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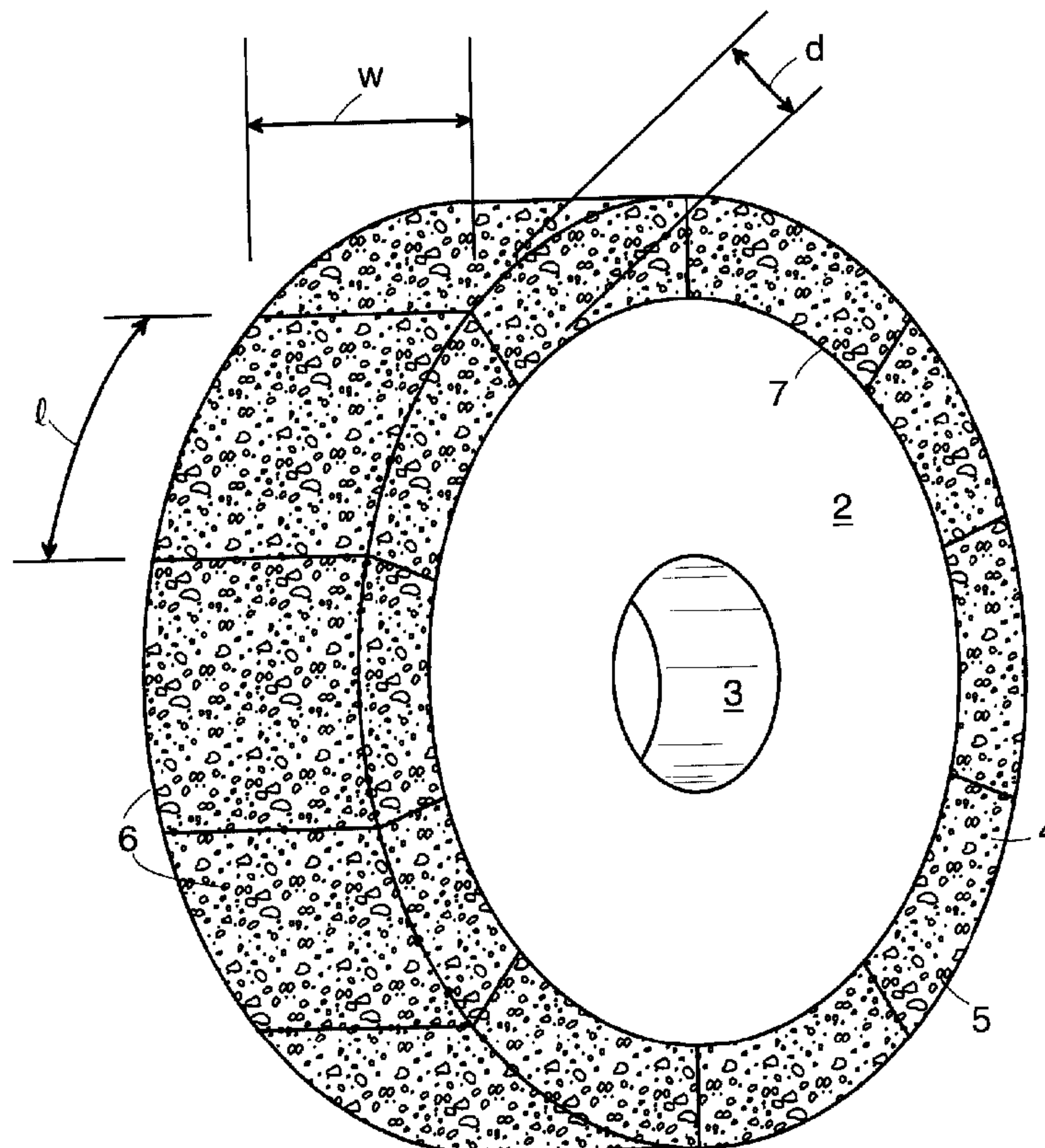
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(57) Abrégé/Abstract:

A method for precision cylindrical grinding of hard brittle materials, such as ceramics or glass and composites comprising ceramics or glass, provides material removal rates as high as 19-380 cm³/min/cm. The abrasive tools used in the method comprise a strong, light weight wheel core bonded to a continuous rim of abrasive segments containing superabrasive grain in a dense metal bond matrix.

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Abstract

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method comprise a strong, light weight wheel core bonded to a
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10 grain in a dense metal bond matrix.

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Method for Grinding Precision Components

This invention relates to a method for precision cylindrical grinding of hard brittle materials, such as ceramics, glass and composites comprising ceramics or glass, at peripheral wheel speeds up to 160 meters/second. The method employs novel abrasive tools comprising a wheel core or hub attached to a metal bonded superabrasive rim. These abrasive tools grind brittle materials at high material removal rates (e.g., 19-380 cm³/min/cm), with less wheel wear and less workpiece damage than conventional abrasive tools.

This invention was made with United States Government support under contract DE-AC05-84-OR21400 awarded by the Department of Energy. The United States Government has certain rights in this invention.

15 Background of the Invention

A method of grinding ceramics and an abrasive tool suitable for grinding sapphire and other ceramic materials is disclosed in U.S.-A-5,607,489 to Li. The tool is described as containing metal clad diamond bonded in a vitrified matrix comprising 2 to 20 volume % of solid lubricant and at least 10 volume % porosity.

A method for grinding cemented carbides using an abrasive tool containing diamond bonded in a metal matrix with 15 to 50 volume % of selected fillers, such as graphite, is disclosed in U.S.-A-3,925,035 to Keat.

A cutting-off wheel made with metal bonded diamond abrasive grain is disclosed in U.S.-A-2,238,351 to Van der Pyl. The bond consists of copper, iron, tin, and, optionally, nickel and the bonded abrasive grain is sintered onto a steel core, optionally with a soldering step to insure adequate adhesion. The best bond is reported to have a Rockwell B hardness of 70.

An abrasive tool containing fine diamond grain (bort) bonded in a relatively low melting temperature metal bond, such as a bronze bond, is disclosed in U.S.-Re-21,165. The low melting bond serves to avoid oxidation of the fine diamond grain. An abrasive rim is constructed as a single, annular abrasive segment and then attached to a central disk of aluminum or other material.

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None of these methods has proven entirely satisfactory in the precision cylindrical grinding of precision components. These methods are limited by prior art tools which fail to meet rigorous specifications for part shape, size and surface quality when operated at commercially feasible grinding rates. Most commercial cylindrical grinding operations employ resin or vitrified bonded superabrasive wheels and these wheels are operated at relatively low grinding efficiencies (e.g., 1 - 5 mm³/s/mm for advanced ceramics) so as to avoid surface and subsurface damage to the precision components. Grinding efficiencies are further reduced due to the tendency of ceramic workpieces to clog the wheel faces of such tools, requiring frequent wheel dressing and truing to maintain precision forms.

As market demand has grown for precision ceramic components in products such as engines, refractory equipment and electronic devices (e.g., wafers, magnetic heads and display windows), the need has grown for an improved method for precision cylindrical grinding of ceramics and other brittle, precision components.

Summary of the Invention

The invention is a method of finishing brittle precision components comprising the steps:

- a) mounting a cylindrical workpiece on a fixture;
- b) mounting an abrasive wheel on a grinding machine, the abrasive wheel comprising a core and a continuous abrasive rim, the core having a minimum specific strength of 2.4 MPa-cm³/g, and a circular perimeter adhesively bonded with a thermally stable bond to at least one abrasive segment in the abrasive rim, the abrasive segment consisting essentially of abrasive grain and a metal bond matrix having a fracture toughness of 1.0 to 6.0 MPa m^{1/2} and a maximum porosity of 5 volume %;
- c) rotating the abrasive wheel at a speed of 25 to 160 meters/second;
- d) contacting the abrasive wheel to an exterior surface of a rotating workpiece; and
- e) grinding the workpiece at a MRR of up to 380 cm³/min/cm to finish the exterior surface of the ceramic component;

whereby after finishing, the ceramic component is substantially free of cracking and subsurface damage from grinding.

f) the metal bond matrix may comprise 35 to 84 wt% copper and 16 to 65 wt% tin.

g) according to another aspect, the abrasive wheel is self-dressing.

h) according to a further aspect, the step of grinding the silicon nitride workpiece with the abrasive wheel draws less than 5% more power as the speed of the abrasive wheel is increased from 56 to 80 meters/second at a constant MRR.

Description of the Drawing

Figure 1 illustrates a continuous rim of abrasive segments bonded to the perimeter of a metal core to form a type 1A abrasive grinding wheel.

Description of the Preferred Embodiments

In the cylindrical grinding method of the invention, a workpiece driven by a positive drive rotates around a fixed axis, and the surface of the workpiece is ground by contact with a rotating abrasive wheel so as to create on the surface of the workpiece a precise shape around the axis of rotation. The cylindrical grinding method of the invention includes a variety of finishing operations, such as traverse grinding of cylindrical surfaces and traverse grinding of tapers; and plunge grinding of cylindrical surfaces, tapers or forms, optionally with multiple or single diameters or adjoining fillets. Fixtures having two ends (live or dead center) to clamp the workpiece are generally needed for grinding workpieces having an aspect ratio of 3:1 or higher. A single end of smaller aspect ratio workpieces may be clamped into a rotating headstock spindle during grinding. Other examples of grinding processes within the invention include rotary surface grinding, crankshaft grinding, cam grinding, cambered

cylindrical grinding and grinding of shapes such as polygons.

5 The grinding operation may be carried out with or without coolant, depending upon the workpiece material, surface finish quality needed, grinding machine design, and other process variables. Truing and dressing operations, while optional, preferably are carried out on the abrasive wheel prior to the grinding operation, and, optionally, as needed during the operation. In the method of the invention some grinding processes may be carried out without dressing the abrasive wheels.

10 During grinding, the workpiece may be rotated in the same direction as the abrasive wheel or in the opposite direction. The workpiece is generally rotated at a speed less than that of the abrasive wheel, preferably at least one order of magnitude less than that of the abrasive wheel. For example, at a wheel

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speed of 80 m/sec, the workpiece speed is preferably 1-12 m/sec, depending upon the shape and composition of the workpiece, the grinding machine used, geometry being ground, material removal rate, and other variables. Smaller workpieces preferably are rotated more rapidly than larger ones. For efficient grinding, harder workpieces (e.g., silicon nitride) require higher normal grinding forces and workpieces with higher mechanical strength (e.g., tungsten carbide) require higher grinding power. One skilled in the art may select appropriate grinding machine settings to achieve maximum efficiency for a given workpiece and grinding operation.

When carrying out the method of the invention to finish ceramic workpieces, conditions that produce cracking and subsurface damage in ceramics, such as high grinding forces, thermal shock, poor removal of heat from the grinding zone, large contact stresses and chatter, or sustained long term vibrations in the grinding zone, are minimized by using the abrasive tools described herein. Acceptable levels of subsurface damage is achieved without loss of grinding efficiency by adjusting the abrasive grain size, shape and concentration to operate in concert with the desired grinding process parameters. Grinding of the ceramic workpiece by brittle fracture is minimized and fine surface finishes having a variability on the order of less than 0.025 microns may be achieved at material removal rates from about 19 to 380 $\text{cm}^3/\text{s}/\text{cm}$. In contrast, prior art resin bonded diamond wheels are capable of maximum MRRs of less than 19 $\text{cm}^3/\text{min}/\text{cm}$ before surface and subsurface damage becomes evident.

The method of the invention employs certain, novel abrasive tools which are grinding wheels comprising a core having a central bore for mounting the wheel on a grinding machine, the core being designed to support a metal bonded superabrasive rim along the periphery of the wheel. These two parts of the wheel are held together with a thermally stable bond, and the wheel and its components are designed to tolerate stresses generated at wheel peripheral speeds of up to at least 80 m/sec, preferably up to 160 m/sec. Best results are obtained at 60 to 100 m/sec. Preferred tools are type 1A wheels designed for mounting on a cylindrical grinding machine.

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The core is substantially circular in shape. The core may comprise any material having a minimum specific strength of 2.4 MPa-cm³/g, preferably 40-185 Mpa-cm³/g. The core material preferably has a density of 0.5 to 8.0 g/cm³, most preferably 2.0 to 8.0 g/cm³. Examples of suitable materials are steel, aluminum, titanium and bronze, and their composites and alloys and combinations thereof. Reinforced plastics having the designated minimum specific strength may be used to construct the core. Composites and reinforced core materials typically have a continuous phase of a metal or a plastic matrix, often in powder form, to which fibers or grains or particles of harder, more resilient, and/or less dense, material is added as a discontinuous phase. Examples of reinforcing materials suitable for use in the core of the tools of the invention are glass fiber, carbon fiber, aramid fiber, ceramic fiber, ceramic particles and grains, and hollow filler materials such as glass, mullite, alumina and Zeolite® spheres.

Steel and other metals having densities of 0.5 to 8.0 g/cm³ are most preferred for making the cores for the tools of the invention. In making the cores used for high speed grinding (e.g., at least 80 m/sec), light weight metals in powder form (i.e., metals having densities of about 1.8 to 4.5 g/cm³), such as aluminum, magnesium and titanium, and alloys thereof, and mixtures thereof, are preferred. Aluminum and aluminum alloys are especially preferred. Metals having sintering temperatures between 400 and 900° C, preferably 570-650°C, are selected if a co-sintering assembly process is used to make the tools. Low density filler materials may be added to reduce the weight of the core. Porous and/or hollow ceramic or glass fillers, such as glass spheres and mullite spheres are suitable materials for this purpose. Also useful are inorganic and non-metallic fiber materials. When indicated by processing conditions, an effective amount of lubricant or other processing aids known in the metal bond and superabrasive arts may be added to the metal powder before pressing and sintering.

The tool should be strong, durable and dimensionally stable in order to withstand the potentially destructive forces

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generated by high speed operation. The core must have a minimum specific strength to operate grinding wheels at very high angular velocity needed to achieve tangential contact speed between 80 and 160 m/s. At such velocities the minimum
5 specific strength parameter needed for the core materials used in this invention is $2.4 \text{ MPa-cm}^3/\text{g}$, and higher parameters in the range of $40\text{--}185 \text{ MPa-cm}^3/\text{g}$ are preferred.

The specific strength parameter is defined as the ratio of core material yield (or fracture) strength divided by core
10 material density. In the case of brittle materials, having a lower fracture strength than yield strength, the specific strength parameter is determined by using the lesser number, the fracture strength. The yield strength of a material is the minimum force applied in tension for which strain of the
15 material increases without further increase of force. For example, ANSI 4140 steel hardened to above about 240 (Brinell scale) has a tensile strength in excess of 700 MPa. Density of this steel is about 7.8 g/cm^3 . Thus, its specific strength parameter is about $90 \text{ MPa-cm}^3/\text{g}$. Similarly, certain aluminum
20 alloys, for example, Al 2024, Al 7075 and Al 7178, that are heat treatable to Brinell hardness above about 100 have tensile strengths higher than about 300 MPa. Such aluminum alloys have low density of about 2.7 g/cm^3 and thus exhibit a specific strength parameter of more than $110 \text{ MPa-cm}^3/\text{g}$. Titanium alloys
25 and bronze composites and alloys fabricated to have a density no greater than 8.0 g/cm^3 , are also suitable for use.

The core material should be tough, thermally stable at temperatures reached near the grinding zone (e.g., about 50 to 270°C), resistant to chemical reaction with coolants and
30 lubricants used in grinding and resistant to wear by erosion due to the motion of cutting debris in the grinding zone. Although some alumina and other ceramics have acceptable failure values (i.e., in excess of $60 \text{ MPa-cm}^3/\text{g}$), they generally are too brittle and fail structurally in high speed
35 grinding due to fracture. Hence, ceramics are not suitable for use in the tool core. Metal, especially hardened, tool quality steel, and metal matrix composites are preferred.

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The abrasive segment of the grinding wheel for use with the present invention is a segmented or continuous rim mounted on a core. A segmented abrasive rim is shown in Fig. 1. The core 2 has a central bore 3 for mounting the wheel to an arbor of a power drive (not shown). The abrasive rim of the wheel comprises superabrasive grains 4 embedded (preferably in uniform concentration) in a metal matrix bond 5. A plurality of abrasive segments 6 make up the abrasive rim shown in Fig. 1. Although the illustrated embodiment shows ten segments, the number of segments is not critical. An individual abrasive segment, as shown in Fig. 1, has a truncated, rectangular ring shape (an arcuate shape) characterized by a length, l , a width, w , and a depth, d .

The embodiment of a grinding wheel shown in Fig. 1 is considered representative of wheels which may be operated successfully according to the method of the invention, and should not be viewed as limiting. Apertures or gaps in the core are sometimes used to provide paths to conduct coolant to the grinding zone and to route cutting debris away from the zone. A wider segment than the core width is occasionally employed to protect the core structure from erosion through contact with swarf material as the wheel radially penetrates the work piece.

The wheel can be fabricated by first forming individual segments of preselected dimension and then attaching the pre-formed segments to the circular perimeter (circumference) 7 of the core with an appropriate adhesive. Another preferred fabrication method involves forming segment precursor units of a powder mixture of abrasive grain and bond, molding the composition around the circumference of the core, and applying heat and pressure to create and attach the segments, *in situ* (i.e., co-sintering the core and the rim).

The continuous abrasive rim may comprise one abrasive segment, or at least two abrasive segments, sintered separately in molds, and then individually mounted on the core with a thermally stable bond (i.e., a bond stable at the temperatures encountered during grinding at the portion of the segments

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directed away from the grinding face, typically from about 50-350°C). Segmented continuous abrasive rims are preferred over a single continuous abrasive rim, molded as a single piece in a ring shape, due to the greater ease of achieving a truly round, planar shape during manufacture of a tool from multiple abrasive segments.

The abrasive rim component contains superabrasive grain held in a metal matrix bond, typically formed by sintering a mixture of metal bond powder and the abrasive grain in a mold designed to yield the desired size and shape of the abrasive rim or the abrasive rim segments.

The superabrasive grain used in the abrasive rim may be selected from diamond, natural and synthetic, and cubic boron nitride and combinations of these abrasives. Grain size and type selection will vary depending upon the nature of the workpiece and the type of grinding process. For example, in the grinding and polishing of sapphire, a superabrasive grain size ranging from 2 to 300 micrometers is preferred. For grinding alumina, a superabrasive grain size of about 125 to 300 micrometers (60 to 120 grit; Norton Company grit size) is generally preferred. For grinding silicon nitride, a grain size of about 45 to 80 micrometers (200 to 400 grit), is generally preferred.

As a volume percentage of the abrasive rim, the tools comprise 10 to 50 volume % superabrasive grain, preferably 10 to 40 volume %. A minor amount of wear resistant material, having a hardness equal to or less than that of the workpiece material, may be added as bond filler to alter the wear rate of the bond. As a volume percentage of the rim component, the filler may be used at 0-15 vol. %, preferably 0.1 to 10 vol. %, most preferably 0.1 to 5 vol. %. Tungsten carbide, cerium oxide, and alumina grain are examples of fillers which may be utilized.

Any metal bond suitable for bonding superabrasives and having a fracture toughness of 1.0 to $6.0 \text{ MPa}\cdot\text{m}^{1/2}$, preferably 2.0 to $4.0 \text{ MPa}\cdot\text{m}^{1/2}$, may be employed herein. Fracture toughness is the stress intensity factor at which a crack initiated in a material will propagate in the material and lead to a fracture of the material. Fracture toughness is expressed as $K_{Ic} =$

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$(\sigma_f) (\pi^{1/2}) (c^{1/2})$, where K_{Ic} is the fracture toughness, σ_f is the stress applied at fracture, and c is one-half of the crack length. There are several methods which may be used to determine fracture toughness, and each has an initial step
 5 where a crack of known dimension is generated in the test material, and then a stress load is applied until the material fractures. The stress at fracture and crack length are substituted into the equation and the fracture toughness is calculated (e.g., the fracture toughness of steel is about 30-
 10 60 MPa.m^{1/2}, of alumina is about 2-3 MPa.m^{1/2}, of silicon nitride is about 4-5 MPa.m^{1/2}, and of zirconia is about 7-9 MPa.m^{1/2}).

To optimize wheel life and grinding performance, the bond wear rate should be equal to or slightly higher than the wear rate of the abrasive grain during grinding operations.

15 Fillers, such as are mentioned above, may be added to the metal bond to decrease the wheel wear rate. Metal powders tending to form a relatively dense bond structure (i.e., less than 5 volume % porosity) are preferred to enable higher material removal rates during grinding.

20 Materials useful in the metal bond matrix of the rim include, but are not limited to, copper, tin, zinc, cobalt and iron, and their alloys, such as bronze and brass, and mixtures thereof. These metals optionally may be used with titanium or titanium hydride, or other superabrasive reactive (i.e., active
 25 bond components) material capable of forming a carbide or nitride chemical linkage between the grain and the bond at the surface of the superabrasive grain under the selected sintering conditions to strengthen the grain/bond interface. Stronger grain/bond interfaces will limit premature loss of grain and
 30 workpiece damage and shortened tool life caused by premature grain loss.

In a preferred embodiment of the abrasive rim, the metal bond matrix comprises 45 to 90 volume % of the rim, more preferably 60 to 80 volume %. When filler is added to the
 35 bond, the filler comprises 0 to 50 volume % of the metal matrix of the rim, preferably 0.1 to 25 volume %. Porosity of the metal bond matrix should be established at a maximum of 5 .

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volume % during manufacture of the abrasive segment. The metal bond matrix preferably has a Knoop hardness of 0.1 to 3 GPa.

In a preferred embodiment of a type 1A grinding wheel, the core is made of aluminum and the rim contains a bronze bond made from copper and tin powders (80/20 wt. %), and, optionally with the addition of 0.1-3.0 wt%, preferably 0.1-1.0 wt%, phosphorus in the form of a phosphorus/copper powder. During manufacture of the abrasive segments, the metal powders of this composition are mixed with 100 to 400 grit (160 to 45 microns) diamond abrasive grain, molded into abrasive rim segments and sintered or densified in the range of 400-550° C at 20 to 33 MPa to yield a dense abrasive rim, preferably having a density of at least 95 % of the theoretical density (i.e., comprising no more than about 5 volume % porosity).

In a typical co-sintering wheel manufacturing process, the metal powder of the core is poured into a steel mold and cold pressed at 80 to 200 kN (about 10-50 MPa pressure) to form a green part having a size approximately 1.2 to 1.6 times the desired final thickness of the core. The green core part is placed in a graphite mold and a mixture of the abrasive grain and the metal bond powder blend is added to the cavity between the core and the outer rim of the graphite mold. A setting ring may be used to compact the abrasive and metal bond powders to the same thickness as the core preform. The graphite mold contents are then hot pressed at 370 to 410°C under 20 to 48 MPa of pressure for 6 to 10 minutes. As is known in the art, the temperature may be ramped up (e.g., from 25 to 410°C for 6 minutes; held at 410°C for 15 minutes) or increased gradually prior to applying pressure to the mold contents.

Following hot pressing, the graphite mold is stripped from the part, the part is cooled and the part is finished by conventional techniques to yield an abrasive rim having the desired dimensions and tolerances. For example, the part may be finished to size using vitrified grinding wheels on grinding machines or carbide cutters on a lathe.

When co-sintering the core and rim of the invention, very little material removal is needed to put the part into its final shape. In other methods of forming a thermally stable

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bond between the abrasive rim and the core, machining of both the core and the rim may be needed, prior to a cementing, linking or diffusion step, to insure an adequate surface for mating and bonding of the parts.

5 In creating a thermally stable bond between the rim and the core utilizing segmented abrasive rims, any thermally stable adhesive having the strength to withstand peripheral wheel speeds up to 160 m/sec may be used. Thermally stable adhesives are stable to grinding process temperatures likely
10 to be encountered during grinding at the portion of the abrasive segments directed away from the grinding face. Such temperatures typically range from about 50-350° C.

The adhesive bond should be very strong mechanically to withstand the destructive forces existing during rotation of
15 the grinding wheel and during the grinding operation. Two-part epoxy resin cements are preferred. A preferred epoxy cement, Technodyne® HT-18 epoxy resin (obtained from Taoka Chemicals, Japan), and its modified amine hardener, may be mixed in the ratio of 100 parts resin to 19 parts hardener. Filler, such as
20 fine silica powder, may be added at a ratio of 3.5 parts per 100 parts resin to increase cement viscosity. The perimeter of the metal core may be sandblasted to obtain a degree of roughness prior to attachment of the segments. The thickened epoxy cement is applied to the ends and bottom of segments
25 which are positioned around the core substantially as shown in Fig. 1 and mechanically held in place during the cure. The epoxy cement is allowed to cure (e.g., at room temperature for 24 hours followed by 48 hours at 60°C). Drainage of the cement during curing and movement of the segments is minimized during
30 cure by the addition of sufficient filler to optimize the viscosity of the epoxy cement.

Adhesive bond strength may be tested by spin testing at acceleration of 45 rev/min, as is done to measure the burst speed of the wheel. The wheels need demonstrated burst ratings
35 equivalent to at least 271 m/s tangential contact speeds to qualify for operation at 160 m/s tangential contact speed under currently applicable safety standards in the United States.

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With these abrasive tools one can carry out the inventive method of precision cylindrical grinding and finishing of hard, brittle, wear resistant materials, such as advanced ceramic materials, glass, components containing ceramic materials or glass, and ceramic composite materials. The brittle, precision components of the invention are materials having a fracture toughness ranging from about 0.6 (silicon) to about 16 (tungsten carbide), with the optimum benefits achieved in grinding ceramics with a fracture toughness of about 2-8 MPa·m^{1/2}.

The method of the invention is preferred for grinding materials including, but not limited to, silicon; mono- and polycrystalline oxides, carbides, nitrides, borides and silicides; polycrystalline diamond; glass; and composites of ceramic in a non-ceramic matrix; and combinations thereof. Examples of typical workpiece materials include, but are not limited to, silicon nitride, silicon carbide, silicon oxide, silicon dioxide (e.g., quartz), aluminum nitride, aluminum oxide-titanium carbide, tungsten carbide, titanium carbide, vanadium carbide, hafnium carbide, aluminum oxide (e.g., sapphire), zirconium oxide, tungsten boride, boron carbide, boron nitride, titanium diboride, silicon oxynitride and stabilized zirconia and combinations thereof. Also included are certain metal matrix composites such as cemented carbides, hard brittle amorphous materials such as mineral glass, polycrystalline diamond and polycrystalline cubic boron nitride. Either single (mono-) crystal or polycrystalline ceramics can be effectively ground. With each type of ceramic, the quality of the ceramic part and the efficiency of the grinding operation in the method of the invention increase as the peripheral wheel speed in the method of the invention is increased up to 160 m/s.

Among the precision components parts improved by using the method of the invention are ceramic engine valves and rods, pump seals, ball bearings and fittings, cutting tool inserts, wear parts, drawing dies for metal forming, refractory components, visual display windows, flat glass for windshields, doors and windows, insulators and electrical parts, and ceramic

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electronic components, including, but not limited to, silicon wafers, magnetic heads, and electronic substrates.

Unless otherwise indicated, all parts and percentages in the following examples are by weight. The examples merely
5 illustrate the invention and are not intended to limit the invention.

Example 1

Abrasive wheels useful in the method of the invention were prepared in the form of 1A1 metal bonded diamond wheels
10 utilizing the materials and processes described below.

A blend of 43.74 wt % copper powder (Dendritic FS grade, particle size +200/-325 mesh, obtained from Sintertech International Marketing Corp., Ghent, NY); 6.24 wt% phosphorus/copper powder (grade 1501, +100/-325 mesh particle
15 size, obtained from New Jersey Zinc Company, Palmerton, PA); and 50.02 wt% tin powder (grade MD115, +325 mesh, 0.5% maximum, particle size, obtained from Alcan Metal Powders, Inc., Elizabeth, New Jersey) was prepared. Diamond abrasive grain (320 grit size synthetic diamond obtained from General
20 Electric, Worthington, Ohio) was added to the metal powder blend and the combination was mixed until it was uniformly blended. The mixture was placed in a graphite mold and hot pressed at 407° C for 15 minutes at 3000 psi (2073 N/cm²) until a matrix with a target density in excess of 95% of theoretical
25 had been formed (e.g., for the #6 wheel used in Example 2: > 98.5% of the theoretical density). Rockwell B hardness of the segments produced for the #6 wheel was 108. Segments contained 18.75 vol. % abrasive grain. The segments were ground to the required accurate geometry to match the periphery of a machined
30 aluminum core (7075 T6 aluminum, obtained from Yarde Metals, Tewksbury, MA), yielding a wheel with an outer diameter of about 393 mm, and segments 0.62 cm thick.

The abrasive segments and the aluminum core were assembled with a silica filled epoxy cement system ^{*}(Technodyne HT-18
35 adhesive, obtained from Taoka Chemicals, Japan) to make grinding wheels having a continuous rim consisting of multiple abrasive segments. The contact surfaces of the core and the
* Trade-mark

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segments were degreased and sandblasted to insure adequate adhesion.

To characterize the maximum operating speed of this new type of wheel, full size wheels were purposely spun to destruction to determine the burst strength and rated maximum operating speed according to the Norton Company maximum operating speed test method. The table below summarizes the burst test data for typical examples of the 393-mm diameter experimental metal bonded wheels.

10 Experimental Metal Bond Wheel Burst Strength Data

Wheel #	Wheel Diameter cm(inch)	Burst RPM	Burst speed (m/s)	Burst speed (sfpm)	Max. Operating Speed (m/s)
4	39.24 (15.45)	9950	204.4	40242	115.8
5	39.29 (15.47)	8990	185.0	36415	104.8
7	39.27 (15.46)	7820	160.8	31657	91.1
9	39.27 (15.46)	10790	221.8	43669	125.7

According to these data, the experimental grinding wheels of this design will qualify for an operational speed up to 90 m/s (17,717 surface feet/min.). Higher operational speeds of up to 160 m/s can be readily achieved by some further modifications in fabrication processes and wheel designs.

Example 2

20 Grinding Performance Evaluation:

Three, 393-mm diameter, 15 mm thick, 127 mm central bore, (15.5 in x 0.59 in x 5 in) experimental metal bonded segmental wheels made according to the method of Example 1, above, (#4 having segments with a density of 95.6 % of theoretical; #5 at 97.9 % of theoretical; and #6 at 98.5 % of theoretical density)

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were tested for grinding performance according to the method of the invention. Initial testing at 32 and 80 m/s established wheel #6 as the wheel having the best grinding performance of the three, although all experimental wheels were acceptable.

5 Testing of wheel #6 was done at three speeds: 32 m/s (6252 sfpm), 56 m/s (11,000 sfpm), and 80 m/s (15,750 sfpm). Two commercial prior art abrasive wheel recommended for grinding advanced ceramic materials served as control wheels and they were tested along with the metal bonded wheels in the method of
10 the invention. One was a vitrified bonded diamond wheel (SD320-N6V10 wheel obtained from Norton Company, Worcester, MA) and the other was a resin bonded diamond wheel (SD320-R4BX619C wheel obtained from Norton Company, Worcester, MA). The resin wheel was tested at all three speeds. The vitrified wheel was
15 tested at 32 m/s (6252 sfpm) only, due to speed tolerance considerations.

Over one thousand plunge grinds of 6.35 mm (0.25 inch) wide and 6.35 mm (0.25 inch) deep were performed on silicon nitride workpieces. The grinding testing conditions were:

20 Grinding Test Conditions:

Machine: Studer Grinder Model S40 CNC

Wheel Specifications: SD320-R4BX619C, SD320-N6V10,
Size : 393mm diameter, 15 mm thickness and
127 mm hole.

25 Wheel Speed: 32, 56, and 80 m/s (6252, 11000, and 15750 sfpm)

Coolant: Inversol 22 @60% oil and 40% water

Coolant Pressure: 270 psi (19 kg/cm²)

Material Removal Rate: Vary, starting at 3.2 mm³/s/mm (0.3
30 in³/min/in)

Work Material: Si₃N₄ (rods made of NT551 silicon nitride, obtained from Norton Advanced Ceramics, Northboro, Massachusetts) 25.4 mm (1 in.) diameter X 88.9 mm (3.5 in.) long

35 Work Speed: 0.21 m/s (42 sfpm), constant

Work Starting diameter: 25.4 mm (1 inch)

Work finish diameter: 6.35 mm (0.25 inch)

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For operations requiring truing and dressing, conditions suitable for the metal bonded wheels of the invention were:

Truing Operation:

5 Wheel: 5SG46IVS (obtained from Norton Company)
 Wheel Size: 152 mm diameter (6 inches)
 Wheel Speed: 3000 rpm; at +0.8 ratio relative to
 the grinding wheel
 Lead: 0.015 in. (0.38mm)
 Compensation: 0.0002 in.

10 Dressing Operation:

 Stick: 37C220H-KV (SiC)
 Mode: Hand Stick Dressing

15 Tests were performed in a cylindrical outer diameter
 plunge mode in grinding the silicon nitride rods. To preserve
 the best stiffness of work material during grinding, the 88.9
 mm (3.5 in.) samples were held in a chuck with approximately 31
 mm (1-1/4 in.) exposed for grinding. Each set of plunge grind
 tests started from the far end of each rod. First, the wheel
 20 made a 6.35 mm (1/4 in.) wide and 3.18 mm (1/8 in.) radial
 depth of plunge to complete one test. The work rpm was then
 re-adjusted to compensate for the loss of work speed due to
 reduced work diameter. Two more similar plunges were performed
 at the same location to reduce the work diameter from 25.4 mm
 25 (1 in.) to 6.35 mm (1/4 in.). The wheel was then laterally
 moved 6.35 mm (1/4 in.) closer to the chuck to perform next
 three plunges. Four lateral movements were performed on the
 same side of a sample to complete the twelve plunges on one end
 of a sample. The sample was then reversed to expose the other
 30 end for another twelve grinds. A total of 24 plunge grinds was
 done on each sample.

 The initial comparison tests for the method of the
 invention were conducted at 32 m/s peripheral speed at three
 material removal rates (MRR') from approximately $3.2 \text{ mm}^3/\text{s}/\text{mm}$
 35 ($0.3 \text{ in}^3/\text{min}/\text{in}$) to approximately $10.8 \text{ mm}^3/\text{s}/\text{mm}$ (1.0
 $\text{in}^3/\text{min}/\text{in}$). Table 1 shows the performance differences, as
 depicted by G-ratios, among the three different types of wheels
 after twelve plunge grinds. G-ratio is the unit-less ratio of
 volume material removed over volume of wheel wear. The data

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showed that the N grade vitrified wheel had better G ratios than the R grade resin wheel at the higher material removal rates, suggesting that a softer wheel performs better in grinding a ceramic workpiece. However, the harder,
5 experimental, metal bonded wheel (#6) was far superior to the resin wheel and the vitrified wheel at all material removal rates.

Table 1 shows the estimated G-ratios for the resin wheel and the new metal bonded wheel (#6) at all material removal
10 rate conditions. Since there was no measurable wheel wear after twelve grinds at each material removal rate for the metal bonded wheel, a symbolic value of 0.01 mil (0.25 μ m) radial wheel wear was given for each grind. This yielded the calculated G-ratio of 6051.

15 Although the metal bond wheel of the invention contained 75 diamond concentration (about 18.75 volume % abrasive grain in the abrasive segments), and the resin and vitrified wheels were 100 concentration and 150 concentration (25 volume % and 37.5 volume %), respectively, the wheel of the invention still
20 exhibited superior grinding performance. At these relative grain concentrations, one would expect superior grinding performance from the control wheels containing a higher volume % of abrasive grain. Thus, the actual results were quite unexpected.

25 Table 1 shows the surface finish (Ra) and waviness (Wt) data measured on samples ground by the three wheels at the low test speed. The waviness value, Wt, is the maximum peak to valley height of the waviness profile. All surface finish data were measured on surfaces created by cylindrical plunge
30 grinding without spark-out. These surfaces normally would be rougher than surfaces created by traverse grinding.

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TABLE 1

Sample	MRR'	Wheel	Tangen	Unit	Specific	G-	Surface	Waviness
	mm ³ /s/ mm	Speed m/s	-tial Force N/mm	Power W/mm	Energy W·s/mm ³	Ratio	Finish Ra μm	W μm
<u>Resin</u>								
973	3.2	32	0.48	40	12.8	585.9	0.52	0.86
1040	6.3	32	0.98	84	13.3	36.6	0.88	4.01
980	8.9	32	1.67	139	9.5	7.0	0.99	4.50
1016	3.2	56	0.49	41	13.1	586.3	0.39	1.22
1052	6.3	56	0.98	81	12.9		0.55	1.52
						293.2		
992	3.2	80	0.53	45	14.2	586.3	0.42	1.24
1064	6.3	80	0.89	74	11.8	293.2	0.62	1.80
1004	9.0	80	1.32	110	12.2	586.3	0.43	1.75
<u>Vitrified</u>								
654	3.2	32	1.88	60	19.2	67.3	0.7	2.50
666	9.0	32	4.77	153	17.1	86.5	1.6	5.8
678	11.2	32	4.77	153	13.6	38.7	1.7	11.8
<u>Metal Experimental</u>								
407	3.2	32	2.09	67	2.1	6051	0.6	0.9
419	6.3	32	4.03	130	20.6	6051	0.6	0.9
431	9.0	32	5.52	177	19.7	6051	0.6	0.8
443	3.2	56	1.41	80	25.4	6051	0.6	0.7
455	6.3	56	2.65	150	23.9	6051	0.5	0.7
467	9.0	56	3.70	209	23.3	6051	0.5	0.6
479	3.2	80	1.04	85	26.9	6051	0.5	1.2
491	6.3	80	1.89	153	24.3	6051	0.6	0.8
503	9.0	80	2.59	210	23.4	6051	0.6	0.8

Table 1 shows the difference in grinding power consumption at various material removal rates for the three wheel types.

- 5 The resin wheel had lower power consumption than the other two wheels; however, the experimental metal bonded wheel and vitrified wheel had comparable power consumption. The experimental wheel drew an acceptable amount of power for ceramic grinding operations, particularly in view of the

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favorable G-ratio and surface finish data observed for the wheels of the invention. In general, the wheels of the invention demonstrated power draw proportional to material removal rates.

5 When grinding performance was measured at 80 m/s (15,750 sfpm) in an additional grinding test, the resin wheel and experimental metal wheel had comparable power consumption at material removal rate (MRR) of $9.0 \text{ mm}^3/\text{s}/\text{mm}$ ($0.8 \text{ in}^3/\text{min}/\text{in}$). As shown in Table 2, the experimental wheels were operated at
10 increasing MRRs without loss of performance or unacceptable power loads. The metal bonded wheel power draw was roughly proportional to the MRR. The highest MRR achieved in this study was $47.3 \text{ mm}^3/\text{s}/\text{mm}$ ($28.4 \text{ cm}^3/\text{min}/\text{cm}$).

Table 2 data are averages of twelve grinding passes.
15 Individual power readings for each of the twelve passes remained remarkably consistent for the experimental wheel within each material removal rate. One would normally observe an increase of power as successive grinding passes are carried and the abrasive grains in the wheel begin to dull or the face
20 of the wheel becomes loaded with workpiece material. This is often observed as the MRR is increased. However, the steady power consumption levels observed within each MRR during the twelve grinds demonstrates, unexpectedly, that the experimental wheel maintained its sharp cutting points during the entire
25 length of the test at all MRRs.

Furthermore, during this entire test, with material removal rates ranging from $9.0 \text{ mm}^3/\text{s}/\text{mm}$ ($0.8 \text{ in}^3/\text{min}/\text{in}$) to $47.3 \text{ mm}^3/\text{s}/\text{mm}$ ($4.4 \text{ in}^3/\text{min}/\text{in}$), it was not necessary to true or dress the experimental wheel. However, different grinding
30 operations might require truing or dressing.

The experimental wheel showed no measurable wheel wear after 168 plunges at 14 different material removal rates. The total, cumulative amount of silicon nitride material ground without any evidence of wheel wear for the experimental metal
35 bond wheel was equivalent to about 271 cm^3 per cm (42 in^3 per inch) of wheel width. By contrast, the G-ratio for the 100 concentration resin wheel at $9.0 \text{ mm}^3/\text{s}/\text{mm}$ ($0.8 \text{ in}^3/\text{min}/\text{in}$) material removal rate was approximately 583 after twelve plunges.

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Table 2 shows that the samples ground by the experimental metal bonded wheel at all 14 material removal rates maintained constant surface finishes between $0.4\text{ }\mu\text{m}$ ($16\text{ }\mu\text{in.}$) and $0.5\text{ }\mu\text{m}$ ($20\text{ }\mu\text{in.}$), and had waviness values between $1.0\text{ }\mu\text{m}$ ($38\text{ }\mu\text{in.}$) and $1.7\text{ }\mu\text{m}$ ($67\text{ }\mu\text{in.}$). The resin wheel was not tested at these high material removal rates. However, at about $9.0\text{ mm}^3/\text{s}/\text{mm}$ ($0.8\text{ in}^3/\text{min}/\text{in}$) material removal rate, the ceramic bars ground by the resin wheel had slightly better but comparable surface finishes (0.43 versus $0.5\text{ }\mu\text{m}$, and poorer waviness (1.73 versus $1.18\text{ }\mu\text{m}$).

Surprisingly, there was no apparent deterioration in surface finish when the ceramic rods were ground with the new metal bonded wheel as the material removal rate increased. This is in contrast to the commonly observed surface finish deterioration with increase cut rates for standard wheels, such as the control wheels used herein.

Overall results demonstrate that in the method of the invention, the experimental metal wheel was able to grind effectively at a MRR which was over 5 times the MRR achievable with a standard, commercially used resin bond wheel. The experimental wheel had over 10 times the G-ratio compared to the resin wheel at the lower MRRs.

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TABLE 2
14 MRRs Tested At 80 m/s Wheel Speed

Sample	MRR' mm ³ /s /mm	Tangen- tial Force N/mm	Unit Power W/mm	Specifi c Energy W•s/mm ³	G- Ratio	Surface Finish Ra μm	Waviness W μm
<u>Resin</u>							
1004	9.0	1.32	110	12.2	586.3	0.43	1.75
<u>Metal</u>							
<u>Invention</u>							
805	9.0	1.21	98	11.0	6051	0.51	1.19
817	18.0	2.00	162	9.0	6051	0.41	0.97
829	22.5	2.62	213	9.5	6051	0.44	1.14
841	24.7	2.81	228	9.2	6051	0.47	1.04
853	27.0	3.06	248	9.2	6051	0.48	1.09
865	29.2	3.24	262	9.0	6051	0.47	1.37
877	31.4	3.64	295	9.4	6051	0.47	1.42
889	33.7	4.01	325	9.6	6051	0.44	1.45
901	35.9	4.17	338	9.4	6051	0.47	1.70
913	38.2	4.59	372	9.7	6051	0.47	1.55
925	40.4	4.98	404	10.0	6051	0.46	1.55
937	42.7	5.05	409	9.6	6051	0.44	1.57
949	44.9	5.27	427	9.5	6051	0.47	1.65
961	47.2	5.70	461	9.8	6051	0.46	1.42

When operated at 32 m/s (6252 sfpm) and 56 m/s (11,000
5 sfpm) wheel speeds (Table 1), the power consumption for the
metal bonded wheel was higher than that of the resin bond wheel
at all of the material removal rates tested. However, at the
high wheel speed of 80 m/s (15,750 sfpm) (Tables 1 and 2), the
power consumption for the metal bonded wheel became comparable
10 or slightly less than that of resin wheel when operated at the
same MRR. Overall, the trend showed that the power consumption
decreased with increasing wheel speed when grinding at the same
material removal rate for both the resin wheel and the
experimental metal bonded wheel. Power consumption during
15 grinding, much of which goes to the workpiece as heat, is less
important in grinding ceramic materials than in grinding

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metallic materials due to the greater thermal stability of the ceramic materials. As demonstrated by the surface quality of the ceramic samples ground with the wheels of the invention, the power consumption did not detract from the finished piece and was at an acceptable level.

For the experimental metal bonded wheel G ratio was essentially constant at 6051 for all material removal rates and wheel speeds. For the resin wheel, the G-ratio decreased with increasing material removal rates at any constant wheel speed.

Table 2 shows the improvement in surface finishes and waviness on the ground samples at higher wheel speed. In addition, the samples ground by the new metal bonded wheel had the lowest measured waviness under all wheel speeds and material removal rates tested.

These tests of the method of the invention utilizing the novel metal bonded wheel demonstrated superior wheel life compared to the control wheels. In contrast to the commercial control wheels, there was no need for truing and dressing the experimental wheels during the extended grinding tests. The experimental wheel was successfully operated at wheel speeds up to 90 m/s in these tests, and was designed to be operated safely and effectively on an appropriate cylindrical grinding machine at speeds up to 160 m/s to carry out the method of the invention.

Example 3

In a subsequent grinding test of the experimental wheel (#6) at 80 m/sec under the same operating conditions as those used in the previous Example, a MRR of 380 cm³/min/cm was achieved while generating a surface finish measurement (Ra) of only 0.5 μm (12 μin) and utilizing an acceptable level of power. The observed high material removal rate without surface damage to the ceramic workpiece which was attained by utilizing the method of the invention has not been reported for any ceramic material grinding operation with any commercial abrasive wheel of any bond type.

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We claim:

1. A method of finishing brittle precision components comprising the steps:

- 5 a) mounting a cylindrical workpiece on a fixture;
- b) mounting an abrasive wheel on a grinding machine, the abrasive wheel comprising a core and a continuous abrasive rim, the core having a minimum specific strength of 2.4 MPa-cm³/g, and a circular perimeter adhesively bonded with a
- 10 thermally stable bond to at least one abrasive segment in the abrasive rim, the abrasive segment consisting essentially of abrasive grain and a metal bond matrix having a fracture toughness of 1.0 to 6.0 MPa M^{1/2} and a maximum porosity of 5 volume %;
- 15 c) rotating the abrasive wheel at a peripheral speed of 25 to 160 meters/second;
- d) contacting the abrasive wheel to an exterior surface of a rotating workpiece; and
- e) grinding the workpiece at a MRR of up to 380 cm³/min/cm
- 20 to finish the exterior surface of the ceramic component; whereby after finishing, the ceramic component is substantially free of cracking and subsurface damage from grinding.
2. The method of claim 1, wherein the core of the abrasive wheel has a density of 0.5 to 8.0 g/cm³.
- 25 3. The method of claim 2, wherein the core is a metallic material selected from the group consisting of aluminum, steel, titanium and bronze, composites and alloys thereof, and combinations thereof.
4. The method of claim 1, wherein the abrasive segments
- 30 consist essentially of 45 to 90 volume % metal bond and 10 to 50 volume % abrasive grain.
5. The method of claim 1, wherein the abrasive grain is selected from the group consisting of diamond and cubic boron nitride and combinations thereof.
- 35 6. The method of claim 1, wherein the metal bond matrix has a Knoop hardness of 0.1 to 3 GPa.
7. The method of claim 1, wherein the metal bond matrix comprises 35 to 84 wt% copper and 16 to 65 wt% tin.

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8. The method of claim 7, wherein the metal bond matrix further comprises 0.2 to 1.0 wt% phosphorus.

9. The method of claim 1, wherein the abrasive segments have an elongated, arcuate shape and an inner curvature selected to mate with the circular perimeter of the core, and each abrasive segment has two ends designed to mate with adjacent abrasive segments such that the abrasive rim is continuous and substantially free of any gaps between abrasive segments when the abrasive segments are bonded to the core.

10. The method of claim 1, wherein the abrasive wheel is a type 1A1 wheel.

11. The method of claim 1, wherein the core is adhesively bonded to the rim with a two-part epoxy adhesive.

12. The method of claim 1, wherein the abrasive wheel is self-dressing.

13. The method of claim 1, wherein the grinding of a silicon nitride workpiece with the abrasive wheel draws less than 30% more power as the peripheral speed of the abrasive wheel is increased from 32 to 80 meters/second at a constant MRR.

14. The method of claim 13, wherein the step of grinding the silicon nitride workpiece with the abrasive wheel draws less than 5% more power as the speed of the abrasive wheel is increased from 56 to 80 meters/second at a constant MRR.

15. The method of claim 1, wherein the abrasive wheel is substantially free of measurable wear over a ranges of MRRs from 9.0 to 47.1 mm³/s/mm at an abrasive wheel speed of 80 meters/second after having removed from a silicon nitride workpiece at least 271 cm³ per cm of abrasive wheel.

16. The method of claim 1, wherein the workpieces consist of material selected from the group consisting essentially of silicon; mono- and polycrystalline oxides carbides, nitrides, borides and silicides; polycrystalline diamond; glass; and composites of ceramic in a non-ceramic matrix; and combinations thereof.

17. The method of claim 16, wherein the workpiece is selected from the group consisting of silicon nitride, silicon carbide, silicon oxide, silicon dioxide, aluminum nitride, aluminum oxide-titanium carbide, tungsten carbide, boron carbide, boron nitride, titanium carbide, vanadium carbide, hafnium carbide,

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aluminum oxide, zirconium oxide, tungsten boride, and titanium boride, and combinations thereof.

18. The method of claim 1, wherein the precision components comprise ceramic engine valves and rods, pump seals, ball
5 bearings and fittings, cutting tool inserts, wear parts, drawing dies for metal forming, refractory components, visual display windows, flat glass for windshields, doors and windows, insulators and electrical parts, silicon wafers, magnetic heads and electronic substrates.

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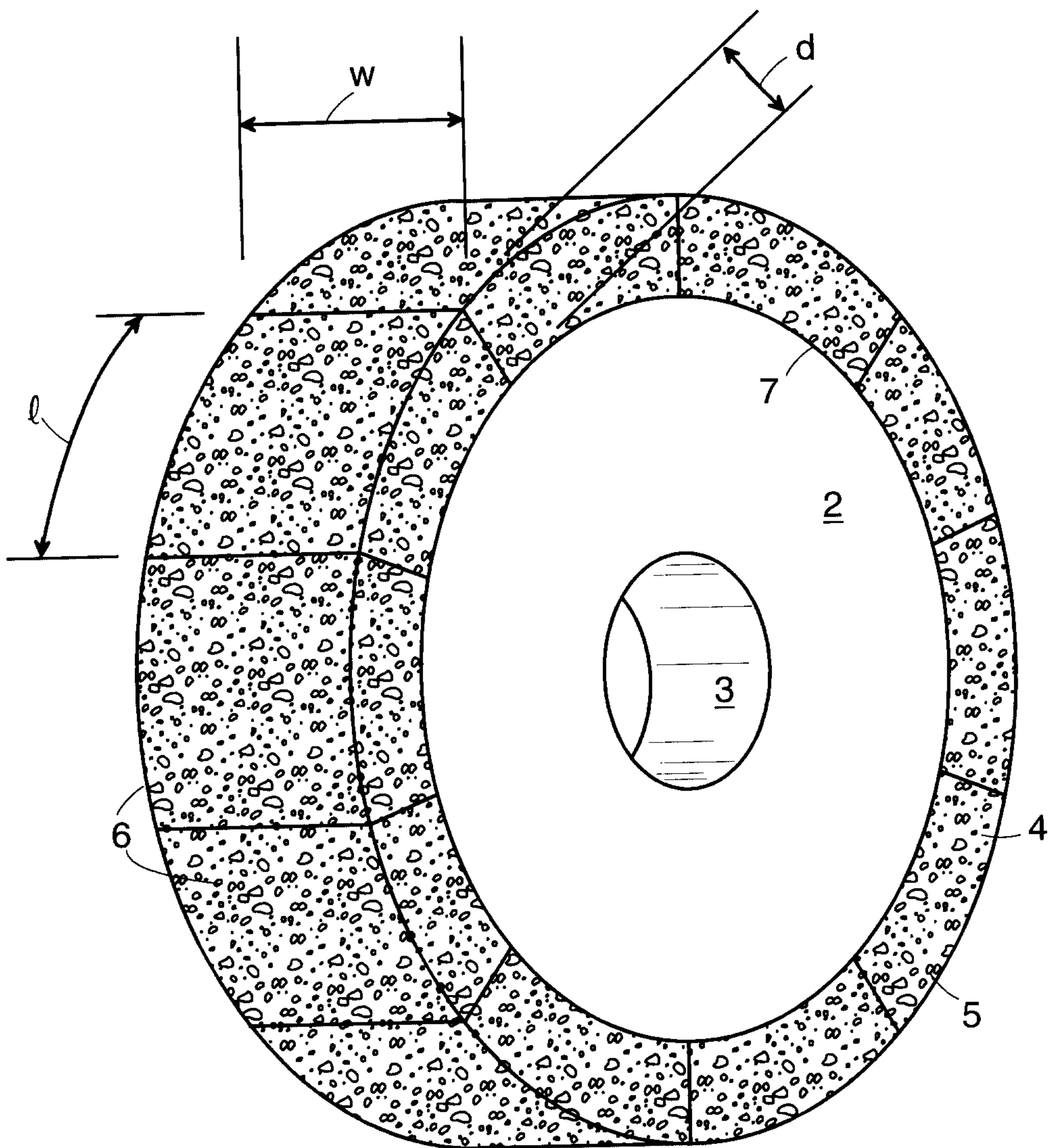


FIG. 1

