27792/89

APPLICATION ACCEPTED AND AMENDMENTS ALLOWED 18-7-90

FORM 1 SPRUSON & FERGUSON

COMMONWEALTH OF AUSTRALIA

PATENTS ACT 1952

APPLICATION FOR A STANDARD PATENT

Dana Corporation, incorporated in Virginia, of 4500 Dorr Street, Toledo, Ohio, UNITED STATES OF AMERICA, hereby apply for the grant of a standard patent for an invention entitled:

Low Grade Material Axle Shaft

which is described in the accompanying complete specification.

Details of basic application(s):-

Basic Applic. No:	Country:	Application Date:
166178	US	10 March 1988

The address for service is:-

Spruson & Ferguson
Patent Attorneys
Level 33 St Martins Tower
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Sydney New South Wales Australia

DATED this SIXTH day of JANUARY 1989

Dana Corporation

By: M.J. Anderson

Registered Patent Attorney

TO: THE COMMISSIONER OF PATENTS
OUR REF: 83060
S&F CODE: 53160

004632 06/01/89

5845/2

SPRUSON & FERGUSON

COMMONWEALTH OF AUSTRALIA
PATENTS ACT 1952

DECLARATION IN SUPPORT OF A CONVENTION APPLICATION FOR A PATENT

In support of the Convention Application made for a patent for an invention entitled:

Low Grade Material Axle Shaft

I/~~We~~, ... Sue A. Griffin .....
[full name of declarant(s)]

care of Dana Corporation, 4500 Dorr St., Toledo, Ohio 43615 USA .....
[full address of declarant(s) - not post office box]

.....

.....

do solemnly and sincerely declare as follows:-

1. I am/~~We are~~ authorised by Dana Corporation, the applicant for the patent to make this declaration on its behalf.

2. The basic application as defined by Section 141 of the Act was made in United States of America on 10 March 1988 by Kent E. Davison and Ervin M. Rickert.

3. Kent Eugene Davison; and Ervin M. Rickert; of Route 8, Morsches Road, Columbia City, Indiana 46725; and 1922 Otsego Dr., Ft. Wayne, Indiana 46825, both in United States of America [respectively], are the actual inventors of the invention and the facts upon which the applicant is entitled to make the application are as follows:

   The said applicant is the assignee of the actual inventors.

4. The basic application referred to in paragraph 2 of this Declaration was the first application made in a Convention country in respect of the invention the subject of the application.

DECLARED at Toledo, OH this 19th day of January 19 89

Sue A. Griffin
.....
Signature of Declarant
Sue A. Griffin, Assistant Secretary
Dana Corporation

TO: THE COMMISSIONER OF PATENTS
AUSTRALIA

JTA:541T

(12) PATENT ABRIDGMENT (11) Document No. AU-B-27792/89
(19) AUSTRALIAN PATENT OFFICE (10) Acceptance No. 602477

(Modified Examination)

(54) Title
LOW GRADE MATERIAL AXLE SHAFT

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(72) Inventor(s)
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(57) Claim

1. In a method of forming an axle shaft with a minimum body diameter of 1.70 inches from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.20% chromium and the balance iron and other materials not affecting the hardenability of the steel, with a critical diameter of 2.1-2.6", the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof, machining said ends to a final configuraton and dimension, and induction hardening said shaft without any intervening annealing or normalizing after forging.

17. In the method of forming an axle shaft having a minimum body diameter of 1.70" and a minimum rated capacity of 30,000 pounds from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.23% chromium, 0.025-0.05% aluminum, 0-0.15% copper, 0-0.20% nickel, 0-0.15% molybdenum, 0.020-0.045% sulfur and 0.035% maximum phosphorous and the balance iron with a critical diameter of 2.1-2.6", the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof; machining said ends to a final configuration and dimension, induction hardening said shaft without any intervening annealing or normalizing after forging, and tempering said shaft.

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19. In a method of forming an axle shaft with a body diameter between 1.70 and 2.05 inches and a rated capacity between 30,000 and 44,000 pounds from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.20% chromium and the balance iron and other materials not affecting the hardenability of the steel, the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof, machining said ends to a final configuration and dimension, and induction hardening said shaft without any intervening annealing or normalizing after forging.

S & F Ref: 83060

FORM 10

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COMPLETE SPECIFICATION

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Complete Specification Lodged:
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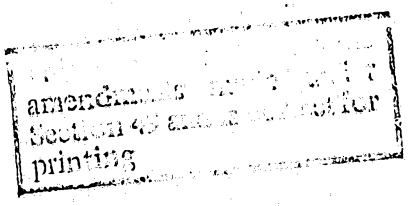

Priority:

Related Art:

Name and Address of Applicant: Dana Corporation
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Toledo Ohio
UNITED STATES OF AMERICA

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Complete Specification for the invention entitled:

Low Grade Material Axle Shaft

The following statement is a full description of this invention, including the best method of performing it known to me/us

LOW GRADE MATERIAL AXLE

BACKGROUND OF THE INVENTION

1. FIELD OF THE INVENTION:

This invention relates to a new alloy composition, and, more particularly, to a new alloy composition and a method of forming drive axle shafts having a minimum diameter of 1.70 inches and a minimum capacity of 30,000 pounds.

2. DESCRIPTION OF THE PRIOR ART:

One of the most important considerations in selection or formulation of a carbon steel alloy for producing a high strength axle shaft is controlling the hardenability of the alloy. Proper hardenability in turn depends upon having an alloy with the proper carbon content, that is, a high enough carbon content to produce the minimum surface hardness measured on the Rockwell C Scale, R_c, and a low enough carbon content to be able to control the hardening process without exceeding maximum desired surface hardness or penetration of hardness into the core of the axle shaft. Hardenability establishes the depth to which a given hardness penetrates, which can also be defined as the depth to which martensite will form under the quenching conditions imposed, that is, at a quenching rate equal to or greater than the critical cooling rate.

Modern day hardenability concepts had their origin around 1930 in the research laboratories of United States Steel Corporation. In 1938 the Jominy Test came into being in the laboratories of General Motors as a means of determining hardenability. The test consists of quenching the end of a one inch round bar and deterimining the

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hardness, R_c, at 1/16" intervals along the bar starting at the quenched end. Grossmann at United States Steel pioneered the calculation of hardenability presenting it in a paper published in the Trans Am. Inst. Mining Met. Engrs., V. 150, 1942, pp. 227-259. Grossmann postulated that hardenability can be based on a bar of ideal diameter, DI, defined as a diameter in inches of a bar that shows no unhardened core in an ideal quenching condition, or further defining it to produce a 50% martensite structure at the center of the bar. The calculation of DI is presented in many metallurgical texts, for example, in "Modern Metallurgy for Engineers" by Frank T. Sisco, second edition, Pitman Publishing Company, New York, 1948 or in the text "The Hardenability of Steels - Concepts, Metallurgical Influences and Industrial Applications" by Clarence A. Siebert, Douglas V. Doane and Dale H. Breen published by the American Society of Metals, Metals Park, Ohio, 1977.

Basically, the critical diameter in inches, DI, is calculated by multiplying together the multiplying factor, MF, for all the elements found in a particular steel either as residuals or purposly added to the steel. For example, a SAE/AISI 1040 carbon steel, using the Grossmann data would have the following multiplying factors for a typical percentage as follows:

Carbon 0.39% MF, =0.23; manganese 0.68%, MF 3.27; silicon 0.11%, MF = 1.08; nickel 0.12%, MF = 1.05, chromium 0.04%, MF = 1.09; molybdenum, 0.02%, MF = 1.06. The ideal diameter is then calculated as DI =0.23 x 3.27 x 1.08 x 1.05 x 1.09 x 1.06 equals 0.98 inches. This would mean that an

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ideal diameter with a perfectly quenched steel would be 0.98 inches; thus, to insure proper hardenability, the maximum diameter of this shaft would be something less than 0.98 inches probably of the order of 3/4".

By utilizing the DI calculations, it can be determined what can be the maximum diameter of the shaft of a particular composition that will have a desirable hardenability profile with 50% Martensite at the center of the core.

It is well established that high manganese carbon steel compositions provide satisfactory hardenability because the manganese allows the carbon to penetrate into the core in solution with the iron to produce the desired martensite as quenched. A SAE/AISI 1541 medium carbon steel having 0.36-0.44% C and 1.35-1.65% Mn will have adequate hardenability for axle shafts with a maximum diameter of less than 1.7 inches to produce a load carrying capacity of less than 30,000 pounds. Axle shafts with a body diameter greater than 1.7 inches for axle load carrying capacities of 30,000, 34,000, 38,000 or 44,000 pounds, cannot be produced with a 1541 steel because the manganese cannot produce a desirable hardness profile into the core of the shaft resulting in at least 50% martensite at the center. A satisfactory solution to this problem is obtained by the use of trace percents of boron in the SAE 1541 steel denoting the steel as SAE 15B41. Such boron percentages, are typically in the range between 0.0005 -0.003% boron.

With the use of boron in the steel to produce the proper hardenability profile, the risk of retaining residual

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stresses after forging the usual spline at one end and flange at the other end of the axle shaft is present. This can greatly reduce the fatigue life of the shaft, producing premature failure by stress cracking. This is true because the boron will precipitate out into the grain boundaries as boron nitride to product brittleness. To counteract this the boron nitride is driven out of the grain boundaries when the axle shafts are normalized by heating to above the transformation temperature and air cooling. This is a time consuming and very expensive process.

SUMMARY OF THE INVENTION

The present invention is directed to the formulation of an alloy which has good hardenability so that axle shafts of 1.70 - 2.05 inch body diameters can be formed as drive axles with a load carrying capacity from 30,000 to 44,000 pounds. With an alloy steel consisting essentially of 0.40~~-0.48%~~ ~0.48% carbon, 1.35-1.61% manganese, 0.16-~~.30%~~ 0.30% silicon, 0-.23% chromium and the balance iron and other materials not affecting the hardenability of the steel, the axle shaft may be formed by forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof, machining the ends to final configuration and dimension, and induction hardening the shaft without any intervening annealing or normalizing after forging.

The alloy steel should contain between 0.025 and 0.05% aluminum to promote a grain size of the steel of ASTM 5 to 8 further assuring the proper hardenability.

The alloy typically will contain 0-.15% copper, ~~0-0.20%~~ 0-0.2% nickel, 0- .15% molybdenum, 0.02-~~.045%~~ 0.045% sulfur and

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0.035% maximum phosphorus.

The axle shaft should have a critical diameter between 2.1 and 2.6 inches.

The axle shaft should also have a maximum hardness at its center of R_c 35 with a surface hardness after tempering of R_c 52 to R_c 59 and a maximum hardness of R_c 40 at a distance of 0.470 inches measured from the surface. This hardness profile should exist when the foregoing composition and critical diameter criteria have been met.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS OF THE INVENTION

In the search for high strength steel alloys having good hardenability, small changes in the chemistry can have a dramatic effect on the ability of the alloy to meet the design criteria, and the method of forming the product, such as an axle shaft, can be substantially changed. An example of such a change in chemistry and the resulting change in product performance and method of forming is envolved in the manufacture of axle shafts. In the forming of automotive axles, primarily for passanger cars and light trucks where the body diameter does not exceed 1.70", the axle shaft can be manufactured with a 1541 alloy steel which will meet hardenability specifications without normalizing or annealing. With axle shafts of 1.70 - 2.05 inch body diameters used in axles with axle load carrying ratings from 30,000 to 44,000 pounds, if a 1541 alloy is used, there will be insufficient hardenability or depth of hardening and the axle shaft will have an unsatisfactory life expectancy. The standard axle shafts in this range of body diameters and

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capacities have heretofore been manufactured utilizing a 15B41 alloy steel which has trace amounts of boron in the steel to increase the depth of hardening to produce the required strengths with adequate fatigue life.

The chemical composition for SAE/AISI 1541 is as follows:

ELEMENT	ANALYSIS RANGE MAXIMUM % BY WEIGHT
Carbon	.36 - .44
Manganese	1.35 - 1.65
Silicon	.15 - .35
Sulfur	.050 max.
Phosphorus	.040 max.

The analysis for the boron added steel 15B41 is the same as presented in the above table with the addition of 0.0005 -0.003 percent boron. With the 15B41 high manganese carbon steel with boron added, axle shafts in industry standard strengths can be produced having adequate fatigue life with the following diameters:

AXLE RATING POUNDS	BODY DIAMETER INCHES
30,000	1.72
34,000	1.84
38,000	1.91
44,000	2.05

While the 15B41 steel composition provides proper hardenability at the required strength levels, the method of manufacturing the axle shaft becomes more complex.

Typically the axle shaft is manufactured from bar

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stock having the desired body diameter. After cutting the rod to the desired axle shaft length, the ends of the shaft are forged to produce a spline at one end and a flange at the other end. The configuration and final dimensions of the spline and flange are determined by the manufacturer or tailored to specification for the original equipment manufacturer or for the replacement parts market. The spline and flange are machined to this final dimension after the forging operation. The hardening of the shaft is accomplished by heating it after machining to above the upper critical temperature and water quenching. Preferably this is accomplished by induction heating either in a one-shot process where the axle is rotated between centers and the induction coil is stationary or by the induction scanning process where the axle shaft is rotated and the induction coil is moved. A rapid water quench produces the desired hardness gradient. The shaft is finally tempered in a continuous tempering furnace to relieve residual stresses, which can reduce the hardness values by a couple points on the Rockwell C scale.

With the use of 1541 for the smaller diameter axle shafts, the foregoing method of forming the axle shaft is followed without the use of any intermediate heat treating between the forging and the machining steps. With the use of 15B41, the boron introduces grain boundary stresses. To reduce these stresses, it is necessary to anneal or normalize the axle shaft after the forging operation and prior to the machining and hardening steps. An annealing or normalizing process is a time consuming and expensive

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procedure, thus increasing the cost of the axle shaft.

Other steel alloys which meet the strength and hardenability requirements such as 50B50 are more expensive and also require normalizing after forging.

In working with various alloy compositions and evaluating the hardenability by performing a hardness profile across the diameter much like the Jominy lengthwise profile, it has been found that a fully adequate hardenability profile will prevail if the shaft has a minimum yield strength of 110,000 pounds per square inch. This will also assure a more than adequate fatigue life. Knowing that chromium, like manganese, can extend the hardness penetration into the core of a shaft, formulations with different manganese and chromium compositions were tested. Too high of a chromium content also tends to produce a steel with too much hardenability. Also if the manganese is on the high side when the carbon is also on the high side, there is a tendency to harden to too great of a degree at the core, causing reduced fatigue life. Starting with the aforementioned composition of a 1541 steel, and partially ignoring the general teaching that increasing both the manganese and the carbon content will increase the hardness penetration or hardenability, it was found that shifting the carbon range slightly higher and lowering to a small degree the higher manganse limit coupled with a judicious addition of a small percent of chromium, a new steel alloy could be formulated which will provide a more than adequate case depth. The chemical composition for this SAE/AISI 1541M steel alloy is as follows:

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ELEMENT	ANALYSIS RANGE OR MAXIMUM PERCENT BY WEIGHT
Carbon	.40 - .48
Manganese	1.35 - 1.61
Chromium	0 - .23
Silicon	.16 - .30
Sulphur	.020 - .045
Phosphorus	.35 max.
Molybdeum	0 - .15
Nickel	0 - .20
Copper	0 - .15

The nickel and copper components of the new 1541M alloy steel are residual percentages which are normally found in melts in this country. Likewise the silicon, suplhur and phosphorus contents are those commonly imposed and accepted for standard carbon alloy steel compositions. Aluminum in the range in 0.025 - 0.05% range can be utilized to assure a fine grain size of ASTM5-8.

It has also been found that if the ideal critical diameter, DI, range is also specified, there is additional assurance that an axle shaft formed by the method which eliminates an annealing or normalizing step after forging, will more than adequately meet the strength and fatigue requirements, and hard ss profiles will not have to be taken to assure this. For the actual diameter range of 1.70 - 2.05 inches, this range is DI = 2.1 - 2.6 inches. The imposition of this ideal diameter range requirement eliminates the rare possibility that all of the elements could be on the minimum side or the maximum side which could

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produce an inadequate life expectancy.

In calculating the DI, the MF for carbon, manganese, nickel, chromium, molybdeum, copper, and silicon is utilized. The multiplying factor MF for aluminum would be 1.0 if it is absent or present in the quantity mentioned above to assure a fine grain size range. The multiplying factors for phosphorus and sulphur are not used in this calculation since they cancel each other out in the composition range given, that is, the factor for phosphorus is about 1.03 and the factor for sulphur is about 0.97.

In formulating the critical diameter range of 2.1 - 2.6 inches, Caterpillar specification 1E - 38 is used to determine the multiplying factor for a given element percentage. This specification is found in the publication "Hardenability Prediction Calculation for Wrought Steels: by Caterpillar, Inc. incorporated herein by reference. If all of the elements were at their minimum or maximum values the corresponding multiplying factors would be as follows:

	LOWEST VALUE		HIGHEST VALUE	
	%	MF	%	MF
Carbon	.40	.213	.48	.233
Manganese	1.35	5.765	1.61	7.091
Chromium	0	1.0	.23	1.497
Silicon	.16	1.112	.30	1.21
Molybdenum	0	1.	.15	1.45
Nickel	0	1.	.20	1.073
Copper	0	1.	.15	1.06

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If the multiplying factors for the lowest values of all elements are multiplied together the DI = 1.3 inches which would be inadequate to meet the additionally imposed minimum DI of 2.1 inches. Likewise if all the highest percentage multiplying factors are multiplied together the DI would be 4.9 inches again beyond the maximum allowable DI of 2.6 inches.

Alternately or additionally, the hardenability can be specified in terms of a minimum hardness gradient, a maximum core hardness, a maximum hardness at a given depth, and a range of surface hardnesses. The requirements for a more than adequate strengh and fatigue life would be a maximum core hardness of R_c 35, a maximum hardness of R_c 40 at a depth of .47 inches and a surface hardness range of R_c 52 to R_c 59. The minimum hardness gradient would be as follows:

DISTANCE IN INCHES	Rc
.050"	52
.100"	52
.200"	52
.300"	45
.400"	33
.500"	22

The foregoing hardenability specification takes into account the fact that the axle shaft is tempered after induction hardening at a temperature not to exceed 350°F for from 1 1/2 to 2 hours. An additional requirement to assure elimination of residual stresses by the tempering is that it be conducted within two hour or the induction hardening.

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The claims defining the invention are as follows:

1. In a method of forming an axle shaft with a minimum body diameter of 1.70 inches from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.20% chromium and the balance iron and other materials not affecting the hardenability of the steel, with a critical diameter of 2.1-2.6", the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof, machining said ends to a final configuraton and dimension, and induction hardening said shaft without any intervening annealing or normalizing after forging.

2. The method according to claim 1 wherein the alloy steel further contains 0.025-0.05% aluminum and the grain size of the steel is ASTM 5 to 8.

3. The method according to claim 1 wherein said steel contains 0-0.15% copper, 0.020-0.20% nickel, 0-0.15% molybdenum, ~~0.020-0.045%~~ 0.020-0.045% sulfur and 0.035% maximum phosphorus.

4. The method according to claim 1 wherein said axle shaft has a rated capacity between 30,000 and 44,000 pounds with a nominal shaft body diameter between 1.70 and 2.05 inches.

5. The method according to claim 4 wherein said axle has a rated capacity of 30,000, 34,000, 38,000, or 44,000 pounds.

6. The method according to claim 3 wherein said critical diameter is calculated by utilizing the multiplying factors for the carbon, manganese, nickel, chromium, molybdenum, copper and silicon.

7. The method according to claim 1 further including the step of tempering said shaft after hardening.

8. The method of according to claim 8 wherein said shaft is tempered at a temperature not to exceed 350° F. for a time between 1 1/2 to 2 hours.

9. The method of according to claim 8 wherein said tempering step is commenced within two hours of said induction hardening step.

10. The method according to claim 7 wherein said axle shaft has a maximum hardness at its center of R_c 35.

11. The method according to claim 8 wherein said axle shaft has a maximum hardness of R_c 40 at a distance of 0.470" measured from the surface.

12. The method according to claim 7 wherein said axle shaft has a surface hardness after tempering of R_c 52 to R_c 59.

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13. The method according to claim 12 wherein said axle shaft has a minimum hardness gradient at distances measured from the surface of R_c 52 at 0.050", R_c 52 at 0.100", R_c 52 at 0.200", R_c 45 at 0.300", Rc 33 at 0.400", and R_c 22 at 0.500".

14. The method according to claim 1 wherein said induction hardening step is accomplished as a single shot introduction process with a water quench.

15. The method according to claim 12 wherein the core of axle shaft body is unaffected by said induction hardening step and the microstructure of the hardened area is approximately 90% martensite and 10% bainite.

16. The method according to claim 1 wherein said axle shaft has at least a 50% martensite structure at its center after induction hardening.

17. In the method of forming an axle shaft having a minimum body diameter of 1.70" and a minimum rated capacity of 30,000 pounds from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.23% chromium, 0.025-0.05% aluminum, 0-0.15% copper, 0-0.20% nickel, 0-0.15% molybdenum, 0.020-0.045% sulfur and 0.035% maximum phosphorous and the balance iron with a critical diameter of 2.1-2.6", the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof; machining said ends to a final configuration and dimension, induction hardening said shaft without any intervening annealing or normalizing after forging, and tempering said shaft.

18. The method according to claim 17 wherein the grain size of said steel is ASTM 5-8, the maximum hardness at its center is R_c 35, and the surface hardness after tempering is R_c 52-R_c 59.

19. In a method of forming an axle shaft with a body diameter between 1.70 and 2.05 inches and a rated capacity between 30,000 and 44,000 pounds from an alloy steel consisting essentially of 0.40-0.48% carbon, 1.35-1.61% manganese, 0.16-0.30% silicon, 0-0.20% chromium and the balance iron and other materials not affecting the hardenability of the steel, the steps of forging the ends of a shaft to form a spline at one end thereof and a flange at the other end thereof, machining said ends to a final configuration and dimension, and induction hardening said shaft without any

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intervening annealing or normalizing after forging.

DATED this TWENTY-FIRST day of FEBRUARY 1990

Dana Corporation

Patent Attorneys for the Applicant
SPRUSON & FERGUSON

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