ternary ceramic thermal spraying powders and their thermal spraying coatings with superior properties. Powders contain at least 30% by weight ternary ceramic, at least 20% by weight binary molybdenum boride, at least one of the binary borides of Cr, Fe, Ni, W and Co and a maximum of 10% by weight of nano and sub-micro-sized boron nitride. The primary crystal phase of the manufactured thermal sprayed coatings from these powders is a ternary ceramic, while the secondary phases are binary ceramics. The coatings have extremely high resistance against corrosion of molten metal, extremely thermal shock resistance and superior tribological properties at low and at high temperatures.
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TERNARY CERAMIC THERMAL SPRAYING POWDER AND COATING METHOD

[0001] This application claims priority to, and the benefit of, U.S. Provisional Patent Application No. 61/798,032, filed March 15, 2013 with the U.S. Patent Office, which is hereby incorporated by reference.

STATEMENT REGARDING FEDERALLY SPONSORED RESEARCH OR DEVELOPMENT

[0002] This invention was made with government support under contract DE-SC0003454 awarded by the U.S. Department of Energy (DOE). The government has certain rights in this invention.

BACKGROUND OF THE INVENTION

[0003] The present invention relates generally to spraying powders, and more specifically, to thermal spray coatings using spray powder.

DESCRIPTION OF THE RELATED ART

[0004] It has become apparent that thermal sprayed coatings used for high temperature applications and in molten metal corrosion atmospheres require high corrosion resistance, thermal shock resistance (resistance to spallation during rapid change in temperature), high wear resistance at low and high temperatures, low friction coefficients and low porosity. With regard to manufacturing and overall performance, current tungsten carbide-cobalt (WC-Co) High Velocity Oxygen Fuel (HVOF) coatings have spallation problems under point loads, corrosion limitations, and are expensive. For example, WC-Co thermal spray coatings are currently used as wear resistant coatings on steel rolls submerged in a molten zinc and aluminum bath, but such coatings dissolve within 48 hours. Accordingly, there is a need to improve roll lifespan beyond its current life. It is also noted that the cost of frequently replacing these rolls is a significant portion of the cost of galvanized metals. Alternate manufacturing materials have drawbacks as well. For example, weld overlays require extensive machining, and still have limited wear and corrosion performance. Similar issues occur in deep drawing dies and die cutters, which wear down and then lead to sticking due to improper clearance as well as in die casting and extrusions.

6,238,807, and US 7,862,911 disclose either thermal sprayed powders or thermal sprayed coatings containing molybdenum boride, nickel, chromium and tungsten. However, the resulting coatings do not demonstrate superior thermal shock resistance, wear and corrosion resistance under molten metal environment.

[0006] Therefore, there is a need to develop a spray powder to form coatings providing improved resistance against molten metal corrosion environment, superior thermal shock resistance and high wear resistance at high temperatures.

SUMMARY OF THE INVENTION

[0007] Particular embodiments of the present invention comprise a thermal spray powder comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about 20-70 % by weight, where Metal is Co, Ni or Fe; one or more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

[0008] Further embodiments of the present invention comprise a method of manufacturing a thermal spray powder, the steps comprising: providing a composition comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about 20-70 % by weight, where Metal is Co, Ni or Fe; one or more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

[0009] Yet further embodiments of the present invention comprise a thermal sprayed coating formed from a thermal spray powder comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about 20-70 % by weight, where Metal is Co, Ni or Fe; one or more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

[0010] Other embodiments of the present invention comprise a method of forming a thermal sprayed coating on a substrate, the steps comprising: providing a thermal spray powder comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about 20-70 % by weight, where Metal is Co, Ni or Fe; one or
more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co; and, thermally spraying the thermal spray powder onto the substrate.

[0011] The foregoing and other objects, features and advantages of the invention will be apparent from the following more detailed descriptions of particular embodiments of the invention, as illustrated in the accompanying drawings wherein like reference numbers represent like parts of the invention.

BRIEF DESCRIPTION OF THE DRAWINGS

[0012] FIG. 1 is an x-ray diffraction scan showing a first pattern of a particular inventive embodiment of a sintered ternary ceramic powder and a second pattern of a particular embodiment of a sprayed coating of the powder.

[0013] FIG. 2 is a schematic view of a thermal spray coating operation, where a thermal spray source is employing thermal spray powder to apply a coating upon a substrate in accordance with a particular embodiment of the invention.

DETAILED DESCRIPTION OF PARTICULAR EMBODIMENTS

[0014] Ceramics have shown the high resistance against the corrosive attacks, such as against the corrosive attacks of molten Zn, for example. Ceramics combine high wear resistance and hardness with high chemical stability at high temperatures - even in molten metal media. For example, ceramic-based coatings are beneficial for use with zinc pot rolls. Still, it has been found that, in particular situations, to provide sufficient resistance to corrosive attack, ceramic-based coatings should be sufficiently dense to provide almost zero porosity (less than 1% porosity) and thereby prevent penetration of the corrosive material to the substrate through porosity. Furthermore, to provide sufficient resistance to corrosive attack in particular instances, it is desirable to provide ceramic-based coatings having an adjusted CTE to the substrate to eliminate cracking and spallation at high temperatures.

[0015] Accordingly, the present invention provides improved thermal sprayed powder consisting of ternary and binary ceramics capable of manufacturing thermal sprayed coatings with improved properties against molten metal environments, superior thermal shock resistance, high wear resistance and low friction properties under low and high temperature conditions, where the primary crystal phase of the powder and of the thermal
sprayed coating is a multi-ceramic containing ternary and binary ceramics.

[0016] In particular embodiments, the invention comprises a thermal spray powder, and in other embodiments a resulting thermal spray coating, comprising:

- one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo2-B2, or Meta^2-MO2-B2 (which includes any combination thereof) from about 20-70% by weight (that is, from about 20% to 70%), where Metal is Co, Ni or Fe;
- one or more binary molybdenum borides, from substantially 0-60% (up to substantially 60%) by weight; and,
- one or more metal binary borides from substantially 0-50% (up to substantially 50%) by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

[0017] In alternative embodiments of the powder and coating, the one or more binary molybdenum borides provided are from substantially 20-60% (that is, substantially 20% to 60%) by weight and/or the one or more metal binary borides range provided are from substantially 10-50% (that is, substantially 10% to 50%) by weight.

[0018] In further embodiments, with regard to any embodiment of the thermal spray powder, the thermal spray powder also includes:

- one or more nano and submicro-sized reinforced solid lubricants from substantially 0-10% (that is, substantially from 0 to 10%) by weight, the one or more nano and submicro-sized reinforced solid lubricants comprising BN, WS2 or MoS2.

[0019] With regard to said powder and the resulting coatings, ternary borides of Metal-Mo-B, Metal-Mo2-B2, and/or Metal2-Mo2-B2 (where Metal is Co, Ni or Fe) exhibit superior lubrication properties due to the presence of borides, a high hardness (Vicker's hardness number (HV 300) of 500-900), stability against high temperatures, and corrosion attacks due to the ternary ceramic structures. In sum, these ternary borides provide enhanced resistance to wear and corrosion. Furthermore, the binary molybdenum borides exhibit high temperature stability and corrosion resistance against molten metal. Moreover, metal binary borides comprising Cr, Fe, Ni, W, Co each generally provide resistance at high temperature, serve as a binder and offer low friction due to solid lubricant effect. Nano and submicro-sized reinforced solid lubricants comprising BN, WS2 or MoS2 reduce friction and operate as a release agent. As used herein, the
abbreviated chemical elements noted in this application are used in accordance with industry-accepted nomenclature. For example, B = boron, C = carbon, Co = cobalt, Cr = chromium, Fe = iron, Mo = molybdenum, N = nitrogen, Ni = nickel, S = sulfur, W = tungsten, and Zn = zinc.

[0020] A particular aspect of the present invention comprises a method of manufacturing a thermal spray powder, the steps comprising: providing a composition comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about substantially 20-70% by weight, where Metal is Co, Ni or Fe; one or more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

[0021] In particular embodiments, manufacturing a thermal spray composite powder includes mixing, milling, agglomeration, and sintering. In particular embodiments, a method of manufacturing a thermal spray composite powder comprises a step of milling a composition mixture to form a milled composition mixture, a step of agglomerating the milled composition to form an agglomerated composition, and then a step of sintering the agglomerated composition to form a thermal spray composite powder. It is appreciated that one or more known machine(s) or process(es) may be employed for mixing, milling, agglomerating, and sintering.

[0022] Milling generally refers to a step of grinding, pulverizing, breaking down, or otherwise reducing the composition mixture into a finer composition comprising particulates or constituents of smaller size, and which may be achieved by any known machine(s) or process(es) to achieve this purpose. For example, in particular embodiments, the step of milling is performed using a ball milling process.

[0023] In particular embodiments, the step of milling includes milling a composition mixture comprising the following:

one or more metals from substantially 0-80% (up to substantially 80%) by weight, where the one or more metals are selected from Mo, Cr, Fe, Ni, W, and Co; and

one or more binary boride ceramics from substantially 0-80% (up to substantially 80%) by weight, where the one or more metal binary borides are selected from Mo, Cr, Fe, Ni, W, and Co.
[0024] In further embodiments, the composition mixture for milling also includes one or more solid lubricants from zero to 10% by weight. It is appreciated that any known lubricant may be employed. For example, the lubricant may comprise one or more nano and submicro-sized reinforced solid lubricants comprising BN, WS₂ or MoS₂.

[0025] Agglomeration generally refers to a process of adhering or joining particles of a composition to form a composition containing agglomerates. It is appreciated that, in particular embodiments, the step of agglomerating includes adding one or more ingredients to the milled composition mixture to form a modified milled composition mixture having a desired characteristic. For example, in particular embodiments, the modified milled composition mixture is a wet composition mixture, where a portion of the wet composition mixture is a liquid and where the desired characteristic of the wet mixture comprises a desired viscosity. Exemplary ingredients that may be added to the milled composition mixture include one or more binders, one or more surfactants, and/or water.

[0026] Subsequent to adding one or more ingredients to the milled composition mixture, if performed, the step of agglomerating transforms the milled composition mixture or the modified milled composition to an agglomerated composition, such as an agglomerated dry composition comprising a particulate or pulverulent (powdery) composition. This may occur by any known process. For example, this transformation may be achieved by a drying process (as a step of drying the modified milled composition) - such as when the composition mixture being transformed is a wet composition mixture. Exemplary drying processes include spray drying, freeze drying, drum drying, and pulse combustion drying. In particular embodiments, a drying process is selected and performed to create spherical, free flowing agglomerated powders.

[0027] Subsequently, the agglomerated composition undergoes a sintering process in a step of sintering, where the application of heat to the agglomerated composition joins or fuses at least certain particles of the agglomerated composition. Additionally, in particular embodiments, before the step of sintering, the agglomerated composition are debounded in a step of debinding the agglomerated composition. Furthermore, after the step of sintering, an additional step of agglomerating is performed in particular variations of these methods.
[0028] In particular embodiments, the sintering process is performed at temperatures not lower than 850°C and not higher than 1300°C for substantially 1 to 5 hours. In other variations, the sintering process is performed from substantially 900°C to 1000°C for substantially 2 to 4 hours. In a more specific embodiment, the sintering process is performed at substantially 900°C for substantially 3 hours. In particular embodiments, the material employed to create the powder (which includes the one or more ternary boride ceramics, binary molybdenum borides, metal binary borides, and any nano and submicro-sized reinforced solid lubricants) has an average particle size at least equal to 0.1 μη and no larger than 10 μη. The powder may be characterized as having these particle sizes up to the agglomerating and sintering operations (that is, through the mixing and milling processes). It is understood that the particle size may be less than 0.1 μη in other embodiments. To ready the powder for thermal spraying, agglomeration and sintering of the powder is performed to achieve any particle size sufficient for the particular spraying operation to be employed.

[0029] In another aspect of this invention, a method of forming a thermal sprayed coating using the above mentioned powder is provided. Particular embodiments comprise a method of forming a thermal sprayed coating on a substrate, the steps comprising: providing a thermal spray powder comprising: one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo₂B₂, or Metal₂Mo₂B₂ from about 20-70 % by weight, where Metal is Co, Ni or Fe; one or more binary molybdenum borides from substantially 0-60% by weight; and, one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co; and thermally spraying the thermal spray powder onto the substrate. It is understood that any powder described herein may be employed in other embodiments of such methods. The substrate may comprise any desired object comprised of any material. For example, in particular embodiments, the substrate is formed at least partially of a metal, including without limitation steel.

[0030] Such methods of forming thermal sprayed coatings include spraying the thermal spray powders mentioned above (in a post-agglomeration and post-sintering state) in the range between 5 and 125 μη (micrometers), and in particular embodiments between 10 to 45 μη. Further, thermal spraying is achieved using any known machines for achieving this purpose (which is interchangeable referred to herein as a "spray source"). In particular embodiments, plasma thermal spraying, flame spraying, detonation spraying, cold gas spraying, arc spraying, or high velocity oxygen fuel spraying (each being
referred to as a "spraying source") are employed to thermally spray the powder. With reference to an exemplary embodiment in FIG. 2, a spray source 30 is shown spraying a coating 10 onto a substrate 20 using the thermal spray powder disclosed herein.

[0031] In particular embodiments, the thermal sprayed coatings consist of a mixture of crystal phases of ternary and binary ceramics with low amounts of amorphous phases. In particular, the primary phase is a ternary ceramic of the Metal-Mo-B, Metal-Mo$_2$B$_2$, and/or Metal$_2$Mo$_2$B$_2$ system, where Metal is Co, Fe or Ni, while secondary phases are binary borides and boron nitride. The coatings contain no less than substantially 30% by weight and no more than substantially 70% by weight primary phase.

[0032] Additionally, in particular embodiments, the thermal sprayed coating includes one or more solid lubricants, including without limitation, for example, boron nitride, from zero to 10% by weight. If the solid lubricant content is too high, such as 20% by weight or more, the thermal shock resistance and adhesion of the coatings will be reduced considerably.

[0033] In certain applications, the coating has a thickness of substantially 50 to 300 µη, and in particular embodiments, a thickness of substantially 100 to 200 µη. If the coating is thinner, the coating exhibits unsatisfactory corrosion protection. If the coating is thicker, thermal shock resistance reduces considerably.

[0034] These coatings will be applied in low and high temperature applications, where low friction and high wear resistance in addition to superior corrosion resistance against molten metal are desired. Additionally, these coatings are applicable for all wear and corrosion resistance applications.

[0035] The ternary boride ceramics discussed above were tested and evaluated. In particular, these powders and coatings were found to exhibit high thermal stability since less transformation has been seen between powder and thermal sprayed coatings. For example, with reference to FIG. 1, an x-ray diffraction (XRD) scan pattern of a sintered ternary ceramic powder is shown together with the XRD scan pattern of a sprayed coating of the powder. The main phases detected in the XRD patterns comprise ternary boride ceramics as well as small amounts of binary ceramics. In particular, the scan pattern of the powder consists of Mo$_2$FeB$_2$, FeMoB, MoB, FeB, Fe$_2$B and Fe$_3$B hard phases. Upon review, the coatings show broad diffraction peaks, which is indicative of amorphous phases. This can be explained by the high cooling rate of sprayed particles of approximately $10^6$ ks$^{-1}$ after HVOF spraying. In conclusion, the ceramics phases are
extremely hard and show high thermal stability since less transformation has been seen between powder and thermal sprayed coatings (see FIG. 1).

[0036] The coatings were also evaluated after concluding over 3000 hours of Zinc immersion testing, where metal substrates spray coated with an inventive ternary boride ceramics powder were immersed in a molten zinc bath. It was observed that the coatings remained hard (Vickers hardness of 895+61 HV0.3), and were observed to feature a dense, superfine structured and homogeneous morphology with a porosity less than 1%, where only very small sized pores on a submicron level were detectable.

[0037] Furthermore, the ternary boride ceramic coatings surpassed over 3000 hours testing without dissolving, spallation or deterioration, which was 8 times longer than commercially available WC-Co coatings and over 860 times longer than uncoated stainless steel.

[0038] In a first example, in manufacturing a thermal spray composite powder, a step of milling comprises ball milling micron-sized Mo powders (a metal constituent) with substantially 70 weight percent (Wt %) FeB (a binary boride ceramic) for substantially 4 hours in a ball mill to form a milled composition mixture comprising a composite powder. In performing a step of agglomerating, after performing the step of milling, an alcohol-based binder and one or more conventional surfactants is added with water to the milled composition mixture to obtain a modified milled composition mixture generally comprising a slurry having a viscosity substantially between 200 and 300 centipoises (a centipoise is equal to a centimeter-gram-second unit of dynamic viscosity equal to one hundredth of a poise). Subsequently, an agglomerated powder is formed by spray drying the milled composition using an atomizer to create spherical, free flowing agglomerated powders where approximately two thirds of the particles have a diameter of 35 microns. These powders were debounded for substantially two hours at temperatures substantially equal to or between 100°C and 200°C and sintered for substantially 3 hours at substantially 900°C to produce a ternary ceramic. These ternary ceramic powders may then be screened to substantially obtain a -230, +500 mesh cut to provide a ternary ceramic powder cut.

[0039] In evaluating this ternary ceramic powder cut, it was sprayed onto a grit-blasted 316L stainless steel substrate using a Sulzer Metco Diamond Jet 2600 HVOF unit mounted on a robot and fed by a powder feeder. The generated coating had a Vicker's
hardness number (HV 300) of about 800, an adhesive strength of greater than about 8,000 pounds per square inch gauge (pursuant to ASTM 622 bonded pin adhesion test). The coating had a 180 degree bend radius of less than about 0.5 inches, and a density of about 8.7 grams per cubic centimeter. In determining the bend radius, a coupon was bent around a 1/2 inch diameter mandrel into a "U" shape without spallation or breaking the coating. Similar results will provide by applying of such coatings to other known high strength steels.

[0040] This coating is at least suitable for applying in molten metal baths including Zn, Zn-alloys or molten steel.

[0041] In a second example, a thermal spray composite powder was produced by mixing the Mo, FeB and BN powders, while mixing is done sometime before or after sintering of ternary ceramics.

[0042] In performing a step of milling, the composite powder was prepared by ball milling micron-sized Mo powders with substantially 70 weight percent (Wt %) FeB and one to 10 wt. % BN for substantially 4 hours in a ball mill. A alcohol-based binder and one or more conventional surfactants are then added along with water to obtain a modified milled composition mixture having a viscosity substantially equal to or between 200 and 300 centipoises. An agglomerated composition comprising a powder is formed by spray drying the generated slurry using an atomizer to create spherical, free flowing agglomerated powders where approximately two third of the particles had a diameter of 35 microns. These powders are debounded for substantially two hours at temperatures substantially equal to or between 100°C and 200°C and sintered for substantially 3 hours for example at substantially 900°C to produce a ternary ceramic. These ternary ceramic powders are screened to substantially obtain a -230, +500 mesh cut.

[0043] In particular embodiments, these powders are mixed with substantially one to 10 wt% BN after ternary ceramic sintering. An agglomerated powder is formed by mixing with alcohol-based binder, conventional surfactants and water. After spray drying the powder, in particular embodiments, it is sintered. The powders are then screened to substantially obtain a -230, +500 mesh cut.
[0044] To evaluate this ternary ceramic powder cut, it was sprayed onto a grit-blasted 316L stainless steel substrate using a Sulzer Metco Diamond Jet 2600 HVOF unit mounted on a robot and fed by a powder feeder. The generated coating had a Vicker's hardness number (HV 300) of 500-700, an adhesive strength of greater than about 6,000 pounds per square inch gauge (pursuant to ASTM 622 bonded pin adhesion test). The coating had a 180 degree bend radius of less than about 0.5 inches, and a density of less than 8 grams per cubic centimeter.

[0045] While this invention has been described with reference to particular embodiments thereof, it shall be understood that such description is by way of illustration and not by way of limitation. Accordingly, the scope and content of the invention are to be defined by the terms of the appended claims.
What is claimed is:

1. A thermal spray powder comprising:
   - one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo$_2$B$_2$, or Metal$_2$Mo$_2$B$_2$ from about 20-70% by weight, where Metal is Co, Ni or Fe;
   - one or more binary molybdenum borides from substantially 0-60% by weight;
   - and,
   - one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

2. The thermal spray powder of claim 1, where the one or more binary molybdenum borides is substantially 20-60% by weight.

3. The thermal spray powder of any claim 1 to 2, where the one or more metal binary borides is substantially 10-50% by weight.

4. The thermal spray powder of claim 1, where the particle size of the one or more ternary boride ceramics, the one or more binary molybdenum borides, and the one or more metal binary borides is at least equal to 0.1 $\mu$m and no larger than 10 $\mu$m.

5. The thermal spray powder of claim 4, where the particle sizes of the powder, in a post-agglomeration and post-sintered state, range substantially between 5 and 125 $\mu$m (micrometers).

6. The thermal spray powder of claim 4, where the particle sizes of the powder, in a post-agglomeration and post-sintered state, range substantially between 10 and 45 $\mu$m (micrometers).

7. The thermal spray powder of claim 1 further comprising:
   - one or more nano and submicro-sized reinforced solid lubricants substantially up to 10% by weight, the one or more nano and submicro-sized reinforced solid lubricants comprising BN, WS$_2$ or MoS$_2$. 
8. The thermal spray powder of claim 7, where the particle size of the one or more ternary boride ceramics, the one or more binary molybdenum borides, the one or more metal binary borides, and the one or more nano and submicrosized reinforced solid lubricants is at least equal to 0.1 μm and no larger than 10 μm.

9. The thermal spray powder of claim 8, where the particle sizes of the powder, in a post-agglomeration and post-sintered state, range substantially between 5 and 125 μm (micrometers).

10. The thermal spray powder of claim 8, where the particle sizes of the powder, in a post-agglomeration and post-sintered state, range substantially between 10 and 45 μm (micrometers).

11. A method of manufacturing a thermal spray powder, the steps comprising:

- milling a composition mixture to form a milled composition mixture, the composition mixture comprising:
  - one or more metals from substantially 0-80% (up to substantially 80%) by weight, where the one or more metals are selected from Mo, Cr, Fe, Ni, W, and Co;
  - one or more binary boride ceramics from substantially 0-80% (up to substantially 80%) by weight, where the one or more metal binary borides are selected from Mo, Cr, Fe, Ni, W, and Co; and,
  - one or more solid lubricants from zero to 10% by weight;

- agglomerating the milled composition to form an agglomerated composition;

- sintering the agglomerated composition to form a thermal spray composite powder comprising:
  - one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo₂-B₂, or Metal₂-Mo₂-B₂ from about 20-70 % by weight, where Metal is Co, Ni or Fe;
  - one or more binary molybdenum borides from substantially 0-60% by weight; and,
  - one or more metal binary borides from substantially 0-50% by weight,
where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

12. The method of manufacturing a thermal spray powder of claim 11, where the one or more ternary boride ceramics, the one or more binary molybdenum borides, and the one or more metal binary borides are provided having particle sizes at least equal to 0.1 μm and no larger than 10 μm.

13. The method of manufacturing a thermal spray powder of claim 12 further comprising:
   agglomerating and sintering such that particles forming the thermal spray composite powder have a particle size at least equal to 5 μm and no larger than 125 μm.

14. The method of manufacturing a thermal spray powder of claim 12 further comprising:
   agglomerating and sintering such that particles forming the thermal spray composite powder have a particle size at least equal to 10 μm and no larger than 45 μm.

15. The method of manufacturing a thermal spray powder of any claim 13 to 14, where sintering is performed at temperatures not lower than substantially 850°C and not higher than substantially 1300°C for substantially 1 to 5 hours.

16. The method of manufacturing a thermal spray powder of claim 11, where the thermal spray composite powder further comprises:
   one or more nano and submicro-sized reinforced solid lubricants from 0-10% by weight, the one or more nano and submicro-sized reinforced solid lubricants comprising BN, WS₂ or MoS₂.

17. The method of manufacturing a thermal spray powder of claim 11, where the one or more ternary boride ceramics, the one or more binary molybdenum borides, the one or more metal binary borides, and one or more nano and submicro-sized reinforced solid lubricants are provided having particle sizes at least equal to 0.1 μm and no larger than 10 μm.

18. The method of manufacturing a thermal spray powder of claim 17 further comprising:
agglomerating and sintering such that particles forming the thermal spray composite powder have a particle size at least equal to 5 µm and no larger than 125 µm.

19. The method of manufacturing a thermal spray powder of claim 17 further comprising:
agglomerating and sintering such that particles forming the thermal spray composite powder have a particle size at least equal to 10 µm and no larger than 45 µm.

20. The method of manufacturing a thermal spray powder of any claim 18 to 19, where sintering is performed at temperatures not lower than substantially 850°C and not higher than substantially 1300°C for substantially 1 to 5 hours.

21. The method of manufacturing a thermal spray powder of any claim 11-20, where the one or more binary molybdenum borides is substantially 20-60% by weight.

22. The method of manufacturing a thermal spray powder of any claim 11-21, where the one or more metal binary borides is substantially 10-50% by weight.

23. A thermal sprayed coating formed from a thermal spray powder comprising:
one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo2-B2 or Meta^-M02-B2 from about 20-70 % by weight, where Metal is Co, Ni or Fe;
one or more binary molybdenum borides from substantially 0-60% by weight; and,
one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co.

24. The thermal sprayed coating of claim 23 further comprising:
one or more nano and submicro-sized reinforced solid lubricants comprising BN, WS₂ or MoS₂, from substantially 0-10% by weight.

25. The thermal sprayed coating of any claim 23 to 24, where the one or more binary molybdenum borides is substantially 20-60% by weight.
26. The thermal sprayed coating of any claim 23 to 25, where the one or more metal binary borides is substantially 10-50% by weight.

27. The thermal sprayed coating of any claim 23 to 26, where the coating has a thickness of substantially 50 to 300 μη.

28. A method of forming a thermal sprayed coating on a substrate, the steps comprising:

   providing a thermal spray powder comprising:

   one or more ternary boride ceramics of Metal-Mo-B, Metal-Mo2-B2, or Meta^-M02-B2 from about 20-70 % by weight, where Metal is Co, Ni or Fe;

   one or more binary molybdenum borides from substantially 0-60% by weight; and,

   one or more metal binary borides from substantially 0-50% by weight, where the one or more metal binary borides are selected from Cr, Fe, Ni, W, and Co; and,

   thermally spraying the thermal spray powder onto the substrate.

29. The method of forming a thermal sprayed coating of claim 28, where the thermal spray powder used in the step of spraying is comprised of particle sizes ranging substantially between 5 and 125 μη (micrometers).

30. The method of forming a thermal sprayed coating of claim 28, where the thermal spray powder used in the step of spraying is comprised of particle sizes ranging substantially between 10 and 45 μη (micrometers).

31. The method of forming a thermal sprayed coating of any claim 28 to 30, where the step of thermally spraying is performed by plasma thermal spraying, flame spraying, detonation spraying, cold gas spraying, arc spraying, or high velocity oxygen fuel spraying.

32. The method of forming a thermal sprayed coating of any claim 28 to 31, where the thermal spray powder further comprises:
one or more nano and submicro-sized reinforced solid lubricants from 0-10% by weight, the one or more nano and submicrosized reinforced solid lubricants comprising BN, WS$_2$ or MoS$_2$.

33. The method of forming a thermal sprayed coating of any claim 28 to 32, where the step of spraying forms a coating on the substrate from the thermal spray powder having a thickness of substantially 50 to 300 µm.

34. The method of forming a thermal sprayed coating of any claim 28 to 33, where the substrate is a metal.

35. The method of forming a thermal sprayed coating of claim 34, where the metal is steel.

36. The method of forming a thermal sprayed coating of any claim 28 to 35, where the coating is characterized as having a corrosion resistance coating against molten metal.

37. The method of forming a thermal sprayed coating of claim 36, where the molten metal is zinc.

38. The method of forming a thermal sprayed coating of any claim 28 to 37, where the coating is characterized as having a superior thermal shock resistance.

39. The method of forming a thermal sprayed coating of any claim 28 to 38, where the coating is characterized as having a dense, superfine structured and homogeneous morphology with a porosity less than 1%.

40. The method of forming a thermal sprayed coating of any claim 28 to 39, where the coating is characterized as having a Vickers hardness of 895+61 HV0.3.

41. The method of forming a thermal sprayed coating of any claim 28 to 40, where the coating is characterized as having high wear resistance and low friction.

42. The method of forming a thermal sprayed coating of any claim 28 to 41, where the one or more binary molybdenum borides is substantially 20-60% by weight.

43. The method of forming a thermal sprayed coating of any claim 28 to 42, where the one or more metal binary borides is substantially 10-50% by weight.
FIG. 1
INTERNATIONAL SEARCH REPORT

PCT/US13/06111

A. CLASSIFICATION OF SUBJECT MATTER

IPC(8) - B05C 5/00; C23C 4/04, 4/10 (2014.01)
USPC - 427/456; 428/702, 704

According to International Patent Classification (IPC) or to both national, classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

IPC(8) Classification(s): B05C 5/00; B32B 3/00; C23C 4/04, 4/10 (2014.01)
USPC Classification(s): 427/446, 456; 428/702, 704

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)


C. DOCUMENTS CONSIDERED TO BE RELEVANT

<table>
<thead>
<tr>
<th>Category*</th>
<th>Citation of document, with indication, where appropriate, of the relevant passages</th>
<th>Relevant to claim No.</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>US 8,206,792 B2 (GOLLOB, DS et al.) 26 June 2012; the entire document</td>
<td>1-20, 23-25, 28-31</td>
</tr>
</tbody>
</table>

Further documents are listed in the continuation of Box C.

* Special categories of cited documents:
  *A* document defining the general state of the art which is not considered to be of particular relevance
  *E* earlier application or patent but published on or after the international filing date
  *L* document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)
  *O* document referring to an oral disclosure, use, exhibition or other means
  *P* document published prior to the international filing date but later than the priority date claimed

Date of the actual completion of the international search

07 July 2014 (07.07.2014)

Date of mailing of the international search report

08 AUG 2014

Name and mailing address of the ISA/US

Mail Stop PCT, Attn: ISA/US, Commissioner for Patents
P.O. Box 1450, Alexandria, Virginia 22313-1450
Facsimile No. 571-273-3201

Authorized officer: Shane Thomas

PCT Helpdesk: 571-272-4300
PCT CSP: 571-272-7774
This international search report has not been established in respect of certain claims under Article 17(2)(a) for the following reasons:

1. □ Claims Nos.: 
   because they relate to subject matter not required to be searched by this Authority, namely:

2. □ Claims Nos.: 
   because they relate to parts of the international application that do not comply with the prescribed requirements to such an extent that no meaningful international search can be carried out, specifically:

3. □ Claims Nos.: 21-22, 26-27, 32-43 
   because they are dependent claims and are not drafted in accordance with the second and third sentences of Rule 6.4(a).

This International Searching Authority found multiple inventions in this international application, as follows:

1. □ As all required additional search fees were timely paid by the applicant, this international search report covers all searchable claims.

2. □ As all searchable claims could be searched without effort justifying additional fees, this Authority did not invite payment of additional fees.

3. □ As only some of the required additional search fees were timely paid by the applicant, this international search report covers only those claims for which fees were paid, specifically claims Nos.:

4. □ No required additional search fees were timely paid by the applicant. Consequently, this international search report is restricted to the invention first mentioned in the claims: it is covered by claims Nos.:

Remark on Protest

□ The additional search fees were accompanied by the applicant's protest and, where applicable, the payment of a protest fee.

□ The additional search fees were accompanied by the applicant's protest but the applicable protest fee was not paid within the time limit specified in the invitation.

□ No protest accompanied the payment of additional search fees.