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(54) Title: FABRICATION OF ELECTRICALLY ACTIVE FILMS BASED ON MULTIPLE LAYERS

(57) Abstract: A continuous film of desired electrical characteristics is obtained by successively printing and annealing two or more dispersions of prefabricated nanoparticles.

FABRICATION OF ELECTRICALLY ACTIVE FILMS
BASED ON MULTIPLE LAYERS

The present invention relates to thin-film materials, their fabrication, and devices
5 made therefrom; and in particular to graded and multi-junction thin-film semiconductor
structures.

Thin-film technologies are currently being developed for the purpose of reducing
the cost of semiconductor devices, in particular photovoltaic (PV) cells. Whereas
conventional solar cells are made of slices of solid crystalline silicon wafers, which have
10 thicknesses of typically a few hundred microns, thin-film materials can be directly
deposited onto a substrate to form layers of $\sim 2 \mu\text{m}$ or less, resulting in lower material as
well as lower manufacturing costs. Moreover, thin-film technologies allow for
monolithic integration, i.e. the *in situ* creation of electrical connections, which further
reduces production costs.

15 Thin-film materials include cadmium-telluride (CdTe), copper indium diselenide
(CIS) and variants thereof, amorphous silicon, and polycrystalline silicon ($< 50 \mu\text{m}$). In
recent years, technical progress has occurred particularly in thin-film technologies based
on CdTe and CIS. Both materials have high absorptivities, so that most of the incident
radiation can be absorbed within 1-2 μm of the film. Used as the absorber layer, in which
20 incoming photons create electron-hole pairs, these materials can be paired with, for
instance, a layer of CdS, to form heterojunctions, and sandwiched between front and back
contacts to form a solar cell.

To gain widespread acceptance, thin-film PV cells must exhibit high conversion
efficiencies of photon energy to electric current, and operate reliably in an outdoor
25 environment over many years, ideally no less than 30 years. Technologies based on
CdTe and CIS have demonstrated long-term stability; however, performance degradation
has also been observed. Efficiencies of current thin-film devices reach 65% of the
theoretical maximum (75% in the laboratory), still lagging behind some monocrystalline
silicon and GaAs cells, which have demonstrated 90% of their ultimate achievable

performance. Improvements in efficiency of thin-film technologies can be achieved through multijunctions and graded materials. For example, studies on CIS have revealed that doping with gallium, to form compounds referred to as CIGS and exhibiting gradients in the concentrations of Ga and In, lead to better efficiencies.

5 The complexities of thin-film technologies, which are essential for high efficiencies, adversely affect cost and manufacturability, establishing a need for improved techniques—in particular low-cost techniques amenable to practice with off-the-shelf equipment. Challenges to the development of low-cost and reliable CIGS and CdTe devices include the standardization of equipment for layer deposition, absorber layers
10 having thicknesses less than 1 μm , and control of film uniformity over large areas.

 In various embodiments, the present invention provides methods for fabricating a continuous film by successively printing and annealing two or more dispersions of prefabricated nanoparticles. In particular, some embodiments of the invention facilitate manufacture of graded and multijunction semiconductor films, which can be used in PV
15 cells and other semiconductor devices. Since the method requires no vacuum, it is cheaper and more conveniently practiced than vacuum-based techniques.

 Nanoparticles according to this invention are particles of specified elemental composition and no more than 100 nm, and preferably no more than 20 nm, in diameter. Typical nanoparticles include metal-oxide particles, which collectively form a powder.
20 Some nanoparticle compositions suitable for semiconductor thin films comprise two or more of the chemical elements Cu, Ag, In, Ga, Al, Te, Se, S, Cd, and As. It should be stressed, however, that the invention is not limited to said elements, but that the method generally applies to any composition of nanoparticles suitable for dispersion, and subsequently printing. One of the advantages of techniques in accordance with the
25 present invention lies in the ability to optimize the composition of the thin film by providing compositional control over the precursor nanoparticles. This facilitates the fabrication of a continuous film comprised of layers of specified chemical composition, which allows for the compositional optimization of these layers, and, as a result thereof,

for improved control over the electric characteristics of the film and, in particular, over the variation of these characteristics throughout the thickness of the film.

Dispersions according to the invention include any (homogeneous) mixture of nanoparticles and a suitable flowable carrier comprising solvents or dispersing agents, whether the mixture is a solution, a colloid, or a suspension. These dispersions of nanoparticles are termed “printing compositions” or “nanoparticle-based inks” herein.

Methods according to the invention can be implemented using a variety of printing techniques and the corresponding printing equipment, including, but not limited to, techniques such as inkjet printing, pneumatic spray printing, screen printing, pad printing, laser printing, dot matrix printing, thermal printing lithography, or 3D printing. This versatility contributes to feasibility and cost-effectiveness. Furthermore, the composition of the nanoparticles can vary through the various deposition and annealing steps. For example, in one embodiment of the invention, the different printable compositions comprise nanoparticles of the same elements in different proportions, for example, the nanoparticles may have the formula $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$, wherein x varies between 0 and 1 (i.e. $0 \leq x \leq 1$), resulting in a concentration gradient of at least one element (in the example In and Ga) through the film.

Accordingly, in a first aspect, the invention provides a method of fabricating a film, which includes the steps of providing a substrate and flowable printing compositions with different dispersions of prefabricated nanoparticles, and successively printing and annealing layers of these printing compositions into one continuous film. In some embodiments, two or more layers are successively printed before they are annealed. Further, in some embodiments an etching step precedes annealing. Some of the individual printed layers may have thicknesses smaller than $1 \mu\text{m}$.

In certain embodiments, the printing compositions contain the same types of nanoparticles in different proportions, or nanoparticles composed of the same elements in different proportions, so that the annealed layers form a film with a concentration gradient of at least one material. In alternative embodiments, each printing composition

includes different types of nanoparticles. In preferred embodiments, the nanoparticles have a size no greater than 20 nm and a low size dispersity.

In some embodiments, the film includes a semiconductor material and interacts electrically with the substrate. Moreover, this structure can be complemented by an electrically conductive superstrate to form a semiconductor device. In a particular
5 embodiment, the device is a solar cell.

In a second aspect, the invention provides flowable printing compositions with a substantially viscosity-independent flow rate. These printing compositions contain a carrier and a dispersion of nanoparticles; the nanoparticles include Cu and/or Ag as a first
10 component and Se, Te, and/or S as a second component. Additionally, the printing compositions may contain In, Ga, and/or Al as a third component.

It should be stressed that embodiments of methods in accordance with the invention are not limited to the printing compositions described above.

The foregoing discussion will be understood more readily from the following
15 detailed description of the invention when taken in conjunction with the accompanying drawings.

FIG. 1 is a flow diagram detailing the steps of a method to manufacture a film by successive printing and annealing.

FIG. 2 schematically depicts a system and method for manufacturing the film in
20 accordance with an embodiment of the invention.

FIG. 3A is a schematic elevational view of a representative solar cell manufactured in accordance with the invention.

FIG. 3B is a schematic elevational view of a graded CIGS film, manufactured by first depositing all constituting layers and then annealing once to produce a continuous
25 film, and of a solar cell made therefrom.

1. Fabrication method

Refer first to **FIGS. 1 and 2**, which illustrate, respectively, a representative process sequence **100** and operative equipment implementing embodiments of the present invention. The process sequence comprises the steps detailed in the flow diagram of **FIG.**

1, utilizing the equipment and resulting in the intermediate and final structures illustrated in FIG. 2. In a first step 110, a substrate 200 and a plurality of flowable printing compositions comprising different dispersions of prefabricated nanoparticles, as further described below, are provided. A printing composition 202a is selected for the first layer
5 in step 112, and in step 114, this composition is printed onto the substrate 200 using a printer 204. In some cases, which will be described below, an etch step 116 is performed after printing. In an optional step 118, the deposited layer is dried and annealed using a heat source 206 to form a continuous film 208. By “annealing” is meant heating of a deposited layer at a sufficient temperature and for a sufficient time that the nanoparticles
10 fuse into a continuous layer of uniform composition. Whether annealing is performed after deposition of a particular layer depends on the specifics of the printing composition, the layer thickness, and desired film characteristics. In general, however, the composition will be dried before a subsequent composition is deposited thereon. Annealing source 206 may be any suitable heat source, e.g., an oven, vacuum oven,
15 furnace, IR lamp, laser, or hot plate, and suitable annealing times and temperatures, which depend on nanoparticle size and composition as well as ink composition, may be obtained without undue experimentation by means of calibration as described below in the context of printing equipment. Anneal temperatures are generally above 200 °C.

The steps 112, 114, and optionally steps 116, 118, are repeated to print a second
20 printing composition 202b, which generally differs from the composition 202a, resulting in a film 208 that now contains two layers. This repetition may involve the same printing and annealing equipment 204, 206, in which case the new printing composition is substituted (e.g., in the form of a cartridge) in printer 204. Alternatively, the process sequence 100 may be carried out in a assembly-line configuration with separate printing
25 and annealing equipment dedicated to each deposition and annealing step. Utilizing the same equipment for multiple steps may be more practical where numerous films are fabricated in parallel (i.e., the same steps are simultaneously performed on multiple substrates); while a line configuration may be preferred where individual substrates 200 are processed serially.

Steps **110-118** may (but need not) be repeated a plurality of times to form a film having three or more layers **202a**, **202b**, **202c**. Once again, a particular printing composition can be utilized once or more than once, although in the case of graded films, the composition will change progressively with each deposition step. After the last layer has been printed, an etching step **120** is again optional, but a final annealing step **122** must take place, whether previous anneals **116** have been performed or not, to form a continuous final film **210** from the deposited layers. The number of layers deposited and annealed in this way is at least two, and is limited only by the desired thickness and composition of the final film **210**. Accordingly, while **FIG. 2** illustrates the manufacturing of a film with three layers, this is merely for illustration.

Each layer may contain a single type of nanoparticle, in which case different layers typically contain nanoparticles having different chemical compositions; or alternatively (or in addition), each layer may contain a plurality of nanoparticle types, in which case the same set of nanoparticles may be used in different proportions in the different layers. In some embodiments of the invention, the method **100** is employed to produce a high-efficiency graded film by using a variety of printing compositions with nanoparticles comprising the same elements, but in different proportions. For example, a CIGS film may be produced with the varying chemical composition $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$, where x varies progressively among successively deposited layers (i.e., successive printing compositions). For example, x may be the distance of a certain location within the film from a boundary surface of the film (e.g., the top surface of the structure or the boundary surface in contact with the substrate), divided by the thickness of the film. Such films have been manufactured by means of chemical vapor deposition (CVD) and used as the absorption layer in a PV cell with the gallium concentration increasing towards a molybdenum back contact, resulting in a particularly high efficiency of 19.5%—a consequence of reduced back-surface recombination due to the quasi-electric field established by the concentration and corresponding bandgap gradient. The method **100** provides an alternative to the CVD process: using a suitable substrate, such as Mo-covered glass, CuInSe_2 may be deposited for the first layer, followed by printing

compositions in which the Ga content is progressively (in a linear or nonlinear fashion) increased in each layer, until the composition of the nanoparticles is primarily CuGaSe₂. Once the layers are printed in the desired sequence, they are annealed to form a high-performance graded CIGS film. Other nanoparticle-based inks may also be introduced as
5 intermediate layers to further tune the band structure of the material in order to optimize the performance of the cell. Techniques in accordance with the present invention offer the additional advantage, compared with CVD, of avoiding the need for vacuum equipment.

In other embodiments of the invention, the printing compositions comprise
10 different types of nanoparticles. The method 100 can, for instance, be used to manufacture a CdS/CdTe thin film. In general, the approach of the invention can be applied to any material for which a suitable nanoparticle source is available.

2. Printing Equipment 204

15 For the implementation of the printing step 114, a variety of well-characterized printing processes can be used to advantage, including, but not limited to, inkjet printing, pneumatic spraying, screen printing, pad printing, laser printing, dot matrix printing, thermal printing, lithography, and 3D printing. Computer-controlled inkjet printers are readily available and particularly attractive for practice of the invention because of the
20 level of control they provide. A commercially available inkjet printer can be used with little or no modification to print nanoparticle-based inks (the printing compositions) as described herein. To avoid problems such as clogging of the printer head or other incompatibilities, the viscosities of nanoparticle-based inks can be adjusted to those of inks produced by the printer's manufacturer, as detailed below in the context of printing
25 compositions. The amenability of the method 100 to readily available, low-cost equipment, such as inkjet printers, constitutes one of its advantages.

To facilitate control over the thicknesses of the individual layers and the film as a whole, the printer can be calibrated as follows. For each printing composition, a sequence of print runs is carried out, each print run involving a different number of

printing passes. Drying and annealing are performed after each printing pass or at the end of a run. The thickness of the film resulting from each of the runs is different, and is determined via scanning electron microscopy (SEM) or transmission electron microscopy (TEM), or any other suitable technique. As a result, a layer having a desired thickness
5 may be made by reference to the number of printing passes corresponding to that thickness. Similar calibration techniques can be used to determine optimal annealing temperatures and times for desired film properties.

3. Printing compositions 202

10 Printing compositions in accordance herewith are flowable dispersions of nanoparticles. Particulate precursor materials simplify compositional control for multi-component materials such as, for example, CIGS, since key components (e.g., Cu, In, Ga) can be precisely mixed in the precursor powders. One method of fabricating these powders involves mixing the constituent elements at the required ratios, dissolving them
15 in acid to form an aqueous mixture, using hydroxide precipitation to form a gelatinous mixture of hydroxides of the elements, and drying the mixture to obtain a fine powder of mixed oxides. Nanoparticle synthesis can also be carried out using techniques described, for example, in U.S. Patent No. 6,379,635 and co-pending U.S. Patent Application Serial Nos. 11/579,050 and 11/588,880.

20 Methods for preparing CdSe, InS, GaS, ZnS, CdS, ZnAs, CdAs and further related nanocrystalline materials are described in Sections 1. to 10. of the experimental section of US Patent No. 6,379,635. This describes how nanocrystal precursors can be provided in a single precursor complex (such as, but not limited to an alkyl carbonate complex of the required metal ions), which is then thermally decomposed under suitable
25 conditions (e.g. temperatures of around 200 to 300° C) to yield the final desired nanocrystalline material.

By way of example, nanocrystalline CdSe can be formed as follows, 1.2 MeCddsc (0.5 mmol) was placed in 10 ml of TOP (98 %, Aldrich) and the mixture formed was filtered after which was injected in 30 g of TOPO at 200° C. The temperature of the

solution was then raised to 250° C and heated for half an hour. The deep red solution that formed was allowed to cool down to 75° C after which a large excess of dry CH₃OH (BDH) was added. A flocculate precipitate formed and was isolated by centrifugation and redispersed in toluene, any insoluble material was then discarded. The toluene was pumped off under vacuum (10⁻² Torr) to give a deep red material which was washed with CH₃OH. The solid was redispersed in toluene to give solutions with a Port wine red colour which remained optically clear for weeks. Size selective precipitation was performed by adding CH₃OH to this solution until turbidity was observed followed by centrifugation the solid. This procedure was successively applied to the supernatant solutions obtained during the fractionation process until no optical absorption was detected.

Similar methods can be employed to produce III-VI nanocrystalline materials (e.g. InS, GaS) by replacing MeCddsc with a complex of the general formula, (AlK1)₂M^{III}E₂CN(AlK2)₂, where AlK1 and AlK2 are individually alkyl groups, such as Me, Et, Np etc, each AlK1 and AlK2 being the same or different, M^{III} is a group III metal ion such as In, Ga etc, and E is a group VI ion, such as S, Se etc; and to produce II-VI or II-V nanocrystalline materials (e.g. ZnS, CdS, ZnAs and CdAs) by replacing MeCddsc with a complex of the general formula [M^{II}[E₂CN(AlK3)₂]₂], where each AlK3 is an alkyl group, such as Me, Et, Pr etc, each AlK3 being the same or different, M^{II} is a group II metal ion, such as Zn, Cd etc, and E is a group V or VI ion, such as As or S, Se etc.

Methods for preparing large amounts of nanocrystalline materials are described in US Patent Application Serial Nos. 11/579,050 and 11/588,880. These applications describe for the first time how molecular clusters can be used to seed controlled growth of any desirable nanocrystalline material, which can then be provided with one or more outer shells of a different material to the material forming the nanocrystal core grown on the molecular cluster seed.

By way of example, US Patent Application Serial No. 11/579,050 describes in Examples 1 to 9, the preparation of the clusters [HNEt₃]₂[Cd₄(SPh)₁₀] and [HNEt₃]₄[Cd₁₀Se₄(SPh)₁₆] which can then be used to produce CdSe nanocrystals by controlled addition

of Cd ions and Se ions (from, e.g. TOPSe and $\text{Cd}(\text{CH}_3\text{CO}_2)_2$) over a suitable time period, in combination with suitably controlled increases in the temperature of the reaction mixture. For example, Example 1 describes the preparation of CdSe nanoparticles from $[\text{Et}_3\text{NH}]_4[\text{Cd}_{10}\text{Se}_4(\text{SPh})_{16}]$ / TOPSe / $\text{Cd}(\text{CH}_3\text{CO}_2)_2$ in HDA. HDA (300g) was placed in
5 a three-neck flask and dried/degassed by heating to 120°C under a dynamic vacuum for 1 hour. The solution was then cooled to 70°C. To this was added 1.0g of $[\text{Et}_3\text{NH}]_4[\text{Cd}_{10}\text{Se}_4(\text{SPh})_{16}]$ (0.311mmol), TOPSe (20ml, 40.00mmol) [previously prepared from dissolving selenium powder in TOP] and $\text{Cd}(\text{CH}_3\text{CO}_2)_2$ (10.66g 40.00mmol) the temperature of reaction mixture was gradually increased from 70°C to 180°C over an 8
10 hour period. The progressive formation/growth of the nanoparticles was monitored by their emission wavelength by taking aliquots from the reaction mixture and measuring their UV-vis and PL spectra. The reaction was stopped when the emission spectra had reached 572nm by cooling the reaction to 60°C followed by addition of 200ml of dry “warm” ethanol which gave a precipitation of nanoparticles. The resulting CdSe were
15 dried before re-dissolving in toluene filtering through Celite followed by re-precipitation from warm ethanol to remove any excess HDA and $\text{Cd}(\text{CH}_3\text{CO}_2)_2$. This produced 9.26g of HDA capped CdSe nanoparticles.

By way of further example, US Patent Application Serial No. 11/588,880 describes the preparation of CdSe nanoparticles from
20 $[\text{HNEt}_3]_4[\text{Cd}_{10}\text{Se}_4(\text{SPh})_{16}]/\text{TOPSe}/\text{Me}_2\text{Cd}/\text{TOP}$ in HDA by dropwise addition of $\text{Me}_2\text{Cd}/\text{TOP}$, and the preparation of ZnS nanoparticles from $[\text{Et}_3\text{NH}]_4\text{Zn}_{10}\text{S}_4(\text{SPh})_{16}$ seeds in HDA by dropwise addition of Et_2Zn and S-octylamine. Nanoparticles formed in this way can then be provided with one or more shell layers to provide CdSe/ZnS-HDA capped nanoparticles, and ZnSe/ZnS nanoparticles. For example ZnS nanoparticles can
25 be formed from $[\text{Et}_3\text{NH}]_4\text{Zn}_{10}\text{S}_4(\text{SPh})_{16}$ seeds in HDA by dropwise addition of Et_2Zn and S-octylamine as follows. A 200 g portion of hexadecylamine (HDA) was placed in a three-neck round-bottomed flask and dried and degassed by heating to 120 °C under a dynamic vacuum for > 1 hour. The solution was cooled to 60 °C, the reaction flask was filled with nitrogen and the following reagents were loaded into the flask using standard

airless techniques: 0.6 g $[\text{HNEt}_3]_4[\text{Zn}_{10}\text{S}_4(\text{SPh})_{16}]$ (0.2 mmol), 4 mL of a 0.5 M solution of Et_2Zn in trioctylphosphine (2 mmol), and 4 mL of a 0.5 M solution of elemental sulphur in octylamine (2 mmol). The temperature was increased to 120 °C and allowed to stir for 2 hours. At this point a programmed temperature ramp from 120 °C to 210 °C at a rate of $\sim 0.2^\circ\text{C}/\text{min}$ was initiated. Simultaneously, 8 mL of 0.5 M Et_2Zn and 8 mL of 0.5 M S-octylamine were added dropwise at a rate of ~ 0.05 mL/min. The reaction was stopped when the PL emission maximum had reached the required emission ($\lambda_{\text{max}} = 391$ nm, FWHM = 95 nm) by cooling to 60° C followed by addition to 300 ml of dry ethanol or acetone to precipitate the particles from solution. This precipitate was isolated by filtration. The resulting ZnS particles were further purified (to remove excess HDA, sulphur and zinc) by re-dissolving in toluene, filtering the solution through Celite and re-precipitating from warm ethanol (product yield: 0.9g). Capping or shelling of CdSe nanoparticles can then be carried out, for example, as follows HDA (800g) was placed in a three neck round-bottom flask, dried and degassed by heating to 120°C under a dynamic vacuum for > 1 hour. The solution was then cooled to 60°C, to this was added 9.23g of CdSe nanoparticles that have a PL maximum emission of 585nm. The HDA was then heated to 220°C. To this was added by alternate dropwise addition a total of 20ml of 0.5M $\text{Me}_2\text{Zn}\cdot\text{TOP}$ and 0.5M, 20ml of sulfur dissolved in octylamine. Three alternate additions of 3.5, 5.5 and 11.0ml of each were made, whereby initially 3.5ml of sulphur was added dropwise until the intensity of the PL maximum was near zero. Then 3.5ml of $\text{Me}_2\text{Zn}\cdot\text{TOP}$ was added dropwise until the intensity of the PL maximum had reached a maximum. This cycle was repeated with the PL maximum reaching a higher intensity with each cycle. On the last cycle, additional precursor was added once the PL maximum intensity been reached until it was between 5 - 10% below the maximum intensity, and the reaction was allowed to anneal at 150°C for 1 hour. The reaction mixture was then allowed to cool to 60 °C whereupon 300ml of dry “warm” ethanol was added which resulted in the precipitation of particles. The resulting CdSe-ZnS particles were dried before re-dissolving in toluene and filtering through Celite followed by re-precipitation from warm ethanol to remove any excess HDA. This produced 12.08g of HDA capped

CdSe-ZnS core-shell nanoparticles. Elemental analysis C = 20.27, H = 3.37, N = 1.25, Cd = 40.11, Zn = 4.43 %; Max PL 590nm, FWHM 36nm.

A method for producing CIGS nanoparticles of any desirable stoichiometry employing a selenol compound is disclosed in U.S. Provisional Application Serial No. 5 60/991,510. Embodiments of the method involve dispersing at least a first portion of a nanoparticle precursor composition (comprising sources of at least one of Al, Ga, and/or In, and at least one of Cu, Ag, Zn, and/or Cd) in a solvent (e.g., a long-chain hydrocarbon solvent); heating the solvent to a first temperature for an appropriate length of time; 10 adding a selenol compound to the solvent and heating the solvent; adding a second portion of the nanoparticle precursor composition to the reaction mixture; heating the mixture to a second temperature higher than the first temperature over an appropriate length of time; and maintaining the temperature for up to 10 hours. Once the particles have been formed, the surface atoms of the particles will typically be coordinated to a 15 capping agent, which can comprise the selenol compound employed in the method. If a volatile selenol compound is used, this capping agent can be driven off with heating to yield 'naked' nanoparticles amenable to capping with other coordinating ligands and further processing. Examples 1 and 2 provide further details regarding the implementation of this method:

20 Example 1: Cu(I) acetate (1 mmol) and In(III) acetate (1 mmol) are added to a clean and dry RB-flask. Octadecene ODE (5 mL) is added the reaction mixture heated at 100 °C under vacuum for 30 mins. The flask is back-filled with nitrogen and the temperature raised to 140 °C. 1-octane selenol is injected and the temperature falls to 120 °C. The resulting orange suspension is heated with stirring and a transparent 25 orange/red solution is obtained when the temperature has reached 140°C. This temperature is maintained for 30 minutes, then 1M tri-octyl-phoshine selenide TOPSe (2mL, 2 mmol) is added dropwise and the solution heated at 160°C. The PL is monitored until it reaches the desired wavelength, after which it is cooled and the resulting oil

washed with methanol/acetone (2:1) 4-5 times and finally isolated by precipitation with acetone.

Example 2 (Large Scale Production): A stock solution of TOPSe was prepared by dissolving Se powder (10.9, 138 mmol) in TOP (60 mL) under nitrogen. To dry, degassed ODE was added Cu(I) acetate (7.89 g, 64.4 mmol) and In(III) acetate (20.0 g, 68.5 mmol). The reaction vessel was evacuated and heated at 140 °C for 10 min, backfilled with N₂ and cooled to room temp. 1-Octane selenol (200 mL) was added to produce a bright orange suspension. The temperature of the flask was raised to 140 °C and acetic acid distilled from the reaction at 120 °C. On reaching 140 °C the TOPSe solution was added dropwise over the course of 1 hour. After 3 hours the temperature was raised to 160 °C. The progress of the reaction was monitored by taking aliquots from the reaction periodically and measuring the UV/Visible and photoluminescence spectra. After 7 hours the reaction was cooled to room temperature and the resulting black oil washed with methanol. Methanol washing was continued until it was possible to precipitate a fine black material from the oil by addition of acetone. The black precipitate was isolated by centrifugation, washed with acetone and dried under vacuum. Yield: 31.97 g.

To optimize particle properties or the selection of suitable dispersants, the nanoparticles can be characterized with respect to their composition, size, and charge by conventional techniques including x-ray diffraction (XRD), UV/Vis/Near-IR spectrometry, scanning or transmission electron microscopy (SEM/TEM), energy dispersive x-ray microanalysis (EDAX), photoluminescence spectrometry, and/or elemental analysis. Inductively coupled plasma atomic-emission spectroscopy (ICPAES) analysis of representative Cu/In/Se core particles, prepared in a 1-octane selenol capping agent which was subsequently removed, provided the following suitable nanoparticle composition: Cu 16.6%; In 36.6 %; Se 48.3 %, corresponding to Cu_{1.00}, In_{1.22}, Se_{2.34}, and a Cu/In ratio of 0.82.

In preferred embodiments of this invention, the nanoparticles have average sizes not greater than 20 nm, and low size dispersities of around ±2 nm or less. Conformance

to these constraints facilitates printing of thin films with control over the band structure through the film, resulting in high conversion efficiencies. Moreover, low size dispersities allow for good packing of the nanoparticles, and uniform melting temperature of the nanoparticle films, which contributes to proper film formation.

5 The nanoparticles are dispersed in a carrier comprising solvents, such as toluene, and dispersing agents to form the printing composition. The dispersion may take the form of a solution, colloid, or suspension, generally depending on the particle size, and may have the consistency of a liquid, paste, or other viscoelastic material, as long as it is flowable. Its viscosity should be within the range from 0.158×10^{11} cP to 2.3×10^{11} cP.

10 In embodiments in which water-based inks are formed with non-soluble nanoparticles, surface area and charge of the particles drive the selection of dispersants suitable for ink formulation. For example, in pigment-based inkjet printing, the overall charge the particles acquire (i.e., the zeta potential) in the medium in which they are dispersed should be sufficiently high to ensure dispersion stability; but excessive
15 dispersion stability can result in flocculation and consequent clogging of the printer head. To ensure the jetting potential of the ink through the nozzle, the average agglomerate size should be minimized. In the printing industry, it is generally recognized that particle sizes of over 500 nm may cause plugging of the inkjet nozzles, compromising print quality.

20 To mitigate print-head blocking concerns, nanoparticles can be coated in water-solubilizing capping agents, such as a mercaptocarboxylic acid (e.g., mercaptoacetic acid). For example, U.S. Patent No. 6,114,038 teaches how to exchange the coating groups of water-insoluble, pyridine-capped nanocrystals with a large excess of neat mercaptocarboxylic acid to obtain water-soluble capped nanocrystals. In brief, the
25 pyridine-capped nanocrystals are precipitated with hexanes and centrifuged; the residue is dissolved in neat mercaptoacetic acid and incubated at room temperature for at least six hours; chloroform is added to precipitate the nanocrystals and wash away excess thiol; and the nanocrystals are again centrifuged, washed with chloroform and hexane, and dried with argon. The viscosity of the printing composition (nanoparticle-based ink) is

desirably adjusted to achieve plastic flow behavior, i.e., where viscosity is essentially independent of flow rate. This facilitates control over coating characteristics. Capping agents needed for solubilization or suspension can be removed to stop the formation of carbon deposits within the film. In some cases, this removal occurs naturally as a result of annealing at elevated temperatures, but if does not, it can be aided by a prior etching step 116, 120.

4. Applications

Semiconductor thin-film structures manufactured according to the method illustrated in FIG. 1 can be used in photovoltaic cells, LEDs, transistors, and other semiconductor devices. FIG. 3A illustrates a representative structure of a solar cell with a CIGS absorber film. The substrate 305 comprises molybdenum on glass, and the submicron Mo layer also provides the back contact of the cell 300. The absorber film 307 comprises a series of annealed layers of CIGS, and exhibits increasing Ga and decreasing In concentration towards the Mo contact 300. This film can be manufactured by printing and annealing each layer consecutively. Alternatively, as illustrated in FIG. 3B, which exemplifies the In and Ga contents of individual layers, it can be manufactured by first depositing all the layers, and subsequently fusing these layers into one continuous film in one annealing step. A buffer layer 312 forms the junction with the CIGS film. Conventionally, this junction comprises CdS. However, due to environmental and health concerns associated with Cd, preferred PV cells are cadmium-free, using ZnS, ZnO(O,OH), or In₂S₃ instead. Accordingly, a ZnO layer 314 over a glass cover 316 provides the superstrate of the cell 300. The performance of a ZnO/ZnO(O,OH)/CIGS/Mo cell can be improved or optimized by introducing layers of other semiconductor materials within the absorber film 307. CIGS variants (e.g., as shown in the following Table 1) in which Se is replaced by S or Te, Cu by Ag, or In or Ga by Al, for example, can be used to manipulate the energies of the valence and conduction bands to aid in electron-hole capture. Embodiments of the invention provide a convenient means to integrate these additional layers. Moreover, if nanoparticle

sources are available for the junction layer and/or the substrate or superstrate, these layers can likewise be integrated into the device by printing and annealing, as long as none of the required annealing temperatures is detrimental to the other layers within the device.

5 **Table 1.**

Low Bandgap		High Bandgap	
Material	E_g (eV)	Material	E_g (eV)
CuInSe ₂	1.0	CuAlSe ₂	2.71
CuInTe ₂	1.0–1.15	CuInS ₂	1.53
CuGaTe ₂	1.23	CuAlTe ₂	2.06
		CuGaSe ₂	1.70
		CuGaS ₂	2.50
AgInSe ₂	1.20	AgGaSe ₂	1.80
AgGaTe ₂	1.1–1.3	AgAlSe ₂	1.66
AgAlTe ₂	0.56	AgInS ₂	1.80
		AgGaSe ₂	1.80
		AgGaS ₂	2.55
		AgAlS ₂	3.13

Although the present invention has been described with reference to specific details, it is not intended that such details should be regarded as limitations upon the scope of the invention, except as and to the extent that they are included in the
 10 accompanying claims.

CLAIMS

1. A method of fabricating a film, the method comprising the steps of:
 - a. providing a substrate and a plurality of flowable printing compositions each comprising a different dispersion of prefabricated nanoparticles of specific elemental composition and having an average size no greater than 100 nm;
 - b. printing a layer of one of the printing compositions onto the substrate;
 - c. annealing the printed layer to form a continuous film; and
 - d. repeating steps (b) and (c) with another or the same printing composition to form the film.
2. The method of claim 1, wherein at least some of the printed layers have a thickness smaller than 1 μm .
3. The method of claim 1 or 2 further comprising repeating step (b) at least once before step (c).
4. The method of claim 1, 2 or 3 further comprising the step of performing an etch after printing and before annealing.
5. The method of any one of claims 1 to 4 wherein the printing compositions are suspensions.
6. The method of any one of claims 1 to 4 wherein the printing compositions are solutions.
7. The method of any preceding claim wherein the printing step is performed by a printing technique selected from the group consisting of inkjet printing, pneumatic

spraying, screen printing, pad printing, laser printing, dot matrix printing, thermal printing, lithography, or 3D printing.

8. The method of any preceding claim wherein each printing composition comprises different types of nanoparticles.

9. The method of any one of claims 1 to 7 wherein at least some of the printing compositions contain the same types of nanoparticles in different proportions, the layers being applied to create a concentration gradient of at least one material through the film.

10. The method of any preceding claim wherein the nanoparticles of each composition comprise the same elements in different proportions, the layers being applied to create a concentration gradient of at least one material through the film.

11. The method of claim 10 wherein a first set of nanoparticles comprises or consists essentially of $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$, where $0 \leq x \leq 1$ and with x varying through the printing compositions.

12. The method of any preceding claim wherein the nanoparticles have a size no greater than 20 nm and a low size dispersity.

13. The method of any preceding claim wherein the thin film comprises a semiconductor and interacts electrically with the substrate.

14. The method of claim 13 further comprising adding an electrically conductive superstrate and external contacts to produce a semiconductor device.

15. The method of claim 14, wherein the semiconductor device is a solar cell.

16. A flowable printing composition comprising a carrier and a dispersion of nanoparticles therein, the nanoparticles comprising first and second components, the first component comprising at least one of Cu or Ag, and the second component comprising at least one of Se, Te, or S, the composition having a viscosity substantially independent of flow rate.

17. The printing composition of claim 16, wherein the first component comprises Cu or Ag, but not both.

18. The printing composition of claim 16 or 17, wherein the second component comprises exactly one of the elements Se, Te, or S.

19. A flowable printing composition comprising a carrier and a dispersion of nanoparticles therein, the nanoparticles each comprising first, second, and third components, the first component comprising at least one of Cu or Ag, the second component comprising at least one of Se, Te, or S, and the third component comprising at least one of In, Ga, or Al, the composition having a viscosity substantially independent of flow rate.

20. The printing composition of claim 19, wherein the first component comprises Cu or Ag, but not both.

21. The printing composition of claim 19 or 20, wherein the second component comprises exactly one of the elements Se, Te, or S.

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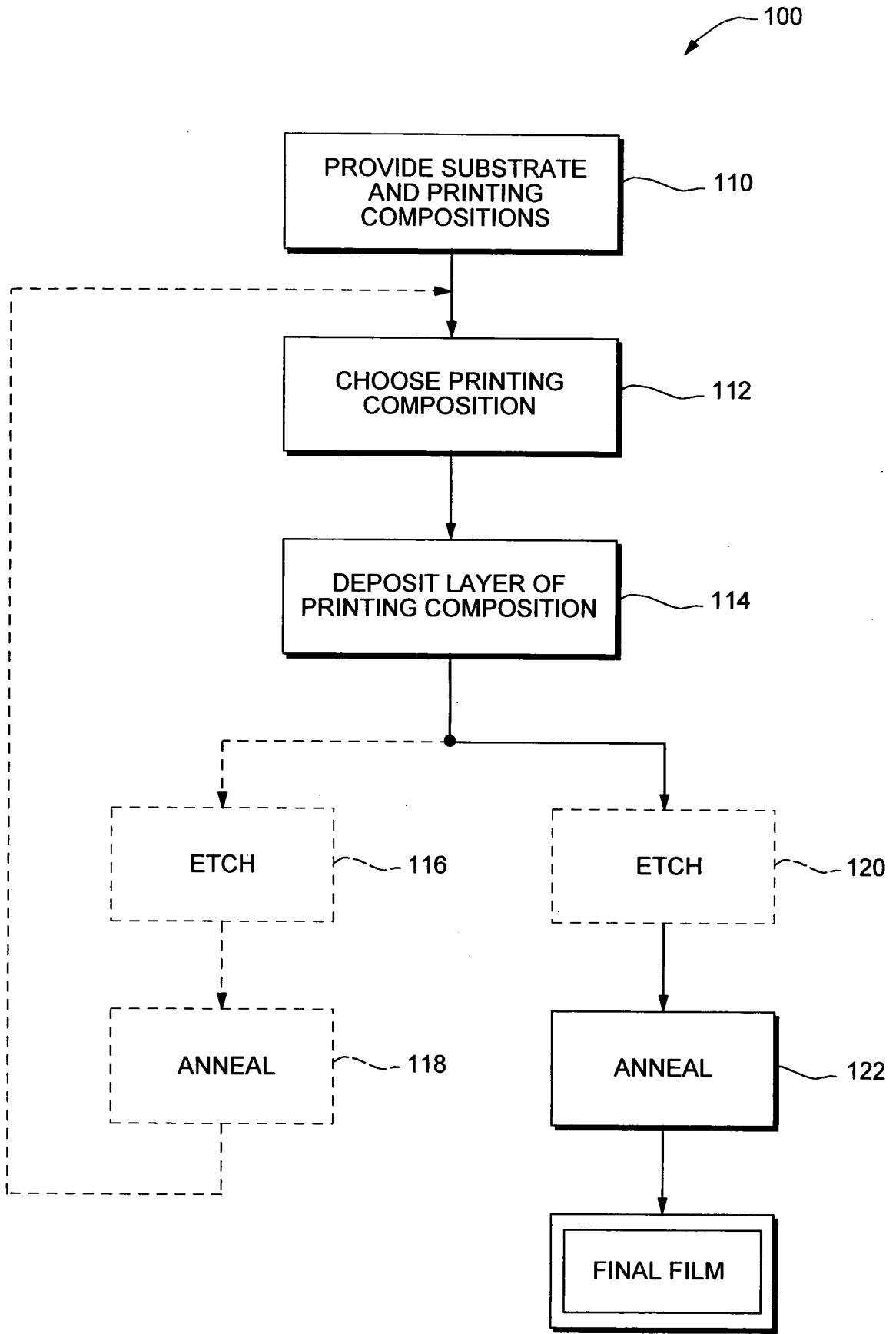


FIG. 1

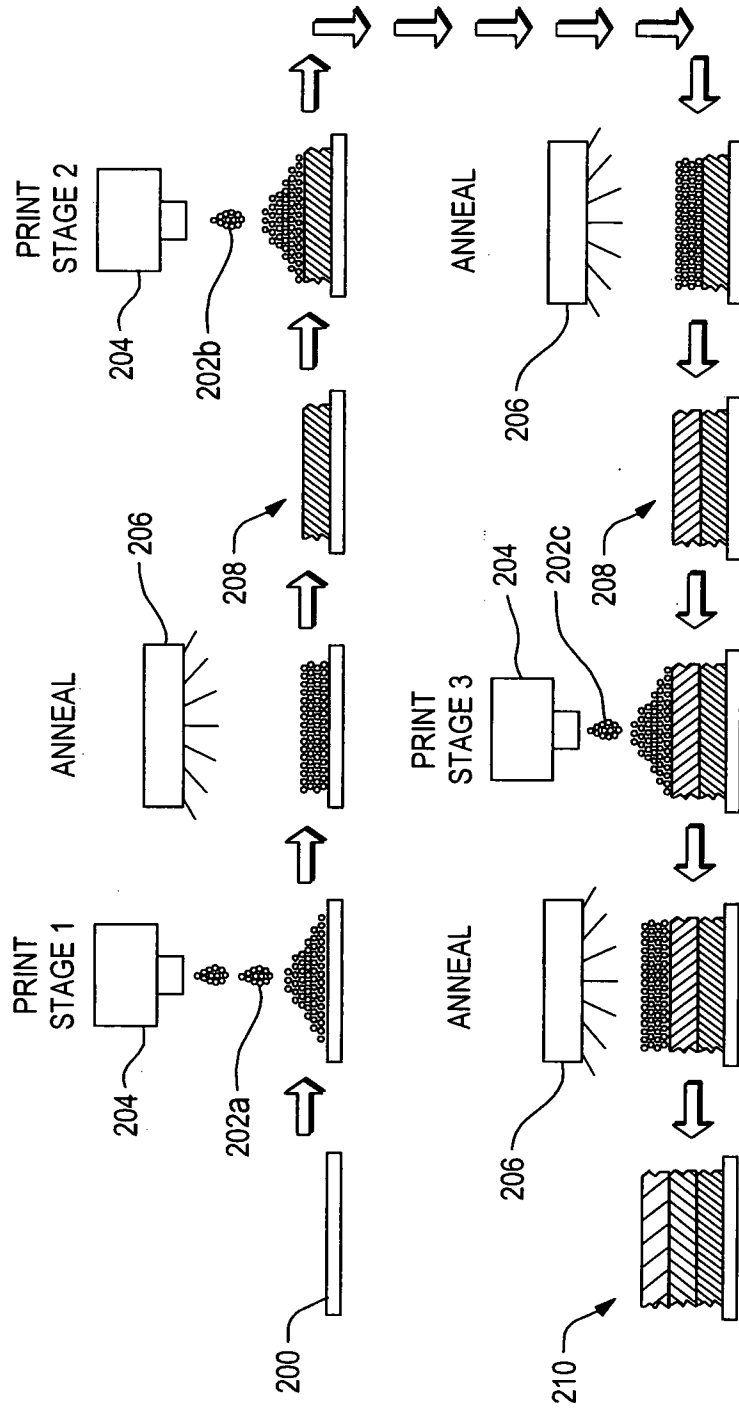


FIG. 2

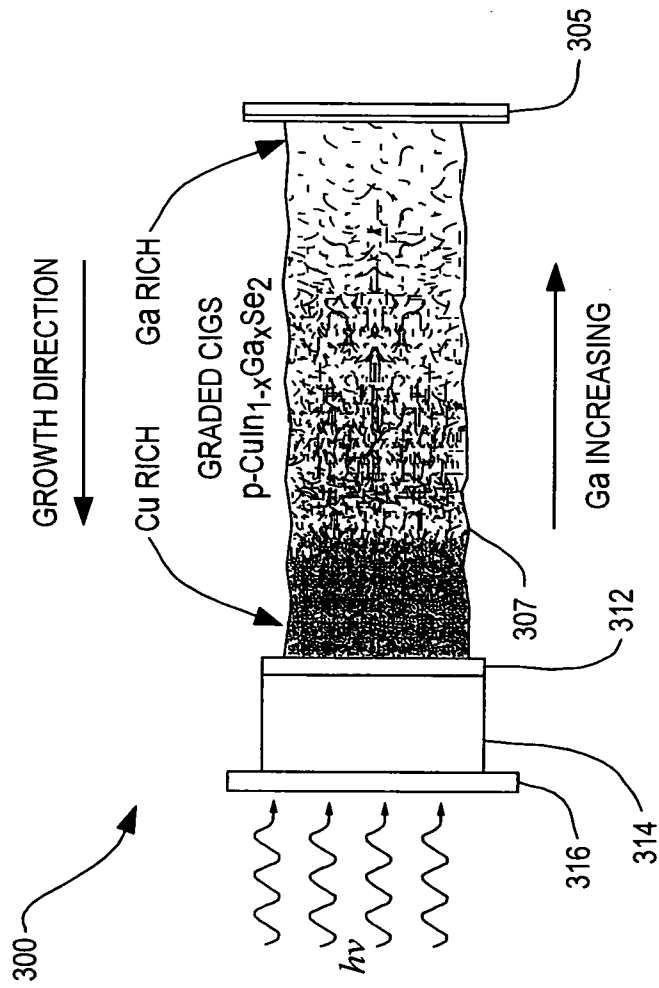


FIG. 3A

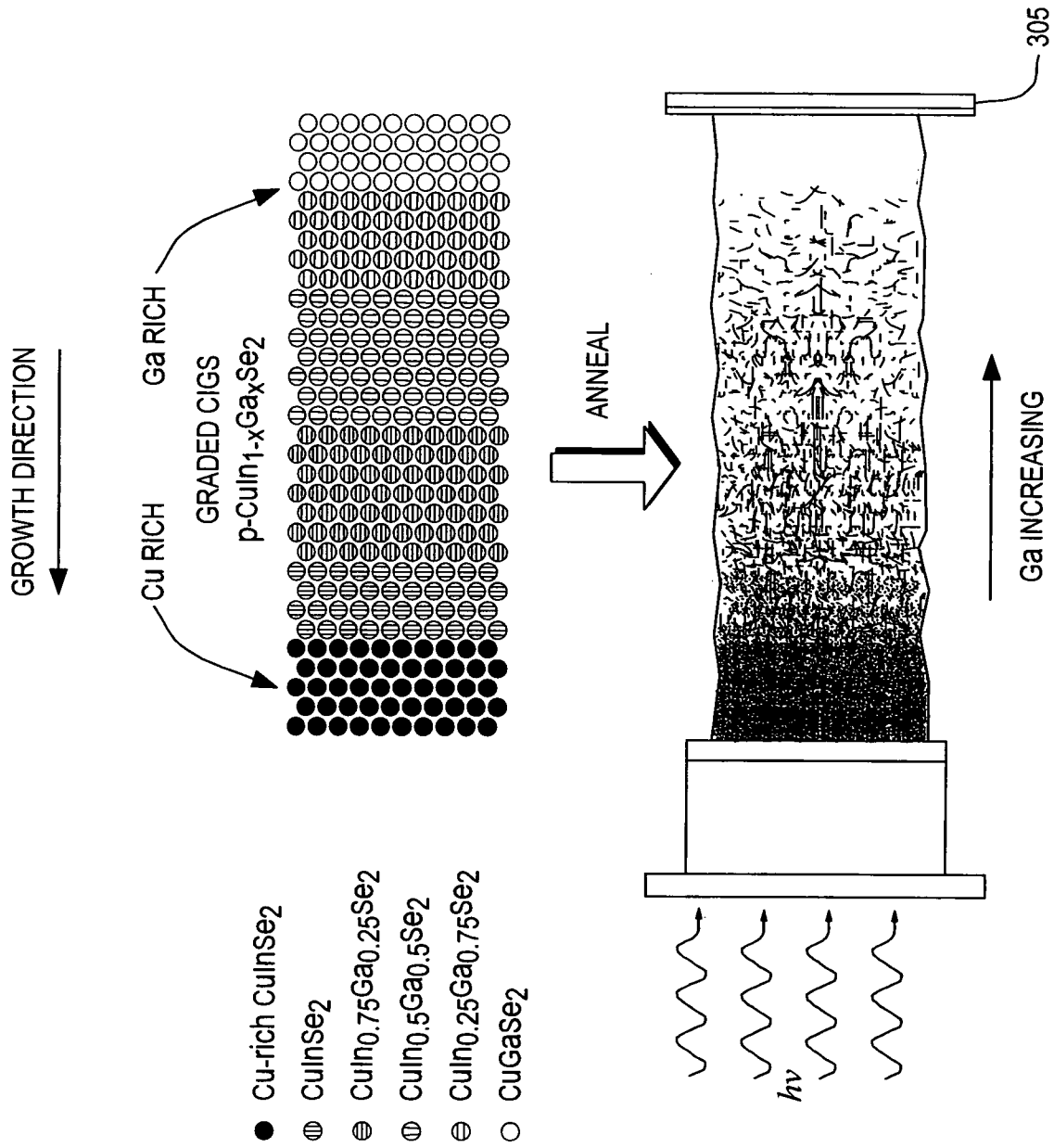


FIG. 3B