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(71) Applicant(s)
Asahi Kasei Kogyo Kabushiki Kaisha

(72) Inventor(s)
Shuichiro Ogawa; Keigo Takeguchi; Atsushi Fujii

(74) Agent/Attorney
WATERMARK PATENT and TRADEMARK ATTORNEYS, Locked Bag 5, HAWTHORN VIC 3122

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<table border="0"><tr><td data-bbox="384 524 879 1070" style="vertical-align: top;"><p>(21) 国際出願番号 PCT/JP98/01715 (22) 国際出願日 1998年4月15日(15.04.98) (30) 優先権データ 特願平9/99113 1997年4月16日(16.04.97) JP 特願平9/266919 1997年9月30日(30.09.97) JP 特願平9/283087 1997年9月30日(30.09.97) JP 特願平9/348114 1997年12月17日(17.12.97) JP (71) 出願人 (米国を除くすべての指定国について) 旭化成工業株式会社 (ASAHI KASEI KOGYO KABUSHIKI KAISHA)[JP/JP] 〒530-8205 大阪府大阪市北区堂島浜1丁目2番6号 Osaka, (JP) (72) 発明者 ; および (75) 発明者 / 出願人 (米国についてののみ) 小川周一郎(OGAWA, Shuichiro)[JP/JP] 〒416-0933 静岡県富士市中丸660-2 Shizuoka, (JP) 竹口圭吾(TAKEGUCHI, Keigo)[JP/JP] 藤井 昌(FUJII, Atsushi)[JP/JP] 〒416-0934 静岡県富士市鮫島351-1 旭化成誠和寮 Shizuoka, (JP)</p></td><td data-bbox="879 524 1386 1070" style="vertical-align: top;"><p>(74) 代理人 弁理士 森 哲也, 外(MORI, Tetsuya et al.) 〒101-0045 東京都千代田区神田鍛冶町三丁目7番地 村木ビル8階 日栄国際特許事務所 Tokyo, (JP) (81) 指定国 AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CU, CZ, DE, DK, EE, ES, FI, GB, GE, GH, GM, GW, HU, ID, IL, IS, JP, KE, KG, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, US, UZ, VN, YU, ZW, ARIPO特許 (GH, GM, KE, LS, MW, SD, SZ, UG, ZW), ユーロパ特許 (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), 欧州特許 (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI特許 (BF, BJ, CF, CG, CI, CM, GA, GN, ML, MR, NE, SN, TD, TG). 添付公開書類 国際調査報告書</p></td></tr></table>			<p>(21) 国際出願番号 PCT/JP98/01715 (22) 国際出願日 1998年4月15日(15.04.98) (30) 優先権データ 特願平9/99113 1997年4月16日(16.04.97) JP 特願平9/266919 1997年9月30日(30.09.97) JP 特願平9/283087 1997年9月30日(30.09.97) JP 特願平9/348114 1997年12月17日(17.12.97) JP (71) 出願人 (米国を除くすべての指定国について) 旭化成工業株式会社 (ASAHI KASEI KOGYO KABUSHIKI KAISHA)[JP/JP] 〒530-8205 大阪府大阪市北区堂島浜1丁目2番6号 Osaka, (JP) (72) 発明者 ; および (75) 発明者 / 出願人 (米国についてののみ) 小川周一郎(OGAWA, Shuichiro)[JP/JP] 〒416-0933 静岡県富士市中丸660-2 Shizuoka, (JP) 竹口圭吾(TAKEGUCHI, Keigo)[JP/JP] 藤井 昌(FUJII, Atsushi)[JP/JP] 〒416-0934 静岡県富士市鮫島351-1 旭化成誠和寮 Shizuoka, (JP)</p>	<p>(74) 代理人 弁理士 森 哲也, 外(MORI, Tetsuya et al.) 〒101-0045 東京都千代田区神田鍛冶町三丁目7番地 村木ビル8階 日栄国際特許事務所 Tokyo, (JP) (81) 指定国 AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CU, CZ, DE, DK, EE, ES, FI, GB, GE, GH, GM, GW, HU, ID, IL, IS, JP, KE, KG, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, US, UZ, VN, YU, ZW, ARIPO特許 (GH, GM, KE, LS, MW, SD, SZ, UG, ZW), ユーロパ特許 (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), 欧州特許 (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI特許 (BF, BJ, CF, CG, CI, CM, GA, GN, ML, MR, NE, SN, TD, TG). 添付公開書類 国際調査報告書</p>
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<p>(54)Title: PROCESS FOR PRODUCING OPTICAL INFORMATION RECORDING MEDIUM AND OPTICAL INFORMATION RECORDING MEDIUM PRODUCED BY THE PROCESS</p> <p>(54)発明の名称 光学情報記録媒体の製造方法、およびこの方法で製造された光学情報記録媒体</p> <p>(57) Abstract An optical information recording medium which can eliminate the necessity for the initialization step. A crystallization-accelerating layer (3) comprising a given material is formed over a substrate (1) on one side thereof, and a recording layer (4) comprising a Ge-Sb-Te alloy is formed directly on the layer (3). Since the recording layer (4) crystallizes immediately after film formation, no initialization step is necessary for the optical information recording medium obtained. Examples of the material of the crystallization-accelerating layer firstly include substances having a face-centered cubic lattice crystal structure. Examples thereof secondly include tellurium-free substances having a rhombohedral lattice crystal structure. An especially preferred crystallization-accelerating layer is a discontinuous film in the form of a layer of islands made of a material comprising bismuth and/or a bismuth compound. Incorporating nitrogen into the crystallization-accelerating layer provides an optical information recording medium which need not be initialized and has excellent suitability for repeated use.</p> <div data-bbox="951 1290 1342 1727"></div>				

ABSTRACT

An optical information recording medium which can eliminate the necessity for the initialization process. A crystallization assisted layer (3) comprising a given material is formed over a substrate (1) on one side thereof, and a recording layer (4) comprising a Ge-Sb-Te alloy is formed directly on the layer (3). Since the recording layer (4) crystallizes immediately after film formation, no initialization process is necessary for the optical information recording medium obtained. Examples of the material of the crystallization assisted layer firstly include materials having a face-centered cubic lattice system crystal structure. Examples thereof secondly include tellurium-free materials having a rhombohedral lattice system crystal structure. An especially preferred crystallization assisted layer is a discontinuous island-like film made of a material comprising bismuth and/or a bismuth compound. Incorporation nitrogen into the crystallization assisted layer provides an optical information recording medium which need not be initialized and has excellent overwrite cyclability.



SPECIFICATION

PROCESS FOR PRODUCING OPTICAL INFORMATION RECORDING 5 MEDIUM AND OPTICAL INFORMATION RECORDING MEDIUM PRODUCED BY THE PROCESS

TECHNICAL FIELD

The present invention relates to phase change optical information
10 recording mediums having a recording layer of changing phase between a
crystalline state and an amorphous state in accordance with the intensity
of an irradiation beam, and in particular, it relates to a method for
producing an optical information recording medium capable of making an
initialization process unnecessary.

15

BACKGROUND ART

Recently, optical information recording mediums have been
extensively studied and developed as means for recording, reading and
erasing an immense quantity of information. Especially, a so-called phase
20 change optical disk which records/erases information, using the fact that
the phase of the recording layer changes reversibly between a crystalline
state and an amorphous state, has the advantage that only by changing the
laser beam power, old information is erased while new information is being
recorded simultaneously (hereinafter referred to as "overwrite"). Thus,
25 such optical disk is regarded as being full of promise.

As the recording materials of such overwritable phase change
optical disk, chalcogen alloys are mainly used which include In-Se alloys



(see "Appl. Phys. Lett. Vol. 50, p. 667, 1987"), In-Sb-Te alloys (see "Appl. Phys. Lett. Vol. 50, p.16, 1987"), and Ge-Te-Sb alloys (see Japanese Patent Laid-Open Publication Sho No. 62-53886), which have a low melting point and a high absorption efficiency for a laser beam.

5 When information is actually recorded/erased on/from such optical disk of a chalcogen alloy, at least one kind of dielectric layer of a material selected from the group consisting of metal or semi-metal oxides, carbides, fluorides, sulfides, and nitrides is generally formed directly above and/or under the recording layer in order to prevent the substrate from being
10 deformed due to heat produced on recording/erasing, to prevent the recording layer from being oxidized, and/or to prevent the substances from moving along the guide grooves or from being deformed.

Optical disks having a three- or four-layered structure which includes a recording layer of a chalcogen alloy, a dielectric layer provided
15 directly under and/or above the recording layer, a reflective layer which also acts as a cooling layer (for example, Al-alloy) provided on an opposite side of a transparent substrate from the recording layer, provided on the substrate, are the mainstream of the phase change optical disks because they are preferable in terms of the recording/erasing characteristics.

20 In general phase change optical disks, when the recording layer is irradiated with a laser beam having a recording power to heat it up to its melting point and is then rapidly cooled, the recording layer material is produced amorphous to thereby form a recording mark. Then, when the recording layer is irradiated with a laser beam having an erasing power to
25 be heated to more than the crystallizing temperature and then gradually cooled, the recording layer material is crystallized to thereby erase the recording mark.



Such phase change optical disks are each produced by sequentially forming thin layers as the respective layers on the substrate by sputtering/evaporation. Since the recording layer present immediately after its layer formation is amorphous, it is irradiated with a laser beam to be wholly crystallized, which is generally referred to as "initialization process", and the optical disks, thus obtained, are then shipped.

However, this initialization process takes a time of a little less than one minute to initialize the whole optical disk having a diameter of 120 mm even with the use of the most-efficient laser beam irradiation, which leads to an increase in the manufacturing cost of the optical disks. For the time required for processing one optical disk in each manufacturing substep (cycle time), the time required for the initialization process is long compared to the substrate molding step or the layer forming step. Thus, in order to eliminate a time loss taken to pass to the initialization process when the cycle time for the layer forming step is 8 seconds, the six or seven very expensive initializing devices are required. As a result, by performing the initialization process, the manufacturing cost of the optical disks is increased.

In order to reduce the time required for the initialization process, for example, Japanese Patent Laid-Open Publication Hei No. 5-342629 discloses providing an auxiliary layer of an easily crystallizable continuous film or discontinuous island-like film adjacent to the recording layer. As the components of the auxiliary layer, compounds are named which include tellurium (Te), Selenium (Se) or Te - Se compounds.

However, according to this method, the time required for initializing the recording layer is reduced, but the initialization process cannot be eliminated as a rule, excluding the case where both of the auxiliary layer



and the recording layer are comprised of extremely easily crystal-growing substances.

It is therefore an object of the present invention to provide an optical information recording medium which eliminates the necessity for
5 the initialization process.

DISCLOSURE OF THE INVENTION

The present invention provides a method for producing an optical information recording medium which has on one side of a substrate a
10 recording layer whose main components comprise germanium (Ge), antimony (Sb) and tellurium (Te) (hereinafter referred to as "Ge-Te-Sb alloy"), comprising the steps of forming a crystallization assisted layer of materials having a face-centered cubic lattice system crystal structure on one side of the substrate, and forming a recording layer directly above the
15 crystallization assisted layer. According to this method, the recording layer immediately after its formation is crystallized.

The Ge-Te-Sb alloys take two types of crystal phases: namely, a face-centered cubic lattice system crystal structure and a hexagonal system crystal structure. It is known that as the temperature of this alloy is
20 raised from its amorphous state, its phase changes from a face-centered cubic lattice crystal structure to a hexagonal structure. In the present invention, the recording layer is easily crystallized when its layer is formed due to the presence of the crystallization assisted layer having the same face-centered cubic lattice system crystal structure as the recording layer.

25 The face-centered cubic lattice system crystal structures include face-centered cubic lattices, and face-centered tetragonal lattice; diamond-shape structures: CuAu-, CuPt-, Ni₂Cr-, Cu₃Au-, Ni₄Mo-, Ag₃Mg-, Ni₃V-,



Cu₃Pd-, and Au₃Mn-type superlattices; NaCl-, NaTl-, ZnS-, CaF₂-, FeS₂-, cristobalite high-temperature-, Laves phase MgCu₂-, Cu₃Au-, Al₃Ti-, Cu₂AlMn-, Al₂MgO₄-, and Bi₂Te₃-type structures; and their interstitial and substitutional solid solutions.

5 The present invention also provides a method for producing an optical information recording medium having on one side of a substrate a recording layer whose main components comprise germanium (Ge), antimony (Sb) and tellurium (Te), comprising the steps of forming on one side of a substrate a crystallization assisted layer of a tellurium (Te)-free
10 material having a crystal structure of a rhombohedral lattice system, and forming a recording layer directly over the crystallization assisted layer. According to this method, the recording layer becomes crystallized immediately after its formation.

 In the present invention, the absolute value of a lattice
15 unconformity between the crystal structure of the crystallization assisted layer and that of the recording layer is preferably not more than 8%. The lattice unconformity is represented by:

$$\text{Lattice unconformity (\%)} = ((B - A)/A) \times 100 \dots (a)$$

20

A: When the recording layer is of a face-centered cubic lattice system crystal, an atomic interval in a direction $\langle 110 \rangle$ of the crystal;

25

B: A particular one of the atomic intervals of a crystallized crystallization assisted layer such that the difference between A and the particular atomic interval B is minimum among the differences each between A and a respective one of the atomic intervals of the



crystallized crystallization assisted layer. In the case of the face-centered cubic lattice system, it is generally the atomic interval in a direction $\langle 100 \rangle$ or $\langle 110 \rangle$.

5 When the crystal comprises two or more kinds of elements, the distance between two adjacent atoms of different kinds may be used as an atomic interval in the expression (a). When A is greatly different from B in the expression (a), the atomic interval B of the crystallization assisted layer may be assumed to be an integer or fraction times the atomic interval.

10 The range of lattice unconformity is preferably - 4.5 to + 8%, and more preferably - 3 to + 7%.

 Examples of materials each of which has a crystal structure of a face-centered cubic lattice system where an absolute value of lattice unconformity between the crystal structure of that material and that of the
15 recording layer is not more than 8% include PbTe and Bi₂Te₃.

 Examples of tellurium-free materials each of which has a crystal structure of a rhombohedral system where an absolute value of lattice unconformity between the crystal structure of that material and that of the recording layer is not more than 8% include antimony (Sb), bismuth (Bi),
20 antimony (Sb) compounds, and bismuth (Bi) compounds. The Sb compounds include Sb alloys, and intermetallic compounds of Sb and other metals or semimetals. The Bi compounds include Bi alloys, and intermetallic compounds of Bi and other metals or semimetals.

 In the present invention, the thickness of the crystallization
25 assisted layer is preferably not more than 200 Å. If the thickness is larger than 200 Å, the record erasing characteristics would be deteriorated. The thickness of the crystallization assisted layer is more preferably not more



than 100 Å. If this layer is excessively thin, the recording layer can be crystallized insufficiently. Thus, it is preferably not less than 1 Å.

In the present invention, the crystallization assisted layer may be in the form of a continuous film or a discontinuous island-like film, which is
5 contacted with the recording layer. Most preferably, it is a discontinuous island-like film of materials which contain bismuth (Bi) and/or a bismuth (Bi) compound.

The optical information recording medium whose recording layer is crystallized by the crystallization assisted layer provided so as to be
10 contacted with a substrate-side surface of the recording layer eliminates the necessity for the initialization process. If a continuous film of materials which comprise bismuth (Bi) and/or a bismuth (Bi) compound is used as the crystallization assisted layer, the CNR (Carrier to Noise-Ratio) in the second or subsequent recording by overwriting is slightly lower than
15 that in the first recording.

In comparative examples, if a discontinuous island-like film of materials containing bismuth (Bi) and/or a bismuth (Bi) compound is used as the crystallization assisted layer, the CNR in the second or subsequent recording by overwriting is substantially the same as that in the first
20 recording.

The discontinuous island-like film is formed, for example, by sputtering such that its thickness is not more than a predetermined value.

When the crystallinity of the recording layer formed on the crystallization assisted layer is insufficient, the recording layer is
25 preferably formed by setting the temperature of the substrate in a range of from 45°C through a temperature inclusive above which temperature the substrate would be deformed (at 110°C when the substrate is produced of



polycarbonate). Thus, the recording layer is placed in a stabilized crystalline state.

The methods for maintaining the substrate at high temperatures under formation of the recording layer include (1) heating the substrate or
5 the crystallization assisted layer which underlies the recording layer immediately before the formation of the recording layer to thereby maintain the substrate at high temperature; (2) starting to heat the substrate or the crystallization assisted layer after the formation of the recording layer has started and continuing to heat the substrate or the
10 crystallization assisted layer during the formation of the recording layer; (3) starting to heat the substrate or the layer surface of the recording layer immediately after the recording layer has been formed; and (4) starting to form the recording layer immediately after the preceding layer has been formed, using the heat produced by the formation of the preceding layer
15 and stored within the substrate.

The heating methods include irradiating a surface of the layer formed on the substrate (a surface of the crystallization assisted layer) with light including heat rays; and heating a substrate holder itself with a heater or the like, or using high frequency induction, flash exposure; or
20 plasma processing.

In the producing method of the present invention, the formation of the crystallization assisted layer is preferably performed within a layer forming atmosphere to which a nitrogen gas is added.

When the producing method of the present invention includes the
25 step of forming a first dielectric layer between the substrate and the crystallization assisted layer and/or the step of forming a second dielectric layer on an opposite side of the recording layer from the crystallization



assisted layer, the formation of the first and/or second dielectric layer is preferably performed within a layer forming atmosphere to which a nitrogen gas and/or an oxygen gas is added.

The present invention also provides an optical information
5 recording medium with a recording layer formed on one side of a substrate, the recording layer comprising materials whose main components are germanium (Ge), antimony (Sb) and tellurium (Te), wherein the recording layer is formed in a crystalline state and wherein the crystallization assisted layer is formed in contact with the substrate side surface of the
10 recording layer, the recording layer and the crystallization assisted layer being produced by the respective above-mentioned producing methods.

In the inventive optical information recording medium, a ratio x, y, and z of the respective elements (Ge, Sb, Te) of the main components of the recording layer (Ge: Sb: Te = x: y: z where $x + y + z = 1$) is preferably in a
15 range shown hatched in a triangular graph of FIG. 2, which satisfies the following expressions (1)-(3) simultaneously:

$$0.1 \leq x \leq 0.4 \quad \dots (1)$$

$$0.08 \leq y \quad \dots (2)$$

$$20 \quad 0.45 \leq z \leq 0.65 \quad \dots (3)$$

When $x < 0.1$, the optical information recording medium is not preferable in terms of stability. When $x > 0.4$, $y < 0.08$, $z < 0.45$ and $z > 0.65$, these conditions are unpreferable because the recording layer is
25 difficult to crystallize.

A preferable range of the ratio x, y and z of the respective elements of the main components of the recording layer (Ge: Sb: Te = x: y: z where $x +$



y + z = 1) should satisfy the following expressions (4)-(6) simultaneously:

$$0.15 \leq x \leq 0.3 \quad \dots (4)$$

$$0.12 \leq y \quad \dots (5)$$

5 $0.5 \leq z \leq 0.6 \quad \dots (6)$

The materials of the recording layer are preferably Ge-Te-Sb-Bi alloys containing Bi in addition to Ge, Te and Sb. The materials may be Ge-Te-Sb or Ge-Te-Sb-Bi alloys, for example, containing hydrogen, nitrogen,
10 oxygen, carbon, Al, Ti, Fe, Co, Ni, Cu, Zn, Ga, Se, Sn, In, Ag, Pd, Rh, Ru, Mo, Nb, Hf, Zr, Ta, W, Re, Os, Ir, Pt, Au, Tl and/or Pb. Those elements may be added from the target during the formation of the recording layer or added in a gaseous state to the atmosphere gas so as to be contained within the recording layer.

15 In the inventive optical information recording medium, the thickness of the recording layer is preferably 50-1000 Å. If it is less than 50 Å, the recording layer could not obtain a satisfactory recording sensitivity. When it exceeds 1000 Å, a problem about the recording sensitivity and resolution would occur, undesirably.

20 The inventive optical information recording medium preferably has a 4-layered structure in which a crystallization assisted layer, a recording layer, a dielectric layer and a reflective layer are formed on the substrate in this order. More preferably, the inventive optical information recording medium has a 5-layered structure in which a first dielectric layer, a
25 crystallization assisted layer, a recording layer, a second dielectric layer and a reflective layer are formed on the substrate in this order. The inventive optical information recording medium may further include other



necessary layers additionally.

As the first and second dielectric layers, materials having high heat resistance and a melting point of not less than 1000°C are preferable; for example, SiO₂; a mixture of ZnS and SiO₂; Al₂O₃; AlN; and Si₃N₄. Although
5 the thickness of the first dielectric layer is not especially specified, the thickness of the second dielectric layer is preferably 50-500 Å. If it is less than 50 Å, it can not provide a satisfactory recording sensitivity. If it exceeds 500 Å, it cannot provide satisfactory overwrite cyclability. The thickness of the reflective layer is preferably not less than 300 Å.

10 The methods of forming the respective layers include evaporation, sputtering and ion plating.

A method of confirming the presence of the crystallization assisted layer in the optical information recording medium will be described next.

The first method is to observe a cross section of the optical
15 information recording medium, with a transmission electron microscope. The elements of the crystallization assisted layer can be specified with the aid of an electron beam diffraction apparatus and an energy dispersion X-ray analysis apparatus. When the crystallization assisted layer is island-like or very thin, it is difficult to confirm its presence, using this method.

20 The second method includes slowly etching layers formed on the substrate of the optical information recording medium, by sputtering, in a direction perpendicular to the substrate surface while analyzing elements present at respective positions in the layers formed on the substrate, using a secondary ion mass spectrometry (SIMS) or Auger electron spectroscopy
25 (AES). This method is effective when the crystallization assisted layer is island-like or very thin.

According to this method, as the recording layer is slowly etched



toward an interface between the recording layer and the crystallization assisted layer while the elements are being analyzed, the quantity of elements which compose the crystallization assisted layer increases toward the interface between the crystallization assisted layer and its underlying
5 layer (generally, the dielectric layer), and after the interface is reached, rapidly decreases. By finding this phenomenon, the presence of the crystallization assisted layer will be known.

As an example, in the case of an optical disk which has a layer structure of a substrate/a first dielectric layer/a crystallization assisted
10 layer/a recording layer/a second dielectric layer/a reflective layer/a UV set resin layer, a method of confirming the presence of the crystallization assisted layer, using the second method, will be explained as follows. First, an adhesive tape is adhered to the UV set resin layer to thereby separate the layered layer from the substrate. At this time, since the second
15 dielectric layer is generally separated from the recording layer, the substrate on which the recording layer and the crystallization assisted layer remain is put into the secondary ion mass spectrometry or Auger electron spectroscopy to analyze the elements using by etching slowly from the recording layer side.

20 In the case of an optical disk where the recording layer comprises Ge, Te and Sb and where the crystallization assisted layer is in the form of a discontinuous island-like film of Bi, the presence of Ge, Te and Sb is first confirmed by analysis of the elements. As the layers on the substrate are further etched, the presence of Bi is recognized. By further etching, the
25 quantity of Bi increases whereas the quantities of Ge, Te and Sb decrease gradually. When the dielectric layer is reached, no presence of Ge, Te, Sb and Bi is recognized. Thus, when such phenomenon is found in the second



method, it can be determined that the optical disk comprises a crystallization assisted layer comprised of a discontinuous island-like film of Bi.

5 An optical disk whose recording layer is crystallized immediately after its formation is easily distinguished from an optical disk whose recording layer is crystallized in the initialization process, with the following method .

10 In the optical disk whose recording layer is crystallized by an initializing device using general laser-beam irradiation, the innermost and outermost peripheries of the recording layer are not initialized to remain amorphous state due to the composition of the initializing device. Thus, the innermost and outermost peripheries of the disk and its intermediate portion are different in reflectivity, which will be visually recognized by those skilled in the art. In comparative examples, in the optical disk
15 whose recording layer is crystallized immediately after its formation and not subjected to the initialization process, there is no such difference in reflectivity because the whole surface of the recording layer is crystallized.

BRIEF DESCRIPTION OF THE DRAWINGS

20 FIG. 1 is a cross-sectional view of a layer structure of an optical information recording medium corresponding to one embodiment of the present invention;

FIG.2 is a triangular graph indicative of a preferable range of a ratio of the respective elements of Ge-Sb-Te as the main components of the
25 recording layer;

FIG. 3 is a graph of diffraction X-ray spectra obtained by X-ray diffraction from X-ray diffraction samples A-C of a second embodiment;



FIG. 4 is a photograph of a thin layer corresponding to a crystallization assisted layer of an example 3-1;

FIG. 5 is a photograph of a thin layer corresponding to a crystallization assisted layer of an example 3-2; and

5 FIG. 6 is a graph of diffraction X-ray spectra obtained by X-ray diffraction from X-ray diffraction samples A-G of a third embodiment.

BEST MOOD FOR CARRYING OUT THE INVENTION

First Embodiment:

10 Phase change optical disks each having a layer structure of FIG. 1 were produced as follows:

First, as each substrate 1, a polycarbonate substrate with a center hole was prepared, having a diameter of 120 mm, a thickness of 0.6 mm, and a helical guide groove of 0.74 μm in width and 1.4 μm in track pitch.

15 Then, formed on this substrate 1 were a first dielectric layer 2 of a mixture of ZnS and SiO_2 (the content of SiO_2 is 30 mole %) and having a thickness of 2500 Å, a crystallization assisted layer 3 produced of each of materials shown in Table 1 and having a corresponding thickness also shown in Table 1, a recording layer 4 of a Ge - Te - Sb alloy having a
20 thickness of 200 Å, a second dielectric layer 5 of a mixture of ZnS and SiO_2 (the content of SiO_2 is 30 mole %) having a thickness of 150 Å, and a reflective layer 6 of an Al - Ti alloy (the content of Ti is 1.1 atom %) having a thickness of 800 Å in this order. In a comparative example 1-1, no crystallization assisted layer 3 was formed. A UV cured resin layer 7 was
25 formed on the reflective layer 6.

The formation of the first and second dielectric layers 2 and 5 was performed by RF sputtering, using a mixture of ZnS and SiO_2 (the content



of SiO_2 was 30 mole %) as the target. An argon gas was used as the atmosphere gas for sputtering.

The formation of the crystallization assisted layer 3 was performed by DC sputtering, using each of the materials whose purity was 99.9 atom % as the target. An argon gas was used as the atmosphere gas.

The formation of the recording layer 4 was performed by DC sputtering, using a Ge - Te - Sb alloy as the target. An argon gas was used as the atmosphere gas. Analysis of the recording layer 4 by a fluorescent X-ray clarified that the composition of the formed recording layer was: Ge = 21 atom %, Te = 24 atom %, and Sb = 55 atom %.

The formation of the reflective layer 6 was performed by DC sputtering, using an Al - Ti alloy (the contents of Ti was 1.1 atom %) as the target. An argon gas was used as the atmosphere gas.

The formation of the UV cured resin layer 7 was performed by spin-coating the reflective layer 6 with a UV curable resin, and then by irradiating the resin with UV rays.

Table 1, attached hereto, shows the materials and thicknesses, crystal structures, the values of B of the expression (a) (atomic intervals of the crystals), and the values of lattice unconformity calculated from the expression (a) of the crystallization assisted layers the respective examples and comparative examples.

The value of A of the expression (a) is $4.325 \text{ \AA}$. When the crystal structure has a hexagonal system, the value of B is one of the atomic interval in the a-axial direction and the atomic interval in the c-axial direction, which has the smaller one of the differences between A and these atomic intervals. When the crystal structure has a body-centered cubic lattice system, the value of B is one of the atomic interval in a $\langle 100 \rangle$



direction and the atomic interval in a $\langle 111 \rangle$ direction, which has the smaller one of the differences between A and these atomic intervals. When the crystal structure has a monoclinic system, the value of B is the atomic interval between the closest atoms.

5 First, the reflectivity and recording characteristics (CNR (Carrier to Noise Ratio) and erase ratio) of each of examples 1-1 to 1-4 and comparative examples 1-1 to 1-7 of uninitialized optical disks, thus obtained, were measured. Then, the reflectivity and recording characteristics (CNR and erase ratio) of each example after its initialization were measured.

10 The measurement of the CNR and erase ratio of the disk of each example was performed after recorded once and by overwriting, 100 times. The measurement of the reflectivity and the recording characteristics was performed with the aid of a laser of a wavelength of 640 nm with an objective lens having a NA of 0.6.

15 The initialization was performed by rotating the disk at a linear velocity of 5 m/s and causing a laser beam of a wavelength of 820 nm to scan the same place on the disk ten times or more. The cross-sectional outline of the laser beam was an ellipse having a longer axis of $96\mu\text{m}$ and a shorter axis of $1.5\mu\text{m}$. Thus, a flat area between two adjacent guide
20 grooves was also initialized so as to have the same state as the guide grooves.

The measurement of the disk reflectivity was performed on guide groove-free flat portions of each disk before and after its initialization. The measurement of the disk reflectivity after its initialization was
25 performed after it was confirmed that the disk was sufficiently crystallized in the initialization process such that the reflectivity was saturated.

The measurement of the C/N ratio of each disk was performed



after a single signal whose mark and space lengths were $0.61\mu\text{m}$ was recorded once and after the same signal was overwritten 100 times during the disk rotation at a linear velocity of 6 m/s.

The measurement of the erase ratio was performed by measuring
5 an amount of a decrease in the carrier wave of a signal whose mark and space lengths were $0.61\mu\text{m}$ after a signal whose mark and space lengths were $0.61\mu\text{m}$ was recorded once or overwritten 100 times and further a signal whose mark and space lengths were $2.85\mu\text{m}$ was overwritten once. The results of those measurements are together shown in Table 2 attached
10 hereto.

As will be seen from Table 2, the crystal structures of the materials which compose the crystallization assisted layers of the examples 1-1 to 1-4 are of a face-centered cubic lattice system and the absolute value of the lattice unconformity is not more than 8 %. Thus, satisfactory results were
15 obtained before the initialization; that is, the C/N ratio was 45 dB or more, the erase ratio was 20 dB or more, and the reflectivity was 15 % or more. The difference between before and after the initialization about the reflectivity, C/N ratio and erase ratio was small. There was no large difference between after recorded once and 100 times about C/N ratio and
20 erase ratio. Thus, the necessity for the initialization process can be eliminated.

In comparative examples, the comparative examples 1-1 to 1-7 exhibited no results which satisfied all of the conditions, that is; 45 dB or more of the C/N ratio, 20 dB or more of the erase ratio, and 15 % or more of
25 the reflectivity as the characteristics before the initialization, but exhibited satisfactory results after the initialization. Thus, the initializing step cannot be eliminated. The crystal structures of the materials which



compose the crystallization assisted layers of the comparative examples 1-4 and 1-5 are of a rhombohedral lattice system, but contain Te. Thus, those results were obtained.

The comparative example 1-4 used materials of the crystallization
5 assisted layer which contains Te and Se. Preferably, Se should not be used because it has toxicity and is required to handle with care when the target is produced or the layer is formed. Thus, the composition of the comparative examples 1-4 is not preferable because the necessity for the initialization process cannot be eliminated as well as the producing step is
10 complicated.

Second Embodiment:

Example 2-1:

A phase change optical disk having the same layer structure as the
15 FIG. 1 one was produced as follows.

First, a first dielectric layer 2 having a thickness of about 1400 Å was formed on a polycarbonate substrate 1 having a diameter of 90 mm and a thickness of 0.6 mm with a guide groove for a laser beam by RF sputtering with the aid of a target of ZnS - SiO₂. A Sb layer having a thickness of 50
20 Å was then formed as the crystallization assisted layer 3 on the first dielectric layer 2 by sputtering with the aid of a target of Sb.

A recording layer 4 having a thickness of 200 Å of Ge₂Te₈Sb₂ was formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 having a thickness
25 of 200 Å was then formed on the recording layer 4 by sputtering with the aid of a target of ZnS - SiO₂. A reflective layer 6 having a thickness of 1500 Å was then formed on the second dielectric layer 5 by sputtering with the



aid of a target of an Al alloy. A UV curable resin was then spin-coated on the reflective layer 6 to be cured to thereby form a UV cured resin layer 7.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb and Te ($\text{Ge} : \text{Sb} : \text{Te} = x : y : z$ where $x + y + z = 1$) was 0.22 : 0.22 : 0.56, which satisfied the expressions (1)-(3) simultaneously.

Comparative Example 2-1:

A phase change optical disk was produced in a manner similar to that in which the example 2-1 was produced, excluding that no crystallization assisted layer 3 was formed.

Example 2-2:

A phase change optical disk having the same layer structure as the FIG. 1 one was produced as follows:

A first dielectric layer 2 having a thickness of about 1400 Å was formed on a substrate 1 similar to that of the example 2-1 in a manner similar to that in which the example 2-1 was produced. A Sb layer having a thickness of 65 Å was then formed as the crystallization assisted layer 3 on the first dielectric layer 2 in a manner similar to that in which the example 2-1 was produced. Then, the crystallization assisted layer 3 was irradiated with light (for five minutes at a power of 500 W) from a halogen lamp to further crystallize the crystallization assisted layer 3.

A recording layer 4 of $\text{Ge}_{31}\text{Te}_{57}\text{Sb}_{12}$ having a thickness 225 Å was then formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 of ZnS - SiO_2 having a thickness of 200 Å, and a reflective layer 6 of an Al alloy



having a thickness of 1500 Å were then sequentially formed by sputtering on the recording layer 4 in a manner similar to that in which the example 2-1 was produced. A UV cured resin layer 7 was then formed and set on the reflective layer 6 as in the example 2-1.

5 In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.31 : 0.21 : 0.57$ where $x + y + z = 1$, and satisfied the expressions (1)-(3) simultaneously.

10 Example 2-3:

A phase change optical disk having the same layer structure as the FIG. 1 one was produced as follows:

A first dielectric layer 2 having a thickness of about 1400 Å was formed on a substrate 1 similar to that of the example 2-1 in a manner
15 similar to that in which the example 2-1 was produced. A Bi layer having a thickness of 50 Å was then formed as the crystallization assisted layer 3 on the first dielectric layer 2 by sputtering with the aid of a target of Bi. Then, the crystallization assisted layer 3 was irradiated with light (for five minutes at a power of 500 W) from a halogen lamp to further crystallize the
20 crystallization assisted layer 3.

A recording layer 4 of $\text{Ge}_{23}\text{Te}_{54}\text{Sb}_{23}$ having a thickness of 225 Å was then formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 of $\text{ZnS} - \text{SiO}_2$ having a thickness of 200 Å, and a reflective layer 6 of an Al alloy
25 having a thickness of 1500 Å were then sequentially formed by sputtering on the recording layer 4 in a manner similar to that in which the example 2-1 was produced. A UV cured resin layer 7 was then formed and set on



the reflective layer 6 as in the example 2-1.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.23 : 0.23 : 0.54$ where $x + y + z = 1$, and
5 satisfied the expressions (1)-(3) simultaneously.

Example 2-4:

A phase change optical disk having the same layer structure as the FIG. 1 was produced as follows:

10 A first dielectric layer 2 having a thickness of about 1400 Å was formed on a substrate 1 similar to that of the example 2-1 in a manner similar to that in which the example 2-1 was produced. A Bi layer having a thickness of 50 Å was then formed as the crystallization assisted layer 3 on the first dielectric layer 2 by sputtering with the aid of a target of Bi.

15 A recording layer 4 of $\text{Ge}_{17}\text{Te}_{66}\text{Sb}_{28}$ having a thickness of 225 Å was then formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 of ZnS - SiO_2 having a thickness of 200 Å, and a reflective layer 6 of an Al alloy having a thickness of 1500 Å were then sequentially formed by sputtering
20 on the recording layer 4 in a manner similar to that in which the example 2-1 was produced. A UV cured resin layer 7 was then formed and set on the reflective layer 6 as in the example 2-1.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and
25 Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.17 : 0.28 : 0.55$ where $x + y + z = 1$, and satisfied the expressions (1)-(3) simultaneously.



Example 2-5:

A phase change optical disk having the same layer structure as the FIG. 1 was all produced in a manner similar to that in which the example 2-4 was produced, excluding formation of a layer of $\text{Ge}_{37}\text{Te}_{54}\text{Sb}_9$ having a thickness of 225 Å as the recording layer 4.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.37 : 0.09 : 0.54$ where $x + y + z = 1$, and satisfied the expressions (1)-(3) simultaneously.

10

Example 2-6:

A phase change optical disk having the same layer structure as the FIG. 1 was all produced in a manner similar to that in which the example 2-1 was produced, excluding that the thickness of the first dielectric layer 2 of $\text{ZnS} - \text{SiO}_2$ was 1200 Å; that the thickness of the crystallization assisted layer 3 of Sb layer was 100 Å; that the thickness of the recording layer 4 of a $\text{Ge}_2\text{Te}_5\text{Sb}_2$ alloy was 250 Å; that the thickness of the second dielectric layer 5 of $\text{ZnS} - \text{SiO}_2$ was 150 Å; and that the thickness of the reflective layer 6 of an Al alloy was 500 Å.

20 An X-ray diffraction sample A was produced which comprised respective layers formed on a smooth glass substrate with each of the respective layers being exactly the same in structure as a corresponding one of the layers of the last mentioned disk.

An X-ray diffraction sample B was also produced in a manner similar to that in which the sample A was produced, excluding that the thickness of a Sb layer as the crystallization assisted layer 3 having a thickness of 180 Å was formed on the smooth glass substrate and that no



layers corresponding to the recording layer 4, reflective layer 6 and UV cured resin layer 7 were formed.

Comparative Example 2-2:

5 A phase change optical disk was produced in a manner similar to that in which the example 2-6 was produced, excluding that no layer corresponding to the crystallization assisted layer 3 was formed.

An X-ray diffraction sample C was produced which comprises respective layers formed on a smooth glass substrate with each of the layers
10 being exactly the same in structure as a corresponding one of the layers of the last-mentioned disk.

Example 2-7:

A phase change optical disk having the same layer structure as the
15 FIG. 1 was all produced in a manner similar to that in which the example 2-4 was produced, excluding formation of a layer of $\text{Ge}_{43}\text{Te}_{52}\text{Sb}_4$ having a thickness of 225 Å as the recording layer 4.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and
20 Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.43 (= 43/99) : 0.04 (= 4/99) : 0.53 (= 52/99)$ where $x + y + z = 1$. The value of x was larger than an upper limit of the range of the expression (1) and the value of y was smaller than a lower limit of the range of the expression (2).

25 Example 2-8:

A phase change optical disk having the same layer structure as the
FIG. 1 was all produced in a manner similar to that in which the example



2-4 was produced, excluding formation of a layer of $\text{Ge}_{25}\text{Te}_{40}\text{Sb}_{35}$ having a thickness of 225 Å as the recording layer 4.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.25 : 0.35 : 0.40$ where $x + y + z = 1$. The value of z was smaller than a lower limit of the range of the expression (3).

Example 2-9:

A phase change optical disk having the same layer structure as the FIG. 1 was all produced in a manner similar to that in which the example 2-4 was produced, excluding formation of a layer of $\text{Ge}_{25}\text{Te}_{32}\text{Sb}_{43}$ having a thickness of 225 Å as the recording layer 4.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.25 : 0.43 : 0.32$ where $x + y + z = 1$. The value of z was smaller than a lower limit of the range of the expression (3).

Example 2-10:

A phase change optical disk having the same layer structure as the FIG. 1 was all produced in a manner similar to that in which the example 2-4 was produced, excluding formation of a layer of $\text{Ge}_{20}\text{Te}_{67}\text{Sb}_{13}$ having a thickness of 225 Å as the recording layer 4.

In the recording layer 4 of the phase change optical disk, the ratio x, y, and z of the respective quantities of the component elements Ge, Sb, and Te was $\text{Ge} : \text{Sb} : \text{Te} = x : y : z = 0.20 : 0.13 : 0.67$ where $x + y + z = 1$. The value of z is larger than an upper limit of the range of the expression (3).



Advantageous Effects Produced
by the Crystallization Assisted layer:

The reflectivity and recording characteristics (CNR (Carrier to Noise Ratio) and erase ratio) of each of the optical disks of the examples 2-1 to 2-10 and comparative examples 2-1 and 2-2, thus obtained, were measured before its initialization. They were also measured after initialization.

The initialization was performed by irradiating the respective disks with a laser beam having wavelength of 680 nm, using "MO Disk Bulk Eraser LK101A" manufactured by K.K. Shibasoku. The measurement of the recording characteristics was performed as follows. First, a laser beam having a wavelength of 680 nm was modulated for each sample between an optimal peak power and an optimal bias power. Each optical disk sample which was rotating at 1800 rpm was irradiated with the modulated laser beam to perform initial recording with a first recording signal and the CNR was then measured. Next, the initial recording signal on the sample was overwritten by a recording signal different from the first recording signal, and an erase ratio of the initial recording signal was then measured.

In the samples of the examples 2-1 to 2-3, 2-8, 2-9 and the comparative example 2-1, the first and second recording signal frequencies were 1.08 and 3.89 MHz, respectively. In the samples of the examples 2-4 to 2-7, 2-10 and the comparative example 2-2, the first and second recording signal frequencies were 3.89 and 1.08 MHz, respectively.

The results of measurement of those characteristics are shown in Table 3 attached hereto. In the samples of the examples 2-8 to 2-10, the reflectivity of the initial records on each disk before the initialization was



excessively low and the CNR and erase ratio could not be measured.

As will be seen from Table 3, the difference between before and after the initialization about the reflectivity, CNR and erase ratio of each of the samples of the examples 2-1 to 2-6 is small compared to the samples of the comparative examples 2-1 and 2-2 which include no crystallization assisted layer. It will be seen in each of the samples of the examples 2-1 to 2-6 that its recording layer has been crystallized by the related crystallization assisted layer and that its recording characteristics have been improved. It will be seen that especially in the examples 2-2 and 2-3 where the crystallization assisted layer was crystallized before the recording layer was formed, there is little difference between before and after the initialization about the reflectivity, CNR and erase ratio. Thus, it will be seen that the necessity for initialization process will be eliminated. In the samples of the examples 2-1 to 2-3, the especially preferable results that the CNR and the erase ratio were 53 and 25 dB or more, respectively, were obtained.

By comparison of the examples 2-3 to 2-5 and 2-7 to 2-10 which each comprise the Bi layer having the same thickness of 50 Å as the crystallization assisted layer but which are different in the composition of the recording layer, it will be seen that since the ratio x : y : z indicative of the composition of the recording layer of each of the samples of the examples 2-7 to 2-10 does not satisfy the expressions (1)-(3) simultaneously, the recording layer is difficult to crystallize and that the reflectivity before the initialization is very low compared to the examples 2-3 to 2-5.

25

Results of X-Ray Diffraction:

The orientations of crystals of the crystallization assisted layers



and recording layers of the X-ray diffraction samples A-C, produced as described above, were examined in the X-ray diffraction apparatus manufactured by K.K. Rigaku with a $\text{CuK}\alpha 1$ as the light source.

Since the crystal structure of the crystal phase of the phase change optical disk's recording layer (Ge - Sb - Te type) had a NaCl structure, the crystal structure of the recording layer was given an index representing the NaCl structure. Since Sb or Bi was used as the crystallization assisted layer material, the crystal structure of crystallization assisted layer was given an index representing a rhombohedral structure. Giving an index representing a rhombohedral structure was performed by converting the rhombohedral structure to a hexagonal lattice system, as is generally performed.

FIG. 3 is a chart of diffraction X-ray spectra where a vertical axis represents a diffraction intensity (I) and a horizontal axis a doubled value of a diffraction angle θ obtained from the respective samples A-C.

It was seen that when the sample C was directly subjected to X-ray diffraction, no diffraction peaks were observed and that its recording layer was amorphous. Thus, this sample was heated in an oven at a temperature of 275 °C for 10 minutes. Then, the resulting sample (new sample C) was again subjected to X-ray diffraction to obtain its spectra shown in FIG. 3.

It will be seen from this chart that the sample A (having its crystallization assisted layer provided immediately under the recording layer) exhibits peaks in (111), (222) and (003) at angles of $2\theta = 25.7, 52.9$ and 23.7 degrees, respectively. It will be also seen that the sample B (including only the first dielectric layer and crystallization assisted layer) exhibits peaks in (003) and (006) at angles of $2\theta = 23.7$ and 48.5 degrees,



respectively. It will be further seen that the sample C (having exactly the same structure as the sample A, excluding that no crystallization assisted layer is provided) exhibits peaks in (200), (220) and (222) at angles of $2\theta = 29.7, 42.7$ and 52.9 degrees, respectively.

5 Since the peaks in (111) (222) of the GeSbTe crystal appeared in the X-ray diffraction pattern of the sample A, it will be seen that the recording layer of the sample A was crystallized after its formation.

 The orientation of Sb atoms which compose the crystallization assisted layer can be known from the chart of the sample B. More
10 particularly, it will be known that powder Sb has high peak intensities in (012) and (104) whereas the Sb atoms of the Sb film formed on the first dielectric layer are oriented in (003) and (006) planes.

 By comparing the samples A and C, it will be known that the orientations of their crystals differ depending on whether they include a
15 crystallization assisted layer even when they have recording layers of the same composition and thickness. In this particular example, the presence of the crystallization assisted layer causes peaks in the (200) and (220) to disappear.

20 Third Embodiment:

Example 3-1:

 A phase change optical disk having the same layer structure as the FIG. 1 was produced as follows.

 First, a first dielectric layer 2 having a thickness of about 100 nm
25 was formed on a polycarbonate substrate 1 having a diameter of 90 mm and a thickness of 0.6 mm with a guide groove for a laser beam by RF sputtering with the aid of a target of $\text{ZnS} - \text{SiO}_2$. A Bi film having a thickness of 3 nm



was then formed as the crystallization assisted layer 3 on the first dielectric layer 2 by sputtering with the aid of a target of Bi.

A recording layer 4 of $\text{Ge}_{20.3}\text{Te}_{66.7}\text{Sb}_{23.0}$ having a thickness of 20 nm was formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 having a thickness of 12 nm was then formed on the recording layer 4 by sputtering with the aid of a target of $\text{ZnS} - \text{SiO}_2$. A reflective layer 6 having a thickness of 70 nm was then formed on the second dielectric layer 5 by sputtering with the aid of a target of an Al alloy. A UV curable resin was then spin-coated on the reflective layer 6 to be set to thereby form a UV cured resin layer 7.

The recording characteristics of the optical disk, thus obtained, were measured without initialization as follows: First, a laser beam having a wavelength of 680 nm was modulated between a peak power of 12 mW and a bias power of 5 mW. The sample of the optical disk was rotated at a linear velocity of 6 m/s while being irradiated with the modulated laser beam to perform its initial recording and its CNR was measured. The recording frequency was 5 MHz and the read power was 1 mW. The CNRs of records on the optical disks were measured after the two, three and four overwriting or recording operations.

The CNRs measured after the one, two, three and four overwriting or recording operations were 49.7, 48.8, 48.7 and 48.3 dB, respectively. It will be seen that the CNR decreased by 0.9 dB after the two recording operations.

The state of the sample present after the formation of its crystallization assisted layer 3 and before the formation of its recording layer 4 was observed with the aid of a high resolution SEM (UHRSEM



which was HITACHI S-5000 produced by Hitachi Seisakusho) at an acceleration voltage of 15 kV. A photograph of its reflective electron image is shown in FIG. 4.

It will be seen from this photograph that the crystallization assisted layer 3 is in the form of a discontinuous island-like film in which most of the islands have a length of 8-40 nm with the longest one being 70 nm. The interval between the islands was 20 nm at a maximum. Measurement by an atomic force microscope (AFM: Topo Metrix TMX-2000) in a contact mode clarified that an average thickness of the islands was 2.04 nm.

Example 3-2:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 3-1 was produced, excluding that a Bi layer having a thickness of about 1.5 nm was formed as the crystallization assisted layer 3.

The recording characteristics of the optical disk, thus obtained, were measured without initialization as in the example 3-1. The CNRs measured after the one, two, three and four overwriting or recording operations were 51.7, 52.1, 51.8 and 52.0 dB, respectively. It will be seen that no decrease occurred in the CNR after the two recording operations or more.

The state of the sample present after the formation of its crystallization assisted layer 3 and before the formation of its recording layer 4 was observed with the aid of the high resolution SEM in a manner similar to that performed in the Example 3-1. A photograph of its reflective electron image is shown in FIG. 5.

It will be seen from this photograph that the crystallization assisted



layer 3 is in the form of a discontinuous island-like film in which most of the islands have a length of 4-20 nm with the longest one being 25 nm and that the interval between the islands was 13 nm at a maximum. Measurement similar to that of the Example 3-1 in a contact mode clarified that an
5 average thickness of the islands was 1.73 nm.

Example 3-3:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 3-1
10 was produced, excluding that a Bi layer having a thickness of about 0.65 nm was formed as the crystallization assisted layer 3.

The measurement of the recording characteristics of the uninitialized optical disk, thus obtained, clarified that its CNR was 50.8 dB after recorded first time, 51.1 dB after recorded twice, 51.3 dB after
15 recorded three times, and 51.3 dB after recorded four times. That is, no decrease in the CNR occurred after recorded twice or more.

Example 3-4:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 3-1
20 was produced, excluding that a Bi layer having a thickness of about 0.3 nm was formed as the crystallization assisted layer 3.

The measurement of the recording characteristics of the uninitialized optical disk, thus obtained, clarified that its CNR was 54.0 dB
25 after recorded first time, 54.1 dB after recorded twice, 53.9 dB after recorded three times, and 54.2 dB after recorded four times. That is, no decrease in the CNR occurred after recorded twice or more.



Although the state of this sample present after the formation of its crystallization assisted layer 3 and before the formation of the recording layer 4 was observed with the aid of the high-resolution SEM as in the example 3-1, no island-like objects could not be recognized. Since the
5 resolution of this device was 0.7 nm, it was considered that the size of the island-like objects as the crystallization assisted layer 3 was less than 0.7 nm. An average thickness of the island-like objects was measured as 1.01 nm in a manner similar to that used in the example 3-1.

10 Example 3-5:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 3-1 was produced, excluding that the first dielectric layer 2 had a thickness of 115 nm, that the crystallization assisted layer 3 of Bi had a thickness of 1.5
15 nm, that the recording layer 4 comprised a $\text{Ge}_{22.1}\text{Te}_{56.4}\text{Sb}_{21.5}$ layer having a thickness of 22 nm, and that the reflective layer 6 of an Al alloy had a thickness of 50 nm.

The measurement of the recording characteristics of the uninitialized optical disk, thus obtained, performed in a manner similar to
20 that performed for the example 3-1, clarified that its CNR was 53.2 dB after recorded first time, 53.3 dB after recorded twice, 53.2 dB after recorded three times, and 53.2 dB after recorded four times. That is, no decrease in the CNR occurred after recorded twice or more.

25 Example 3-6:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 3-1



was produced, excluding that a Bi layer having a thickness of about 5.0 nm was formed as the crystallization assisted layer 3.

The measurement of the recording characteristics of the uninitialized optical disk, thus obtained, clarified that its CNR was 50.7 dB
5 after recorded first time, 47.5 dB after recorded twice, 47.9 dB after recorded three times, and 47.2 dB after recorded four times. That is, a decrease 3.2 dB in the CNR occurred after recorded twice.

When the state of this sample present after the formation of its crystallization assisted layer 3 and before the formation of the recording
10 layer 4 was observed with the aid of the high-resolution SEM as in the example 3-1, it was found that the crystallization assisted layer 3 was formed in a continuous film.

Advantageous Effects Produced by
15 the Crystallization Assisted layer:

It will be seen from the results of the examples 3-1 to 3-6 that when the crystallization assisted layer 3 of Bi was formed as a discontinuous island-like film, the necessity for the initialization process for crystallizing the recording layer formed on the crystallization assisted layer 3 was
20 eliminated, and that the CNRs measured after the two or more recording operations by overwriting were not lower than that measured in the initial recording operation.

It will also be known that the crystallization assisted layer 3 of Bi was formed so as to have a thickness of less than 3 nm to thereby provide a
25 discontinuous island-like film by general sputtering (when the sputtering conditions were: for example, the sputtering gas was Ar, the sputtering pressure was 0.5 Pa, and the power consumed was DC 10 W).



When the length of the island-like crystallization assisted layer is less than 100 nm, its discontinuous island-like film is easily obtained by general sputtering which lacks patterning, for example. For the island-like crystallization assisted layer 3, preferably, its length is less than 80 nm, and the maximum gap between two adjacent islands is less than 50 nm.

Crystal Structures of the Recording Layer and
Crystallization Assisted layer:

An X-ray diffraction sample A was produced having the same layer structure as the example 3-5, excluding that no UV cured resin layer 7 was formed on its cover glass.

An X-ray diffraction sample B was produced having the same layer structure as the sample A, excluding that no crystallization assisted layer 3 was formed.

An X-ray diffraction sample C was produced having the same layer structure as the sample A, excluding that the recording layer 4 had a composition of $\text{Ge}_{18.7}\text{Sb}_{26.9}\text{Te}_{54.4}$.

An X-ray diffraction sample D was produced having the same layer structure as the sample A, excluding that the recording layer 4 had a composition of $\text{Ge}_{26.4}\text{Sb}_{17.6}\text{Te}_{56.0}$.

An X-ray diffraction sample E was produced having the same layer structure as the sample A, excluding that the recording layer 4 had a composition of $\text{Ge}_{30.8}\text{Sb}_{12.9}\text{Te}_{56.3}$.

An X-ray diffraction sample F was produced having the same layer structure as the sample A, excluding that the recording layer 4 had a composition of $\text{Ge}_{36.5}\text{Sb}_{9.3}\text{Te}_{54.2}$.

An X-ray diffraction sample G was produced having the same layer



structure as the sample A, excluding that the recording layer 4 had a composition of $\text{Ge}_{49.7}\text{Te}_{51.3}$.

Since the samples A, C-F each had a discontinuous island-like film of Bi as the crystallization assisted layer 3, its recording layer 4 was
5 crystallized by the crystallization assisted layer 3. Since the sample B lacked the crystallization assisted layer 3, its recording layer 4 had not been crystallized. Thus, the sample B was heated at 275°C for 10 minutes in an oven to crystallize the recording layer. The sample B, thus obtained, was subjected to X-ray diffraction. The sample G had the crystallization
10 assisted layer 3, but its recording layer 4 was not crystallized immediately after its formation.

Those X-ray diffraction samples A-G were each put on the X-ray diffraction apparatus to examine the orientation of the crystals of the crystallization assisted layer and the recording layer. In this case, the X-
15 ray diffraction apparatus produced by K.K. Rigaku, and $\text{CuK}\alpha 1$ as the light source were used. Since the crystal structure of the phase change optical disk recording layer (Ge - Sb - Te) in its crystalline phase was a NaCl structure, the crystal structure of the recording layer was regarded as having the NaCl structure and given a corresponding index.

20 FIG. 6 is a chart of the diffraction X-ray spectra of the respective samples A-G where the vertical axis represents a diffraction intensity (I) and the horizontal axis a doubled value of the diffraction angle θ .

It will be seen from this chart that the samples A-F exhibit peaks in (200) at $2\theta = 29.7$ degrees whereas the sample G has no such peak. The
25 intensity of the peak tends to decrease as the Ge content of the recording layer increases and its Sb content decreases.

The sample B has peaks in (220) and (222) at $2\theta = 42.7$ and 52.9



degrees, respectively, in addition to the peak in (200) at $2\theta = 29.7$. This is considered to be due to the sample B lacking a crystallization assisted layer 3. It will be seen that by forming a discontinuous island-like film of Bi as the crystallization assisted layer 3 and further forming a recording layer 4 of Ge - Sb - Te on the crystallization assisted layer 3, the crystals of the recording layer 4 is easily oriented strongly in a surface (200) alone.

Fourth Embodiment:

Example 4-1:

10 A phase change optical disk having the same layer structure as the FIG. 1 was produced as follows.

First, a first dielectric layer 2 having a thickness of about 100 nm was formed on a polycarbonate substrate 1 having a diameter of 90 mm and a thickness of 0.6 mm with a guide groove for a laser beam and irregularities for address signals by RF sputtering with the aid of a target of ZnS - SiO₂. A Bi layer having a thickness of 1.5 nm was then formed as the crystallization assisted layer 3 on the first dielectric layer 2.

A recording layer 4 of Ge₂₃Te₅₄Sb₂₃ having a thickness of 23 nm was formed on the crystallization assisted layer 3. The temperature of the substrate 1 was 35 °C before the formation of the recording layer 4 during which the temperature of the substrate 1 did not rise. A second dielectric layer 5 of ZnS - SiO₂ having a thickness of 20 nm and a reflective layer 6 of Al alloy having a thickness of 150 nm were then formed sequentially on the recording layer 4 by sputtering. A UV curable resin was then spin-coated on the reflective layer 6 and set to thereby form a UV cured resin layer 7.

Example 4 - 2:



A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 4-1 was produced, excluding that after the crystallization assisted layer 3 was formed, the substrate 1 was irradiated with light from a halogen lamp to heat the substrate up to 75°C, and that the recording layer 4 was then formed immediately. The temperature of the substrate was maintained in a range of from 45 through 75°C inclusive during the formation of the recording layer 4.

10 Example 4 - 3:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 4-1 was produced, excluding that after the crystallization assisted layer 3 was formed, the substrate 1 was heated up to 55°C with a heater. The temperature of the substrate was maintained in a range of from 45 through 55°C inclusive during the formation of the recording layer 4.

Example 4 - 4:

A phase change optical disk having the same layer structure as the FIG. 1 was produced, excluding that immediately after the disk was formed up to the crystallization assisted layer 3 in a manner similar to that in which the example 4 - 1 was produced, a recording layer 4 of $\text{Ge}_{21}\text{Te}_{64}\text{Sb}_{25}$ alloy having a thickness of 23 nm was formed, that the temperature of the substrate was 45°C immediately after the formation of the crystallization assisted layer 3, and that the temperature of the substrate was maintained 45°C during the formation of the recording layer 4. The other conditions were all identical to those of the example 4 - 1.



Example 4 - 5:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 4-1 was produced, excluding that after the formation of the crystallization assisted layer 3 the substrate 1 was irradiated with light from a halogen lamp to heat the substrate up to 115°C, and that a recording layer 4 of $\text{Ge}_{23}\text{Te}_{53}\text{Sb}_{24}$ alloy having a thickness of 23 nm was then formed immediately.

10 Effects Produced by the Heating of the Substrate
when the Recording Layer was Formed:

The reflectivity and recording characteristics (CNR and erase ratio) of each of the optical disks of the examples 4 - 1 to 4 - 5, thus obtained, before the initialization were measured.

15 The reflectivity of each optical disk was measured, using a laser having a wavelength of 680 nm immediately after the disk was obtained as such and after its accelerating test was performed (in which the disk was held within a tank of a humidity of 90 % at a temperature of 80°C for 300 hours).

20 The recording characteristics of each disk were measured as follows. First, the laser beam having a wavelength of 680 nm was modulated between an optimal peak power and an optimal bias power for each sample. By irradiating the optical disk sample which was rotating at 1800 rpm with the modulated laser beam, it was subjected to initial recording (its
25 recording frequency was 1.08 MHz), and then its CNR was measured.

Then, the initial recording signal recorded on the disk was overwritten by a recording signal (having a frequency of 3.89 MHz)



different from the initial recording signal (having a frequency of 1.08 MHz), and then the erase ratio of the initial recording signal was measured.

Those results are together shown in Table 4, attached hereto, on which numerical values appearing on the right of respective arrows in the
5 reflectivity column are obtained immediately after the acceleration test. The optical disk of the example 4 - 5 was measured by the measuring device, but address signals could not be read well in many places, and cannot be used.

As will be seen from Table 4, the reflectivities of the optical disks of
10 the examples 4 - 2 to 4 - 4 whose recording layers were formed at a substrate temperature in a range of from 45 through 110°C inclusive were not substantially changed by the acceleration test. In comparative examples, the optical disks of the example 4 - 1 whose recording layer was formed at a substrate temperature of 35°C was insufficiently crystallized.
15 Thus, changes in its reflectivity due to the acceleration test were found. The erase ratio of the optical disks of the examples 4 - 2 to 4 - 4 were satisfactory compared to the optical disk of the example 4 - 1.

As will be known from the above, in this embodiment, when the optical disks whose recording layer were formed at a substrate temperature
20 in a range of from 45°C through a substrate deforming temperature inclusive were used without the initialization process, those optical disks exhibited stabilized characteristics compared to those disks whose recording layers were formed at a substrate temperature of 35°C. In many cases, however, even when the recording layers were formed at a substrate
25 temperature of less than 45°C, the recording layers were sufficiently crystallized to thereby provide stabilized characteristics.



Fifth Embodiment:

Example 5 - 1:

A phase change optical disk having the same layer structure as the FIG. 1 was produced as follows.

5 First, a first dielectric layer 2 having a thickness of about 80 nm was formed on a polycarbonate substrate 1 having a diameter of 120 mm and a thickness of 0.6 mm with a guide groove for a laser beam by RF sputtering with the aid of a target of ZnS - SiO₂. A Bi layer having a thickness of 1.5 nm was then formed as the crystallization assisted layer 3
10 on the first dielectric layer 2 by sputtering with the aid of a target of Bi.

A recording layer 4 of Ge₂₃Te₅₄Sb₂₃ having a thickness of 22 nm was formed on the crystallization assisted layer 3 by sputtering with the aid of a target of a Ge - Te - Sb alloy. A second dielectric layer 5 having a thickness of 12 nm was then formed on the recording layer 4 by sputtering with the
15 aid of a target of ZnS - SiO₂. A reflective layer 6 having a thickness of 100 nm was then formed on the second dielectric layer 5 by sputtering with the aid of a target of an Al alloy. A UV curable resin was then spin-coated on the reflective layer 6 and set to thereby form a UV cured resin layer 7.

During the formation of the recording layer 4, the temperature of
20 the substrate was maintained at 85°C. The first dielectric layer 2, crystallization assisted layer 3, recording layer 4, second dielectric layer 5 and reflective layer 6 were all formed within an atmosphere of an argon gas.

25 **Example 5 - 2:**

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 1



was produced, excluding that the crystallization assisted layer 3 was formed within an atmosphere of an argon gas plus a nitrogen gas of 4 volume %.

5 Example 5 - 3:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 1 was produced, excluding that the crystallization assisted layer 3 was formed within an atmosphere of an argon gas plus a nitrogen gas of 8
10 volume %,.

Example 5 - 4:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 -
15 1 was produced, excluding that the crystallization assisted layer 3 was formed within an atmosphere of an argon gas plus a nitrogen gas of 16 volume %.

Example 5 - 5:

20 A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 1 was produced, excluding that the crystallization assisted layer 3 was formed within an atmosphere of an argon gas plus a nitrogen gas of 24
25 volume %.

Example 5 - 6:

A phase change optical disk having the same layer structure as the



FIG. 1 was produced in a manner similar to that in which the example 5 - 1 was produced, excluding that the crystallization assisted layer 3 was formed within an atmosphere of an argon gas plus a nitrogen gas of 32 volume %.

5

Effects Produced by Addition of Nitrogen
to the Crystallization Assisted layer:

The reflectivities and recording characteristics of each of the optical disks of the examples 5 - 1 to 5 - 6 were measured before its initialization.
10 First, the laser beam having a wavelength of 680 nm was modulated between a peak power of 11 mW and a bias power of 5 mW depending on the recording signal. By irradiating the optical disk sample rotating at a linear velocity of 6 m/s with the modulated laser beam, the initial recording was performed. The resulting records on the disk sample were read and
15 its reflectivity and jitter were then measured.

Next, the disk was overwritten by a recording signal, which produced records identical to the initial ones, 10, 1000, 20000, 30000, 50000, and 100000 times, the respective records were read, and then the corresponding jitters were measured. The results of those measurements
20 are together shown on Table 5 attached hereto.

As will be seen from Table 5, the optical disks of the examples 5 - 2 to 5 - 6 whose crystallization assisted layers were formed within the atmosphere containing a nitrogen gas were low in jitter value compared to the optical disk of the example 5 - 1 example whose crystallization assisted
25 layer was formed within a nitrogen gas-free atmosphere, when the overwriting was performed repeatedly 20, 000 times or more. The optical disks of the examples 5 - 2 to 5 - 6 are excellent in overwrite cyclability



compared to the optical disk of the example 5 - 1.

The crystallization assisted layer formed within the atmosphere containing the nitrogen gas contains nitrogen, which is considered to improve the overwrite cyclability.

5 The reason why the overwrite cyclability of the disks are improved due to nitrogen being contained in the crystallization assisted layer can be presumed as follows. The nitrogen contained in the crystallization assisted layer gradually exudes out into the recording layer as the recording/erasing operations are repeated to change the recording layer
10 present in the crystal state to a finer-crystal layer, to increase the viscosity of the recording layer present in the crystal state and to increase the crystallizing temperature of the recording layer. Especially, it is considered that nitrides which will be formed on the interface due to the changing of the recording layer to the finer-crystal layer restrains the
15 recording layer from partially thinning due to movement of materials of the recording layer (flow of the materials of the recording layer in its direction of rotation during melting).

As a quantity of nitrogen contained in the crystallization assisted layer increases, the overwrite cyclability improve. If the quantity of
20 nitrogen increases excessively, the crystallization becomes insufficient to thereby decrease the reflectivity. It is to be noted that since 15.5 and 15.2 % of the reflectivities of the examples 5 - 5 and 5 - 6 are in an usable range, there are no problems. The optical disks of the examples 5 - 5 and 5 - 6 where the content of a nitrogen gas in the layer forming atmosphere was
25 high have high jitter values compared to other optical disks after the recording was performed 1,000 times, but are less than 15 %, which is considered as not raising any questions practically.



Joint MORIS (Magneto-Optical Recording International Symposium/ISOM (International Symposium on Optical Memory) '97 Technical Digest p.292 reported that addition of nitrogen to the recording layer greatly improved the overwrite cyclability.

5 The present invention is intended to eliminate the necessity for the initialization process by providing a crystallization assisted layer on an optical information recording medium to crystallize the recording layer when the same is formed. When the recording layer of the optical information recording medium having such crystallization assisted layer
10 contains nitrogen, the recording layer cannot be crystallized when formed.

Actually, a phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 1 was produced, excluding that the recording layer 4 was formed within an atmosphere which contained an argon gas plus a nitrogen
15 gas of 4 volume %. The reflectivity of the disk measured before its initialization was 4.3 %, and the recording layer was not be crystallized.

As described above, in an optical information recording medium having a crystallization assisted layer, the necessity for the initialization process is eliminated and the overwrite cyclability are improved by adding
20 nitrogen to the crystallization assisted layer without adding nitrogen to the recording layer.

Sixth Embodiment:

Example 6 - 1:

25 A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 4 of the fifth embodiment was produced, excluding that the first dielectric



layer 2 was formed within an atmosphere which contained an argon gas and an oxygen gas of 0.1 volume %.

Example 6 - 2:

5 A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 4 of the fifth embodiment was produced, excluding that the first dielectric layer 2 was formed within an atmosphere which contained an argon gas and a nitrogen gas of 0.5 volume %.

10

Example 6 - 3:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 4 of the fifth embodiment was produced, excluding that the second dielectric layer 5 was formed within an atmosphere which contained an argon gas and an oxygen gas of 0.1 volume %.

15

Example 6 - 4:

A phase change optical disk having the same layer structure as the FIG. 1 was produced in a manner similar to that in which the example 5 - 4 of the fifth embodiment was produced, excluding that the second dielectric layer 5 was formed within an atmosphere which contained an argon gas and a nitrogen gas of 0.5 volume %.

20

25 Effects Produced by Addition of Oxygen or Nitrogen to the Dielectric Layer:

The reflectivities and recording characteristics of the respective



optical disks of the examples 6 - 1 to 6 - 4 before their initialization were measured in a manner similar to that used for the fifth embodiment. The results of those measurements are together shown in Table 6, attached hereto, which also contains the results of the measurement of the example 5 - 4 of the fifth embodiment for the sake of comparison.

As will be seen from Table 6, attached hereto, the jitter values of the optical disks of the examples 6 - 1 to 6 - 4 which contain nitrogen in their crystallization assisted layer 3 as well as oxygen or nitrogen in their first or second dielectric layer are restricted to small ones compared to the optical disk of the example 5 - 4 which contains only nitrogen in its crystallization assisted layer 3.

It will be seen that especially when the number of the overwriting is large (for example, after recorded 100, 000 times), the difference in jitter between each of the disks of the examples 6 - 1 to 6 - 4 and a respective one of the optical disk of the example 5 - 4 is large, and that the overwrite cyclability of the disks are further improved by incorporating oxygen or nitrogen into the first or second dielectric layer.

INDUSTRIAL APPLICABILITY

As described above, in an optical information recording medium obtained by the inventive making method, the recording layer becomes crystallized immediately after its formation due to the existence of the crystallization assisted layer of predetermined materials. Thus, the necessity for the initialization process required in the past is eliminated. As a result, the process for making the optical information recording medium is simplified and the cost is reduced.



Especially, by forming the crystallisation assisted layer in the form of a discontinuous island-like film of materials which comprise bismuth (Bi) and/or bismuth (Bi) compounds, an optical information recording medium having stabilised recording characteristics is obtained.

An optical information recording medium excellent in overwrite cyclability is obtained which eliminates the necessity for initialisation process by incorporating nitrogen into the crystallisation assisted layer.

The term "comprises", and grammatical variations thereof such as "comprising" when used in the description and claims does not preclude the presence of additional features, integers, steps or components, or groups thereof.



Table 1

	material & thickness	crystal structure	B-value	lattice unconformity
Example 1-1	PbTe 10Å	face-centered cubic lattice system NaCl type	4.5637 Å <110> direction	5.519%
Example 1-2	PbTe 50 Å	as above	as above	as above
Example 1-3	Bi ₂ Te ₃ 10Å	face-centered cubic lattice system Bi ₂ Te ₃ type	4.2931 Å <110> direction	-0.738%
Example 1-4	Bi ₂ Te ₃ 50Å	as above	as above	as above
Comp.Ex. 1-1	no crystallization assisted layer	-----	-----	-----
Comp.Ex. 1-2	W 10Å	body-centered cubic lattice system, body-centered cubic lattice	3.1653 Å <100> direction	-26.8%
Comp.Ex. 1-3	Te 30Å	hexagonal system selenium type	4.4579 Å a-axial direction	3.07%
Comp.Ex. 1-4	Sb ₂ TeSe ₂ 30Å	rhombohedral lattice system	4.121 Å	-4.717%
Comp.Ex. 1-5	Sb ₂ Te ₃ 30Å	rhombohedral lattice system	4.2463 Å	-1.809%
Comp.Ex. 1-6	Ag ₂ Te 30Å	monoclinic system	4.48 Å	3.58%
Comp.Ex. 1-7	CrTe 30Å	hexagonal system	3.98 Å	-9.13%



Table 2

	initialization	reflectivity (%)	CNR (dB) once /100times	erase ratio(dB) once /100 times
Example 1-1	before	18	50/51	22/24
	after	20	52/52	23/24
Example 1-2	before	21	53/53	27/30
	after	22	53/53	31/31
Example 1-3	before	18	49/50	28/30
	after	21	52/52	30/30
Example 1-4	before	21	50/50	29/29
	after	22	53/53	30/30
Comp.Ex. 1-1	before	5	36/49	7/29
	after	20	53/53	30/30
Comp.Ex. 1-2	before	6	33/49	7/28
	after	22	50/50	29/29
Comp.Ex. 1-3	before	9	33/50	8/30
	after	21	51/52	30/30
Comp.Ex. 1-4	before	12	39/42	6/23
	after	21	50/43	22/23
Comp.Ex. 1-5	before	9	50/51	20/20
	after	18	50/51	21/20
Comp.Ex. 1-6	before	8	38/46	6/22
	after	19	50/51	20/21
Comp.Ex. 1-7	before	7	32/42	9/18
	after	18	48/47	18/17



Table 3

	initialization	reflectivity	CNR(dB)	erase ratio (dB)
Example 2-1	before	19	53	32
	after	22	54	34
Comp.Ex. 2-1	before	6	35	5
	after	21	53	35
Example 2-2	before	20	53	27
	after	21	54	27
Example 2-3	before	20	54	32
	after	22	54	33
Example 2-4	before	22	49	25
	after	23	53	27
Example 2-5	before	12	46	20
	after	18	50	23
Example 2-6	before	17	52	25
	after	18	56	27
Comp.Ex. 2-2	before	5	45	17
	after	17	53	25
Example 2-7	before	6	40	12
	after	17	47	17
Example 2-8	before	4	unmeasurable	unmeasurable
	after	15	unmeasured	unmeasured
Example 2-9	before	4	unmeasurable	unmeasurable
	after	13	unmeasured	unmeasured
Example 2-10	before	4	unmeasurable	unmeasurable
	after	12	unmeasured	unmeasured



Table 4

	reflectivity (%)	CNR (dB)	erase ratio (dB)
example 4 - 1	12 → 18	52	29
example 4 - 2	19 → 20	53	32
example 4 - 3	20 → 21	53	33
example 4 - 4	20 → 21	54	32

Table 5

	addition of N ₂ in film formation	reflectivity (%)	jitter (%)						
			after recorded						
			once	10 times	1,000 times	20,000 times	30,000 times	50,000 times	100,000 times
example 5 - 1	none	18.3	5.6	6.9	7.8	12.2	18.5	28.8	28.6
example 5 - 2	4 vol %	18.7	5.7	7.2	7.8	11.1	14.5	23.2	28.1
example 5 - 3	8 vol %	19.2	5.8	7.4	7.8	9.2	12.1	18.7	22.7
example 5 - 4	16 vol %	18.6	6.3	7.5	8.3	8.8	11.1	11.9	16.9
example 5 - 5	24 vol %	15.5	7.5	7.5	10.0	8.5	9.1	10.1	14.5
example 5 - 6	32 vol %	15.2	7.5	7.7	10.6	8.9	9.3	10.2	13.2



Table 6

	addition of N ₂ or O ₂ dielectric layer formation	reflectivity (%)	jitter(%)						
			after recorded						
			once	10 times	1,000 times	20,000 times	30,000 times	50,000 times	100,000 times
example 6 - 1	1st dielectric layer O ₂ 0.1 vol %	16.6	6.1	6.8	8.0	9.6	10.0	10.2	12.4
example 6 - 2	1st dielectric layer N ₂ 0.5 vol %	17.2	6.0	7.1	7.8	9.0	10.3	11.3	12.9
example 6 - 3	2nd dielectric layer O ₂ 0.1 vol %	17.6	6.2	7.6	7.9	8.8	11.2	11.3	13.2
example 6 - 4	1st dielectric layer N ₂ 0.5 vol %	16.9	6.4	7.7	7.9	9.0	10.2	11.9	12.3
example 5 - 4	no addition to dielectric layer	18.6	6.3	7.5	8.3	8.8	11.1	11.9	16.9



CLAIMS

1. A method for producing an optical information recording medium which has on one side of a substrate a recording layer of materials whose
5 main components comprise germanium (Ge), antimony (Sb) and tellurium (Te), comprising the steps of:

forming a crystallization assisted layer of materials having a crystal structure of a face-centered cubic lattice system on one side of the substrate; and

- 10 forming a recording layer directly over the crystallization assisted layer.

2. A method for producing an optical information recording medium which has on one side of a substrate a recording layer of materials whose
15 main components comprise germanium (Ge), antimony (Sb) and tellurium (Te), comprising the steps of:

forming on one side of the substrate a crystallization assisted layer of a tellurium (Te)-free material having a crystal structure of a rhombohedral lattice system; and

- 20 forming a recording layer directly over the crystallization assisted layer.

3. The method according to claim 1 or 2, wherein an absolute value of lattice unconformity between the crystal structure of the crystallization
25 assisted layer and that of the recording layer is not more than 8%.



4. A method for producing an optical information recording medium

which has on one side of a substrate a recording layer of materials whose main components comprise germanium (Ge), antimony (Sb) and tellurium (Te), comprising the steps of:

forming on one side of the substrate a crystallization assisted layer
5 of materials which contain at least one of antimony (Sb), bismuth (Bi), an antimony (Sb) compound, and a bismuth (Bi) compound; and

forming a recording layer directly over the crystallization assisted layer.

10 5. A method for producing an optical information recording medium which has on one side of a substrate a recording layer of materials whose main components comprise germanium (Ge), antimony (Sb) and tellurium (Te), comprising the steps of:

forming on one side of the substrate a crystallization assisted layer
15 in the form of a discontinuous island-like film of materials which contain Bi (bismuth) and/or a bismuth (Bi) compound; and

forming a recording layer directly over the crystallization assisted layer.

20 6. The method according to any one of claims 1-5, wherein the recording layer is formed at a substrate temperature in a range of from 45°C through a temperature inclusive above which temperature the substrate will be deformed.

25 7. The method according to any one of claims 1-5, wherein the substrate is produced of polycarbonate; and wherein the recording layer is formed at a substrate temperature in a range of from 45°C through 110°C



inclusive.

8. The method according to any one of claims 1-7, wherein the formation of the crystallization assisted layer is performed within a film
5 forming atmosphere to which a nitrogen gas is added.

9. The method according to any one of claims 1-8, further comprising the step of:

forming a first dielectric layer between the substrate and the
10 crystallization assisted layer; and/or

forming a second dielectric layer on an opposite side of the recording layer from the crystallization assisted layer,

wherein the formation of the first and/or second dielectric layer is performed within a film forming atmosphere to which a nitrogen gas and/or
15 an oxygen gas is added.

10. An optical information recording medium with a recording layer formed on one side of a substrate, the recording layer comprising materials whose main components are germanium (Ge), antimony (Sb) and
20 tellurium (Te),

the medium being produced by any one of the methods according to claims 1-9,

wherein the recording layer is formed in a crystalline state; and comprising:

25 a crystallization assisted layer formed in contact with the substrate side surface of the recording layer.



11. The optical information recording medium according to claim 10, wherein a ratio x, y, and z of the elements (Ge, Sb, Te) of the main components of the recording layer (Ge: Sb: Te = x: y: z where $x + y + z = 1$) satisfies the following expressions (1)-(3) simultaneously:

5

$$0.1 \leq x \leq 0.4 \quad \dots (1)$$

$$0.08 \leq y \quad \dots (2)$$

$$0.45 \leq z \leq 0.65 \quad \dots (3)$$



FIG. 1

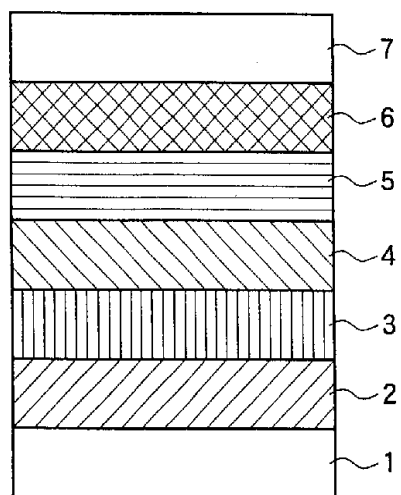


FIG. 2

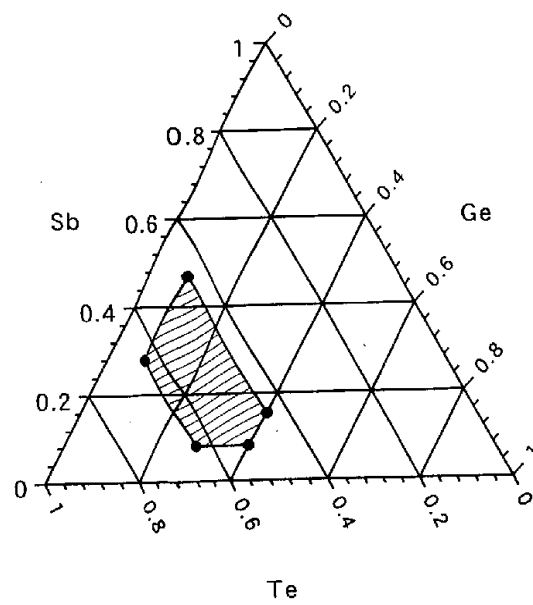


FIG. 3

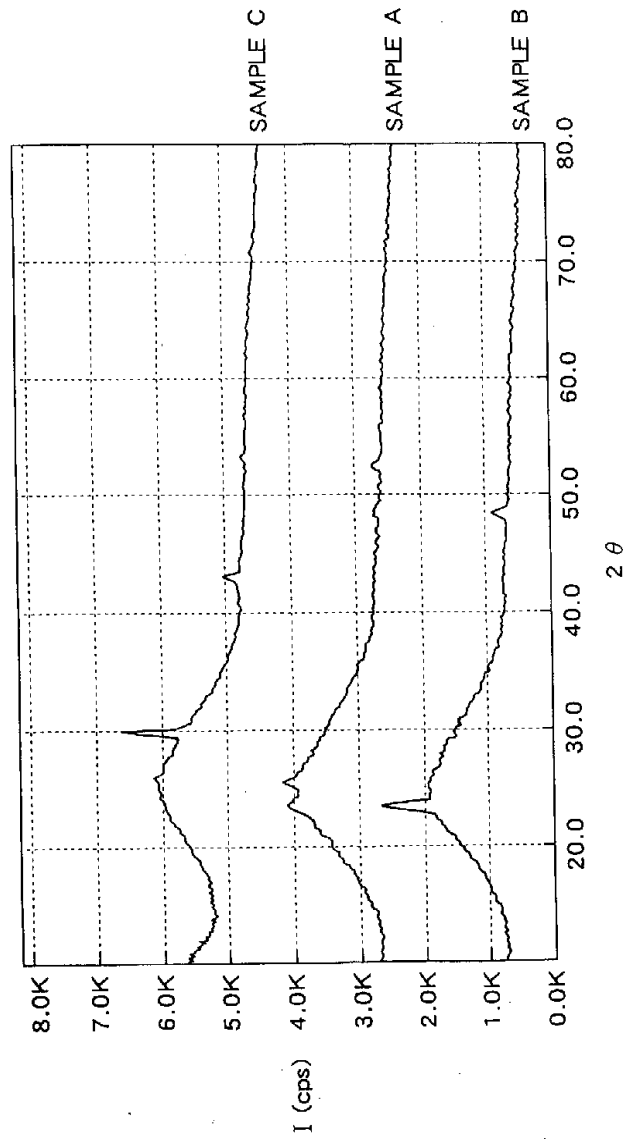


FIG. 4

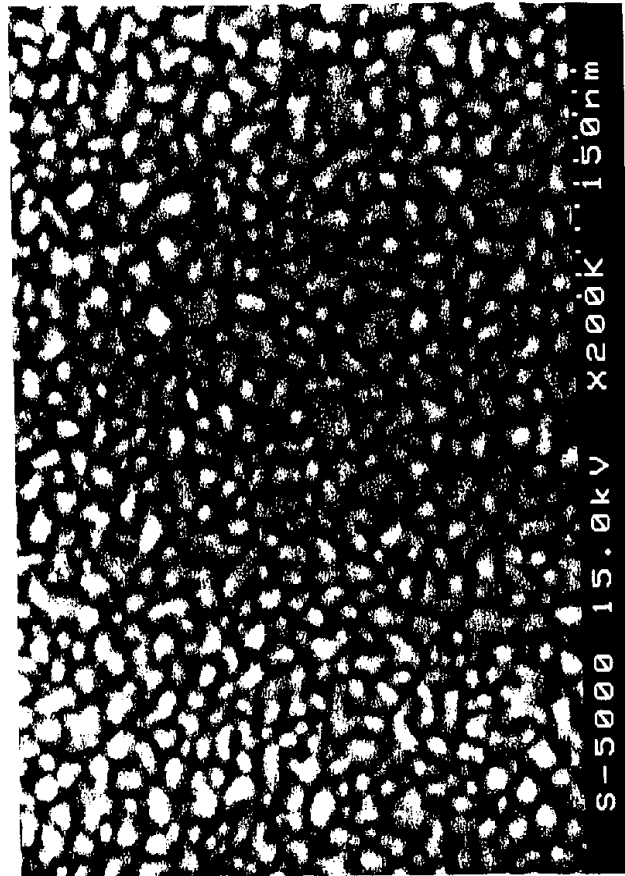


FIG. 5

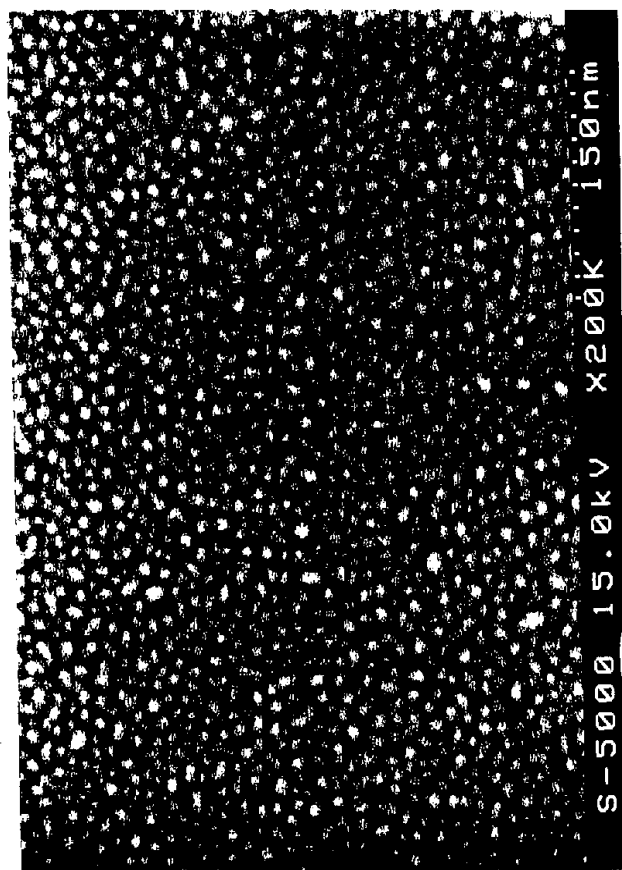


FIG. 6

