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- (73) Patenthaver: **AS Valmieras Stikla Skiedra, Cembu Iela 13, 4201 Valmiera, Letland**
- (72) Opfinder: **PREISS-DAIMLER, Heinz-Jürgen, Wilsdruffer Straße 11, 01723 Wilsdruff, Tyskland**
- (74) Fuldmægtig i Danmark: **Zacco Denmark A/S, Arne Jacobsens Allé 15, 2300 København S, Danmark**
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The invention concerns temperature-resistant aluminosilicate glass fibers and method for the production thereof and use thereof.

There are many inorganic fibers in the high temperature segment. Examples include silica fibers, glass fibers, ceramic fibers, biosoluble fibers, polycrystalline fibers and quartz fibers. Temperature-resistant fibers find use wherever high temperatures need to be controlled. Furthermore, fire protection in buildings is one area of application. Besides use in large industrial foundry facilities for metallic ores, steel and aluminum production, and industrial furnaces, one also finds temperature-resistant glass fibers increasingly in areas such as household appliances, the automotive industry, as well as the aerospace industry.

In modern high tech applications, besides the function of thermal insulation and/or isolation, fibers are also increasingly playing an important role in the reinforcement of plastics and concrete. The reinforcement fibers used here must have high tensile strength, along with their functionalized surface for better binding to their surrounding medium.

Many fiber materials are further processed by textile methods such as those for yarn, twine, weaves and other fabrics. Here as well, the mechanical parameters are of great importance, since these products are used primarily for reinforcement.

Temperature-resistant mineral fibers consist predominantly of the oxides SiO_2 , Al_2O_3 and CaO with weight fractions of SiO_2 over 40 wt. %. Depending on their area of application, they can be specifically modified in their chemical composition by the addition of alkaline and alkaline earth oxides (such as Li_2O , Na_2O , K_2O , MgO , CaO) and transitional metal oxides (such as TiO_2 , ZrO_2 and Y_2O_3). One distinguishes roughly between aluminum silicate fibers or RCF (refractory ceramic fiber), high-temperature glass fibers, AES (biosoluble fibers), polycrystalline fibers made through sol-gel processes, and silicate fibers.

For the production of glass fibers, one uses glass raw materials, recycled glass, volcanic stone or lime, with the designations indicating the raw material base. The melts of glass and stone mixtures are processed via fiber formation equipment into fibers with a diameter of 5 to 30 μm , with basically four different methods for the production of glass fibers. The filaments are bundled into a hundred or more and drawn onto a drum as so-called spin threads.

In the nozzle drawing process, the homogeneously melted glass mass flows continuously through hundreds of nozzle holes of a platinum nozzle vat. By utilizing gravity and drawing force, glass fibers are produced with a diameter of 5 to 30 μm . Thanks to gravitation, the quantity of

replenishing glass melt remains constant, and by varying the rate of drawing the diameter of the glass filament can be controlled. The emerging filaments are cooled down under the action of convective cooling or water cooling and wound onto a drum. Before the winding process, the filaments are coated.

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In the rod drawing process, several glass rods with a diameter of 2 to 8 mm are clamped together and the lower end is heated by a torch flame until it softens. The viscous glass melted at the lower end of the glass rod is drawn into a glass thread by gravity force and drawing force. Glass fiber fleece and textile glass yarns are made preferably by the rod drawing process.

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In the centrifugal process, the glass melt is broken up into mineral fibers by means of centrifugal force under the action of an air current, which are collected as raw felt in collection chambers or gravity shafts.

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With the nozzle blowing method, very fine and short glass fibers can be obtained. The glass melt here is pushed at high pressure and speed of up to 100 m/s through nozzles at the bottom of the melt vat. The fibers are broken up into short pieces here.

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The naturally brittle glass when drawn out into a thin thread has a high flexibility and tensile strength at room temperature. Unlike aramide fibers or carbon fibers, the glass fibers are characterized by an amorphous structure. As with compact window glass, the molecular orientation is chaotic. A glass can therefore be viewed as a congealed liquid. After passing a certain temperature, known as the glass temperature or transformation temperature (T_g), a decoupling of the networks occurs, such that every glass undergoes a change in its shape stability. In this process, entirely or partially amorphous regions change into a rubber-elastic and highly viscous state. Above the transformation temperature, the strength and rigidity of amorphous glass fibers drop significantly.

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The person skilled in the art understands by the term "transformation temperature" (T_g), by definition, the temperature used to characterize the position of the transformation region of a glass. The transformation temperature is a boundary between the brittle-elastic behavior of a solidified glass and the viscoplastic behavior of softened glass. The transformation temperature on average lies at a viscosity of $10^{13.3}$ dPa·s and can be determined per DIN ISO 7884-8:1998-02. The transformation region thus forms the transition from the elastic-brittle behavior to the highly viscous fluid behavior of the glass. The change in length of a glass is greater above the so-called transformation region, whose mean value is characterized by the transformation point T_g , than below it.

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As a result, glass types can only withstand mechanical stresses below the transformation temperature, since they are highly viscous and fluid above the transformation temperature. For products which must have an elevated temperature resistance, there is thus an enormous demand for glass fibers characterized by a high transformation temperature.

WO 96/39362 and DE 2 320 720 A1 describe glass mixtures free of boric acid and fluorine for the production of glass fibers, so that the environmental burdens are minimized as compared to the production of glass fibers based on E glass. In order to still achieve the properties, the melting and the processing conditions of E glass types, a high fraction of MgO is added to the glass mixture as a substitute for the oxides CaO or TiO₂ of at least 2.0 wt. %. Yet due to the high fraction of MgO, such glass compositions have a strong tendency to form mixed crystals, so that the resulting glass types have a coarse crystalline structure. The poor chemical and thermal resistance as well as the tendency to stress cracks are a drawback with these glass types.

US 3,847,627 A discloses a glass composition with a large CaO content in the range of 17 to 24 wt. % and a MgO content in the range of 1.5 to 4.0 wt. %, whose fiber formation temperature lies at least at a temperature of 1228 °C. No values for the transformation temperature are to be found in this document.

From EP 2 321 231 A1 there are known high temperature and chemically resistant glass fibers based on a low fraction of Fe₂O₃, but an alternative addition of Cr₂O₃, having a good light transmission/refraction index. The temperature resistance of the described glass composition is at around 760 °C. The temperature resistance is not satisfactory for a number of applications. A further drawback is the fiber formation temperature needed for the production of glass fibers, being over 1270 °C.

At present, two types of glass fibers are known commercially whose temperature resistance is already substantially above the transformation temperature of 760 °C.

First of all are the so-called S-glass fibers or HM-glass fibers, which are characterized by a high strength and a high E modulus and therefore can be used for the reinforcement of structural parts subject to rather high requirements for their strength and especially their rigidity. As a drawback, for some glass types very pure and costly oxides are used instead of the usual glass raw materials, and at the same time the high melting temperatures of this oxide mixture at around 1700 °C causes increased corrosion of the glass melting vats and their component parts. A

heightened corrosion on the one hand shortens the service life of the glass melting vat and on the other hand causes worse glass quality, so that special melting methods are required.

5 In order to achieve economically attractive lifetimes of the component parts of melting vats, the melt temperature of a glass composition should be below 1400 °C. However, the glass compositions presently known have the drawback that, when the melt temperature is lowered, the characteristic transformation temperature for the temperature resistance of a glass is also lowered.

10 On the other hand, chemically after-treated temperature-resistant glass fibers are known that are made from both E glass and also from special glass fibers. The special glass fibers prior to the chemical treatment consist primarily of SiO₂ and Na₂O. In additional steps, certain oxides (Na₂O) are entirely or partly extracted from the glass fibers over a lengthy time in hot acid, after which they are neutralized, chemically after-treated and finished. Such after-treated glass fibers can be
15 stressed up to a temperature of 1000 °C. Such glass types are costly to produce, due to the complex manufacturing process.

Thus, there continues to exist an elevated demand for temperature-resistant aluminosilicate glass fibers with improved properties. In particular, there is a need to provide temperature-resistant
20 aluminosilicate glass fibers which fill the gap in terms of their temperature resistance between the conventional C, E and ECR glass types and the costly chemically after-treated glass types on the one hand, and which can be stressed up to a temperature of 1000 °C, on the other hand.

Therefore, the problem which the invention proposes to solve is to provide a temperature-resistant aluminosilicate glass fiber which is characterized by a transformation temperature of
25 >760 °C, while the melt temperature (T_s) and the fiber formation temperature (T_f) as well as the liquid temperature (T_l) are as low as possible. For emission protection reasons, the use of boron and fluorine compounds is to be avoided.

30 According to the invention, the problem is solved by a temperature-resistant aluminosilicate glass fiber according to claim 1. In regard to the particular oxide, a fraction of 0 wt. % means that the oxide may be present with a fraction below the limit of detection. Impurities related to the raw materials or the process technology are excluded from this.

35 The temperature-resistant aluminosilicate glass fiber consists of a composition free of boric acid, which is melted without the addition of raw materials containing boroxide.

Surprisingly, it has been found that the amorphous SiO_2 network of the aluminosilicate glass fibers can be influenced specifically by doping with strontium and/or copper and/or zirconium atoms, which results in a change in the physical parameters of the material, especially the transformation temperature (T_g), melt temperature (T_s) and fiber formation temperature (T_f). The mentioned weight fractions of these oxides have proven to be especially suitable for enhancing mechanical characteristics (such as tensile strength, modulus of elasticity, elasticity, elongation, breaking strength, flexibility, etc.) of the glass fibers of the invention as compared to the glass fibers known from the prior art (E glass, ECR glass and C glass).

Upon cooling of the melt, the doping of the amorphous SiO_2 network with foreign ions demonstrably hinders the transition from the metastable amorphous modification to the energy-favored crystalline modification. Surprisingly, dopings with network transformers such as strontium and/or copper and/or barium atoms have proven to be especially advantageous for this.

By a doping of the SiO_2 network of known glass compositions with the mentioned network transformers, T_g can be increased to over 760°C , while at the same time T_s and T_f can be lowered or kept constant. Thanks to the chosen composition, such a glass melt is suitable for the production of continuous glass fibers at low temperature.

The addition of ZrO_2 increases the transformation temperature higher than Al_2O_3 , but at the same time it raises the melt temperature.

Surprisingly, it has been found that the transformation temperature is hardly influenced by the oxides CaO , SrO and BaO , while the oxides SiO_2 , Al_2O_3 , MgO , ZrO_2 and TiO_2 increase the transformation temperature. On the other hand, the oxides Na_2O , K_2O and CuO even in small amounts very substantially lower the transformation temperature.

Furthermore, it was found that the oxides SiO_2 , Al_2O_3 and ZrO_2 raise the melt temperature T_s and the fiber formation temperature T_f . By contrast, the oxide Fe_2O_3 , which gets into the glass without influence via the raw materials, lowers both the transformation temperature as well as the melt temperature T_s and fiber formation temperature T_f .

The addition of TiO_2 raises the transformation temperature and lowers the fiber formation temperature and melt temperature.

On the other hand, an added fraction of CuO contributes to a lowering of T_s and T_f .

ZrO₂ at the expense of SiO₂ raises T_g, as well as the melt and fiber formation temperature.

The glass fibers of the invention can be present both in the form of filaments and in the form of staple fibers.

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The fiber diameter of the glass fibers of the invention is preferably 5 - 30 μm, especially preferably 5 - 25 μm.

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In one embodiment of the invention, the composition of the aluminosilicate glass fibers of the invention has the following fractions (in terms of the overall composition) of oxides:

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52 - 60 wt. %	SiO ₂
14 - 16 wt. %	Al ₂ O ₃
< 0.4 wt. %	Fe ₂ O ₃
0.03 - 0.3 wt. %	Na ₂ O
0.3 - 0.7 wt. %	K ₂ O
20 - 22 wt. %	CaO
0.4 - 0.8 wt. %	MgO
1 - 5 wt. %	TiO ₂
0.5 - 3 wt. %	BaO
0 - 2 wt. %	SrO
0 - 3 wt. %	ZrO ₂ ,
0 - 1 wt. %	CuO

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wherein the total fraction of the alkaline earth metal oxides (Na₂O and K₂O) is at most 1.0 wt. % in total,

wherein the total fraction of the oxides SrO, CuO, ZrO₂ lies in a range of 0.1 to 4.0 wt. %, and

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wherein the temperature-resistant aluminosilicate glass fiber has a transformation temperature >760 °C and a fiber formation temperature (viscosity of 10^{3.0} dPa·s) <1260 °C, preferably ≤1230 °C.

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The aluminosilicate glass fiber according to the invention has the following properties after its production:

a) a transformation temperature >760 °C,

- b) a fiber formation temperature $<1260\text{ }^{\circ}\text{C}$, preferably $\leq 1230\text{ }^{\circ}\text{C}$,
- c) a melt temperature $<1400\text{ }^{\circ}\text{C}$.

Surprisingly, it has been found that the initial tear strength of the glass fibers according to the invention and the fabric made from them after their production is around 15% above the initial tear strength of the E glass types and ECR glass types known in the prior art.

Especially advantageous is the remaining residual strength (relative residual tear strength) of the glass fibers of the invention with a diameter in the range of 9 to 15 μm and the fabric made from them after a temperature stress of $760\text{ }^{\circ}\text{C}$ in the range of 10% to 15% compared to the initial tear strength at room temperature.

Strength is a material property which describes the mechanical resistance which a material presents to a plastic deformation. According to the invention, strength refers to the tensile strength. The tensile strength is the highest resistance of the glass fiber to a tensile stress without breaking. The tensile strength and elongation at maximum force are measured in a pull test, which is familiar to the skilled person.

By definition, the residual tear strength is the remaining tear strength of a glass fiber or a fabric made therefrom after a thermal or chemical stress. The remaining residual tear strength (relative residual tear strength) after the thermal or chemical stressing of a glass fiber or a fabric made therefrom can be indicated as a percentage with regard to the initial tear strength of the glass fiber or the fabric.

The residual tear strength of a glass fiber or a fabric made therefrom is determined before and after a temperature stress by clamping it in a suitable tear testing machine and under the action of a constant rate of feed until the glass fiber or fabric is torn.

For the temperature treatment, test fabric strips (5 x 30 cm) are exposed to a constant temperature for 1 h in a thermal cabinet. After cooling, the tear strength is determined by measuring the force in Newtons and the change in length in millimeters of this test fabric.

The initial strength of the test fabric without thermal stress and the tear strength of the heat-treated test fabric are determined. The relative residual tear strength is found from the percentage ratio of the tear strength of the heat-treated test fabric to the initial strength of the non-heat-treated test fabric.

Surprisingly, moreover, it has been found that the aluminosilicate glass fibers with the composition of the invention, containing the oxides SrO, ZrO₂, and/or CuO, have a good resistance to alkali.

5 Methods for determining the alkali resistance of glass fibers are quite familiar to the skilled person and can be found in corresponding guidelines, such as ETAG 004 (External Thermal Insulation Composite Systems with Rendering – Edition 08/2011 – long-term determination) or DIN EN 13496:1999-06 (short-term determination).

10 Fabrics of aluminosilicate glass fibers of the composition according to the invention advantageously have a residual tear strength of at least 70% after a short-term alkali treatment (per DIN EN 13496:1999-06) and at least 65% after a long-term alkali treatment (per ETAG 004).

15 It has been found that Na₂O and K₂O are water soluble oxides, which contribute to an unwanted lowering of the transformation temperature T_g. In the preferred embodiment of the invention, the glass composition according to the invention has the alkaline earth oxides Na₂O and K₂O together with a maximum combined fraction of 1.0 wt. %. Preferably, the glass composition of the invention has the alkaline earth oxide Na₂O with a maximum fraction of 0.25 wt. %.

20 However, as a complicating factor, it has been found that most oxides react with each other and thus the effects of individual oxides in the glass composition of the invention are very greatly dependent on their fraction. An especially preferred glass composition of the aluminosilicate glass fibers of the invention is therefore characterized in that the fraction of SiO₂ (in terms of the overall composition) lies in a range of 54.0 to 58.0 wt. %.

25 A glass composition of the aluminosilicate glass fiber of the invention has a fraction of Al₂O₃ in the range of 14.0 and 16.0 wt. % and a fraction of CaO in the range of 20.0 to 22.0 wt. %.

30 In the same context, the glass composition according to the invention has the required oxides MgO and Fe₂O₃, preferably with a fraction of MgO in the range of 0.5 to 0.8 wt. % and of Fe₂O₃ at a maximum of 0.3 wt. %.

In an especially advantageous embodiment of the invention, the glass composition of the invention has the oxides TiO₂ and BaO combined with a total fraction in the range of 4.0 to 6.0 wt. %.

35 Glass fibers according to the invention with an especially preferred glass composition have a transformation temperature of at least 765 °C, most especially advantageously of at least 770 °C.

Thanks to the high transformation temperature, the glass fibers of the invention can especially advantageously withstand higher stresses.

At the same time, the glass compositions according to the invention can be economically melted
5 and formed into glass fibers.

A temperature stressing of glass essentially results in formation of defects in the SiO₂ network. This structural damage to the SiO₂ network remains intact after the cooling to room temperature.

10 Thanks to the composition of the oxides according to the invention, the glass filaments obtained from the melt after a temperature stressing of 760 °C are characterized by a remaining tear strength which is equal to or higher than the tear strength of E glass, ECR glass and C glass after the same temperature stress.

15 The temperature-resistant aluminosilicate glass fibers according to the invention after a temperature stress of 760 °C have less structural damage to the SiO₂ network than the glass fibers known from the prior art (E glass, ECR glass and C glass). The aluminosilicate glass fibers according to the invention are therefore characterized after a temperature stress of 760 °C by a remaining tear strength of at least 10% with respect to the initial strength (initial tear strength) at
20 room temperature without temperature stress.

The glass fibers of the invention can be present both in the form of filaments and in the form of staple fibers.

25 The invention also concerns a method for the production of a temperature-resistant aluminosilicate glass fiber which has the following steps:

a. Preparation of a glass melt according to claim 1

30 b. Converting the melt into filaments or staple fibers,
c. Cooling the resulting filaments or staple fibers,
d. Coiling the filaments into spin threads or making textiles,
e. Drying of the resulting filaments or staple fibers or textiles.

35 The method according to the invention has the advantage that temperature-resistant glass fibers are produced wherein the residual strength of the threads and fabrics after a temperature stress of 760 °C is still 10% with respect to the initial strength at room temperature.

Now, it has also been shown to be advantageous that the residual tear strength of the glass fibers of the invention with a diameter in the range between 9 and 15 μm and of the fabric made from them after a temperature stress of 760 $^{\circ}\text{C}$ is in the range between 10% and 15% with respect to the initial tear strength at room temperature.

The invention has the further advantage that the melt temperature (T_s), the liquidus temperature (T_l) and the fiber formation temperature (T_f) are lowered for an economical production and a stable process in the fiber manufacturing.

Thus, the glass composition according to the invention has the following properties:

- a) a transformation temperature $>760^{\circ}\text{C}$,
- b) a fiber formation temperature $<1260^{\circ}\text{C}$,
- c) a melt temperature $<1400^{\circ}\text{C}$.

Surprisingly, it has been found that thanks to the fraction of SrO according to the invention the viscosity of the glass melt is lowered at high temperatures for T_s and T_f , and thus the flow behavior (rheology) of the glass melt is advantageously improved.

It has been found surprisingly that the fraction of TiO_2 according to the invention lowers the melt temperature of the glass composition. Moreover, TiO_2 , SrO and CuO act advantageously as a flux at higher temperatures, which increases the viscosity of the glass composition in the low temperature region (transformation region T_g). Too high a fraction of TiO_2 appears to be disadvantageous, as it supports the unwanted crystallization.

In one especially preferred embodiment of the invention, the glass composition according to the invention has TiO_2 with a fraction of 1 to 5 wt. %, most preferably 2.5 to 3.5 wt. %.

Preferably, the glass melt according to the invention has the alkaline earth oxide Na_2O with a maximum fraction of 0.25 wt. %.

An especially preferred composition of the glass melt of the invention is therefore characterized in that the fraction of SiO_2 (in terms of the overall composition) lies in a range of 54.0 to 58.0 wt. %.

The composition of the glass melt according to the invention has a fraction of Al_2O_3 in the range of 14.0 and 16.0 wt. % and a fraction of CaO in the range of 20.0 to 22.0 wt. %.

5 The glass melt according to the invention has the required oxides MgO and Fe_2O_3 , preferably with a fraction of MgO in the range of 0.5 to 0.8 wt. % and of Fe_2O_3 at a maximum of 0.3 wt. %.

10 In an especially advantageous embodiment of the invention, the composition of the glass melt according to the invention has the oxides TiO_2 and BaO combined with a total fraction in the range of 4.0 to 6.0 wt. %.

Above the liquidus temperature (T_l), the glass is completely melted and there are no longer any crystals.

15 The fiber formation temperature (T_f) is the temperature of a glass melt at which the viscosity of the melt is 10^3 dPa·s. A low T_f simplifies the drawing process for converting the melt into filaments. At this viscosity, the stress during the fiber production is the lowest, which increases the strength of the fiber. Furthermore, less energy is required and thus the production costs can be kept low.

20 According to the invention, an oxide blend is prepared which is heated in a melting vat by means of gas and/or electric melting until liquefied. After this, the homogeneous glass melt is converted into glass filaments or staple fibers.

25 After the complete melting of the mixture and the homogenization of the glass melt, the glass melt is purified before being converted into filaments. The purification serves to drive out and reduce the gas fractions from the glass melt. Additives for the purification are often prescribed and therefore are basically known to the skilled person. Thus, besides ammonium nitrate, one generally adds sodium nitrate or sodium sulfate for the purification of the glass melt.

30 Surprisingly it has now been found that the addition of BaO does not affect the transformation temperature, but can advantageously lower the temperatures T_s and T_f .

35 In one especially preferred embodiment of the method of the invention, when preparing the glass melt one adds instead of sodium sulfate or sodium nitrate a fraction of the total fraction of BaO as barium sulfate with a fraction of 0.4 wt. %. Advantageously, the adding of barium sulfate serves as a purification agent.

The converting of the melt into filaments occurs by the nozzle drawing method, wherein the filaments are cooled as they emerge from the nozzles. The dissipation of heat is preferably done by convective and/or water cooling.

- 5 Due to the high drawing speeds acting on the glass threads emerging from the nozzles during the converting of the glass melt into glass filaments, a glass structure is formed which is especially prone to near-surface defects (such as Griffith cracks).

- 10 According to one embodiment of the method of the invention, the glass filaments obtained from the glass melt are therefore treated with a size after the cooling process, which can repair or close up the near-surface defects. The elimination of near-surface defects hinders the propagation of open structures, which reduces the cracking tendency of the glass filaments. The strength of the material is also increased by the sizing of the glass filaments.

- 15 The main purpose of the sizing is to protect the glass fibers for the later process steps. Glass fibers according to the invention and their products (such as fabrics) which are not desized are already provided by the size with bonding agents for the respective applications.

- 20 More coarse textiles from direct rovings have a size which is compatible with the matrix. For this reason, these textiles are not desized.

- 25 Textiles of finer threads normally have a size of predominantly organic, partly fatty substances, which need to be removed. The removal of the size is done by heat treatment at temperatures over 400 °C. After this desizing, another substance is deposited on the textile which is compatible with the particular matrix. The loss of strength is low in the case of textiles made from temperature-resistant aluminosilicate glass fibers which are thermally desized and provided with a finish.

- 30 According to one embodiment of the method of the invention, the size preferably contains inorganic substances, such as silanes or substances from sol-gel methods. A silane size or sol-gel sizing can be carried out in the production process at glass fiber temperatures up to 100 °C.

Glass threads which have been treated with a silane size are distinguished by a higher strength than glass threads which were treated with a size without silanes.

- 35 Finally, the present invention concerns the use of temperature-resistant aluminosilicate glass fibers as are described by the invention.

The temperature-resistant aluminosilicate glass fibers according to one preferred embodiment of the invention find uses in the production of high-tensile glass fibers, twine, fleece, fabric or textiles, or fabric for catalysts, filters or other fiber products.

- 5 The temperature-resistant aluminosilicate glass fibers can be texturized for the use of the temperature-resistant aluminosilicate glass fibers of the invention as fabrics for catalysts, for example.

- 10 Preferably, moreover, the temperature-resistant aluminosilicate glass fibers of the invention find use in the production of textiles, where the textiles consist of temperature-resistant aluminosilicate glass fibers that are thermally desized after the weaving and treated with a finish, and have a low strength loss.

Sample embodiment 1:

With the aid of the following sample embodiments, the invention will be explained more closely:

- 5 In order to illustrate the influences of the fractions of the oxides SrO, CuO, ZrO₂ according to the invention on the transformation temperature and the melt temperature, the six following glass melts were produced, having identical fractions of Fe₂O₃, Na₂O, K₂O, CaO, MgO, TiO₂ and BaO in their composition.
- 10 The following table 1 shows a summary of the currently used chemical compositions of aluminosilicate glass fibers (reference glass types) as compared to the chemical composition of the temperature-resistant aluminosilicate glass fibers according to the invention (glass No. 1-6). All information is in wt. %.

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Table 1: influence of the oxides on the temperature parameters of glass types

Components	Sample embodiments, glass No.:						Reference glass types		
	1	2	3	4	5	6	ECR	E glass	C glass
SiO ₂	48.6	48.6	48.6	46.6	48.6	46.6	59.2	53.4	68.3
Al ₂ O ₃	15.6	15.6	13.6	15.6	13.6	15.6	14.1	14.8	2.8
Fe ₂ O ₃	0.3	0.3	0.3	0.3	0.3	0.3	0.3	0.2	0.1
Na ₂ O	0.0	0.0	0.0	0.0	0.0	0.0	0.1	0.2	13.7
K ₂ O	0.6	0.6	0.6	0.6	0.6	0.6	0.6	0.2	3.0
CaO	19.9	19.9	19.9	19.9	19.9	19.9	22.8	22.4	5.4
MgO	0.6	0.6	0.6	0.6	0.6	0.6	0.3	0.1	5.0
TiO ₂	7.9	7.9	7.9	7.9	7.9	7.9	2.6	0.2	0.2
BaO	0.5	0.5	0.5	0.5	0.5	0.5	0.0	0.0	0.0
SrO	6.0		6.0	6.0			0.0	0.0	0.0
CuO		6.0			6.0	6.0	0.0	0.0	0.0
ZrO ₂			2.0	2.0	2.0	2.0	0.0	0.0	0.0
B ₂ O ₃							0.0	8.5	1.5
T _g	775	767	783	785	775	778	753	675	527
T _f	1125	1169	1236	1232	1180	1112	1300	1198	1180
T _s	1358	1246	1375	1355	1261	1245	1459	1344	1426

- 20 The glass blends for the glass types per table 1 are heated until liquid in a melting vat. Using the force of gravity and pulling force, glass threads are created with a nozzle drawing method and wound onto a rotating spool. For cooldown, the glass fibers emerging from the nozzles are treated by means of convective and water cooling.

- 25 The transformation temperature is the boundary between the brittle-elastic behavior of solidified glass and the viscoplastic behavior of softened glass. On average, it lies at a viscosity of 10^{13.3}

dPa·s and was determined per DIN ISO 7884-8:1998-02 at the intersection of the lines of tangency traced at the legs of the inflected curve of elongation.

5 It follows from table 1 that the fractions of the oxides have an influence on the temperature parameters (T_g , T_f and T_s) of the individual glass fibers. As compared to the reference glass types, all experimental glass types of the invention have a higher T_g , while T_g is greater than 760 °C. At the same time, T_s and T_f of the experimental glass types of the invention are lowered on average by 100 °C and 50 °C.

10 As regards the experimental glass types of the invention among themselves, a fraction of 6 wt. % of SrO leads to an increasing of T_s , T_f and T_g . On the other hand, an added fraction of 6 wt. % of CuO contributes to a lowering of T_s and T_f . A fraction of 2 wt. % of ZrO_2 at the expense of the fraction of SiO_2 leads to a raising of T_g , while the temperature parameters T_f and T_s are decreased by the fraction of CuO. TiO_2 acts like SrO, increasing T_g and decreasing T_f and T_s .

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Sample embodiment 2:

In order to illustrate the influences of the fractions of the oxides SrO, CuO, ZrO_2 according to the invention on the transformation temperature and the fiber formation temperature, the seven
20 following glass melts were furthermore prepared. Table 2 gives the corresponding compositions for the glass types with numbers 8 to 13. As purifying agent, only barium sulfate was added to the glass melts with a fraction of 0.4 wt. % in terms of the total fraction of BaO.

25 Table 2 shows the chemical glass compositions of three commercially available aluminosilicate glass fibers (reference glass types) as compared to the seven sample glass compositions of the temperature-resistant aluminosilicate glass fibers according to the invention (glass No. 7-13). All information is in wt. %.

30 The adding of ZrO_2 (0.3 wt. % in glass No. 8) increases T_g , but at the same time also leads to a raising of T_f . Thanks to the addition of SrO (4.0 wt% in glass No. 10), T_s can be significantly lowered to 1363 °C, while at the same time T_g rises slightly. If both oxides are used in combination (see glass No. 11 and 12), their effects on T_g , T_f and T_i depend on the respective total fraction of the glass composition, while the adding of CuO (0.1 wt. % in glass No. 13) permits a fine tuning of the characteristic temperatures.

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Moreover, glass No. 11 contains TiO_2 with a total concentration of 8.3 wt. %, which increases T_g and at the same time lowers the melt and fiber formation temperature.

Table 2: influence of the oxides on the temperature parameters of glass types

Components	Sample embodiments, glass No.:							Reference glass types		
	7	8	9	10	11	12	13	ECR	E glass	C glass
SiO ₂	58.3	56.0	58.2	52.5	46.0	54.5	54.5	59.2	53.4	68.3
Al ₂ O ₃	13.5	15.2	15.4	15.5	12.0	14.8	14.8	14.1	14.8	2.8
Fe ₂ O ₃	0.3	0.3	0.3	0.3	0.3	0.3	0.3	0.3	0.2	0.1
Na ₂ O	0.0	0.2	0.2	0.0	0.0	0.2	0.2	0.1	0.2	13.7
K ₂ O	0.6	0.7	0.7	0.7	0.6	0.7	0.7	0.6	0.2	3.0
CaO	22.0	22.0	20.0	22.0	20.0	20.4	20.3	22.8	22.4	5.4
MgO	0.6	0.6	0.6	0.5	0.6	0.6	0.6	0.3	0.1	5.0
TiO ₂	2.6	2.6	2.6	2.4	8.3	3.5	3.5	2.6	0.2	0.2
BaO	2.0	2.0	2.0	2.0	0.3	2.0	2.0	0	0	0
SrO				4.0	8.0	1.0	1.0	0	0	0
ZrO ₂		0.3			4.0	2.0	2.0	0	0	0
CuO							0.1	0	0	0
B ₂ O ₃								0	8.5	1.5
T _g [°C]	757	765	760	764	763	772	763	753	675	527
T _f [°C]	1161	1186	1212	1151	1127	1230	1215	1285	1156	1011
T _s [°C]	1449	1421	1482	1363	1350	1450	1448	1458	1345	1425

5 Sample embodiment 3 – determination of the residual tear strength after temperature stress

To determine the initial tear strength, test fabrics in strip form (5 x 30 cm in the warp direction and 5 x 30 cm in the weft direction) are tested in a triple determination on a tear testing machine (Zwick GmbH & Co. KG) with a maximum tearing force of 10 kN with a distance of 10 cm between the clamps and a constant feed rate of 100 mm/min and the average of 3 test fabrics was calculated.

Temperature stress

For the determination of the temperature resistance, the test fabrics in strip form (5 x 30 cm; 9 µm glass threads) are treated in a thermal cabinet at 400 °C for 1 h. The test fabrics are then removed from the thermal cabinet and cooled down to around 20 °C at room temperature.

In keeping with the above, test fabrics are treated each time in strip form (5 x 30 cm; 9 µm glass threads) in a thermal cabinet at 500 °C, 600 °C, 650 °C, 700 °C, 750 °C or 800 °C for 1 h and then cooled down at room temperature to around 20 °C.

The testing of the residual tear strength of the heat-treated and cooled down test fabrics is done similar to the determination of the initial tear strength.

The following table 3 shows the relative tear strength values for the individual temperatures, the initial tear strength being assumed to be 100% and the relative residual tear strengths being calculated [in %] as a percentage of the initial tear strength.

- 5 Test fabrics of E glass and ECR glass served as references.

Table 3: relative residual tear strength [in %] after temperature stress

Glass	400 °C	600 °C	650 °C	700 °C	750 °C	800 °C
E glass	13	8	6	1	-	-
ECR	19	10	10	9	5	-
Glass No. 8	20	15	14	14	11	1

- 10 Table 3 shows that the relative residual tear strength of all three test fabrics decreases with increasing temperature stress (from 400 to 700 °C). While test fabrics of E glass after a temperature stress of 750 °C have no residual strength, test fabrics of ECR glass still have a relative residual tear strength of 5% as compared to the initial tear strength. Furthermore, test fabrics of glass fibers of the composition according to the invention after a temperature stress of
- 15 750 °C still have a relative residual tear strength of 11% and after a temperature stress of 800 °C a remaining relative residual tear strength of 1% as compared to the initial tear strength.

Sample embodiment 4 – alkali resistance

- 20 By analogy with sample embodiment 3, the initial tear strengths of the glass fiber fabrics made from glass fibers of the invention of glass No. 8 (see table 2, sample embodiment 2) were determined at a constant feed rate of (50 ± 5) mm/min. Each time, test fabrics of E glass or ECR glass fibers served as the references.

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Short-term alkali treatment per DIN EN 13496:1999-06

- For the determination of the residual tear strength after a short-term alkali treatment per DIN EN 13496:1999-06, the test fabrics as strips (5 cm x 30 cm; 9 µm glass threads) are dipped into an
- 30 alkaline solution (1 g NaOH, 4 g KOH, 0.5 g Ca(OH)_2 per one liter of distilled water) in the weft direction and kept there for 24 hours at a temperature of (60 ± 2) °C. The determination of the alkali resistance is done each time as a seven-fold determination per test fabric.

- As reference, the respective test fabrics were kept under ambient conditions for at least 24 h at
- 35 (23 ± 2) °C and $(50 \pm 5)\%$ relative humidity.

After being kept in the alkaline solution, the test fabrics are rinsed with running tap water at a temperature of $(20 \pm 5) ^\circ\text{C}$ until the pH value on the surface, measured with a pH indicator paper, is less than pH 9. After this, the test fabrics are kept in 0.5% hydrochloric acid for 1 h. After this,
 5 the test fabrics are rinsed in running tap water without much movement, until a pH value of 7 is achieved, measured by pH indicator paper. The test fabrics are dried for 60 min at $(60 \pm 2) ^\circ\text{C}$ and then kept for at least 24 h at $(23 \pm 2) ^\circ\text{C}$ and $(50 \pm 5)\%$ relative humidity before being tested.

To determine the residual tear strength (see table 4), the test fabrics are clamped in the tear
 10 testing machine and pulled at a constant feed rate of $(50 \pm 5) \text{ mm/min}$ until the test fabric is torn. During the testing, the force is determined in Newtons and the change in length in millimeters.

After the alkali treatment per DIN EN 13496:1999-06, a comparable relative residual tear strength of 75% and 76% was determined for all the test fabrics.

Long-term alkali treatment per ETAG 004

The long-term alkali resistance of the test fabrics (fabrics) is determined by ETAG 004 (Edition 08/2011), section 5.6.7.1.2. For this, the test fabrics are dipped as strips (5 cm x 5 cm; 9 μm glass
 20 threads) with the glass composition according to the invention per glass No. 8 (see table 2) into an alkaline solution (1 g NaOH, 4 g KOH, 0.5 g $\text{Ca}(\text{OH})_2$ per one liter of distilled water) at $(28 \pm 2) ^\circ\text{C}$ in the weft direction for 28 days.

After this, the test specimens are rinsed by five minutes of dipping into an acid solution (5 ml of
 25 35% HCl diluted to 4 liters of water) and then placed in succession into 3 water baths (each one 4 liters). The test fabrics are left in each water bath for 5 minutes.

After this, the test fabrics are dried for 48 hours at $(23 \pm 2) ^\circ\text{C}$ and $(50 \pm 5)\%$ relative humidity. The residual tear strengths found after the alkali treatment are given in table 4. For textile glass
 30 lattices, the residual tear strength must be at least 50% of the initial tear strength.

For test fabric made from the glass fibers of the invention per glass No. 8 (1618.6 N/5 cm), a comparable relative tear strength of 69% was determined as for test fabric of ECR glass (1488.4 N/5 cm or 70%). On the other hand, test fabrics of E glass only exhibited a relative residual tear
 35 strength of 64% as compared to the untreated test fabrics.

The higher initial tear strength of the glass fibers according to the invention as compared to glass fibers of E glass or ECR glass should be pointed out as especially advantageous, as is shown by the comparison of glass No. 8 to E glass and ECR glass.

5 **Table 4:** relative residual tear strength [in %] after alkali stress.

Glass	Initial tear strength [N/5 cm]	Residual tear strength after 24 h, 60 °C DIN EN 13496, [N/5 cm]/[%]	Residual tear strength after 28 d, ETAG 004, [N/5 cm]/[%]
E glass	1986.7	1484.4 / 75	1264.8 / 64
ECR	1952.5	1607.4 / 76	1488.4 / 70
Glass No. 8	2345.5	1769.0 / 75	1618.6 / 69

P a t e n t k r a v

1. Temperaturbestandig aluminosilikat-glasfiber, med følgende sammensætning:

5

52-60 vægt-%	SiO ₂
14-16 vægt-%	Al ₂ O ₃
< 0,4 vægt-%	Fe ₂ O ₃
0,03-0,3 vægt-%	Na ₂ O
0,3-0,7 vægt-%	K ₂ O
20-22 vægt-%	CaO
0,4-0,8 vægt-%	MgO
1-5 vægt-%	TiO ₂
0,5-3 vægt-%	BaO
0-2 vægt-%	SrO
0-3 vægt-%	ZrO ₂
0-1 vægt-%	CuO,

hvor den samlede andel af alkalimetaloxyderne Na₂O og K₂O i sum udgør maks. 1,0 vægt-%,

hvor den samlede andel af oxiderne SrO, CuO, ZrO₂ ligger i et område mellem 0,1 til 4,0 vægt-%, og

10

hvor den temperaturbestandige aluminosilikat-glasfiber har en transformations-temperatur på >760°C og en fiberdannelsestemperatur på <1260 °C,

hvor reststyrken af den temperaturbestandige aluminosilikat-glasfiber med en diameter i området mellem 9 til 15 µm efter en temperaturbelastning på 760 °C ligger i området mellem 10 % og 15 % i forhold til udgangstyrken ved rumtemperatur.

15

2. Temperaturbestandig aluminosilikat-glasfiber ifølge krav 1, **kendetegnet ved, at** denne omfatter Na₂O med en andel på maks. 0,25 vægt-%.

20

3. Temperaturbestandig aluminosilikat-glasfiber ifølge krav 1 eller 2, **kendetegnet ved, at** denne omfatter SiO_2 med en andel i området fra 54,0 til 58,0 vægt-%.

5 4. Temperaturbestandig aluminosilikat-glasfiber ifølge et af kravene 1 til 3, **kendetegnet ved, at** denne omfatter MgO med en andel på 0,5 til 0,8 vægt-% og Fe_2O_3 med en andel på maks. 0,3 vægt-%.

10 5. Temperaturbestandig aluminosilikat-glasfiber ifølge et af kravene 1 til 4, **kendetegnet ved, at** denne foreligger i form af filamenter eller i form af stabelfibre.

6. Fremgangsmåde til fremstilling af en temperaturbestandig aluminosilikat-glasfiber ifølge et af kravene 1 til 5, der omfatter de følgende trin:

15 a. tilvejebringelse af en glassmeltemasse, som har de følgende andele af oxider:

52-60 vægt-%	SiO_2
14-16 vægt-%	Al_2O_3
< 0,4 vægt-%	Fe_2O_3
0,03-0,3 vægt-%	Na_2O
0,3-0,7 vægt-%	K_2O
20-22 vægt-%	CaO
0,4-0,8 vægt-%	MgO
1-5 vægt-%	TiO_2
0,5-3 vægt-%	BaO
0-2 vægt-%	SrO
0-3 vægt-%	ZrO_2
0-1 vægt-%	CuO ,

hvor den samlede andel af alkalimetaloxyderne Na_2O og K_2O i sum udgør maks. 1,0 vægt-%,

20 hvor den samlede andel af oxiderne SrO , CuO , ZrO_2 ligger i et område fra 0,1 til 4,0 vægt-%,

- b. overførsel af smeltemassen til filamenter eller stabelfibre,
- c. afkøling af de tilvejebragte filamenter eller stabelfibre,
- d. skylning af filamenterne eller fremstilling af fladestrukturer,
- e. tørring af de tilvejebragte filamenter eller stabelfibre eller fladestrukturer.

5

7. Fremgangsmåde ifølge krav 6, **kendetegnet ved, at** der ved tilvejebringelsen af glassmeltemassen tilsættes en andel af den samlede andel af BaO som bariumsulfat med en andel på 0,4 vægt-%.

10

8. Fremgangsmåde ifølge krav 6 eller 7, **kendetegnet ved, at** filamenterne og stabelfibrene, der er tilvejebragt ud fra glassmeltemassen, behandles med et slettemiddel.

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9. Fremgangsmåde ifølge et af kravene 6 til 8, **kendetegnet ved, at** slettemidlet indeholder uorganiske substanser.

20

10. Anvendelse af en temperaturbestandig alumosilikat-glasfiber tilvejebragt ifølge et af kravene 1 til 5 eller ifølge et af kravene 6 til 9 til fremstilling af trækfaste glasfibre, snoede garner, non-wovens, tekstiler eller fladestrukturer til katalysatorer, filtre eller andre fiberprodukter.

11. Anvendelse ifølge krav 10, **kendetegnet ved, at** tekstilet består af temperaturbestandige alumosilikat-glasfibre, som er termisk afslettede og er udstyret med en finish.