interpenetrating network according to claim 1 or 2, wherein the carbon fibrils are oxidized carbon fibrils produced by oxidizing carbon fibrils with sulfuric acid, potassium chlorate or nitric acid.

5 6. The interpenetrating network according to claim 1, 2 or 5, wherein the polymer is derived from a monomer selected from the group consisting of styrene, substituted styrene, methyl methacrylate and alpha olefins.

7. The interpenetrating network according to claim 6,
10 wherein the monomer is styrene.

8. The interpenetrating network according to claim 6, wherein the monomer is an alpha olefin.

9. The method according to claim 3 or 4, wherein the carbon fibrils are oxidized carbon fibrils produced by
15 oxidizing carbon fibrils with sulfuric acid, potassium chlorate or nitric acid.

10. The method according to claim 3, 4 or 9, wherein the monomer is selected from the group consisting of styrene, substituted styrene, methyl methacrylate and alpha olefins.

20 11. The method according to claim 3, 4 or 9, wherein the monomer is styrene and the initiator is employed.

12. The method according to claim 3, 4 or 9, wherein the monomer is an alpha olefin and the catalyst is employed.

13. The method according to claim 3 or 4 or any one of
25 claims 9 to 12, wherein the rigidized three-dimensional macroscopic assemblage is formed by:

(i) intimately mixing the carbon fibrils with a glue or a carbonizable resin in a kneader, to form a mixture; and

51096-6

18

(ii) extruding or pellitizing and then pyrolyzing the mixture.

14. The method according to claim 15, wherein a phenolic resin is used as the carbonizable resin.

5 15. The method according to claim 3 or 4 or any one of claims 9 to 12, wherein the rigidized three-dimensional macroscopic assemblage is formed by:

(i) forming a gel or paste of the carbon fibrils in a solvent having a critical temperature;

10 (ii) heating the gel or paste to a temperature above the critical temperature in a vessel;

(iii) removing the solvent to form a porous mat or plug; and

(iv) removing the formed porous mat or plug from
15 the vessel.

16. The method according to claim 15, wherein the solvent used is a C₂-C₇ alkane.

17. The method according to claim 16, wherein the solvent is n-pentane.

SMART & BIGGAR
OTTAWA, CANADA
PATENT AGENTS